

# Machine Learning for Analysis of Experimental Scattering and Spectroscopy Data in Materials Chemistry

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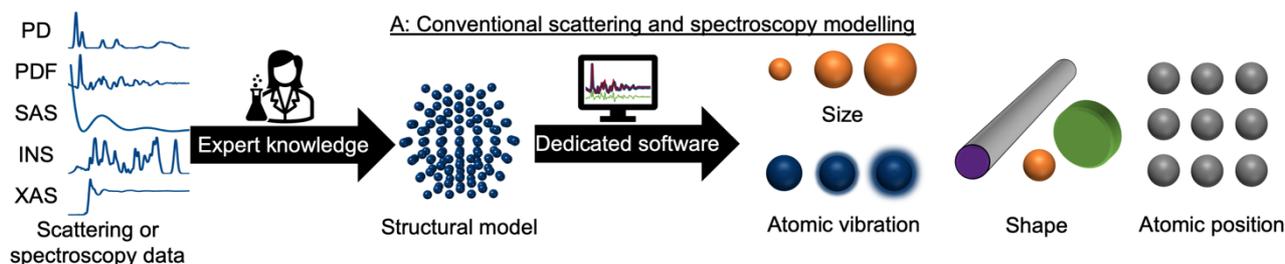
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## Abstract

The rapid growth of materials chemistry data, driven by advancements in large-scale radiation facilities as well as laboratory instruments, has outpaced conventional data analysis and modelling methods, which can require enormous manual effort. To address this bottleneck, we investigate the application of supervised and unsupervised machine learning (ML) techniques for scattering and spectroscopy data analysis in materials chemistry research. Our perspective focuses on ML applications in powder diffraction (PD), pair distribution function (PDF), small-angle scattering (SAS), inelastic neutron scattering (INS), and X-ray absorption spectroscopy (XAS) data, but the lessons that we learn are generally applicable across materials chemistry. We review the ability of ML to efficiently and accurately identify physical and structural models and extract information from experimental data. Furthermore, we discuss the challenges associated with supervised ML and highlight how unsupervised ML can mitigate these limitations, thus enhancing experimental materials chemistry data analysis. Our perspective emphasises the transformative potential of ML in materials chemistry characterisation and identifies promising directions for future applications. The perspective aims to guide newcomers to ML-based experimental data analysis, alerting them to the potential pitfalls and offering guidance for success.

## Introduction

During recent decades, materials science research has been accelerated by the rapid development of large-scale radiation facilities<sup>1,2</sup> and the advancement of laboratory instruments.<sup>3-5</sup> It is now common to acquire large amounts of data from e.g. *in situ* or *operando* experiments.<sup>6-8</sup> The combination of scattering and spectroscopy with computed tomography, allowing detailed position-resolved studies of e.g. batteries and catalysts also results in larger and larger amounts of data with the continued development of synchrotron and neutron sources.<sup>9-12</sup> As illustrated in Figure 1, analysis of scattering- and spectroscopy data can provide invaluable structural information on e.g., lattice parameters, oxidation states, atomic positions, and crystallite or particle size and shape. In conventional data modelling approaches, data are often analysed using minimisation techniques such as least-squares fitting algorithms, where the difference between experimental data and simulated data is minimised by refining parameters in a physical model, e.g., representing the atomic structure. This process is known as structure refinement. Identification of the structure model to use in structure refinement can be a limitation in data analysis: to identify the structural model, extensive database and literature searches are often needed. Automated screening of large numbers of structure models can be combined with structure refinement methods, and have been used for e.g., identifying a cluster<sup>13</sup> or crystal structure<sup>14</sup> from, e.g., PDF data. However, least-squares fitting algorithms are computationally expensive, which limits their use for structure model identification. Consequently, data analysis is often a major bottleneck for materials chemistry research.<sup>15, 16</sup> With the continuing advancement of modern radiation facilities,<sup>17</sup> the need for tools that can aid scientists in structural analysis is in increasing demand. ML has emerged as a powerful tool for automating several aspects of scattering- and spectroscopy data analysis.<sup>18-27</sup> In this perspective, we describe the application of supervised and unsupervised ML to experimental scattering (PD, PDF, and SAS) and spectroscopy (INS and XAS) data. For a short introduction to supervised and unsupervised ML and the most popular ML algorithms, we refer to *Machine Learning Algorithms - A Review*, Batta Mahesh.<sup>28</sup>



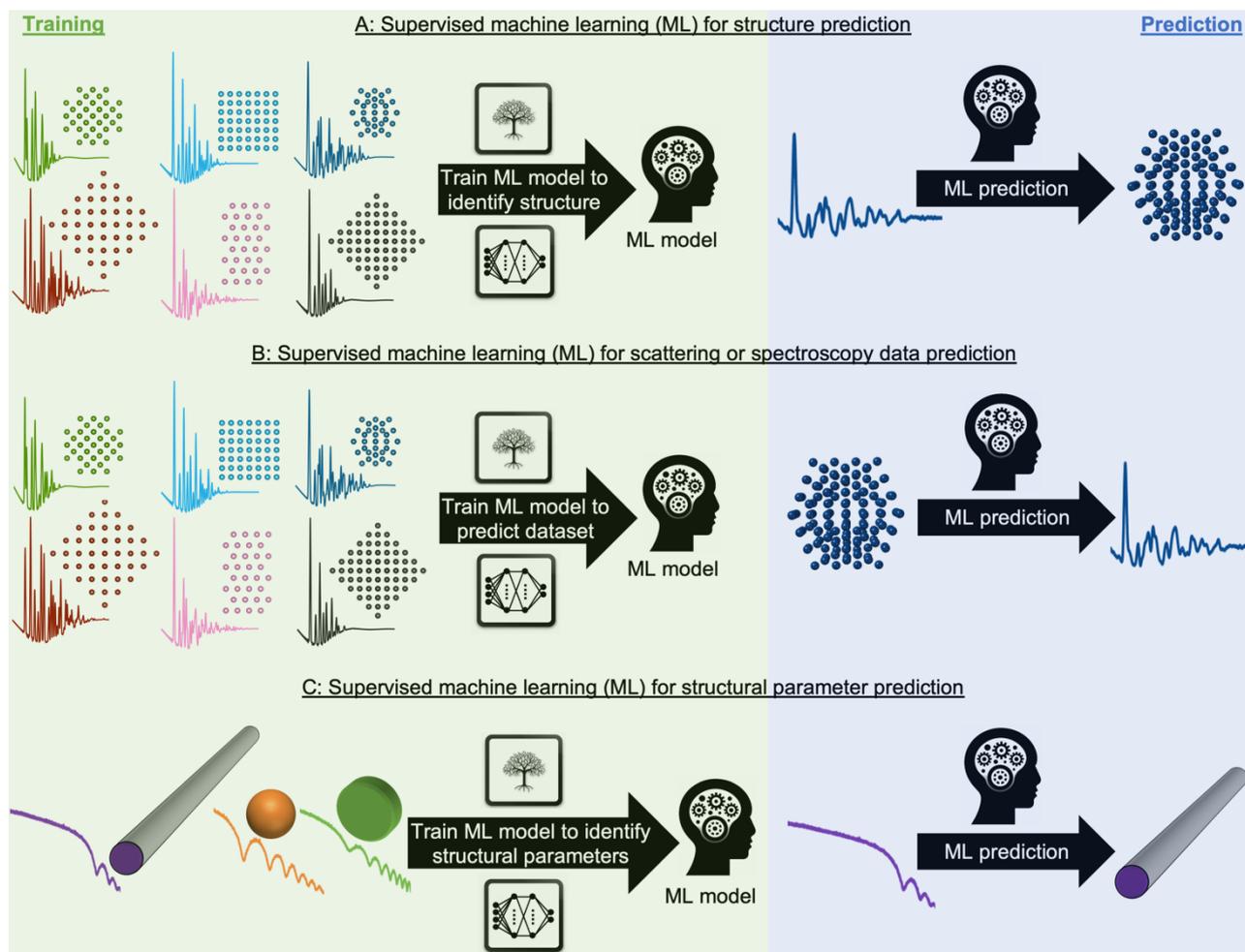
**Fig. 1** | Traditional scattering and spectroscopy modelling workflows involve an expert scientist manually creating a structural model using input data from a database or the literature. This model is refined using dedicated software to extract structural information from the dataset such as atomic positions, crystallite size, crystallite shape, and atomic vibrations. This process is repeated for each new dataset measured.

Most applications of ML in materials' chemistry apply supervised ML methods. Supervised ML is broadly the task of predicting a label based on a given set of input features. As will be exemplified throughout the perspective, we observe three main applications of supervised ML for the analysis of scattering and spectroscopy data: 1) Identifying a physical model from a scattering or spectroscopy dataset (Figure 2A). Here, experimental datasets are the input features, and the model is supervised to relate the datasets to the physical models, which are the labels. 2) Predicting scattering or spectroscopy data from a physical model. This can be achieved by using the data as labels and the physical model as input features (Figure 2B). 3) Bypassing the model refinement step to directly obtain structural information (Figure 2C). This is done by training the supervised ML model on data with varying structural parameters.

To train an ML model using supervised methods, one needs a dataset consisting of many pairs of labels and input features. This dataset, consisting of e.g., structure models and simulated data, is generally split into a training, validation, and test set, often in a 3:1:1 ratio. The model is trained on the training set, while being continuously evaluated on the validation set, using a user-defined objective function, called the loss function. Depending on the chosen class of models, the training will often improve on the training set until it can fit any trends in the data, including noise (overtraining). The validation set is used to ensure that the model training is stopped before it is overtrained. Once training is complete, the test set, which has not been used during training or validation, is employed to estimate the accuracy of the model on future unseen data (generalisation). The quality and size of the training set thus plays a crucial role in the model's efficiency and accuracy, with larger, higher-quality datasets typically yielding better results. A model's ability to interpolate and extrapolate, or make predictions within the range of the training data and beyond it, is generally influenced by the ML algorithm and the range and diversity of the training set. Many factors therefore need consideration when selecting and training an ML model. These include the choice of ML

algorithm (tree-based methods, neural networks (NNs), genetic algorithms, etc.),<sup>28, 29</sup> the number of parameters in the ML model, and both the quality and quantity of the training set. The model's ease of training and deployment can be influenced by the choice of ML algorithm. Interpretability of the model depends strongly on the algorithm used, for example an individual decision tree is easily interpretable, whereas a deep neural network with millions of parameters is not, and requires *post hoc* methods to understand its operation.<sup>30</sup> When it comes to scalability, NNs have many more trainable parameters compared to tree-based methods. This makes tree-based method efficient learners in small data regime, however, NNs often prove more effective at handling larger datasets. NNs are today commonly trained on large datasets, as used in for example, chatGPT<sup>31</sup> and AlphaFold.<sup>32</sup> This superiority in scalability might explain why NNs have become the predominant ML algorithm for structural analysis as large databases of training data have become increasingly available.

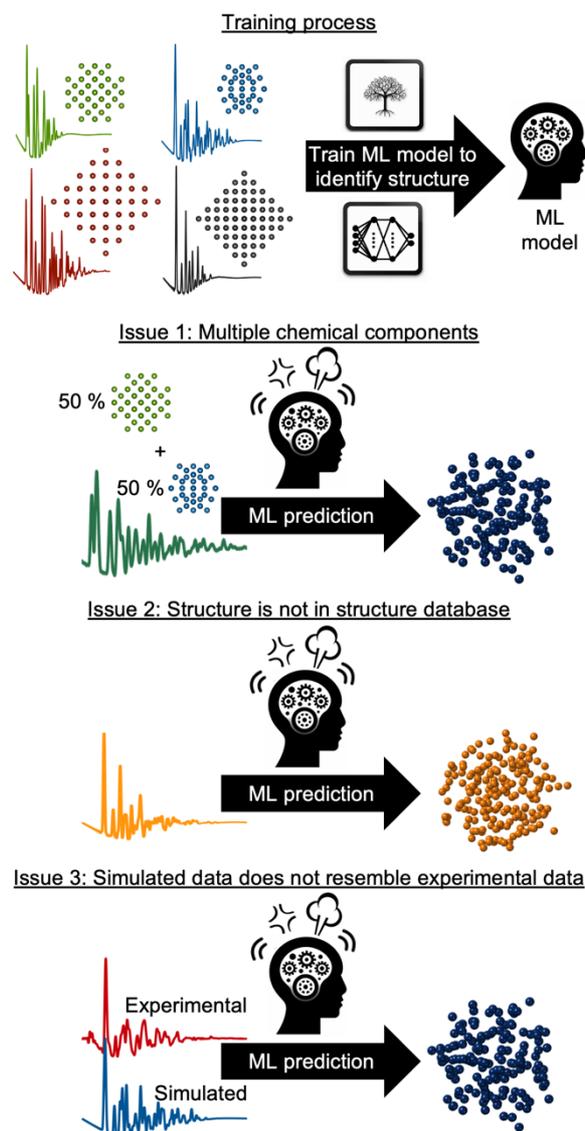
While training an ML model can be computationally expensive, this is a one-time cost. Subsequent predictions using the ML model can be computationally inexpensive and integrated into web-based solutions, or can be done at synchrotron or neutron facilities for real-time structure prediction while the experiments are going on.



**Fig. 2** | A) Use of ML for identifying a structural model. During training, a supervised ML model is trained on pairs of structural models and scattering or spectroscopy simulations, here shown for PDF data. Afterwards, the ML model can quickly and computationally inexpensively identify the structural model from (experimental) scattering or spectroscopy data. B) Use of ML for predicting scattering or spectroscopy data from a structure model. During training, a supervised ML model is trained on pairs of structural models and scattering or spectroscopy simulations, here shown for PDF data. After training, the ML model quickly and computationally inexpensively predicts scattering or spectroscopy data from a structural model. C) Use of ML to predict structural parameters. During training, a supervised ML model is trained on pairs of scattering or spectroscopy simulations with varying structural parameters, here shown for SAXS data. After training, the ML model quickly and computationally inexpensively identifies the structural information from the (experimental) scattering or spectroscopy data.

However, supervised ML is limited by its reliance on paired input data and labels for training, which can be challenging to obtain for experimental data analysis. As will be discussed and exemplified below, we observe three common issues encountered when analysing experimental scattering and spectroscopy data with supervised ML. These are illustrated in Figure 3: 1) Handling data with contributions from multiple chemical components. 2) Handling data arising from structures not present in the training database and 3) accounting for experimental data that contain signals not included in the simulated data. In all three scenarios, the labelled data are inadequate for solving the problem at hand, making unsupervised ML methods a more suitable alternative, or complementary tool. Unsupervised ML models work without paired labels and input features, using only input features or intermediate input-derived labels, such as in autoencoders.<sup>33</sup> Unsupervised ML is often used to present complex data in a low-dimensional space (dimensionality reduction), enabling the analysis of high-dimensional dataset similarities, clustering, and the extraction of underlying data trends that are difficult to comprehend from the input representation space.<sup>28</sup> Unsupervised methods can also be applied to 'demix' data, i.e., separating the signal from each component in a multi-phase scattering or spectroscopy dataset.

In the following sections, we use selected examples to provide an overview of how supervised ML has been used to identify structural models and structural information from experimental PD, PDF, SAXS, INS and XAS data or predict the dataset from the structure. We also outline and exemplify how unsupervised ML has been applied to address the three issues presented in Figure 3, and we discuss the potential future impact of ML in the analysis of experimental materials chemistry data.

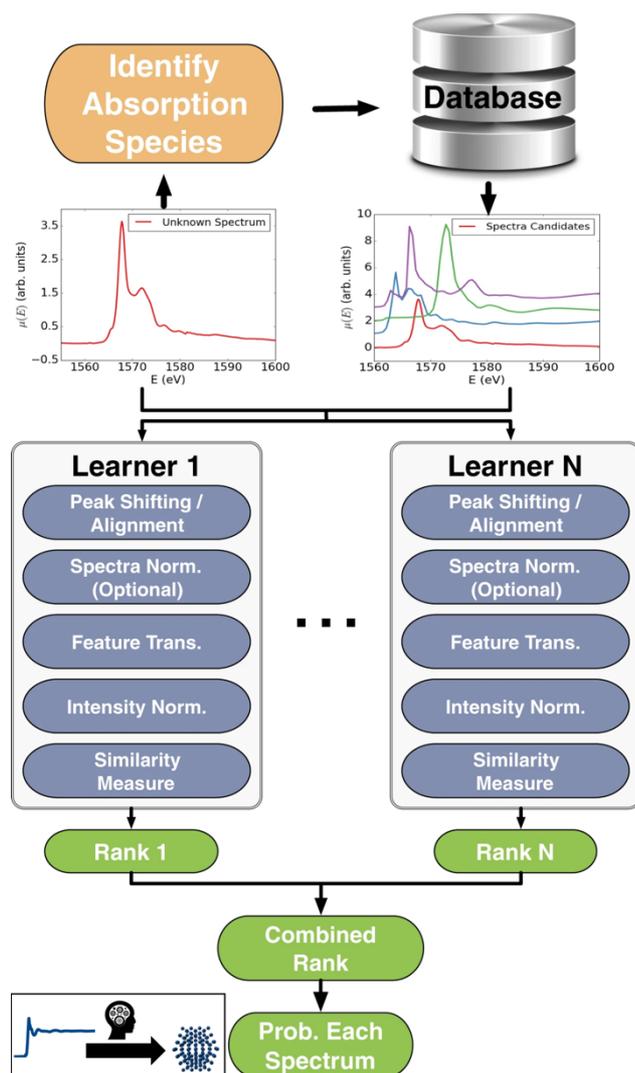


**Fig. 3** | Three common issues when analysing experimental scattering and spectroscopy data. Supervised ML models are trained on pairs of structural models from a structural database and scattering or spectroscopy simulations. However, supervised ML methods are challenged by; Issue 1: The experimental data are obtained from a system containing multiple chemical species, which is not taken into account in the ML model. Issue 2: The required structural model is not included in the structure database used for training the ML model. Issue 3: The experimental data contains background noise, instrumental effects or other phenomena not encountered by the simulated data.

### **X-ray Absorption Spectroscopy: Large XAS Databases Accelerate Supervised ML Structure Identification**

XAS is a powerful experimental technique for investigating the electronic and atomic structure of materials. In XAS, a sample is exposed to a monochromatic X-ray beam, whose energy is varied in a range of ca. 10–100 eV around the K-edge or L-edge of the elements in question, i.e. the energy needed to eject electrons from the 1s or 2s orbital. This causes the sample to absorb some of the X-rays. By measuring and analysing the absorbed X-rays as a function of energy, it is possible to obtain information about the local electronic structure and chemical environment of the atoms in the sample. Information about oxidation state and coordination environment can be obtained through X-ray Absorption Near Edge Spectroscopy (XANES), while Extended X-ray Absorption Fine Structure (EXAFS) can provide knowledge of local atomic structure.

Conventional analysis of XAS data requires expertise in the complex data analysis as well as manual work. To address this, Zheng et al. created a large XANES database, XASdb, with more than 800,000 computed reference XANES entries from over 40,000 materials from the open-science Materials Project database.<sup>34</sup> Their supervised ML model, illustrated in Figure 4, was used for the analysis of XANES data. Given a XANES spectrum as input, it outputs a list of the chemical compounds whose spectra are most similar to the target spectrum (The task is illustrated in Figure 2A). From these compounds, chemical information such as oxidation state and coordination environment can be extracted. It predicts the chemical compound with 69.2 % top-5 accuracy on a test set of 13 simulated XANES spectra. However, the correct oxidation state is within top-5 with 84.6 % accuracy and the coordination environment with 76.9 % accuracy. On six experimental XANES spectra, it predicts oxidation state with 83.3 % accuracy, coordination environment with 83.3 % accuracy and the chemical compound with 33.3 % top-5 accuracy.<sup>35</sup> These results highlight the impact of large databases like the open-science Materials Project<sup>34</sup> and JARVIS.<sup>36</sup> As these databases grow, they will likely catalyse supervised ML analysis of scattering and spectroscopy data in materials chemistry.



**Fig. 4** | Workflow schema of the Ensemble-Learned Spectra Identification (ELSIE) algorithm. The ELSIE algorithm consists of two steps. In the first step, the absorbing species is identified and used to narrow down the candidate computed reference spectra. In the second step, the algorithm yields a rank-ordered list of computational spectra according to similarity with respect to the target spectrum. The figure is adapted from Zheng et al.<sup>35</sup> (Under Creative Commons Attribution 4.0 International License <https://creativecommons.org/licenses/by/4.0/>).

The above example shows that large XANES databases, like XASdb, can be used to address the spectrum-to-structure problem, as illustrated in Figure 2A. However, it can also be used to address the inverse problem: structure-to-spectrum, as illustrated in Figure 2B. Calculating a XANES spectrum from a structure can be computationally demanding but by using a supervised ML model to predict the spectrum, this process can be done in milliseconds to seconds.<sup>37-39</sup> Supervised ML can also be used to directly predict chemical information such as average size, shape, morphology and

oxidation states of first metallic nanoparticles<sup>40,41</sup> and metallic oxides,<sup>42</sup> Bader charge,<sup>43</sup> mean nearest neighbour distances,<sup>43</sup> and local chemical environment,<sup>44</sup> from an XANES spectrum or predict the radial distribution function from the experimental EXAFS data.<sup>45-49</sup>

Analysing XAS data from samples containing more than one chemical species remains a challenge, as supervised ML models trained on data simulated from a single chemical species are constrained to be used on experimental data from individual chemical species, and attempting to account for all possible chemistries, e.g. by training on simulated data from mixed samples, leads to a combinatorial explosion. Instead, linear unsupervised ML techniques like principal component analysis (PCA) and non-negative matrix factorisation (NMF) have been used to discover trends in large XAS datasets and separate them into signals from their respective chemical components.<sup>50-55</sup> For example, Tanimoto et al. used NMF to identify and map spatial domains from absorption spectra in 2D-XAS images of lithium ion batteries. Using NMF for this was successful despite the small differences in the spectra between the various compounds present in the sample.<sup>53</sup> The authors recognised that NMF can be challenged by background effects as these can be predominant in some of the NMF-extracted components. Therefore, they subtracted a reference X-ray absorption spectrum obtained on  $\text{Li}_{0.5}\text{CoO}_2$ , which also includes that background signal. This trick enables the NMF method to distinguish small differences in the spectra.

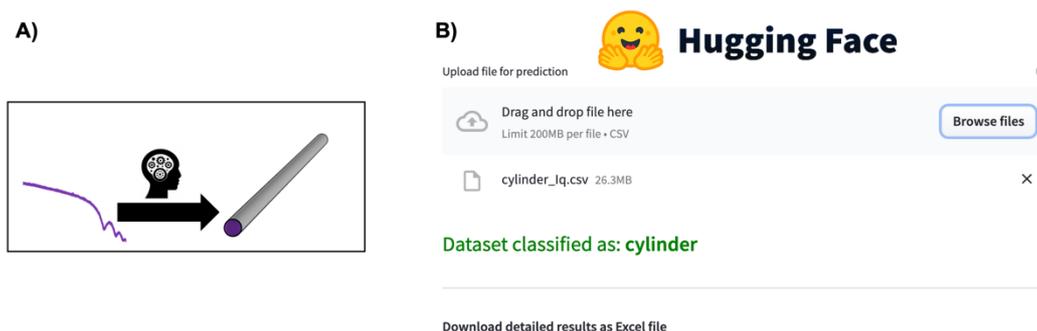
### **Small-angle scattering: Supervised ML for Dataset and Parameter Prediction**

SAS is a strong technique to obtain information about the morphology, orientation and size distribution of e.g., nanoparticles in solution and solids.<sup>56</sup> In a SAS experiment, X-ray or neutron scattering data are measured at small scattering vectors, e.g., the Q-range from ca.  $0.001 \text{ \AA}^{-1}$  to  $1 \text{ \AA}^{-1}$ . This region of the scattering signal contains structural information about the species in the sample on the nanometer scale.

Traditional SAS data fitting is done by refining a model against the data. The model must describe the particle shape, size, and size distribution as well as possible agglomerations of e.g. nanoparticles or large molecules in the sample, and much work is often needed in deciding on a suitable model. This step can be time-consuming and prone to errors.<sup>57</sup> Here, ML can assist by providing a more efficient approach to fitting experimental data to theoretical models. The Computational Reverse-Engineering Analysis for Scattering Experiments Genetic Algorithm (CREASE-GA) tool, developed by members of Prof. A. Jayaraman's research group, can reconstruct 3D structures from SAS patterns

using a genetic algorithm.<sup>58-65</sup> CREASE-GA compares the goodness-of-fit between the experimental SAXS pattern and a population of 3D structures. A genetic algorithm<sup>29</sup> is then used to update the 3D structure population to better describe the experimental SAS pattern. This process continues until convergence, determining the 3D structure of the sample in question. Originally, the SAS patterns from the 3D structure population were calculated using the Debye scattering equation. This posed a computational bottleneck for CREASE-GA.<sup>58</sup> However, the authors have recently managed to accelerate CREASE by over 95 % by employing NNs to estimate the SAS patterns.<sup>60, 63</sup>

Supervised ML has also proved to be an efficient tool for direct parameter extraction from SAS data, which might be difficult or time-consuming for humans to detect, such as orientation,<sup>66</sup> shape,<sup>67-69</sup> or the model for SAS form factor fitting.<sup>70-72</sup> Thereby, the ML model can bypass the cumbersome step of identifying the structural model for structure refinement. For example, the Scattering Ai aNalysis (SCAN) tool can predict the model for SAS form factor fitting from a SAXS pattern obtained from a nanoparticle. With the SCAN tool, the user can choose from a range of ML algorithms including tree-based algorithms and NNs. These algorithms individually achieve accuracies between 27.4 % and 95.9 % based on a test set of simulated SAXS data. However, when combined, they achieve an accuracy of 97.3 %. We are grateful to the authors for making SCAN open source, which has made it possible to implement it as a Hugging Face app.<sup>73</sup> This makes it easily useable, also for users without programming experience, as illustrated in Figure 6.



**Figure 6** | A) The SCAN<sup>72</sup> tool directly predicts structural information such as particle shape from a SAXS pattern. B) Overview of the SCAN<sup>72</sup> tool's ease of use for predicting structural information from a SAS pattern. Simply click “Browse files”, wait for the model to predict the structural information, and, if needed, download the detailed information in an Excel sheet. All of this is possible through the Hugging Face app.<sup>73</sup>

## **Powder Diffraction: Structure Identification**

PD is a fundamental technique in materials chemistry that is used to analyse the crystal structure of a powdered sample. In PD, a powder of crystalline particles is exposed to an X-ray or neutron beam, causing Bragg diffraction due to the periodic atomic arrangement in the sample. By analysing the Bragg peak position, intensity and shape, information about the crystalline structure of the material can be obtained.

Recent advances in ML techniques offer promising new opportunities in PD data analysis. For example, it has been demonstrated that a sample's crystal system and space group can be predicted from X-ray PD data using NNs<sup>26, 74-76</sup> and tree-based techniques.<sup>77</sup> Suzuki et al. demonstrated that an advantage of the tree-based ML approaches is that they are interpretable.<sup>77</sup> Interpretability enables us to understand the ML model's prediction mechanism and thus analyse when it predicts differently from a human expert. This can either show when the ML model is wrong and need to be corrected or reveal when it uncovers unexpected correlations that may lead to scientific insights. In contrast, Park et al. used an NN which is easier to train on possible larger future datasets containing millions or billions of PD patterns.<sup>75</sup>

However, these methods can only be used to determine the crystal system or the space group from PD data. To identify the full structural model for e.g., structure refinement, the unit cell, and unit cell content is also needed. Garcia-Cardona et al.<sup>78</sup> made progress towards this for neutron diffraction data, where the crystal system (cubic, tetragonal, trigonal, monoclinic, and triclinic) could be predicted with an accuracy of 92.65 % (Figure 2A) using convolutional NNs, which are a type of NNs that capture the relationship between neighbouring data points e.g. neighbouring intensities in the diffraction pattern. Subsequently, another supervised tree-based ML model was used to predict unit cell parameters (unit cell length and angles) from the data.<sup>78</sup> The authors note that the ML models possess good performance on simulated data but more sophisticated models are required before it is applicable on experimental data. Furthermore, an ML model that is capable of precisely predicting a full structural model, including unit cell content, as required for e.g., Rietveld refinement of experimental PD data, is yet to be developed.

If working in a more restricted chemical space with well-defined components, it is possible to use supervised ML models for direct prediction of structural parameters for the phases included in the space. Dong et al. demonstrated that it is possible to directly predict structural information such as scale factor, lattice parameter and crystallite size (Figure 2C) from PD patterns from a system of 5

different metal oxides using a convolutional NN that they call Parameter Quantification Network (PQ-Net).<sup>2</sup> They obtained an experimental X-ray diffraction computed tomography dataset of a multiphase Ni-Pd/CeO<sub>2</sub>-ZrO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> containing about 20,000 diffraction patterns, each pixel containing a multiphase pattern. Treating such a large quantity of Rietveld refinements takes significant computer time. To overcome this limitation, PQ-Net was trained on simulated PD data with varying scale factors, lattice parameters and crystallite size for NiO, PdO, CeO<sub>2</sub>, ZrO<sub>2</sub> and theta-Al<sub>2</sub>O<sub>3</sub>. A 2<sup>nd</sup> degree Chebyshev polynomial background and Poisson noise were also added to the training data. After training, PQ-Net can identify the crystalline phase, scale factor, lattice parameter and crystallite size for each experimental PD pattern in the dataset, orders of magnitudes faster than done using conventional Rietveld refinement. As seen in Figure 7, the results of using PQ-Net are comparable to those determined through Rietveld methods on experimental data.

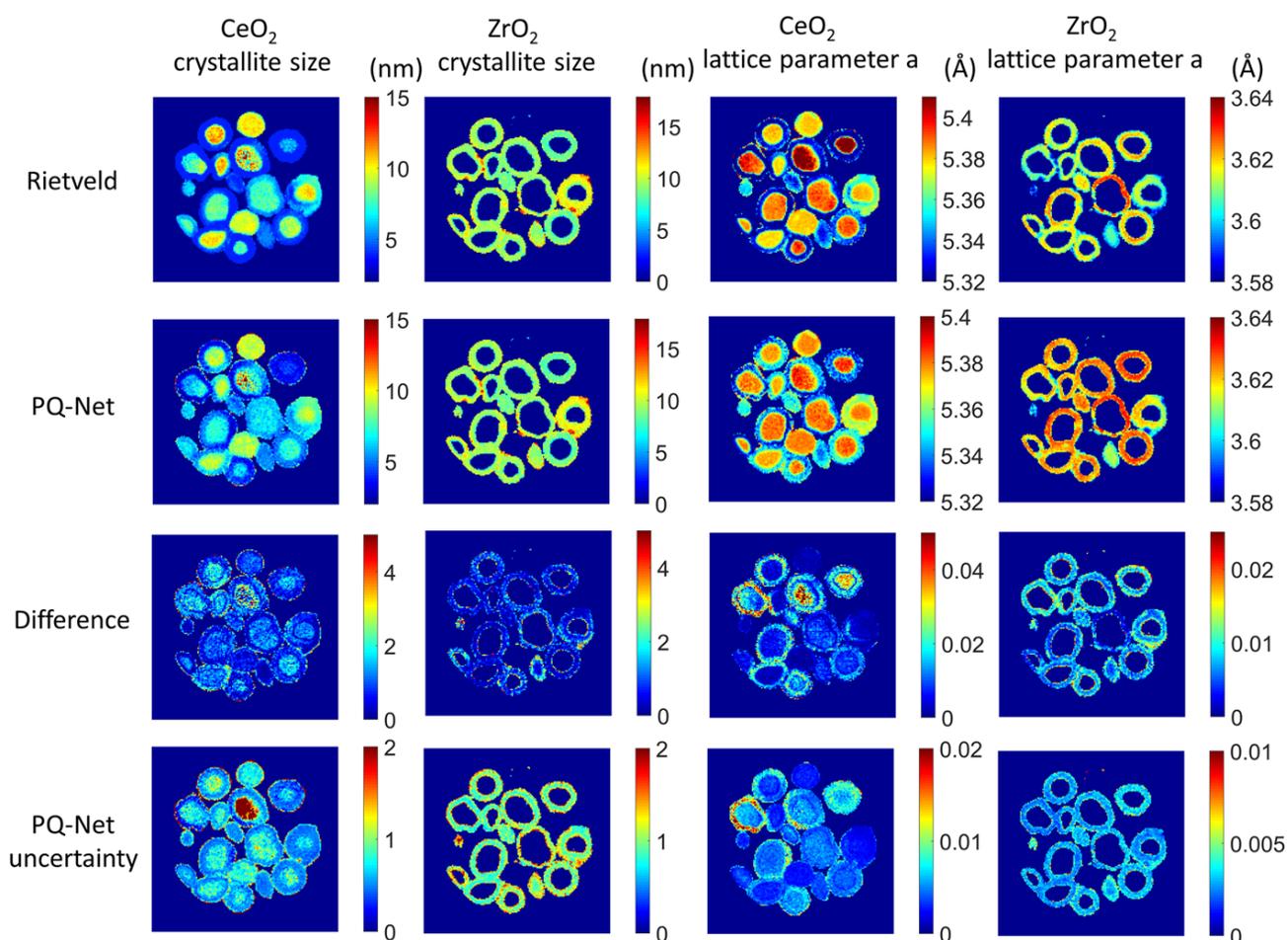


Figure 7 | Crystallite size (colourbar axis corresponding to nm) and lattice parameter *a* (colourbar axis corresponding to Å) maps for CeO<sub>2</sub> and ZrO<sub>2</sub> obtained with the Rietveld method, results obtained

with the PQ-Net, their absolute difference for the experimental multi-phase NiO-PdO-CeO<sub>2</sub>-ZrO<sub>2</sub>-Al<sub>2</sub>O<sub>3</sub> system and the uncertainty maps of the deep ensemble PQ-Net.<sup>2</sup> (Under Creative Commons Attribution 4.0 International License <https://creativecommons.org/licenses/by/4.0/>).

Dong et al. required 100,000 datasets for training to achieve good results predicting structural information on experimental data from a chemical system with five components. For larger and complex systems with more possible components, the supervised ML model must be trained on even more data of both individual structural models and combinations of these. An example of this approach is the work by Lee et al., who used a supervised ML model for phase identification in a quaternary chemical system consisting of Sr, Al, Li, and O. This system thus spans simple to ternary oxides and multiple different polymorphs, and in total 170 inorganic compounds appear in the chemical space.<sup>79</sup> Here, the ML model was trained on 1,785,405 synthetic combinatorically mixed PD patterns. After training, the model was able to phase identify and give rough estimates of phase fractions of multicomponent systems from XRD data.

Instead of training the supervised ML model on large databases of combinations of phases, unsupervised ML methods, such as PCA and NMF, can demix multiphase PD patterns into individual phase patterns, as also addressed for XAS data above.<sup>80, 81</sup> Here, a set of experimental diffraction patterns are given as input to the unsupervised ML algorithm, which decomposes it into its constituent parts. However, the PCA and NMF algorithms may encounter difficulties if the PD pattern of a chemical phase changes during the reaction, for example, through peak shifting from a unit cell change, variations in peak intensity from a change in thermal vibrations, or a change in the crystalline size, leading to different peak widths. Stanev et al. addressed the peak shifting problem by combining NMF with cross-correlation analysis of the demixed PD patterns. They were hence able to cluster the demixed PD patterns originating from the same chemical phase.<sup>82</sup>

Chen et al. employed deep reasoning networks (an unsupervised method) to map the crystal-structure phase diagram of Bi-Cu-V oxide using experimental PD data.<sup>83</sup> Based on PD data from Bi-Cu-V oxides prepared in various compositions, the ML model was trained to demix the phases in the PD patterns, and subsequently map the crystal-structure phase diagram of Bi-Cu-V oxide.<sup>83</sup> Once trained, the deep reasoning network can take a PD pattern from a sample in the composition space as input, and demix signals from multiple phases into their constituent components. Using a linear combination of the components, the PD pattern can be reconstructed, and the phase diagram can be

constructed with phase concentrations. The authors demonstrated this process on X-ray PD patterns from a phase diagram of Bi–Cu–V containing 19 chemical phases.

### **Total Scattering with Pair Distribution Function: Nanoparticle structure determination**

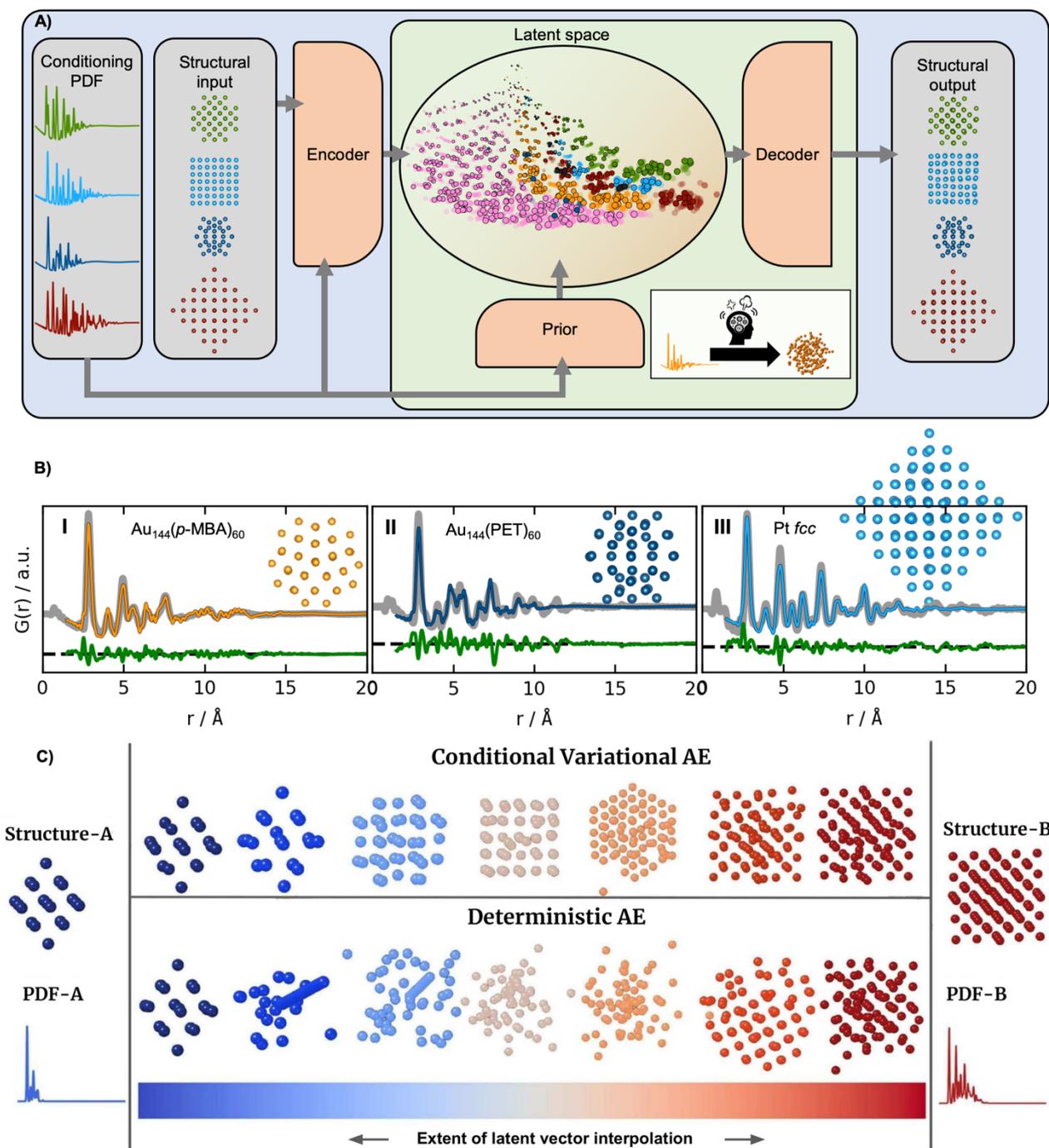
Total scattering experiments are similar to PD, as the scattering of X-rays or neutrons from a sample is measured. However, for total scattering, it is not only the Bragg diffraction peaks that are analysed but also the diffuse scattering arising from local structural order. This enables structural analysis of both crystalline and non-crystalline samples. Total scattering data are often analysed in real space through PDF analysis. A PDF is generated by normalising, correcting, and Fourier-transforming the total scattering signal, and the PDF represents a histogram of interatomic distances. Like PD, PDF can be used to characterise crystalline materials, but has especially emerged as a powerful technique to characterise the atomic arrangement in non- or poorly crystalline materials such as clusters in solution and disordered, amorphous, and nanomaterials.<sup>84, 85</sup> As for XAS, SAS and PD discussed above, ML models can help accelerate the PDF modelling process. For example, supervised ML models have been developed to predict space groups from PDF data of crystalline materials.<sup>86</sup> We have furthermore shown that supervised ML can be used to identify the structure of polyoxometalate clusters.<sup>87, 88</sup>

In a slightly different application of supervised methods, we have recently demonstrated how explainable supervised ML can be used to extract information on the local atomic arrangement present in a sample.<sup>89</sup> The aim of PDF analysis of e.g. nanostructured materials is often to identify models for the main structural motifs in a material. Our algorithm, ML-MotEx provides this information by using SHAP (SHapley Additive exPlanation)<sup>90, 91</sup> values to identify which atoms in a given starting model are important for the fit quality. The starting model should be chosen to contain the main atomic arrangements expected to be found in the sample. If analysing the structure of e.g. an amorphous materials, the starting model may be a related crystalline structure. Based on the model, thousands of structure fragments are generated by iteratively removing atoms from the starting model, and a PDF fit is done for each of the fragments. A supervised ML model is then trained on the thousands of PDF fits, and ultimately, each atom can be assigned an *atom contribution value* which describes how much it contributes to the goodness-of-fit. By analysing the SHAP values, it is thus possible to identify which motifs in the starting model are important in the material to describe the data. ML-MotEx has so far been used to identify the structure of ionic clusters in solution,<sup>89</sup> extract

structural motifs in amorphous metal oxides,<sup>92</sup> and identify stacking faulted domains on MnO<sub>2</sub> from both X-ray PD and PDF data.<sup>93</sup>

The supervised ML methods used for structure identification for both PD and PDF data discussed so far are limited to identifying structural models that are part of the structural database on which they have been trained. Ultimately, the aim of a PDF experiment may be to solve the structure of new nanomaterials. To explore structural models beyond any existing structural database (issue 2 in Figure 3), some classes of unsupervised ML could be useful. We have recently used a graph-based conditional variational autoencoder,<sup>33</sup> DeepStruc (Figure 8A) to determine the atomic structure of metallic nanoparticles up to 200 atoms in size from PDF data.<sup>94,95</sup> Given a PDF, DeepStruc can output a particle structure, and we obtained mean absolute errors of  $0.093 \pm 0.058$  Å on the atomic positions in metallic nanoparticles from simulated PDFs. Figure 8B shows the results of applying DeepStruc to experimental PDFs obtained from three chemical systems, consisting of two magic-sized clusters I) Au<sub>144</sub>(*p*-MBA)<sub>60</sub><sup>96</sup> and II) Au<sub>144</sub>(PET)<sub>60</sub>,<sup>96</sup> and III) a 1.8 nm Pt nanoparticle.<sup>97</sup> All three structures match the structures found in the literature and provide good data fits. Although DeepStruc is supervised in the sense that it is trained on structure and PDF pairs, it also has abilities from unsupervised ML as it learns to probabilistically map cluster structures and PDFs into a two-dimensional chemically meaningful space, which we refer to as the latent space. By inspecting the latent space, it is possible to find relations between different types of cluster models. DeepStruc places decahedral (orange) structures in the latent space between face-centered cubic (fcc) (light blue) and hexagonal closed packed (hcp) (pink) structures. This spatial arrangement can be explained by considering that decahedral structures are constructed from five tetrahedrally shaped fcc crystals, separated by {111} twin boundaries.<sup>13,98,99</sup> The twin boundaries, resembling stacking faulted regions of fcc, justify their location in the latent space between fcc and hcp.<sup>48,95,96</sup> The capability of DeepStruc to interpolate between cluster structures arises from each structure in the latent space being probabilistically rather than deterministically predicted. This has been demonstrated in Anker & Kjær et al.,<sup>94</sup> where we show that generative models<sup>28</sup> are necessary to go beyond the structural database used for training the ML model. Specifically, we showed that a generative model, like DeepStruc, can interpolate between structural models, as shown in Figure 8C, while still yielding sensible results. More traditional deterministic models, which are not probabilistic, could not interpolate between structures and thereby not go beyond the structural database when predicting a structural model from a PDF.

DeepStruc is integrated with the Hugging Face platform, enabling users to rapidly determine the structure of small metallic nanoparticles from PDFs using a simple two-click process.<sup>100</sup> The Hugging Face integration provides a user-friendly experience, without requiring data storage or complex software installations.



**Figure 8** | DeepStruc is a Conditional Variational Autoencoder that can solve the structure of a small mono-metallic nanoparticle from a PDF. A) DeepStruc predicts the xyz-coordinates of the mono-

metallic nanoparticle structure with a PDF provided as the conditional input. The encoder uses the structure and PDF as input, while the prior only takes the PDF as input. A latent space embedding is given as input to the decoder to obtain the structural output, which produces the corresponding mono-metallic nanoparticle xyz-coordinates. During the training of DeepStruc, both the blue and green regions are used, while only the green region is used for structure prediction during the inference process. B) PDF fit of the reconstructed structure from three different nanoparticle systems: I)  $\text{Au}_{144}(\text{p-MBA})_{60}$  PDF,<sup>96</sup> the II)  $\text{Au}_{144}(\text{PET})_{60}$  PDF<sup>96</sup> using a reconstructed structure icosahedral structure and III) a 1.8 nm Pt nanoparticle PDF from Quinson et al.<sup>97</sup> Figure 8A and B are adapted from Kjær & Anker et al.<sup>95</sup> (*Under Creative Commons Attribution 3.0 Unreported License <https://creativecommons.org/licenses/by/3.0/>*). C) Structures generated by decoding different extents of interpolation of the latent variables obtained for PDF-A and PDF-B. The generated structures start from Structure-A and progressively evolve towards Structure-B. This work uses a Conditional Variational Autoencoder similar to DeepStruc and we compare it with a Deterministic Autoencoder. Figure 8C is from Anker & Kjær et al.<sup>94</sup> (*Under Creative Commons Attribution-NonCommercial-NoDerivatives 4.0 International License <https://creativecommons.org/licenses/by-nc-nd/4.0/>*).

Unsupervised ML algorithms have also been employed to either uncover trends in PDFs obtained from multiple samples or to separate the signal from different phases in a PDF.<sup>101-103</sup> NMF has proven to be especially useful, and has been used to analyse PDFs obtained from various materials and conditions, including battery materials, amorphous solid dispersions, or data collected under high-pressure. It has also been used to extract the interface PDF between a Fe and a  $\text{Fe}_3\text{O}_4$  phase.<sup>104-108</sup> Recently, efforts have also been made to develop an efficient and accurate NMF algorithm that can be used during data measurement.<sup>109, 110</sup> This NMF algorithm is available at PDF-in-the-cloud.<sup>111, 112</sup>

### *Inelastic Neutron Scattering: Extraction of a Materials Hamiltonians*

INS is an inelastic experimental technique for investigating the vibrational and magnetic properties of atoms in materials. During an INS experiment, a neutron beam interacts with the atomic nuclei and the magnetic moment of the electrons in the material. By measuring the initial and final neutron energy, one can determine the energy of the interaction or excitation, which allows to study of both the atomic and magnetic structure. Analysing the inelastically scattered neutrons thus provides information about the Hamiltonian, which governs atomic and magnetic interactions. However, interpreting experimental INS data or extracting the Hamiltonian can be challenging due to a large

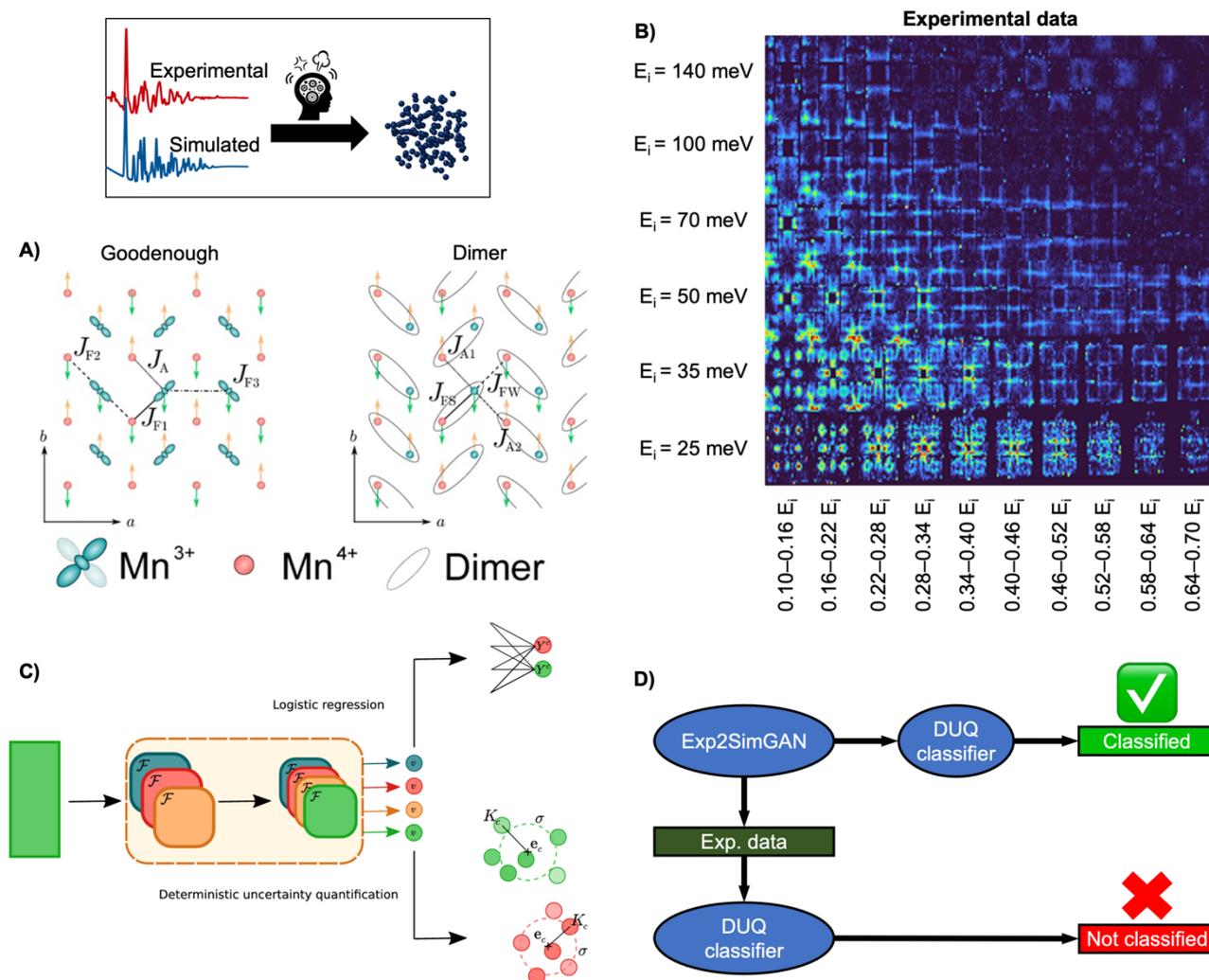
amount of measured data and the complexity of simulating accurate INS data that resembles the scattering process.

For example, determining the appropriate spin wave model of the half-doped bilayer manganite,  $\text{Pr}(\text{Ca}, \text{Sr})_2\text{Mn}_2\text{O}_7$  (abbreviated as PCSMO) has been debated, with a Goodenough spin wave model (Figure 9A),<sup>113, 114</sup> or a Dimer spin wave model (Figure 9A) being considered.<sup>115</sup> Due to the subtle differences between the two models, determining which model the INS spectra correspond to has been challenging, as it requires a meticulous manual fitting process. After extensive experimentation and careful data treatment, it was ultimately determined that the Goodenough spin wave model best describes the experimental data (Figure 9B).<sup>116</sup>

To ease this task, a supervised ML model has been developed to assist in analysing INS data. By training supervised ML models on simulated INS spectra calculated using physics-driven equations, Hamiltonians can be predicted from INS data. Specifically, Butler et al. demonstrated that NNs can predict magnetic Hamiltonians or classify the spin wave model from simulated INS data of PCSMO, saving significant time compared to manual data analysis.<sup>27</sup> They first used a logistic regression<sup>117</sup> model, illustrated in Figure 9C, which makes a simple binary classification, either Goodenough or Dimer, but gives no indication of the reliability of the prediction. It was thereby challenging to judge when to trust the model. To resolve this problem, they used a deterministic uncertainty quantification (DUQ) classifier (Figure 9C),<sup>27, 116</sup> to perform uncertainty classification instead. The DUQ classifier outputs a weight vector associated with the input that is correlated to the class predictions. If all the weights in the weight vector are close to a class, the prediction has a large certainty, while the certainty is larger with a larger spread of weight vectors.

To reliably predict the spin wave model from experimental INS data (Figure 9D), the DUQ classifier was trained on computationally expensive resolution convoluted INS spectra. Physics-driven simulations may not always capture the experimental noise, instrumental effects or other artefacts from phenomena not described by the underlying theory (issue 3 in Figure 3). In this example, the computationally inexpensive resolution unconvoluted INS simulations did not capture any instrumental effects. To address this issue, we introduced an unsupervised image-to-image algorithm, Exp2SimGAN, which is a generative adversarial network<sup>118</sup> (GAN) capable of learning the simulated and experimental data distributions and transforming between them, e.g. transforming a simulated dataset into one that resembles an experimental dataset, or vice versa.<sup>119</sup> By using Exp2SimGAN to convert experimental INS spectra into simulated-like data, the DUQ classifier, trained on computationally inexpensive resolution unconvoluted INS spectra, can be applied to the experimental

INS data (Figure 9E).<sup>119</sup> This approach helps bridge the gap between simulations and experimental data, allowing for more accurate and efficient analysis.



**Fig. 9** | Determining the spin wave model from experimental INS data using ML. A-B) Two magnetic exchange models in a single sheet of Mn ions in a half-doped manganite. (Left) Goodenough model (Right) Dimer model.<sup>27</sup> C) Schematic representation of the DUQ method. The input initially passes through a series of convolutional NNs (orange block) to extract features. In standard logistic regression, the outputs from the convolutional NNs are classified by summing the weights connecting each filter,  $f_i$ , to the class C of interest. This is a simple binary classification. The DUQ classifier instead outputs a weight vector associated with the input that is correlated to the class predictions. If all the weights in the weight vector are close to a class (based on distances,  $K_c$ , from the weight vector to the centre,  $e_c$ , of clusters of training examples), the prediction has a large certainty, while the certainty is larger with a larger spread of weight vectors. Figure 9C is from Butler et al.<sup>27</sup> (Under

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<https://creativecommons.org/licenses/by/4.0/>). D) 2D representation of experimental data of PCSMO measured at 4 K using the MAPS spectrometer.<sup>116</sup> The INS spectra are arranged in terms of incident neutron energy ( $E_i$ ) and bins of energy transfer  $\omega = 0.10-0.16E_i$ , etc.<sup>119</sup> E) The DUQ classifier cannot identify the spin wave model of an experimental dataset with high certainty. However, Exp2SimGAN matches the experimental dataset to the simulated training set of the DUQ classifier enabling the classification of the spin wave model with high certainty. Figure 9A+B+D are from Anker et al.<sup>119</sup> (Under Creative Commons Attribution 3.0 Unreported License <https://creativecommons.org/licenses/by/3.0/>).

Samarakoon et al. demonstrated an alternative approach for the analysis of INS data using autoencoders.<sup>120-122</sup> They showed that autoencoders can eliminate background signals and artefacts from the experimental INS spectrum by compressing them into a latent space. Once in the latent space, the magnetic behaviour can be categorised, and the autoencoder can solve the inverse problem by extracting the Hamiltonians from the experimental INS spectrum. This is achieved by decoding the INS spectrum from the latent space positions. As a result, the autoencoder works as a fast surrogate model for INS simulations accelerating the fitting procedure of the experimental INS spectrum. Later work integrates ML modelling approaches into the INS experiments enabling real-time analysis of INS data.<sup>123</sup>

### Remaining challenges and future outlook

In this perspective, we have shown how analysis of scattering and spectroscopy data is becoming increasingly feasible using supervised and unsupervised ML approaches. Especially, the progress of large open-source databases has catalysed the use of supervised ML.

Supervised ML is now widely used to identify structural models (Figure 2A) from data, to predict data from structural models (Figure 2B), or to directly provide structural information from data (Figure 2C). However, we have highlighted three issues that supervised ML faces in automating the analysis of scattering and spectroscopy data (Figure 3). Issue 1: handling datasets originating from a mix of chemical phases. Here, unsupervised ML, especially NMF, has successfully been used to demix datasets into constituent components. We anticipate the emergence of combination methods, where unsupervised ML firstly demixes complex datasets whereafter they are independently analysed using supervised ML. Issue 2: handling data from a structural model that is not part of a

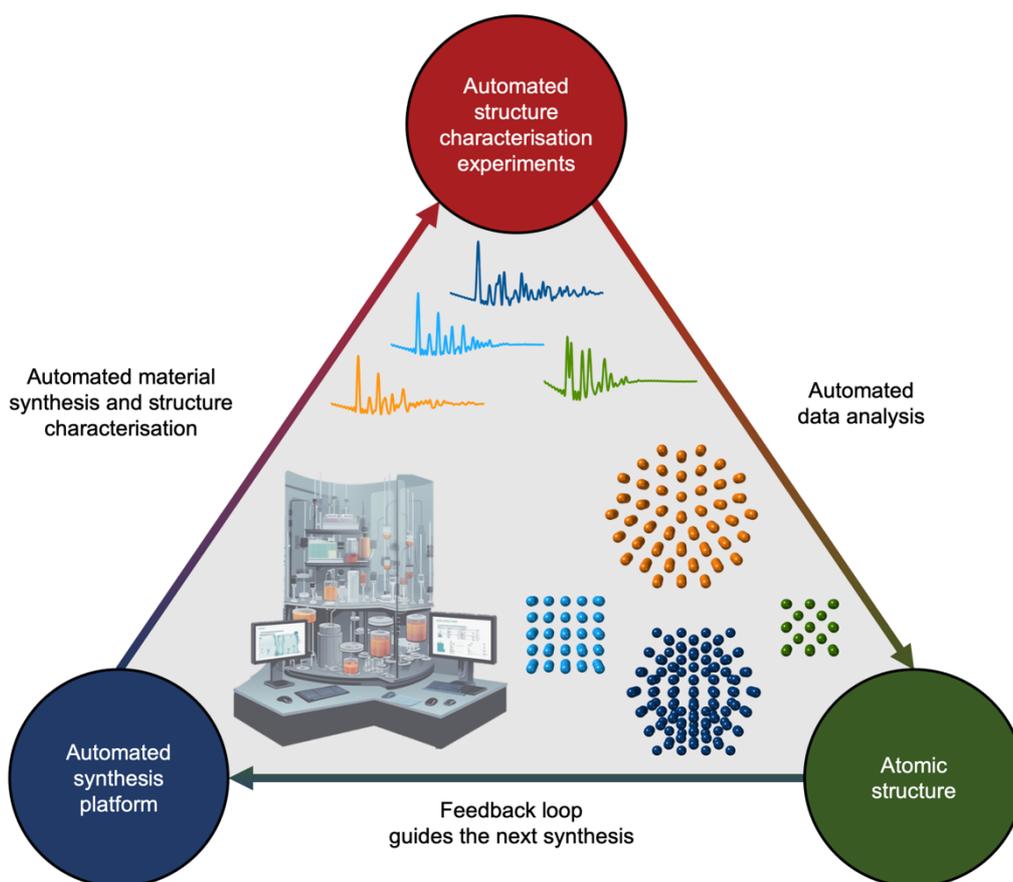
database. Here, generative modelling appears promising for interpolating between structural models in a database. Issue 3: handling experimental data. For the ML models to significantly impact the data analysis of scattering and spectroscopy data, they must perform well on experimental data and not only on simulated data. Often in materials chemistry, supervised ML models are trained on physics-driven simulations which do not include instrumental artefacts, noise or other phenomena not directly described by the underlying physics. Here, new methods are needed to make simulated data resemble experimental data. Unsupervised image-to-image algorithms could potentially address this challenge.<sup>119</sup>

However, using ML to resolve more complicated challenges in materials chemistry is still challenged by limited sizes of datasets connecting structure and spectroscopy/scattering signal. One way to handle limited data is to constrain the ML model with chemical knowledge. Here, physics-informed NNs serve as an inspiration, as they embed partial differential equations as constraints into the NN optimisation problem, for example, when using an NN as a surrogate model for the Schrödinger equation.<sup>124</sup> As a result, the range of potential solutions is limited to a manageable size for ML to handle with the available information. However, not all chemistry can be expressed as differentiable equations, necessitating the development of similar approaches that can incorporate chemical knowledge into the ML architecture as 'chemistry-driven ML'. Equivariant graph-based NNs show promise, as they leverage group representation theory to design architectures that are equivariant to specified symmetry groups, making them well-suited for analysing chemical systems with underlying symmetries.<sup>125</sup> We expect another impact to come from interpretable and explainable ML which enables researchers to understand the underlying mechanisms behind predictions, build trust in ML model outcomes, and uncover unexpected correlations that may lead to scientific insights. For those interested, we refer to a recent review paper by Oviedo et al.<sup>30</sup> for more about interpretable and explainable ML in materials chemistry.

Currently, it is not mandatory to publish data, code, and software requirements alongside research papers, making it difficult for other researchers to apply trained ML models to their own experimental data. A paradigm shift from publishing *papers with code* to publishing *code with papers* may thus be needed. For the ML developer, we refer to N. Artrith et al.<sup>126</sup> for best practices in ML for chemistry. We suggest that publishing *code with papers* would greatly benefit the field, allowing materials chemists to analyse data easily or automatically without domain expertise.

If we unceasingly share ML models, expand open-source databases, and bridge the gap between simulated and experimental data, the next decade holds promise to integrate analysis of scattering

and spectroscopy data with ML into self-driving laboratories. Self-driving laboratories are currently receiving much attention for e.g. identifying new, improved photocatalysts for hydrogen production from water,<sup>131</sup> synthesising pharmaceutical compounds,<sup>132</sup> and optimising nanostructure synthesis based on their optical properties.<sup>133, 134</sup> As illustrated in Figure 10, the self-driving laboratory will synthesise a material, perform a scattering or spectroscopy experiment, and the data can be automatically analysed with ML. The findings will then be fed into an active learning framework that suggests the next experiment based on structural insight.



**Fig. 10** | The proposed self-driving laboratory to integrate analysis of scattering and spectroscopy data with ML into self-driving laboratories. The automated experiment platform synthesises a material and performs one or multiple structure characterisation experiments. The data from these experiments will be analysed using ML. The analysed data will be automatically fed into an active learning framework that will suggest the next experiment enabling a directed synthesis of functional materials via insight at the atomic level.

## AUTHOR CONTRIBUTIONS

All authors were involved in the preparation of the manuscript and approved the final manuscript.

## COMPETING INTERESTS

The authors declare no competing interests.

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