Electrocatalytic enhancement of CO methanation at the metal-electrolyte interface studied by *in situ* X-ray photoelectron spectroscopy

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Abstract: For the direct reduction of CO₂ and H₂O in solid oxide electrolysis cells (SOECs) with cermet electrodes toward methane, a fundamental understanding of the role of elemental carbon as a key intermediate within the reaction pathway is of eminent interest. The present synchrotron-based *in situ* near-ambient-pressure X-ray photoelectron spectroscopy (NAP-XPS) study shows that alloying of Ni/yttria-stabilized-zirconia (YSZ) cermet electrodes with Cu can be used to control the electrochemical accumulation of interfacial carbon, and to optimize its reactivity toward CO₂. In the presence of syngas, sufficiently high cathodic potentials induce excess methane on the

studied Ni/yttria stabilized zirconia (YSZ)-, NiCu/YSZ- and Pt/gadolinium doped ceria (GDC) cermet systems. The hydrogenation of carbon, resulting from CO activation at the triple-phase boundary of Pt/GDC, is most efficient.

The co-electrolysis of $CO_2(g)$ and water to methane (reaction 1) in high-temperature SOECs is a subject of mitigating issues related to climate change by utilizing the main greenhouse gas for electric energy storage in the form of a renewable energy carrier¹:

$$CO_2 + 2H_2O \rightleftharpoons CH_4 + 2O_2$$
 (1)

The most common reaction pathway does not realize reaction (1) directly, as it deals with the simultaneous electrochemical reduction of CO_2 to CO and H_2O to H_2 . The resulting CO/H_2 syngas mixture can be utilized to form a variety of hydrocarbons. For this, an adjustable CO/H_2 ratio is required, e.g. 1:3 to form methane (reaction 2).

$$CO + 3H_2 \rightleftharpoons CH_4 + H_2O$$
 (2)

Based on thermodynamics, temperatures up to ~673 K are beneficial for CO methanation². In this study we (i) provide a proof-of-principle that reactive carbon species are formed particularly on Ni(Cu)/8 mol% yttria stabilized-zirconia (8-YSZ) cermets via electrolysis of CO, and (ii) show that additional methane is formed generally on metal-electrolyte interfaces such as Ni(Cu)/8-YSZ and Pt/10 mol% Gd-doped ceria (GDC-10) even at high temperatures (~973 K) via electrocatalytic promotion of CO methanation. If an additional non-faradaic nature (i.e., change in the metal's work function and adsorbate chemistry) of the reaction, according to the concept of "electrochemical promotion of catalysis" (EPOC) or "non-faradaic electrochemical modification of catalytic activity" (NEMCA)³, is present or not remains open.

In the co-electrolysis of CO₂ and H₂O, surface carbon chemistry plays a major role. Carbon can be deposited via different pathways, involving electrochemical (reaction (3)) or conventional

thermal reduction of CO (reaction (4), Boudouard reaction (5)). Commonly, these pathways are held responsible for surface coking and blocking of the triple-phase-boundary (TPB)⁴.

 $CO + 2e^{-} \rightleftharpoons C + O^{2-} \quad (3)$ $CO + H_2 \rightleftharpoons C + H_2O \quad (4)$ $2CO \rightleftharpoons C + CO_2 \quad (5)$

Vice versa, de-coking can proceed via the inverse reactions of (3,4,5) or, alternatively, via the reaction of carbon with co-generated hydrogen from electrochemical water splitting to form methane (reverse CH₄ pyrolysis reaction (6)):

$$C + 2H_2 \rightleftharpoons CH_4$$
 (6)

It is crucial for both, the efficiency of de-coking and of direct hydrocarbon formation, how reactive the material- and site-specific deposited carbon species are. In other words, equilibrium (6) imposes the need for a distinction between reactive *intermediate* and site-blocking *side product* carbon forms⁵. The carbon reactivity argument holds equally for the reversal of reaction (1), which needs to be realized for direct methane SOFCs⁶.

For the electrochemical promotion of CH₄ synthesis according to reaction (2), both the kinetic reactivity and the chemical potential of carbon formed directly at the TPB via reaction (3) are essential. Electrode materials, which allow for modulation of the bond strength of this TPB-generated carbon, denoted as C_{TPB} in the following, are the key toward a knowledge-based reactivity control. Ni_xCu_y is considered as a promising alloy, fulfilling these carbon-chemistry related requirements⁷. Moreover, it features decreased carbon solubility, which is essential for the suppression of unreactive graphene/graphitic deposits^{8, 9}. However, considering the thermo-catalytic methanation reaction of CO or CO₂, NiCu-alloyed catalysts perform worse than Ni^{10, 11}. Nonetheless, it is important to note that alongside the hydrogenation of intermediate C, the rate-limiting step is the dissociation of CO or CO₂¹². Anyway, the latter can be promoted by means of

electrochemistry (reaction (3)). As a result, it enables the utilization of electrocatalyst materials beyond the current scientific boundaries.

Taking the thermodynamic equilibrium constant K of reaction (6) into account, the carbon activity a(C) influences the equilibrium pressure p of CH₄:

$$K = \frac{p(CH_4)}{a(C) \cdot p(H_2)^2}$$
(7)

If additional carbon is induced electrochemically via reaction (3) in a H₂-containing atmosphere, the methane yield can be enhanced both by increasing the carbon activity via increasing the cathodic polarization, and by increasing the H₂ pressure¹³. Of course, the efficiency of the latter in terms of thermodynamic promotion of CH₄ yield, is limited by the kinetics of the reaction (6). In the following, a comparative approach, with respect to the C_{TPB} growth and clean-off on Ni/8-YSZ vs. Ni₈₀Cu₂₀/8-YSZ working electrodes (WE) and the C_{TPB}-promoted formation of methane on Ni(Cu)/8-YSZ (WE) vs. Pt/GDC-10 counter electrode (CE), is presented. This specifically designed CE is electrocatalytically highly active, featuring high surface area and an optimized number of TPB sites to minimize its overpotential contribution relative to the WE to negligible values¹⁴. The investigation of the interfacial carbon chemistry at defined atmospheres over cermet WEs was carried out via *in situ* near-ambient-pressure X-ray photoelectron spectroscopy (NAP-XPS) at the ISISS end station at the BESSY II synchrotron facility.

(i) Carbon-active electrodes

To highlight the difference in the carbon reactivity at the interface, visible in the kinetics of carbon growth and clean-off reactions, *in situ* XP spectra of the C 1s region of the Ni/8-YSZ and Ni₈₀Cu₂₀/8-YSZ electrodes were recorded at 973 K (Figure 1). In the order of the spectra from bottom to top, we applied initially 0.5 mbar CO at open circuit potential (OCP) to the electrode and detected a minor amount of mixed carbon-oxygenates and elemental carbon species in the C1s region (i.e., components are assigned to O-C=O_{ads}, C=O_{ads}, C-O-C_{ads} and C-C_{ads} species¹⁵). As the

Boudouard reaction (5) is strongly shifted toward CO formation under the chosen experimental conditions, we observed hardly any growth of the C1s signals after 15 min of isothermal exposure. Upon applying a voltage of -1.5 V, graphitic-like C_{TPB} at a BE of 284.7 eV can be electrochemically grown via reaction (3). After 15 min of CO electrolysis, the C_{TPB} growth was terminated by returning to OCP and CO pump-off. To study the efficiency of the oxidative clean-off reaction of this carbon species (reversal of reaction (5)), the electrode was then exposed to 0.13 mbar pure CO₂ under OCP conditions. For Ni/8-YSZ (left panel in Figure 1), only a slow decrease of the C_{TPB} component to ~87% of the initial value was observed after 5 min exposure. By increasing the CO₂ partial pressure up to 0.19 mbar and an exposure for additional 5 min, a further, but still rather slow decrease of the C_{TBP} peak area to ~58% of the initial value was achieved.



Figure 1. In situ XP spectra series of the C 1s region recorded on the Ni/8-YSZ and Ni₈₀Cu₂₀/8-YSZ WE at 973 K under the following conditions (from bottom to top): (a) 0.500 mbar CO, OCP (b) 0.500 mbar CO, OCP, after 15 min (c) 0.500 mbar CO, OCP, after 15 min -1.5 V (d) 0.13 mbar CO₂ for Ni/8-YSZ and 0.10 mbar CO₂ for Ni₈₀Cu₂₀/8-YSZ, both at OCP (e) 0.19 mbar CO₂ and 0.08 mbar CO₂ for the respective electrode.

In contrast, the amount of analogously grown C_{TPB} on the Ni₈₀Cu₂₀/8-YSZ electrode (right side in Figure 1) is diminished to ~44% already at a CO₂ pressure of 0.10 mbar after 5 min. After another 5 min (topmost spectra), the C_{TPB} peak started to vanish (i.e., 28% residual intensity),

although the CO₂ partial pressure was kept at an even lower value of 0.08 mbar. These results indicate that C_{TPB} is more reactive on the Ni₈₀Cu₂₀/8-YSZ electrode, at least for the oxidative carbon clean-off reaction (5). To confirm that carbon is accumulated via the TPB onto the cermet surface a XP spectra series (Figure S1) of Cu 2p Ni 2p, C 1s and Zr 3d were recorded before and after electrochemically induced carbon growth, showing a decrease in the cermet components. The Ni/Cu ratio in S1 changes from 2.95 before carbon growth toward 5.40 after. If this change is a consequence of the known alteration in the Ni to Cu surface composition upon electrochemically induced adsorbate chemistry¹⁶ or if it is geometrically effect due to carbon growth on a porous electrode shielding Cu/Ni areas differently or if it arises from a carbon growths predominantly localized on Ni sites remains open.

To elucidate the C_{TPB} reactivity toward H₂ on Ni/8-YSZ and Ni₈₀Cu₂₀/8-YSZ, the reaction mixture was changed to 0.125 mbar CO and 0.375 mbar H₂ (total pressure 0.5 mbar; CO:H₂ ratio = 1:3). The XP spectra of the C1s region shown in Figure 2, recorded *in situ* at 973 K, show no carbon growth at open-circuit-potential (OCP). After applying a cathodic potential of -1.0 V for 30 min on either system, a graphitic-like C_{TPB}-related peak appears only on the Ni₈₀Cu₂₀/8-YSZ electrode (BE ≈284.7 eV). According to pronounced differences regarding carbon reactivity and bulk solubility on NiCu vs. pure Ni⁸, we interpret this result in terms of hampered anti-segregation especially of C_{TPB} and its clean-off with H₂ (reaction (6)) on NiCu vs. Ni¹¹, leading to the accumulation of carbon at the surface. This raises the question whether this additional carbon species is kinetically available at all for enhanced methane formation through reaction (6). Equilibrium thermodynamics generally favor the reversal of (6) at increasing temperatures², but can be counteracted by the law of mass action (7), which predicts increasing methane pressures with increasing carbon activity a(C). In turn, a(C) can be varied via polarization through equilibrium (3). Figure S2 in the supporting information provides the respective calculated equilibrium partial pressures of CH₄ at 973 K as a function of a(C), based on data from¹⁷.



Figure 2. C1s XP-spectra recorded *in situ* at 973 K in 0.500 mbar CO:H₂ = 1:3 atmosphere at t = 0 s at OCP and after -1.0 V applied to the working electrode for 30 min (dark and light green traces, respectively). Panel A and B compare electrochemically induced C_{TBP} growth on Ni/8-YSZ vs. Ni₈₀Cu₂₀/8-YSZ (vertical dashed line).

(ii) Electrochemically promoted methane formation

In principle, the promotion of a(C) through reaction (3) via cathodic polarization can increase CH₄ formation thermodynamically through equilibrium (7). Whether the kinetics are in favor of a faster methanation, depends on the reactivity of the specific TPB sites, the H₂-reducibility of the electrolytically generated carbonaceous species, and the prevailing EPOC effect at the metal.

Cyclo-voltammetry (CV) combined with quadrupol mass spectrometry (QMS) allows to monitor methane formation on-line under high temperature conditions (973 K, Figure 3). The bottom region of Figure 3 represents the chosen sequence of linear sweeps of the potential (E_{WE} , scan rate = 50 mV/s, blue trace) and the resulting current (I, pink trace) as a function of time. As the SOEC is operated in a single reactant gas atmosphere (i.e., no separated anode and cathode compartments), the QMS detects gaseous species produced by both electrode sides, which makes a distinction between products formed at the WE vs. CE necessary (anode reactions: CO or H₂ oxidation). The potential was swept between -4 V and +4 V to generate high current densities and thus, appropriate quantities of products for the QMS detection both at the Ni₈₀Cu₂₀/8-YSZ WE and the Pt/GDC-10 CE (Figure 3B). In panel A, the CV sweeps were limited to a strong cathodic polarization of the WE only, in order to suppress superimposed QMS intensity via cathodic polarization of the CE. The time-dependent intensity changes of the CH₄-specific (m/z = 15) signal measured during the sweeps can be directly correlated to the alternating potentials applied to the WE/CE (violet shaded area for the CE) and the respective electrolysis current changes. For details regarding QMS data analysis and I vs. E curves we refer to the SI (Figure S3).



Figure 3. Top panels, light orange traces (raw and smoothed experimental data): electrochemically promoted methane formation caused by alternating the cathodic polarizations of the WE vs. CE. A: light blue traces: preferential cathodic polarization of Ni₈₀Cu₂₀/8-YSZ WE with max. 3V; B: alternating polarization of WE and CE with max. 4V. The respective absolute current changes are shown in the bottom panels as pink traces. To highlight the alternating cathodic operation of the CE vs. WE and the time-correlated changes of the m/z=15 mass signal of CH₄, negative potentials at the CE are highlighted by a violet-shaded background.

The main results derived from the data of Figure 3 are: 1) on both, the WE and the CE, the methanation reaction is promoted via cathodic polarization; 2) ongoing reaction (6) suggests the contribution of a faradaic enhancement in activity on both electrodes; 3) methane formation is much more promoted on the Pt/GDC-10 CE than on the WE, irrespective of Ni or $Ni_{80}Cu_{20}$ used as the metallic WE phase; 3) methane promotion at the CE is associated with a much smaller charge transfer for the CO electrolysis reaction (3) as compared to the WE, suggesting a relatively strong EPOC contribution on Pt/GDC; 4) the first cathodic polarization cycle of the CE yield no measurable CH₄ formation, but is expected to activate the CE for CH₄ formation for the second cycle; 5) once the CE is activated, the time-response of formation of additional CH₄ from C_{TPB} is much faster on the CE, as deduced from the synchronicity of cathodic WE polarization and the accelerated CH₄ increase; 6) CH₄ formation on the WE starts immediately in the beginning of the first cathodic polarization cycle, but then increases rather slowly and is associated with a smaller amount of CH₄ yielded from C_{TPB}, especially in view of the much larger integral charge transfer. The response of CH₄ formation to the alternating potential is very sluggish on the WE as compared to the CE.

In summary, the oxidative de-coking reaction of pre-deposited C_{TPB} in the presence of CO_2 ($C_{TPB} + CO_2 \rightarrow 2$ CO) works more efficiently on Ni₈₀Cu₂₀, supporting its previously reported, enhanced coking resilience in the presence of CO_2 relative to clean Ni⁸.

In contrast, the CO methanation results suggest that more C_{TPB} must have been formed electrochemically on Ni₈₀Cu₂₀/8-YSZ than on Pt/GDC-10, but remains kinetically less reactive toward methane formation. Considering the strongly different properties of the WE and CE materials, it appears reasonable that for both, less C_{TPB} accumulates on Pt in the presence of H₂, as carbon is weaker bonded and thus more reactive on Pt relative to Ni, and a more pronounced EPOC effect prevails on Pt. Moreover, H₂ activation is highly efficient on the carbon-free Pt surface. The fact that the onset of CH₄ formation is hardly delayed on the WE, but needs a full cathodic activation cycle on the CE, can be explained by the much higher oxygen exchange capacity of GDC in comparison to 8-YSZ. Whereas any oxygen vacancy in 8-YSZ generated electrochemically at the WE side is immediately quenched by splitting of CO toward lattice oxygen and C_{TPB} , GDC likely requires an enhanced reductive activation by withdrawing sufficient lattice oxygen before CO can become dissociated at the TPB.

Unfortunately, the original goal of this study, namely the quantitative distinction of the electrochemical enhancement on CO-methanation on clean Ni vs. Ni₈₀Cu₂₀, is not sufficiently reliable in view of the signal-to-noise ratio of the available QMS data. Ni/8-YSZ and Ni₈₀Cu₂₀/8-YSZ show almost indistinguishable CH₄ formation properties, as shown in Figure S4 of the SI. These operando XPS measurements validate the accumulation of additional CTPB only on Ni₈₀Cu₂₀/8-YSZ during the chronoamperometric experiments in a CO/H₂ atmosphere at -1.25 V, whereas on Ni/8-YSZ no C_{TPB} is visible at -1.00 V. However, the methane QMS signal (m/z = 15) increases similarly on both electrodes until the polarization is turned off. Figure S5 and S6 provides evidence of the metallic state of Ni and Cu anticipated under reductive CO/H_2 conditions. Together with the O1s region, no significant changes in the XP-spectra upon polarization (OCP vs. -1.25 V) were detected. Basically, one would expect NiCu to be a poorer thermal/non-electrochemical CO and/or CO₂ methanation catalyst than pure Ni^{10, 11}, but it remains unclear whether the proposed combination of faradaic and non-faradaic (EPOC) promotional effects can revert this trend. The preferential coking of NiCu under cathodic polarization in CO/H₂ may rather represent a hint for a delayed reaction of C_{TPB} toward CH₄, suggesting that NiCu may offer little, if any, advantages relative to pure Ni. If the electrochemically induced adsorbate and carbon chemistry affects the Cu/Ni segregation behavior and therefore alter gas surface interaction¹⁶ remains open and calls for non-porous model electrode systems.

Nevertheless, the remaining central message of this work refers to the appropriate choice of materials to create a (bi)metal – mixed oxide electrolyte interface for optimized thermodynamic enhancement of CH₄ yiel with cathodically promoted kinetic EPOC action. The combination of activated bound carbon and hydrogen on the metallic component with superior properties of the metal-electrolyte interface for C-O bond activation is the key for electrochemical enhancement of the CH₄ yield in the co-electrolysis of CO₂ and water via intermediate CO and H₂. Obviously, these preconditions are rather matched by Pt/GDC than by Ni(Cu)/YSZ.

Experimental Methods

For cell preparation, sample mounting, equipment and XPS data analysis we refer to the supporting information (Figure S7).

Supporting Information

XPS comparison of Ni/8-YSZ vs. Ni₈₀Cu₂₀/8-YSZ upon carbon accumulation, Thermodynamic promotion of CH₄ formation, QMS data analysis and Cyclovoltammetry I vs. E curves, *Operando* XPS comparison of Ni/8-YSZ vs. Ni₈₀Cu₂₀/8-YSZ in CO/H₂ atmosphere, SOEC cell preparation, sample mounting, equipment and XPS data analysis

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