

Closing the Loop for Benzylic Polyacylhydrazone(PAcHy) Soft Materials

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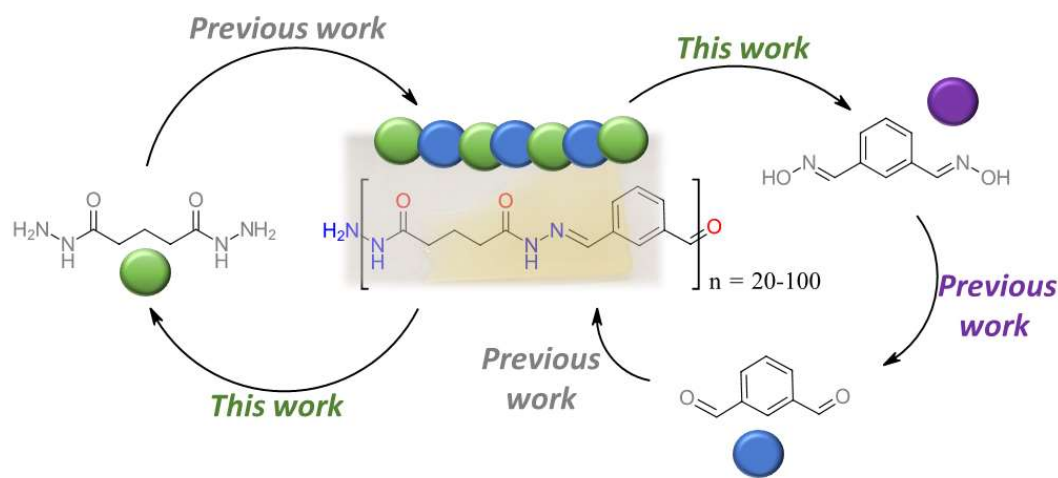
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1 **Abstract**

2 The earth faces a pressing environmental challenge with the annual pollution of millions of tons
3 of plastic waste. This issue is exacerbated by the inherent characteristics of plastics, including their
4 high thermal stability and low biodegradability. In response, the scientific community has dedi-
5 cated significant efforts to combat plastic pollution and develop environmentally sustainable soft
6 materials. A promising approach is the creation of soft materials that can be efficiently recycled
7 within a closed-loop system, thereby minimizing their chemical impact on the environment. This
8 study introduces polyacylhydrazones as a novel class of soft materials with the potential for closed-
9 loop recycling. Closed-loop polymers are sought after as sustainable materials of the future. Dy-
10 namic covalent polymers have the potential to act as closed-loop materials because of their re-
11 versible linkages. However, only a few materials currently exist.¹⁻³ Polyacylhydrazones have been
12 explored as dynamic covalent polymers with multiple uses. Herein we demonstrate the first study
13 that shows their potential as closed-loop materials. We show synthetic techniques that produce

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14 hydrophobic polyacylhydrazones with unique length and morphology under kinetic control. Fi-
15 nally, we show the facile decomposition of a benzylic hydrazone bond down to monomeric units
16 and extraction-based separation, without the need for chromatography. This work advances our
17 understanding of these innovative materials and validates benzylic polyacylhydrazones as “closed-
18 loop” materials for a more sustainable future.
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24 1. Introduction

25 Over the past decade, extensive research efforts have been dedicated to unraveling the far-reaching
26 consequences of plastic pollution^{4,5}. Alarming statistics underscore the severity of the issue, with
27 approximately 29 million tons of plastic waste being disposed of annually in the United States
28 alone^{5,6}. This unrelenting influx of new plastic waste, compounded by the absence of controlled
29 decomposition pathways⁷, has created a crisis that imperils the existence of countless marine spe-
30 cies^{8,9}. Given that marine organisms constitute a substantial portion of our global protein supply,
31 this crisis transcends ecological concerns, extending into the realm of food security and potential
32 humanitarian catastrophes¹⁰. The gravity of the situation is further exacerbated by the role of mi-
33 croplastics, originating from common plastic additives like bisphenol A (BPA)¹¹ and Perfluoroal-
34 kyl Substances (PFAS), in endocrine disruption and the proliferation of estrogen-dependent

35 cancers ¹². To combat the plastic pollution crisis, a coalition of chemists, biochemists, and engi-
36 neers has rallied to develop technologies aimed at mitigating the millions of tons of plastic waste
37 generated annually ^{6,13}. While existing approaches such as catalytic degradation and solvent sepa-
38 ration methods show promise ^{14,15}, a broader array of strategies is required to create future materi-
39 als with minimal environmental impacts ^{16,17}. The emergence of biodegradable materials like pol-
40 ylactic acid (PLA) represents a step in the right direction ¹⁸. However, the proliferation of PLA,
41 without a concurrent diversification of biodegradable plastics, may inadvertently bestow an evo-
42 lutionary advantage upon bacteria capable of digesting them ^{19,20}. This looming challenge under-
43 scores the urgent need for a two-fold approach: **(1)** A diverse array of biodegradable plastics that
44 must be made and dumped in a way that feeds environmental bacteria proportionally; **(2)** The
45 reverse engineering of materials, enabling selective degradation and reusability, thereby achieving
46 a net zero chemical impact on the environment. Consequently, there is a growing interest within
47 the scientific community in the development of materials that mimic plastic properties while being
48 amenable to "closed-loop" recycling. ^{1,21}

49 Previous research achievements have indicated that plastics featuring dynamic covalent bonds pos-
50 sess the inherent potential for "closed-loop" recycling ^{22,23}. Among these dynamic covalent bonds,
51 the acylhydrazone linkage has emerged as a promising dynamic covalent polymer system²⁴. While
52 polyacylhydrazones have more recently gained recognition for their role in the development of
53 crosslinked soft materials and hydrogels ²⁵, our study presents, the pioneering synthesis of hydro-
54 phobic hydrolytically stable polyacylhydrazones through condensation in dimethyl sulfoxide
55 (DMSO). The simplicity of the benchtop synthesis process is elucidated, along with some potential
56 applications of these materials. Most significantly, we unveil the selective conditions conducive to
57 the degradation of these polymers to the monomeric units. We also demonstrate the extraction-
58 based separation of monomer units, closing the loop for benzylic polyacylhydrazones. In this con-
59 text, these findings could help the efforts towards addressing the pressing problem of plastic pol-
60 lution and contribute to the creation of environmentally responsible materials.

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62 2. Results and discussion

63 We began our studies with an interest in creating hydrophobic, water-stable, degradable materials.
64 Polyacylhydrazones were appealing as sustainable polymers because several papers reported the
65 reversibility of the acylhydrazone bond.^{26–28} Polyacylhydrazones have a long history of being used
66 as hydrophilic polymers. These studies demonstrate their robust polymerization along with their
67 hydrolytic stability at neutral pHs. Furthermore, polyacylhydrazones are known to be water-stable
68 from their uses in protein chemistry.²⁹ Therefore, we started our project by targeting a few hydra-
69 zone polymers with potentially hydrophobic properties. Additionally, previous research has shown
70 differences in kinetic exchange between different ketones and aldehydes with hydrazides.³⁰ Ex-
71 change rate difference could prove critical for potential decomposition pathways. Therefore, com-
72 mercially available aliphatic and benzylic diketones/dialdehydes that could produce hydrophobic
73 materials with different decomposition kinetics were chosen (**Figure 1**, Materials A-C). Dialde-
74 hyde monomers (**a**, **b**, and **c**) were coupled with the dihydrazide monomer (**1**) (**Figure 1**). The
75 reaction was then transferred to a silicon mold and dried in the fume hood at 60 °C. As shown in
76 **Figure 1**, the resulting materials gave drastically different physical appearances. **Material A** gave
77 the appearance of a deep-brown material. This is likely due to enolate/enol formation that is com-
78 mon with beta keto aldehydes, esters, and ketones. In contrast, **Material B** had a semi-transparent
79 amber appearance and easily took the shape of the cast when dried. **Material C** formed a less
80 flexible semi-transparent material that also took the shape of its silicon mold. This preliminary
81 screening underscores the profound impact that the choice of dialdehyde monomer can exert on
82 the resulting properties of these polyacylhydrazone block polymer systems. This small group of
83 polymers demonstrated that different materials can be generated just by changing the dialdehyde
84 of these polyacylhydrazone polymer systems. This initial investigation into the synthesis of poly-
85 acylhydrazone-based materials not only underscores the diversity of materials that can be gener-
86 ated within the polyacylhydrazone framework but warrants further exploration into the structure-
87 property relationships of this class of materials.

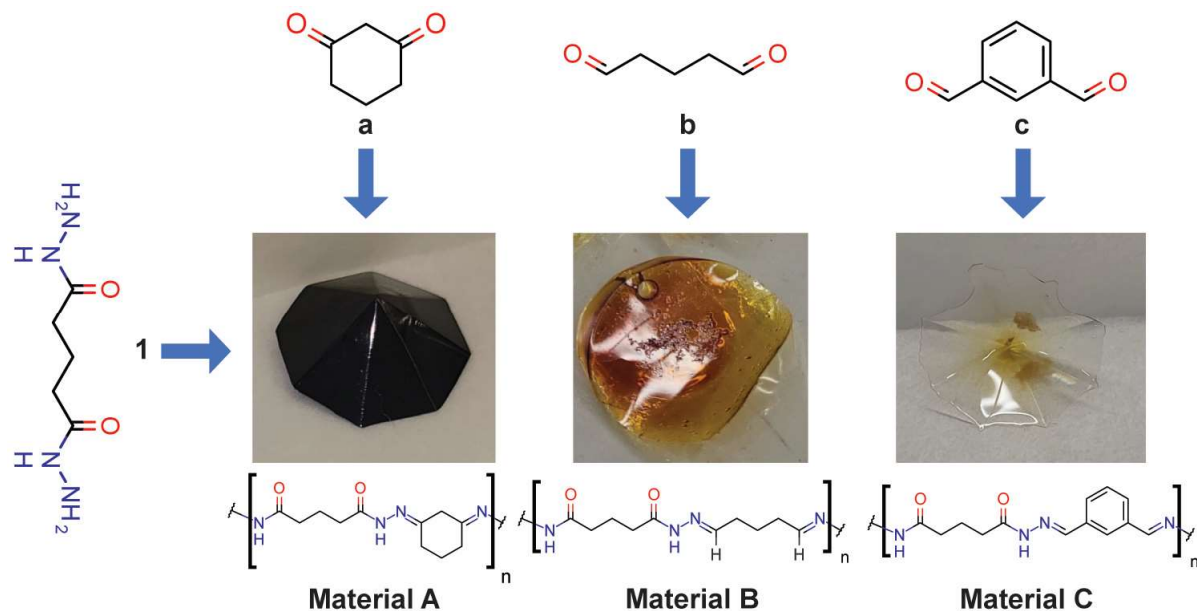


Figure 1. Dihydrazide (**1**) is combined with dialdehydes (**a-c**) at 0.2 M concentration in 10% AcOH in DMSO. Reactions are then solvent casted in silicon molds to form **Materials A-C**.

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90 2.1. Solvent Casting vs. Precipitation

91 We noticed that the material properties we witnessed did not match the powdery materials that
 92 were previously reported in literature.²⁴ In our quest for a deeper understanding of the solvent
 93 casting effect on polymer properties, we embarked on a comparative study, pitting our solvent
 94 casting methodology (performed at 60 °C in a fume hood, (**Figure 2a**) against a precipitation
 95 method (**Figure 2c**) mentioned in the literature²⁴. In our hands the precipitation method aggregated
 96 while drying, creating a porous morphology for the polyacylhydrazones (**Figure 2d**). In contrast
 97 the solvent casting method created a smoother continuous surface morphology (**Figure 2b**). This
 98 comparative analysis not only underscores the profound influence of drying methodology on the
 99 resulting polymer morphology but also hints at the potential role of drying conditions in shaping
 100 the material's physical attributes.

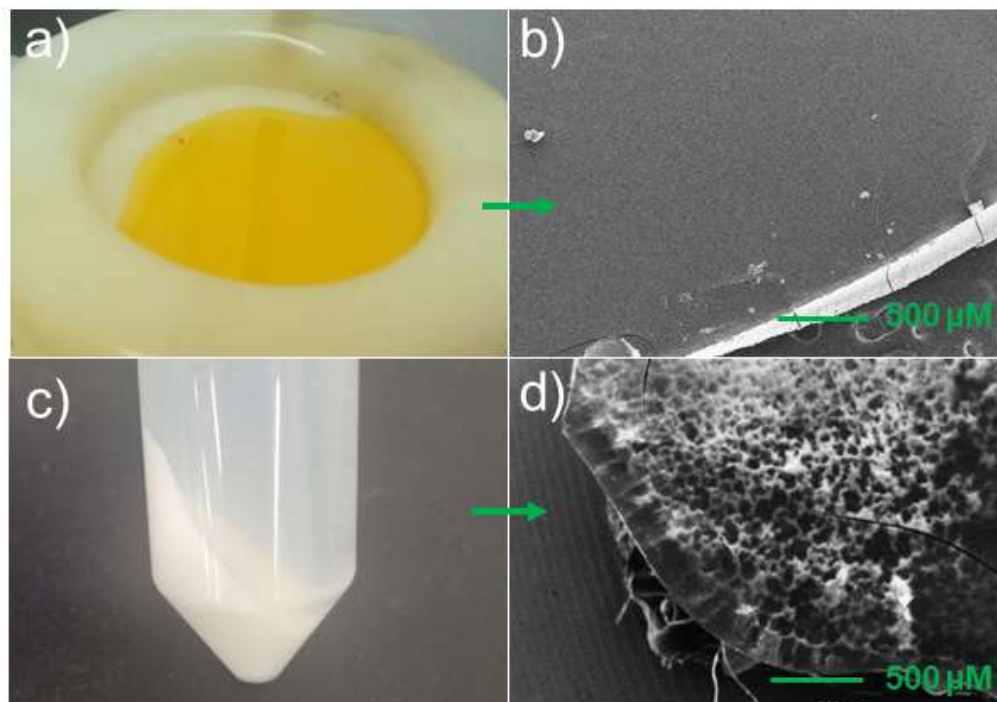


Figure 2. (a) *New Method* - The reaction mixture for polymer **Material C** is condensed on a silicon mold (b) Once dried, the resulting material, **Material C**, is taken from the silicon mold and imaged by SEM (c) *Previous method* - **Material C** is precipitated from the reaction mixture, filtered, and left at room temperature to dry (d) once dried, the precipitated **Material C** is imaged by SEM.

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103 2.2. Effect of Reaction Concentration and DMSO

104 **Materials B** and **C** were previously reported *in situ* in literature.²⁴ However, we noticed that the
 105 plastic and film-like properties that we saw were not mentioned (**Figure 1**). Furthermore, the mor-
 106 phologies appeared to be different due to our drying method (**Figure 2**). We suspected the slow
 107 condensation of the living polymers (**Materials A-C**) in (10%AcOH in DMSO) caused further *in*
 108 *situ* reactivity of the polymers. Furthermore, the remaining concentrations of DMSO (10 % w/w)
 109 resulted in the smooth and flexible materials that were witnessed. Therefore, we investigated the
 110 effect of concentration on polymerization. **Material C** was the only one that could be completely
 111 reconstituted in DMSO or DMF after condensation. As a result, the polymerization of **Material C**
 112 was monitored at several different concentrations by GPC (**Figure 3a**). It was seen that all

113 reactions halted within 24 hours, reaching a kinetic limit at each concentration. Additionally, plot-
114 ting the log of the reaction concentration vs. the M_w of the polymer shows a direct correlation
115 between reaction concentration and polymer size (**Figure 3b**). We hypothesize that as concentra-
116 tion increases, the polymer size increases, and the effective concentration of the reactive ends de-
117 creases. Eventually, the reactive ends are too far away to collide with each other at the reaction
118 temperature and concentration, causing the polymerization to stop. Polymers formed and dried at
119 different concentrations show different molecular weights and appearances (**Figure 3**, inset). This
120 data is important because it suggests that even though the polyacylhydrazone polymerization is
121 reversible and thermodynamically controlled at higher temperatures, the polymer mass can be con-
122 centration-dependent at lower temperatures ($< 60\text{ }^\circ\text{C}$). This discovery suggests that the molecular
123 weight of the polymers can be controlled beyond at lower temperatures.

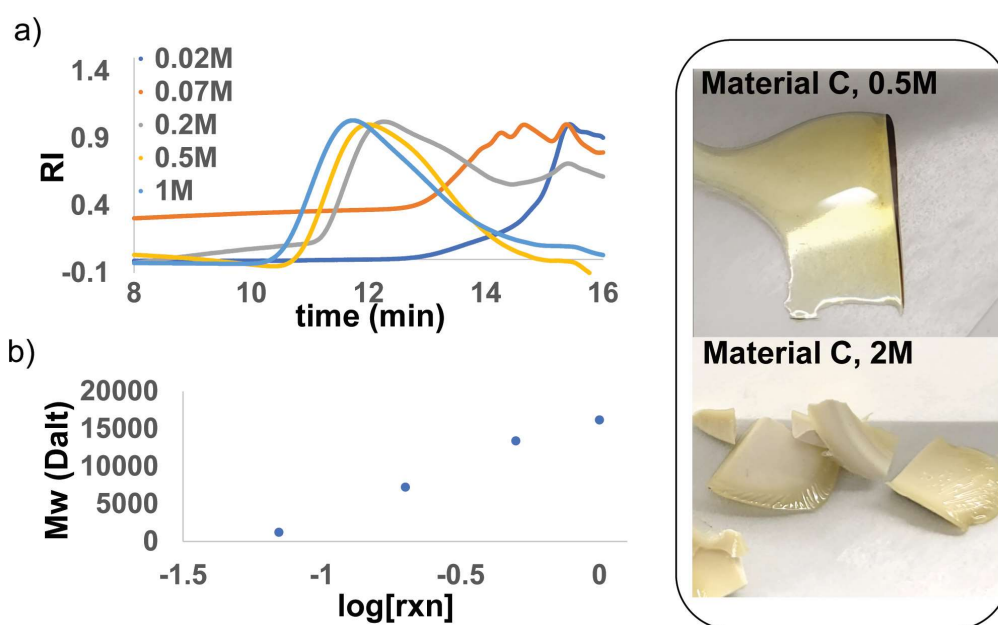


Figure 3. (a) GPC traces of **Material C** under polymerization conditions at different concentrations (b) M_w as determined by GPC is plotted against the log of the reaction concentration ($\log[\text{rxn}]$) (Inset) **Material C** is solvent casted after being synthesized at 0.5M (top) and 2M (bottom) concentrations.

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125 **2.3. End Capping: Controlling molecular mass**

126 Next, our focus shifted towards gaining control of the polymer size by capping the reactive end
127 with a terminal monomer. It is well known that molecular weight and morphology play a critical
128 role in material properties. Therefore, being able to control the polymer size for each material
129 would allow for a wider range of potential properties and applications. We envisioned capping the
130 polymer with a terminal hydrazide to create controlled access to smaller oligomers (**Figure 4a**).
131 Literature suggests that aldehyde and ketone exchange does not readily occur until 100 °C.²⁴ How-
132 ever, hydrazine and hydrazide exchange seemed to readily occur around 50 °C. Therefore, we
133 chose to use benzoic hydrazide, because it could readily exchange with **Materials A-C** at mild
134 temperatures. When incubated with benzoic hydrazide **Materials A** and **B** were not soluble in 10%
135 AcOH/DMSO. Over time **Material A** was partially dissolved, and partially exchanged with ben-
136 zoic hydrazide in 10% AcOH/DMSO. However, **Material B** remained insoluble, barely exchang-
137 ing with benzoic hydrazide. **Material C** was soluble in DMSO. Therefore, the effects of hydrazide
138 end capping could only be assessed by GPC for **Materials A and C**.

139 GPC analysis of **Materials A and C** showed a significant difference in end capping response
140 between materials. The DMSO soluble **Material C** exhibited different molecular weights that were
141 dependent upon the concentration of benzoic hydrazide added (**Figure 4b-c**). **Material A** re-
142 sponded differently to benzoic hydrazide. The solution-soluble components of end-capped **Mate-**
143 **rial A** did not vary in mass depending on benzoic hydrazide concentration (**Figure S2**). Instead,
144 the appearance of a low molecular weight compound ($\leq 1\text{kD}$) only increased with increasing ben-
145 zoic hydrazide (**Figure S2**). This data suggests that once soluble **Material A** quickly breaks down
146 into small molecular weight oligomers. Based upon these results we can conclude that a hydrazide
147 end capping strategy can help to control molecular weight, but the success of this strategy is de-
148 pendent upon the structure of the parent polymer chain. While the terminal hydrazide strategy was
149 effective for controlling the molecular weight of the benzoic polyacylhydrazone **Material C**, it
150 was ineffective for accessing different molecular weights of aliphatic aldehyde and ketone-based
151 polyacylhydrazones, **Materials A and B**.

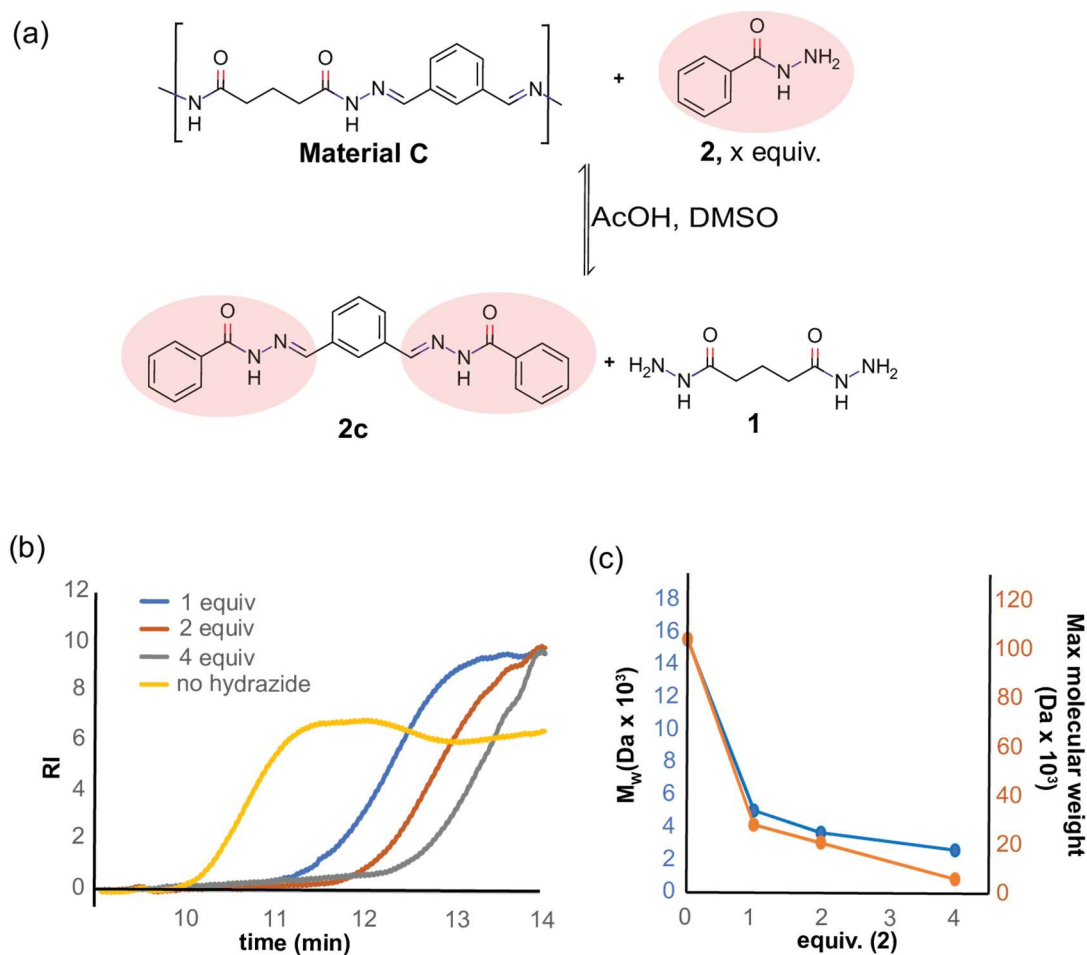


Figure 4. (a) Cartoon of terminal hydrazide exchanging with the dihydrazide 1 in Material C (b) GPC traces of **Material C** after 14 h reaction with **2**. (c) M_w and Max molecular weight (orange) are plotted against equivalents of hydrazide used in the decomposition re-

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153 2.4. Degradation

154 The potential success of terminal hydrazide capping strategies highlights the dynamic nature of
 155 polyacylhydrazones at low temperatures ($< 50\text{ }^\circ\text{C}$) with excess terminal hydrazide. As foremen-
 156 tioned, we were interested in utilizing the dynamic nature of closed-loop recycling. We envisioned
 157 using a terminal nucleophile to degrade these polymers down to monomer units and then separate
 158 them (**Figure 5a**). Previous kinetic studies conducted by the Raines group showed that hydroxyl-
 159 amine is 100 times faster at breaking existing hydrazone bonds than acyl hydrazide.³¹

160 Hydroxylamine is also terminal because the hydroxyl side of the molecule cannot form hydrazone
161 bonds. Therefore, we investigated the degradative capacity of hydroxylamine with **Materials A-**
162 **C (Figure 5a)**. **Materials A-C** were incubated with hydroxylamine HCl with 10% AcOH in
163 DMSO as the solvent and monitored over time by NMR (**Figure 5b**). After 10 minutes of incuba-
164 tion, the NMR peaks for **Material C** became narrower for both the aromatic and aliphatic regions
165 (**Figure 5b-c**). In addition, the oxime peaks appeared around 8.1 ppm (**Figure 5b**). Both the nar-
166 rowing of NMR peaks and the emergence of oxime peaks suggested that depolymerization was
167 occurring.

168 Although **Material C** demonstrated rapid decomposition under DMSO conditions, the decomposi-
169 tions of **Materials A and B** were not as fast. **Material A** completely dissolved over time (48 h).
170 Peak overlap between the diketone and dihydrazide monomers of **Material A** prevented clear
171 NMR analysis of decomposition. **Material B** would not dissolve in solution even after 48 h of
172 incubation with excess hydroxylamine. We can conclude that the complete decomposition to mon-
173 omers did not happen for **Material B** because the monomers for **Material B** are soluble in DMSO.
174 These results highlight that the structure of the dialdehyde and diketone monomer significantly
175 affects the material's response to hydroxylamine-based degradation conditions.

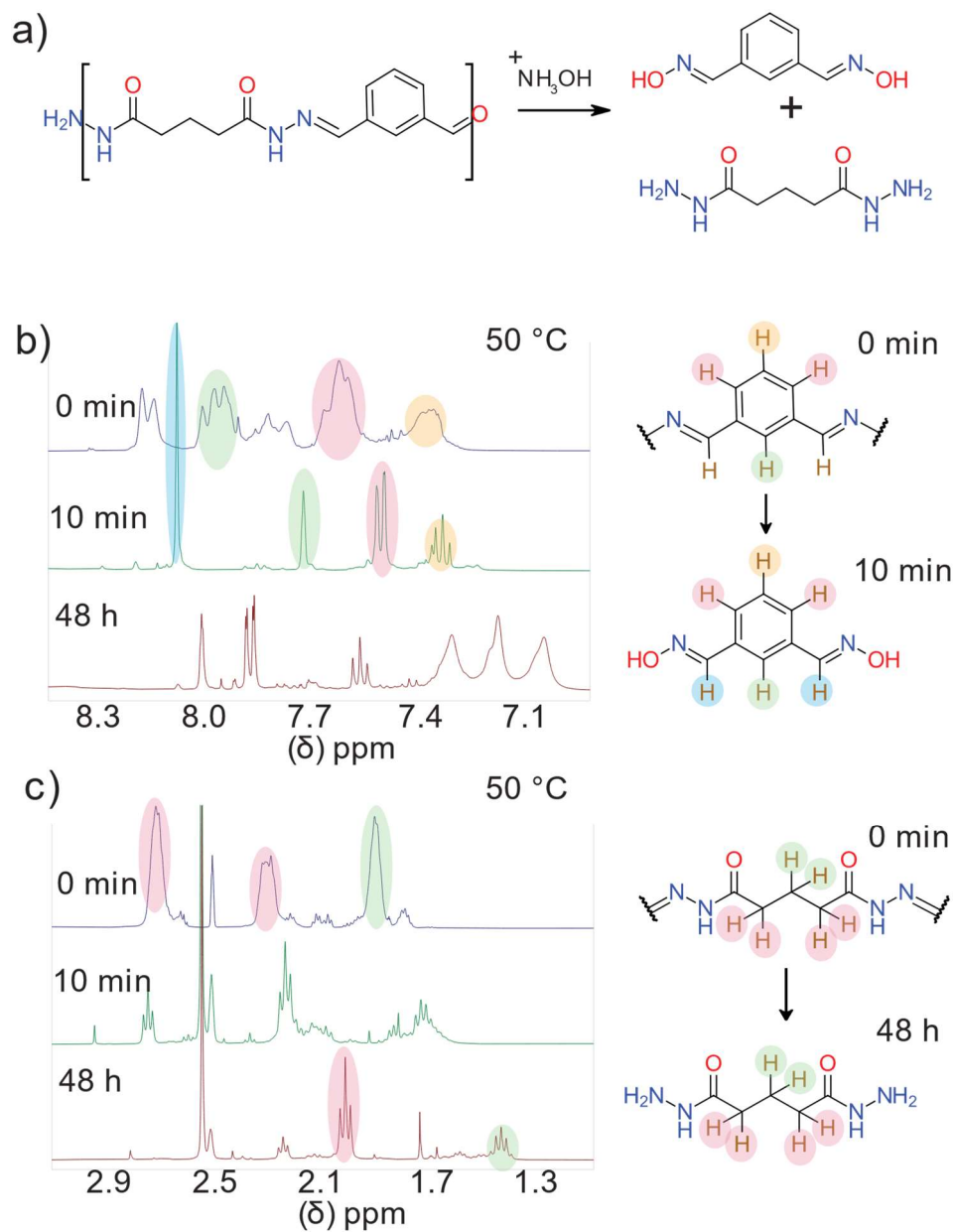


Figure 5. (a) **Material C** completely decomposes into monomeric units. (b) aromatic region (8.4 to 7.0 ppm) is monitored as **Material C** decomposes over time. (c) aliphatic region (3.0 to 1.2 ppm) is monitored as **Material C** decomposes over time.

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177 2.5. Isolation

178 Once it was clear that we could achieve degradation for **Materials A and C**. We strove to separate
 179 the degradation monomers through a simple extraction process. We first optimized degradation to
 180 use 20% H₂O in MeCN as a solvent system instead of DMSO, because MeCN is easier to remove

181 by vacuum. Incubation of **Material C** for 24 h at 37 °C resulted in the generation of the desired
182 dioxime product while **Material A** failed to undergo complete decomposition. Through solvent
183 screening, it was found that **3c** dissolved in CH₂Cl₂ while the pentane dihydrazide was insoluble.
184 Therefore, we performed extractions of the decomposed mixtures with CH₂Cl₂(**Figure 6b**). NMR
185 results show a preferential removal of **3c** by CH₂Cl₂washing. After 6 washes a small amount of **3c**
186 remained (**Figure 6c**). After 12 washes only **1** remained in the aqueous solution. Analysis of the
187 CH₂Cl₂ extract shows that only **3c** was carried into the CH₂Cl₂ layer (**Figure S3**). These results
188 demonstrate that can **Material C** be completely broken down into small molecules. However, the
189 resulting small molecules can be easily separated without the need for chromatography. The hy-
190 drolysis of oximes to aldehydes is well-established in literature. Therefore, by decomposing the
191 polymer into monomeric units and separating the monomers through extraction the recycling loop
192 for this benzylic polyacylhydrazone, **Material C**, was closed through this study. Future research
193 must be done to establish how general this finding is for different benzylic polyacylhydrazones.

194 Being encouraged by the extraction-based separation of **Material C** decomposition products, we
195 were interested in the extraction of the **Material A** decomposition products. Material A was also
196 incubated in the 20% H₂O solution with 4 equivalents of hydroxylamine HCl for 24 h. After 24 h,
197 some insoluble material remained in solution. This result suggests that the decomposition of **Ma-**
198 **terial A** once again was less robust than the decomposition of **Material C**. Additionally only a
199 small portion (<10%) of diketone could be extracted by CH₂Cl₂. This poor extraction suggests
200 incomplete decomposition because 1,3-cyclohexanedioxime is soluble in CH₂Cl₂. In previous
201 studies, the aliphatic ketone forms a more stable hydrazone than the aliphatic aldehyde with slower
202 kinetics.²⁶ This study shows that the benzylic hydrazone is less stable than both the aliphatic
203 diketone and dialdehyde-based hydrazones in the presence of strong nucleophiles.

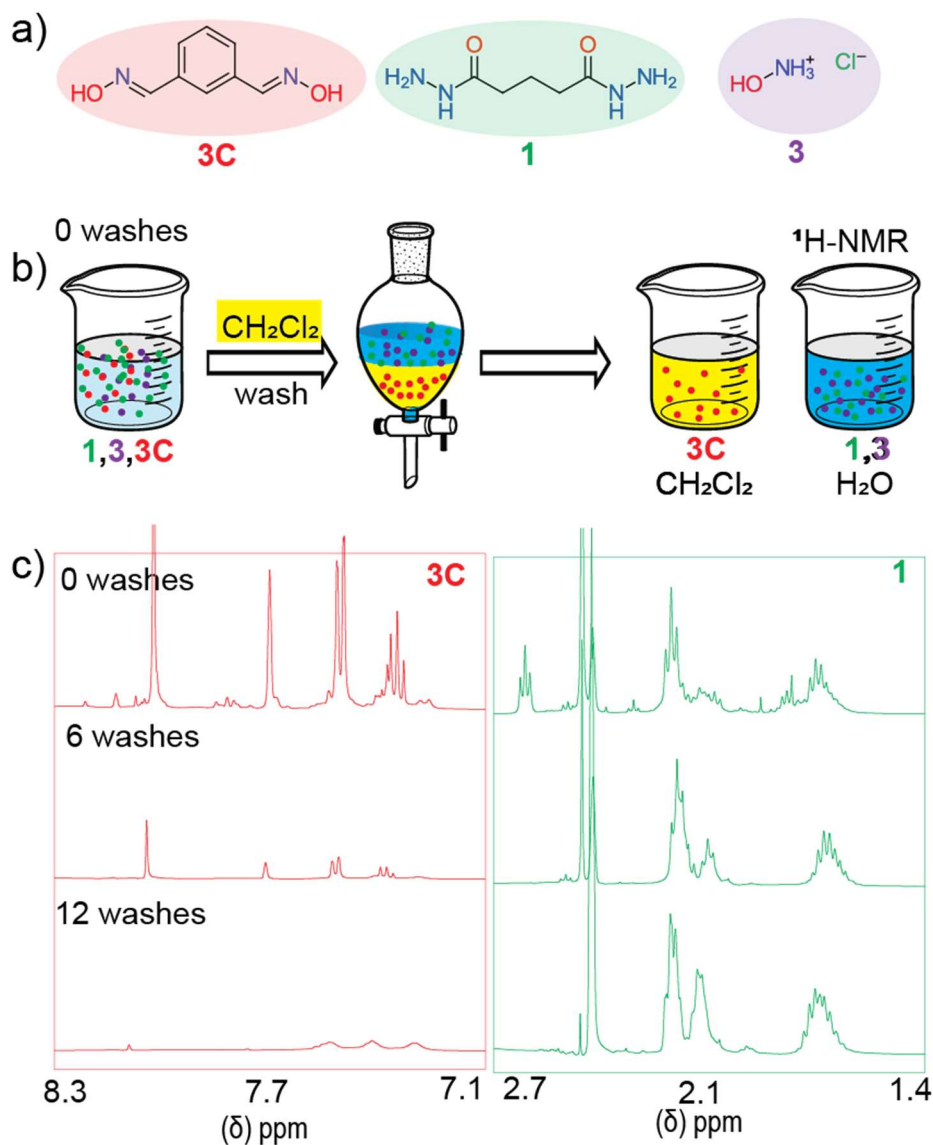


Figure 6. (a) molecules in crude decomposition mixture of **Material C** (b) cartoon of extraction-based separation of **3c** from monomer **1**. (c) NMR of crude decomposition mixture before, during, and after CH_2Cl_2 extractions. Small molecule **3c** disappears from aqueous solution after consecutive CH_2Cl_2 washing.

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206 **3. Conclusion: An advance in Sustainable Materials**

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208 The creation of more sustainable materials is a pressing concern for the survivability of the
 209 current ecosystem, this research in combination with previous studies highlights polyacylhydra-
 210 zone materials as useful dynamic covalent polymers with the potential for sustainability. We

211 discovered that the condensation of polyacylhydrazones in dimethyl sulfoxide (DMSO) yields soft
212 polyacylhydrazones with different morphologies than the previously reported materials due to their
213 retention of DMSO during the drying process. We show that polymer size can be controlled with
214 the stoichiometric addition of terminal hydrazide.

215 In our opinion, the most important discovery of this work lies in the validation of a benzylic
216 polyacylhydrazone as a “closed-loop” material. We have demonstrated their degradability under
217 mild reaction temperatures (37 - 50°C). We have shown that acylpolyhydrazones with benzylic
218 hydrazone linkages can be decomposed to monomer using hydroxylamine HCl in just 10 minutes.
219 We have also shown that the resulting dioxime and dihydrazide products can be separated with
220 simple extraction methods. Pairing these new results with previous chemical methods^{24,32} affirms
221 the candidacy of benzylic polyacylhydrazones for large-scale "closed-loop" recycling.

222 In the wake of this research, our future endeavors are poised to delve deeper into the intricate
223 structure-property relationships of these materials. Through a better structure property understand-
224 ing we aim to unlock their potential as recyclable plastics that seamlessly integrate with emerging
225 manufacturing technologies, including 3D printing. Additionally, we look forward to expanding
226 the potential for closed-loop recycling to more benzylic and aliphatic polyacylhydrazones, thus
227 increasing the scope of polyacylhydrazones with the potential for "closed-loop" recycling.

228

229 **Credit authorship contribution statement**

230 **Matthew B. Minus:** Conceptualization, Investigation, Methodology, Modeling, Simulation, For-
231 mal analysis, Validation, Software, Data curation, Visualization, Writing - original draft, Writing
232 – review & editing, Supervision, Project administration, Resources. **Chukwuzubelu Okenwa**
233 **Ufodike:** Conceptualization, Investigation, Methodology, Modeling, Simulation, Formal analysis,
234 Validation, Software, Data curation, Visualization, Writing - original draft, Writing – review &
235 editing, Supervision, Project administration, Resources. **Al Mazedur Rahman:** Experimentation,
236 Formal analysis. **Evan McHenry:** Experimentation, Formal analysis, **Marco D. Giles:** Method-
237 ology, Experimentation, Formal analysis. **Gaius Chukwuka Nzebuka:** Conceptualization, Data
238 curation, Visualization

239

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244

245 **Declaration of Competing Interest**

246 The authors declare that they have no known competing financial interests or personal relation-
247 ships that could have appeared to influence the work reported in this paper.

248

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255

256

257 **Appendix**

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