Interpretable Data-Driven Modeling Reveals Complexity of Battery Aging

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Abstract

To reliably deploy lithium-ion batteries, a fundamental understanding of cycling and aging behavior is critical. Battery aging, however, consists of complex and highly coupled phenomena, making it challenging to develop a holistic interpretation. In this work, we generate a diverse battery cycling dataset with a broad range of degradation trajectories, consisting of 363 high energy density commercial Li(Ni,Co,Al)O2/Graphite + SiOx cylindrical 21700 cells cycled across 218 unique cycling protocols. We consolidate aging via 16 mechanistic state-of-health (SOH) metrics, including cell-level performance metrics, electrode-specific capacities/state-of-charges (SOCs), and aging trajectory descriptors. Through the use of interpretable machine learning and explainable features, we deconvolute the high-dimensional correlations that contribute to battery degradation. This generalizable data-driven mechanistic framework reveals the complex interplay between cycling conditions, degradation modes, and SOH, representing a holistic approach towards understanding battery aging.

Keywords: lithium-ion batteries, machine learning, data analytics

1 Introduction

Lithium-ion batteries are a key enabler for electrifying transportation and decarbonizing the electricity grid [1–8]. Optimizing new battery designs is challenging due to the need to simultaneously meet many

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performance targets while satisfying design constraints. Improving battery lifetime is especially difficult due to the slow, nonlinear, and coupled physics of the aging process [9–18]. It is time and resource consuming to observe the impact that design choices have on battery life and understand why one battery degrades more rapidly than another.

Characterization at the materials and cell level generates a mechanistic understanding of battery aging [19–22]; however, the throughput is relatively low [23]. In recent years, machine learning (ML) techniques have been developed to analyze battery aging through a data-driven lens [24–39]. While ML techniques are high in throughput, a purely data-driven approach overlooks key scientific and engineering insights. Despite the predictive power of complex black-box ML models (e.g., deep learning), the relationships between cycling conditions and battery aging mechanisms are unclear. On the other hand, physics-based electrochemical simulations, such as the Doyle-Fuller-Newman model [40–42], are physically interpretable. Nonetheless, predicting battery lifetime under unseen conditions remains challenging due to the complexity of interconnected aging phenomena [14] and model parameter identifiability [43]. Yet another approach are mechanistic models which involve estimation of electrode capacities and lithium inventory [44–46]. These models capture aggregate physical mechanisms with fewer model parameters than physics-based simulations [47–51]. Tracking electrode capacities independently provides a clear picture of what types of degradation occur under various operating conditions [23, 52].

A challenge with developing and benchmarking battery aging models is that publicly available datasets do not contain a wide range of operating conditions. Existing datasets are typically collected with specific applications in mind [53–57]. For example, Attia, Severson, and colleagues focused on optimizing electric vehicle fast charging protocols [58, 59]. Diao et al. examined different temperatures to understand how temperatures accelerate battery aging [60]. Paulson, Ward, and colleagues tested various cell chemistries to understand the differences in their aging and build transferable ML models [34, 61]. As a final example, Wildfeuer et al. examined different state-of-charge (SOC) ranges and temperatures in both cycling and calendar aging tests to investigate different experimental factors, but did not apply ML techniques to analyze the large dataset [62]. There remains a significant gap in interpretable data-driven models that can be comprehensively applied to large datasets. The lack of available data spanning many use cases, including a wide range of SOC, charging, and discharging protocols, further compounds this challenge.

In this work, we develop a physically interpretable, data-driven understanding of lithium-ion battery aging. We generate a large dataset consisting of 363 cells under 218 unique cycling conditions spanning diverse use cases and aging trajectories. We apply interpretable ML with explainable features to track
16 mechanistic SOH metrics. With this framework, we begin to answer three principal questions: 1) how
do cells degrade? 2) when will cells degrade? and, most critically, 3) what factors influence degradation?

We demonstrate that physically meaningful features must be used in combination with methods that
robustly extract feature importance \cite{63-67}. Our approach of using interpretable features also reveals
which and how mechanistic SOH metrics can be predicted from early cycle data, addressing the challenge
that features used for early prediction tasks are difficult to meaningfully interpret, such as the features
employed in Severson et al. \cite{58}. With our explainable data-driven model, we analyze and understand
battery aging further than would be possible with either a data-driven or physics-based approach alone.

More generally, constraining ML models to use features that have clear physical meaning dramatically
enhances interpretability and explainability, complementing purely data-driven featurization approaches.
Fig. 1: Overview of dataset. 

a) The scope of our dataset across various cycling conditions is highlighted in the inscribed spider plot in blue compared to other large, publicly available battery cycling datasets [34, 53, 58–62, 68]. All batteries in this dataset are cycled at 25°C. The cycling experiment structure is shown schematically with the loop surrounding the spider plot. Individual cells go through a diagnostic “checkup” cycle, followed by 100 aging cycles repeating until end of life (EOL). 

b) The diagnostic cycle consisting of a reset cycle, a hybrid pulse power characterization (HPPC) [69], and three rate performance tests (RPTs) at 0.2C, 1C, and 2C discharge currents (see SI Table S2 for full conditions). Mechanistic SOH metrics are extracted from various parts of this diagnostic cycle data (see SI Section S.3 for further details).

c) The distribution of rate-dependent capacities at beginning of life (BOL). Means and coefficients of variation are included in the plot showcasing the tight distribution at BOL.

d) The distribution of rate-dependent capacities at end of life (EOL, defined by 0.2C RPT capacity reaching 80% of the nominal capacity, 4.84Ah). The broadened distribution showcases diverse aging and highlights the limitations of using a single mechanistic SOH metric such as the low-rate capacity (for further information on BOL to EOL variability see SI Section S.5) [70].
2 Comprehensive Aging Characterization

Our dataset contains electrochemical data from 363 Li(Ni,Co,Al)O$_2$/Graphite + SiO$_x$ cylindrical 21700 cells cycled for over 2 years (Fig. 1a). To induce diverse aging trajectories, we explore a broad range of cycling voltage windows and charging and discharging rates (see Section S.2.2 for details). To cleanly compare the effects of different cycling conditions, we apply a standardized, periodic diagnostic cycle to comprehensively probe SOH over cell lifetime (typically every 100 aging cycles – Fig. 1b). Given the variation of voltage windows and charge and discharge rates throughout the dataset, we compare cell lifetimes using capacity throughput-based equivalent full cycles (EFCs) [71]. In total, we examine 218 unique aging protocols, with EFCs at end of life (EOL) ranging from 44 to 994 (or 63 to 4,641 cycles). EOL is defined as when the 0.2C rate-specific capacity, $Q_{RPT.0.2C}$, reaches 80% of the nominal capacity (where 1C is 4.84A). Fig. 1a compares the diversity of our cycling conditions to other public datasets. Critically, we realize that a single health metric, such as low-rate capacity, does not capture all facets of degradation (Fig. 1c,d, and SI Section S.5). To address this gap, we automatically calculate and track 16 mechanistic SOH metrics (see SI Section S.1.1 for summary of abbreviations).

We first quantify six cell-level performance metrics: 1) total EFCs at EOL, 2) 1C rate-specific capacity: $Q_{RPT.1C}$, 3) 2C capacity: $Q_{RPT.2C}$, 4) ohmic resistance: $R_{ohm}$, 5) charge transfer resistance: $R_{ct}$, and 6) polarization resistance: $R_P$. We calculate resistances through pulse measurements performed during the hybrid pulse power characterization (HPPC) sequence of the diagnostic cycle at various SOCs and timescales (see SI Section S.3.2 for definitions and calculation details for resistance metrics). Unless otherwise specified, the resistances reported are at 50% SOC.

Second, to determine electrode-specific capacities/SOCs, we implement a mechanistic model-fitting algorithm to extract seven interpretable quantities (see Methods for details): 1) negative electrode capacity: $Q_{NE}$, 2) positive electrode capacity: $Q_{PE}$, 3) lithium capacity: $Q_{Li}$, 4) State of charge of the negative electrode near the full cell charged state: $SOC_{NE.4.0V}$, 5) state of charge of the negative electrode in the discharged state: $SOC_{NE.2.7V}$, 6) state of charge of the positive electrode near the charged state: $SOC_{PE.4.0V}$, and 7) state of charge of the positive electrode in the discharged state: $SOC_{PE.2.7V}$. We select the latter four quantities because the electrode-specific SOC near the fully discharged and fully charged states can dominate aging.

These cell-level and electrode-specific metrics are calculated at every diagnostic cycle for each cell and tracked from beginning of life (BOL) to EOL. As would be expected for commercial cells, these metrics have low variability at BOL (Fig. 1c). Importantly, this low variability also confirms that the staggered start of cycling (resulting in different calendar aging in the discharged state) does not contribute
significantly to the initial conditions of the cells (SI Section S.5). However, by EOL there is high variation in the rate capability, resistance, and electrode-specific capacities/SOCs (Fig. 1d, and SI Fig. S8). This observation underscores the importance of using a comprehensive set of SOH metrics, and confirms that the cycling conditions in this work induce a wide range of degradation trajectories.

In addition to probing cell-level and electrode-specific metrics with each diagnostic cycle, we also quantify the aging trajectory over the entire battery lifetime [9, 72]. We define three trajectory descriptors:

1) knee indicator: **Knee**, 2) resistance growth factor: **R”**, and 3) negative/positive capacity (N/P) ratio: **NP Ratio**. The knee indicator describes a sudden and accelerated capacity-based degradation (i.e., a knee in the capacity vs. cycle number curve) with knee indicator > 0 if a knee exists at any point in the cell lifetime. The resistance growth factor captures the curvature of resistance with respect to EFCs, indicating whether resistance grows at an accelerating or decelerating rate during cycling. Finally, the NP Ratio captures the ratio of the estimated $Q_{NE}$ and $Q_{PE}$. SI Section S.7 details the calculations of these trajectory descriptors.

We combine these 16 cell-level performance metrics, electrode-specific capacities/SOCs, and trajectory descriptors (collectively called mechanistic SOH metrics) and comprehensively quantify battery aging. By concurrently assessing these metrics, we reveal their relationships to 218 cycling conditions to develop a holistic understanding of aging. Fig. 2 visualizes selected metrics calculated on all cells in the dataset.
Fig. 2: Mechanistic SOH metric trajectories. a) The cell-level performance metrics column show the trajectories of selected performance metrics: the 0.2C RPT discharge capacity ($Q_{RPT0.2C}$) (top) and the combination of $R_{ohm}$, $R_{ct}$, $R_p$ ($R_{tot}$) at 50% SOC (bottom). b) The electrode-specific capacities/SOCs column depict the trajectories of electrode-specific capacities, $Q_{NE}$, $Q_{PE}$, and $Q_{Li}$, on the left. A utilization plot showing electrode-specific SOCs at the charged and discharged state is shown on the right. c) The trajectory metrics row shows histograms of the values for the NP ratio, resistance growth factor, and knee indicator. The highlighted protocol (in dark blue) represents CC$_{\text{discharge}}$ = 0.2C, CC$_1$ = CC$_2$ = 0.2C, V$_{\text{charge}}$ = 4.2V and V$_{\text{discharge}}$ = 2.7V aging conditions. This protocol has four experimental repeats shown by the scatter markers with the solid line representing the mean trajectory. The gray lines in the background showcase the mean trajectory of all other unique protocols. This protocol appears as a blue vertical bar in the trajectory metric histograms.

3 Impact of Cycling Conditions on Mechanistic SOH Metrics

By varying six cycling parameters across this dataset (SI Table S4), we induce a diverse range of EOL states and trajectory descriptors (Fig. 2a-c). To understand the impact of cycling conditions on mechanistic SOH metrics, we construct nonlinear random forest ML models, and then employ Shapley additive explanations (SHAP) analysis [73] to interpret these models.

We first develop descriptive models using cycling protocol parameters alone as inputs. These “protocol models” predict a single cell-level performance metric, such as an electrode-specific capacity or a
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trajectory descriptor at EOL conditions. Fig. 3a schematically depicts the structure of these models, with the cycling protocol conditions as the input features and the EOL mechanistic SOH metrics as the target outputs. With only cycling conditions as input features, the EFC model attains good performance on the training/test set (SI Section S.9.5 for performance, and SI Section S.9.1 on train/test split). Well-performing models are critical in order to extract the correct feature importance. To understand the impact of the various cycling conditions, we investigate feature importance using SHAP analysis [74]. Fig. 3b shows an example of SHAP feature importance for predicting EFC. In Fig. 3c, we devise a 96 element aging matrix representation that comprehensively visualizes how cycling conditions affect each mechanistic SOH metric, where the color indicates the magnitude of feature importance for all cycling conditions.

While no single cycling parameter dominates all mechanistic SOH metrics, it was surprising that many of the metrics are either primarily determined by a single cycling parameter, or a combination of features (e.g., the NP Ratio, a combination of \(Q_{PE}\) and \(Q_{NE}\)) that are primarily influenced by a single cycling parameter. For example, the cell-level performance metrics \(Q_{RPT,1C}\) and \(Q_{RPT,2C}\) are dominated by \(CC_2\), the resistances \(R_{ohm}\) and \(R_{ct}\) are dominated by \(V_{charge}\), while \(R_p\) is dominated \(CC_1\). Some more convoluted metrics, such as the EFC, depend on multiple parameters; both \(CC_1\) and \(V_{discharge}\) are about equally important. Surprisingly, \(V_{discharge}\) and \(t_{CV}\) do not dominate aging (within the bounds of this dataset) for most of the mechanistic SOH metrics (except for EFC), despite previous reports stating their importance [75, 76].

For the electrode-specific capacities, both the positive and negative electrode are strongly affected by the magnitude of the current in the direction of lithiation. This current is \(CC_{discharge}\) for \(Q_{PE}\), and \(CC_1\) for \(Q_{NE}\). Since \(Q_{Li}\) also depends most strongly on \(CC_1\), it is possible that \(CC_1\) triggers mechanisms that age both \(Q_{NE}\) and \(Q_{Li}\), such as solid-electrolyte interface (SEI) growth. The electrode-specific SOCs, calculated from electrode-specific capacities, depend most strongly on \(CC_1\) and \(CC_{discharge}\) (the most important features of the electrode-specific capacities), approximately equally. Finally, for the trajectory metrics, the knee indicator depends most strongly on \(CC_1\) and \(CC_{discharge}\), the resistance growth factor (\(R^*\)) on \(CC_1\) and \(CC_2\), and NP ratio on the values it was constructed from, in this case, both the dominant feature from \(Q_{NE}\) (\(CC_1\)), and from \(Q_{PE}\) (\(CC_{discharge}\)). For detailed information on influence of cycling conditions on mechanistic SOH metrics, as well as model performance see SI Section S.9.5

With our aging matrix representation generated by interpretable ML, a battery cell designer could more intelligently identify aging mechanisms and design cycling limits. For example, if it is important
to prevent capacity knees, from this analysis, we see that modifying CC\(_1\) and CC\(_{\text{discharge}}\) will have the greatest impact, whereas modifying the V\(_{\text{discharge}}\) would not be effective.

Fig. 3: Impact of cycling conditions. a) Schematic of inputs and outputs of the protocol models. Gray rectangles indicate cycling parameters, and red rectangles indicate the EOL mechanistic SOH metrics. b) SHAP feature importances for the protocol model predicting EFCs, marked by a dashed box in a. The color indicates the feature value, and horizontal location indicates the SHAP value impact on EFCs. Features are listed in descending order of importance. c) Replicating this approach for each mechanistic SOH metric, the matrix shows the mean absolute SHAP value of each cycling condition for each degradation metric. Darker hue indicates stronger dependence. Additionally, the RAE (relative absolute error) column indicates the test error of the models trained to predict a particular mechanistic SOH metric. SI Section S.9.5 shows the parity plot and SHAP beeswarm plot for each mechanistic SOH metric. This degradation matrix representation visualizes the impact of cycling conditions on degradation in a high-dimensional space.
4 Fundamental Investigation of Performance Metric Degradation

Having revealed the relationship between 16 mechanistic SOH metrics and cycling conditions using an aging matrix, we now demonstrate the explanatory nature of our framework by answering one important, exemplar question: “how does degradation at specific electrodes contribute to resistance growth in a battery?” Resistance growth during aging can limit the discharge capacity and energy of a battery. However, it is challenging to understand where inside a battery resistance growth originates using only full cell measurements because of the convolution of multiple effects from both electrodes. The resistances of individual electrodes are highly dependent on their respective electrode’s lithiation state and degradation. In addition, as cells age under diverse usage conditions, individual electrodes can go through various degradation pathways such as cathode structural changes [77] and anode solid electrolyte interface (SEI) formation [78]. These changes lead to varying degrees of electrode slippage or SOC shifts, adjusting the relative lithium composition of the cathode and anode at a given full cell SOC (SI Fig. S22).

To understand the complex relationship between electrode degradation and resistance growth in a full cell, we expand on the “protocol model” discussed in the previous section to include EOL electrode-specific capacities/ SOCs metrics as input features. This “explanatory model” aims to learn the relationship between the physically meaningful electrode-level features and the mechanistic SOH metric of interest (Fig. 4a). In this section, we investigate the changes in the electrode-specific capacities/SOCs and resistances with cycling. As such, the model inputs are the changes in SOH metrics from BOL to EOL (represented by ∆). The model output here is the low SOC (30%) total resistance (summation of $R_{\text{ohm}}$, $R_{\text{ct}}$, and $R_p$, SI Section S.3.2). We choose this health metric as the example target of our explanatory model because resistance at low SOCs are typically the largest and limit the discharge capacity.

Fig. 4b lists the most dominant features contributing to the observed total resistance growth. From the SHAP analysis, we observe that two electrode-specific features, $\Delta S$OC$_{2.7V}$ and $\Delta S$OC$_{2.7V}$, are dominant features impacting the total resistance but show opposite relationships with resistance growth (Fig. 4b,c). Surprisingly, negative electrode over-discharging ($\Delta S$OC$_{2.7V} < 0$) leads to lower resistance increase. This is unexpected because electrode kinetics are typically most sluggish at the SOC extremes; therefore, at low SOC, we expect that resistance should increase in the direction of deeper discharge for an electrode [76].

To understand the origin of this effect, we recall how $\Delta S$OC$_{2.7V}$ and $\Delta S$OC$_{2.7V}$ are calculated. These quantities are calculated at a specified full cell voltage (2.7V for this example) and, as a result, are
highly correlated (Fig. 4d, SI Section S.8.1). This correlation arises because when one electrode’s SOC shifts, regardless of the aging mechanism, the other electrode’s SOC must shift in the opposite direction to produce the same measured full cell voltage (SI Fig. S22 explores this in further detail). In general, SHAP is unable to differentiate between highly correlated features, and repeating the SHAP analysis multiple times reveals that either $\Delta \text{SOC}_{\text{PE},2.7V}$ or $\Delta \text{SOC}_{\text{NE},2.7V}$ can emerge as the most dominant feature (SI Fig. S20). However, if $\Delta \text{SOC}_{\text{NE},2.7V}$ is removed from this explanatory model, for example, $\Delta \text{SOC}_{\text{PE},2.7V}$ appears as the dominant feature (SI Fig. S21). From this analysis, we understand that, while negative electrode over-discharging ($\Delta \text{SOC}_{\text{NE},2.7V} < 0$) leads to lower resistance increase, the correlated metric positive electrode over-discharging ($\Delta \text{SOC}_{\text{PE},2.7V} < 0$) leads to higher resistance increase, in line with the understanding that electrode kinetics are most sluggish at SOC extremes. Combining statistical analysis with scientific understanding of battery materials, we rationalize that low SOC resistance rise is dominated by the over-discharging of the positive electrode.

Our framework exemplifies the value of SHAP as a tool for identifying correlations between input features and the target mechanistic SOH metrics. While the ML method alone does not differentiate between the contributions from two highly correlated electrodes, the explainable features together with scientific knowledge helps to hypothesize causation. Although we choose in this section to highlight and analyze low SOC resistance as one example, we emphasize that the approach generalizes to any mechanistic aging feature of interest (SI Section S.8.3).
5 Early Prediction Using Explainable Features

Finally, we quantify and rationalize the predictive power of explainable features in early cycles, and demonstrate the value of features extracted from the early diagnostic cycles for early prediction of the 16 EOL mechanistic SOH metrics. Building upon our protocol model in which random forest regression...
models were employed to correlate EOL mechanistic aging features to cycling parameters, we construct a “diagnostic-aided model” that uses both cycling parameters and early values of the mechanistic SOH metrics (specifically, the evolution between the 1st and 3rd diagnostic cycle, Fig. 5a) as inputs to our interpretable ML model. The inclusion of features from early diagnostic cycles differentiates between cells with the same cycling parameters, giving insight into cell-to-cell variability in fixed aging conditions.

We perform similar SHAP analysis as demonstrated in the previous sections on our diagnostic-aided model and present the results in an aging matrix plot in Fig. 5b (see SI Section S.9.5 for parity plots and full shap analysis). For the mechanistic SOH metrics, the diagonal entries of the degradation matrix correspond to self prediction (i.e., predicting the EOL value of a given metric using its early value). Interestingly, while the features on this diagonal might be expected to consistently be the most predictive, this is not always the case. For example, the early prediction of $R_{ct}$ is dominated by $V_{charge}$. Additionally, the early prediction of EFC is dominated by $R_p$, rather than by $Q_{RPT_{0.2C}}$; the latter is the metric used to define the EOL cutoff, and thus EFC at EOL. The result highlights the importance of a detailed tracking of battery SOH. While a given degradation mode might dominate the EOL values of certain mechanistic SOH metrics, the best early indicators for the onset of that mode may be a different metric or set of metrics.

Since SHAP analysis cannot differentiate between correlated input features, in order to draw robust conclusions about the importance of early cycle features, it is necessary to also consider a “diagnostic-only” model, excluding cycling parameters as input features (SI Section S.9.4). In principle, this may affect the relative feature importance of the early cycle features which correlate with specific cycling parameters. In addition, this type of model may be preferred in cases where you either do not directly have access to cycling conditions, cycling conditions are kept constant, or the relationship to cycling conditions is not the focus [79]. In this case, the exclusion of cycling conditions does not meaningfully affect the ranking of the feature importances (see SI Section S.9.4 for further details).
6 Conclusions

In this study, we develop a holistic framework for revealing and explaining coupled battery aging pathways by combining interpretable ML, physically-derived mechanistic SOH metrics, and a diverse dataset spanning over 200 distinct cycling conditions. By tracking a comprehensive set of 16 mechanistic
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aging features, we fully describe the battery SOH through an aging matrix, provide insight into battery degradation mechanisms and also identify mechanistic features from early cycles that enable early predictions.

Through our interpretable ML framework, we deepen our physical intuition on battery degradation with a diverse dataset. While interpretable ML tools can be used to generate hypotheses and summaries of the dataset, the findings must be further validated with physical characterization to gain confidence. We urge the field to use the dataset presented here to expand upon this work while keeping interpretability in mind as to enrich our understanding of battery degradation.

7 Methods

7.1 Data Cycling and Generation

All cells in this study were harvested from a newly purchased 2019 Tesla Model 3. These 21700 cylindrical cells were manufactured by Panasonic and tested to have a low-rate capacity of 4.84Ah. The positive electrode is NCA (approximately 90-5-5 composition) and the negative electrode is a graphite-SiO\textsubscript{x} blend.

Cells were cycled in CSZ ZP-16-2-H/AC environmental chambers set to 25\textdegree C, and fitted with 4-point contact cylindrical cell fixtures from Korea Thermo-Tech Co. Ltd. assembled by SpectraPower. The cells were cycled using two 96 channel Maccor Series 4000 battery cyclers.

The cells are subject to two types of cycling: aging cycles and diagnostic cycles. The aging cycle consists of a multi-step CC-CV charge and a CC discharge. Information on cycling protocol, parameters varied and their distribution see SI Section S.2.2. The diagnostic cycle consists of three main portions: a reset cycle, a hybrid pulse power characterization (HPPC) cycle [69], and a rate performance test (RPT) sequence. The reset cycle, resets the transient kinetics due to the aging cycles, HPPC probes resistance at different SOC increments, and the RPT extracts rate-dependent capabilities (Fig. 1b). For information on diagnostic cycle protocol see SI Table S2. This cycling data is automatically backed up to an S3 bucket and subsequently processed through the BEEP processing pipeline for use in analysis [80].

7.2 Differential Voltage Fitting

We implement differential voltage fitting (DVF) to estimate properties of the battery at the electrode level. Similar methodologies have been implemented by other groups [23, 48, 50, 81, 82]. This method extracts electrode capacities and lithium inventory: $Q_{PE}$, $Q_{NE}$, and $Q_{Li}$. Additional information, such as the SOC of either electrode at a full cell specified voltage is further calculated: $SOC_{PE,2.7V}$, $SOC_{NE,2.7V}$, $SOC_{PE,4.0V}$ and $SOC_{NE,4.0V}$. The DVF routine employed non-invasively probes degradation by fitting
the measured 0.2C RPT full cell differential voltage profile with an emulated full cell profile by stretching and translating the voltage profiles of the cathode and anode. Details of the fitting methodology and feature extraction are provided in SI Section S.3.3. Additionally, a comparison to DVF performed at C/40 is provided in S.6.1.

Reference voltage profiles for the cathode and anode are acquired through destructive tear down of the full cell to extract cathode and anode sheets. Portions of the sheet are then cycled in a pouch cells with a lithium counter electrode at various low rates. Details of the experimental electrode extraction and measurement procedure are in SI Section S.4.

7.3 Machine Learning Models

Random forest regression was chosen as the machine learning model of choice for all models in this work due to its ability in capturing non-linear relations with input features. We first generate a train/test split of the data, and cross validation folds on the training split. The cells that go in to the different splits are chosen randomly for the explanatory model, but an inside-of-domain testing scenario for the protocol only, diagnostic-only, and diagnostic-aided models (see SI Section S.9.1 for details). Random forest hyper parameters are optimized via grid search cross validation. From the subsequent trained model we report the RAE metric to accurately compare the prediction performance on different mechanistic SOH metrics of different scales and distributions. To determine feature importances we then use the SHAP python library on the fitted model to extract SHAP values for all features and datapoints (see SI Section S.9.5 for full parity plots and SHAP analysis). To summarize this information, we then take the absolute mean feature importance and report this value in the matrix plots.

8 Data Availability

Raw data and data structured via the BEEP pipeline [80] will be available at the time of publication

9 Code Availability

Code for figure generation, and random forest model building will be available at the time of publication.

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11 Author contributions

V.L., B.V.V., X.C., W.E.G., W.C.C., conceived and conducted the full-cell cycling experiments. V.L. compiled the manuscript, figures and edits from authors. P.A.A. and B.V.V. conceptualized, implemented, and constructed visualizations for the protocol-only, diagnostic-aided, and diagnostic-only machine learning models and their SHAP analysis. P.A.A., B.V.V., V.L., and X.C. conceptualized the mechanistic SOH metrics used. X.C. conceptualized and implemented the explanatory model and its SHAP analysis. P.A.A. conceptualized and implemented the DVF algorithm in strong collaboration with B.V.V., P.K.H., C.B.G., S.S, A.T. performed data management and data processing pipeline support. N.G., V.L., and H.L.T. designed the methodology for full cell disassembly, pouch cell assembly, and half cell voltage extraction. N.G. and V.L. conducted the pouch cell experiments. D.G., V.L., B.V.V., X.C. and P.A.A. contributed to conceptualizing and implementing the visual representations of the work. All authors edited, reviewed and discussed the work. W.C.C. and R.D.B. supervised the work.
12 Competing interests


13 Additional Information

Supplementary information will be available for this paper at publication. Correspondence and requests for materials should be addressed to W.C.C. or R.D.B.