# **Ethanol dehydrogenation to acetaldehyde with mesoporous Cu-SiO<sup>2</sup> catalysts prepared by aerosol-assisted sol-gel**

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#### **Abstract:**

Copper based catalysts are central for carrying dehydrogenation reactions. However, these materials are prone to deactivation by sintering and coke deposition. Irreversible sintering occurring during reaction (under the effect of temperature) is known to decrease both activity and selectivity, where the unwanted dehydration activity of the support might also play an important role. From this perspective, the quite unreactive silica supports may be attractive. However, using classical catalyst preparation methods (e.g. impregnation), it is a challenge to obtain a stable and homogeneous dispersion of Cu over  $SiO<sub>2</sub>$  owing to the weak support-active phase interactions. Taking a sidestep, aerosol-assisted sol-gel is a promising alternative for the facile preparation of mesostructured metallosilicates with high metal dispersion. Here we report, for the first time, Cu-SiO<sup>2</sup> made by the aerosol-assisted sol-gel method and exploited in the ethanol non-oxidative dehydrogenation to acetaldehyde. These catalysts are compared with a series of catalysts made by impregnation to investigate, through a thorough characterization survey, the effect of the synthesis procedure as well as the effect of Cu loading. We show that aerosol-made catalysts do not suffer heavy sintering, reach high ethanol conversions with acetaldehyde selectivity above 75%, and only slowly deactivate upon time due to a (reversible) coking phenomenon.

# **1. Introduction**

Nowadays, many efforts are being pursued to reduce our dependence on fossil ressources consumption and to obtain chemicals and fuels from renewable resources. In this field, bioethanol is a prominent platform chemical that can be produced from non-edible biomass feedstocks through fermentation<sup>1</sup>. Further upgrading of bioethanol gives access to several important chemical intermediates  $2,3$  such as ethylene, acetaldehyde, acetone and hydrogen. Certainly, acetaldehyde is an important target that can then be further transformed to produce valuable fine and bulk chemicals such as carboxylic acids, ketones, and esters <sup>4</sup>. Industrially, it is produced via the well-established Wacker-Hoechst process (oxidizing ethylene to acetaldehyde in the presence of an aqueous solution of palladium chloride and copper chloride). Suitable alternatives – starting from bioethanol – are desirable but stable and robust catalysts for this application are still lacking<sup>5</sup>.

Noble and non-noble metal based catalytic systems have been widely studied for this reaction, and copper appears as the most suitable compromise when considering performances on the first hand and metals availability and price on the second hand  $6$ . Bueno et al.  $7-9$  reported that the catalytic activity of Cu-ZrO<sub>2</sub> materials depends strongly on Cu nanoparticles size and on their interaction with the support. Hanukovich et al. <sup>10</sup> observe that Cu surface over  $ZrO_2$ , TiO<sub>2</sub>, AlZrO<sub>2</sub> supports becomes poisoned with reactive intermediates, pointing out the need to choose support with moderate Lewis acidity, to avoid the formation of side products that trigger fouling. Copper-based catalysts supported on Al<sub>2</sub>O<sub>3</sub>, ZnAl<sub>2</sub>O<sub>4</sub>, MgAl<sub>2</sub>O<sub>4</sub>, have also been investigated  $11-15$ . These catalysts are active and selective to acetaldehyde (>90%) also at high conversion, but a progressive deactivation appears inevitable, owing to an interplay between Cu sintering and carbon deposition <sup>16</sup>. While sintering is an irreversible effect, mild oxidative treatments could efficiently rejuvenate the catalysts by burning carbonaceous residues. Carbon supported Cu- catalysts  $17-19$  showed good activity levels at 623 K, but dehydration reactions - mainly associated with the presence of oxygen-containing acidic groups on the carbon surface – limit the selectivity of the reaction.

Silica-supported Cu-based catalysts are reported as highly selective to acetaldehyde because of the relative inertness of silica towards side reactions such as dehydration  $^{20}$ . Thus, Cu/SiO<sub>2</sub> catalysts are also suitable model materials to study the behaviour of Cu nanoparticles <sup>9,21,22</sup>. Using such approach, a marked structure-relationship was demonstrated  $2<sup>3</sup>$ . Cu particle size strongly influence selectivity because it dictates the density of (catalytically active) coordinatively unsaturated sites, such as corners and kinks. Nanoparticles around 10 nm have been found to be an optimum for acetaldehyde production while larger particles seem to favour side products such as ethyl acetate <sup>24</sup>. Unfortunately, achieving high active phase (Cu) dispersion on silica is complicated with classical synthesis methods and intricate grafting or colloidal catalyst synthesis procedures have to be implemented  $25-27$ . Moreover, these catalysts remain subject to deactivation by coking and are very prone to deactivation by Cu sintering <sup>28</sup>. Silica, indeed, as a non-reducible oxide-support, is known to establish only weak interactions with supported metal nanoparticles, and the latter are therefore particularly prone to sintering <sup>29</sup>. Thus, controlling active phase dispersion and strengthening the interaction with the support is difficult, in particular when applying conventional impregnation methods on pre-formed silica supports. Therefore, the development of new synthetic procedures allowing to (i) control active site speciation and dispersion and (ii) achieve suitable morphological and textural properties  $30$  is needed.

Along this line, the aerosol-assisted sol-gel process (AASG) – which is emerging as a powerful route to produce various nanomaterials and in particular tailored heterogeneous catalysts  $31,32 -$  is primed to solve the above-mentioned challenges. The method is based on the atomization and fast reactive drying of a precursor's solution. During processing, inorganic polycondensation reactions take place very rapidly, possibly coupled with the templating action of a sacrificial pore-generating agent. This allows the one-step and continuous production of nanomaterials with desired properties. Aerosol processes have been used to develop catalysts for a wide range of reactions, such as Mo-based catalysts for olefin metathesis  $^{33}$ , TiO<sub>2</sub> catalyst supports for CO<sub>2</sub> methanation  $^{34}$ , Cu-based materials for  $CO_2$  hydrogenation to methanol <sup>35</sup>, Au based catalysts for CO oxidation <sup>36</sup> or RuO<sub>2</sub> based catalysts for electrocatalysis <sup>37</sup>. While this technique has never been used to synthesize Cu-based catalysts for ethanol non-oxidative dehydrogenation, we surmised that the control offered in terms of Cu dispersion and texture could offer decisive advantages regarding intrinsic activity and stability. Moreover, it must be recalled that aerosol processes are continuous, low-waste, and easily scalable 31 .

The present work aims at exploiting Aerosol Assisted Sol Gel process (AASG) to prepare low loading Cu-based catalysts for (bio)ethanol dehydrogenation. These new catalysts are compared with conventional impregnated catalysts. With the teachings gained from the deep characterisation of fresh and used catalysts we identify the key parameters that govern the high activity of the AASG-made catalysts, and we identify the causes of deactivation and suitable rejuvenation procedures.

# **2. Experimental section**

#### *2.1 Catalysts preparation*

Copper-based catalysts have been synthesized following two different procedures: one-step aerosolassisted sol-gel (AASG) and incipient wetness impregnation (IWI) of pre-synthesized silica support.

2.1.1 One-pot preparation of  $Cu-SiO<sub>2</sub>$  by AASG technique

Solution A is prepared by mixing 0.977 mol of ethanol (VWR, >99.8 v/v%), 0.444 mol of distilled water and 0.0003 mol of Pluronic<sup>®</sup> F127 (Sigma Aldrich,  $\simeq$  12600 g/mol)]. A second solution (B) is prepared by mixing 0.057 mol of TEOS (TCI Chemicals, >97.0%), 0.548 mol of an aqueous HCl solution with a pH=2 obtained by dilution of fuming HCl (Roth, 37 wt.%). After preparation, solution A and B were stirred overnight and then mixed together.  $Cu(NO<sub>3</sub>)<sub>2</sub> * 2.5 H<sub>2</sub>O$  was added in the desired amount to obtain the targeted Cu loading. The obtained solution was then sprayed with an atomizer, supplied by "TSI incorporated<sup>®</sup>", by applying an air pressure of 207 kPa. The aerosol droplets were dried by passing through a quartz tube heated by a tubular furnace set at 723 K. Then, the dried material was collected on a nitrocellulose filter (Sartorius Stedim, 0.45 μm). Powders were calcined in a muffle furnace under static air, first at  $623 \text{ K}$  (1 K min<sup>-1</sup>) for 3 h, and then at  $823 \text{ K}$  (1 K min<sup>-1</sup>) for an additional 3 h. These catalysts were denoted as A-CuXSi, where X corresponds to the nominal Cu loading on support basis and  $X = 2, 5, 7.4$  and 9.1 wt.%.

# 2.1.2  $Cu/SiO<sub>2</sub>$  catalysts prepared by impregnation

Reference catalysts were prepared by incipient impregnation, using a silica support obtained by AASG. The latter was prepared by the same protocol as the CuXSi catalysts but omitting the addition of Cu nitrate. Here, copper is introduced onto the preformed silica support by wetting the latter with an aqueous solution of  $Cu(NO<sub>3</sub>)<sub>2</sub> * 2.5 H<sub>2</sub>O$  (Alfa Aesar, 98 wt.%). The amount of employed water corresponded to the volume of the support pores, evaluated by  $N_2$  physisorption, incremented by a 20% v/v. Powders have been dried and calcined with the same protocol described above. These catalysts were denoted as I-CuXSi, where X corresponds to the nominal Cu loading on support basis and where I stands for "impregnation".

# *2.2 Materials Characterization*

Textural properties were measured by  $N_2$ -physisorption at 77 K using a Micrometrics Tristar 3000 instrument. Prior to analysis, calcined samples were degassed overnight under vacuum at 443 K. The Brunauer-Emmett-Teller (BET) model was used to determine the Specific Surface Area (SSA,  $m^2/g$ ) in the relative pressure range of 0.05–0.30. Total pore volume  $(V_p)$  was estimated from the adsorption branch of the isotherm at  $p/p_0=0.98$  and the average pore diameter  $(D_p)$  was estimated from the BJH model applied on the adsorption isotherms.

XRD analysis was carried out by means of a Bruker D8 Advance diffractometer (Bragg–Brentano geometry) on both fresh and spent catalysts. The diffractometer works with a Cu K $\alpha$  source  $(\lambda=0.15418 \text{ nm})$  at 1200 W (30 mA, 40 kV). Diffraction patterns have been acquired by setting the

following parameters:  $2\theta$  range  $5^{\circ}$ -100°, step size  $0.05^{\circ}$  (2 $\theta$ ) and 1.5 s each step. The detector was a Bruker Lynxeye XE-T. Identification of the phases was carried out using Pearson's Crystal Database <sup>38</sup>. Crystallites sizes of CuO and of Cu have been evaluated on the most intense peak by Scherrer formula, via DIFFRAC.EVA V4.2.1 software. For CuO, peak at  $2\theta = 35.66$  has been considered, for Cu, peak at  $2\theta = 43.40$  has been considered.

ICP-AES analyses were performed on an ICP Thermo Scientific 6500 instrument after dissolution of samples ( $\simeq$ 100 mg) by metaborate Li - tetraborate Li fusion.

Scanning Electron Microscope Zeiss SUPRA 40 VP, equipped with a field emission gun (FE-SEM) was used to investigate sample morphology and composition for both fresh and spent catalysts. This microscope is equipped with a high sensitivity "InLens" secondary electron detector and with a EDX (Energy Dispersive X-Ray) Spectrometer OXFORD "INCA Energie  $450 \times 3$ ". Samples were suspended in ethanol. A drop of the resultant mixture was deposited on a Lacey Carbon copper grid and the dried sample was then imaged.

Samples composition was investigated by using a scanning electron microscope (SEM) Zeiss Evo 40 equipped with a Pentafet Link Energy Dispersive X-ray Spectroscopy (EDXS) system managed by the INCA Energy software (Oxford Instruments, Analytical Ltd., Bucks, U.K.). Samples powders were directly mounted on a high purity conductive double sided adhesive carbon tab.

TEM and STEM-EDS measurements were carried out on a FEI Titan Themis instrument with a combination of a spherical aberration image (Cs) corrector, a monochromator system, sensitive ChemiSTEM technology, and a high-end GATAN GIF Quantum Energy Filter for EELS and EFTEM with a new enhanced piezo stage, FEI and GATAN software, and a FEI Ceta 16-megapixel CMOS camera. Powders have been dispersed in cyclohexane and  $4 \mu L$  deposed on a gold grid covered by QuantiFoil® holey carbon and then dried in air.

X-ray photoelectron spectroscopy (XPS) analyses were carried out at room temperature with an SSI-X-probe (SSX 100/206) photoelectron spectrometer from Surface Science Instruments (USA), equipped with a monochromatic Al K X-ray microfocused source (1486.6 eV). Samples were deposed onto small sample holders with adhesive tape and then placed on an insulating ceramic carousel (Macor®, Switzerland). Charge effects were avoided by placing a nickel grid above the samples and using a flood gun set at 8 eV. The binding energy scale was calibrated by fixing the Si 2p peak at 103.5 eV  $39$ . Data treatment was performed using the CasaXPS program (Casa Software Ltd., UK). The peaks were decomposed into a sum of Gaussian/Lorentzian (85/15) after subtraction of a Shirleytype baseline.

Infrared (IR) spectra have been obtained with a Nicolet 380 Fourier Transform IR spectrometer. Sample powders were pressed into thin wafers with KBr and spectra were recorded in air (0.5 wt% sample in KBr). Spectra have been collected with 100 scans with a resolution of  $2 \text{ cm}^{-1}$ .

DR-UV–vis-NIR spectra were collected with a JASCO V570 instrument equipped with an integrating sphere. The powders were gently pressed in the sample holder.

Spent catalysts were studied by thermo-gravimetry, using a TGA/DSC 3+ apparatus from Mettler-Toledo, coupled with a Pfeiffer Vacuum ThermoStar<sup>TM</sup> mass spectrometer. Samples ( $\simeq 20$  mg) were placed in alumina crucibles (70  $\mu$ L) and heated at 10 K min<sup>-1</sup> from room temperature to 1173 K under a dry air flow (100 mL/min). The following m/z have been recorded over the whole experiment: 28  $(N_2)$ , 18 (H<sub>2</sub>O), 32 (O<sub>2</sub>) and 44 (CO<sub>2</sub>).

#### *2.3 Ethanol dehydrogenation to acetaldehyde*

Catalytic tests were carried out in a fixed bed reactor (stainless steel, 0.6 cm internal diameter). 50 mg of calcined catalysts (pressed, crushed and sieved in the 0.20–0.40 mm range) were diluted with glass beads (0.5–1.0 mm), by keeping a constant catalytic bed volume. Before reaction, catalysts were pre-reduced in situ by feeding hydrogen (20 vol.%  $H_2$  in  $N_2$ , total flow rate of 175 mL/min) for 30

min at 573 K with a ramping rate of 5 K min<sup>-1</sup>. Then, the reactor was set at the desired reaction temperature (473 K) and the reaction feed was introduced: absolute ethanol was fed through a syringe pump (NE-300) in a stream on  $N_2$ , achieving the desired molar fraction in the gas phase (2.5% mol, total flow rate of 120 mL/min). All measures were carried out at atmospheric pressure while temperatures were increased in a stepwise manner, from 473 K to 673 K. Every temperature step was kept for 1 hour, ensuring steady state conditions. Effluent gas was analysed online with a VARIAN 3800 Gas Chromatograph (Varian Medical Systems, Machelen, Belgium). Four injections were made at each temperature. The injector was set at 423 K and the GC was equipped with a flame ionization detector (FID) kept at 463 K, a Restek Rt-U-BOND column (30 m long, internal diameter of 0.32 mm, film thickness of 10  $\mu$ m) and a programmable oven (setup: 383 K for 5 minutes, then to 423 K with a ramping rate of 5 K/min, dwelled for 5 minutes).

Stability tests on A-Cu7.4Si and I-Cu7.4Si have been carried out at 573 K for 8 hours in the same conditions. In the case of A-Cu7.4Si catalysts, only 40 mg of catalyst was introduced in order to have iso-conversion comparison with the impregnated catalyst (I-Cu7.4Si)

Ethanol conversion  $(X_{EtOH})$  is defined as:

$$
X_{EtOH} = (n_{EtOH(in)} - n_{EtOH(out)})/n_{EtOH(in)} (1)
$$

selectivity to product i is defined as:

$$
S_i = n_i / (v_{i(}\eta_{EtOH(in) -} n_{EtOH(out))}) (2)
$$

And yields are defined as:

$$
Y_i = n_i / v_i n_{EtOH(in)}(3)
$$

where  $n_i$  is the molar flow of compound i, and  $v_i$  is the ratio of stoichiometric reaction coefficients. The error in the steady state carbon balance is  $\pm$  5%.

#### *2.4 Regeneration test*

After the stability test (8 h at 573 K) an oxidation step was carried out by flowing a gaseous stream (50 mL/min) of 20 vol %  $O_2$  in N<sub>2</sub>. Temperature was raised with a ramping rate of 5 K min<sup>-1</sup> up to 788 K, and then dwelled for 15 minutes. After cooling down up to room temperature, the catalyst has been tested again in the same experimental conditions.

#### **3. Results and discussion**

#### *3.1 Catalysts characterisation*

The copper content has been experimentally verified by ICP-AES (Table 1). Experimental data are in good agreement, even if the observed loading is slightly lower than the nominal one (for both aerosol-made catalysts and impregnated catalysts), suggesting a contribution of the hygroscopic properties of the metal precursor used in the preparation.

Nitrogen physisorption curves (Figure 1A, C) revealed that all the calcined catalysts showed Type IV isotherms indicating the presence of open mesopores. The shape of the hysteresis loop corresponds to the H2-type (asymmetrical with a steeper desorption branch than the adsorption one) characteristic of restrictions within the porous network. The forced closure of the hysteresis loop at  $p/p_0$  values of 0.4–0.5 indicated the presence of mesopores smaller than  $\sim$  4 nm <sup>40</sup>. The Barrett-Joyner-Halenda (BJH) model applied on the adsorption branch (Fig. 1B, D) of the isotherm revealed the presence of mesopores in the 6-12 range (centred at about 8 nm) for all catalysts. The nitrogen uptake at low relative pressure also indicates the presence of micropores whose contribution remains relatively small, as underlined by textural values summarised in Table 1.

Sample	$Cu wt.$ % - <b>ICP-AES</b>	Surf. area $[m^2 \cdot g^{-1}]$	$V_p$ <sup>[a]</sup> [cm <sup>3</sup> ·g <sup>-1</sup> ]	$V_{micro}$ <sup>[b]</sup> [cm <sup>3</sup> g <sup>-1</sup> ]	$D_p$ <sup>[c]</sup> ${\rm [nm]}$
$A-SiO2$		380	0.45	0.06	7
I-Cu <sub>2</sub> Si	1.3	329	0.44	0.05	7
I-Cu <sub>5</sub> Si	3.9	300	0.37	0.03	$\overline{7}$
$I-Cu7.4Si$	6.4	290	0.54	0.03	11
$I-Cu9.1Si$	8.0	297	0.40	0.03	$\overline{7}$
A-Cu <sub>2</sub> Si	1.6	505	0.70	0.09	9
$A-Cu5Si$	3.6	458	0.42	0.11	6
$A-Cu7.4Si$	6.7	458	0.62	0.10	8
$A-Cu9.1Si$	8.2	418	0.66	0.09	9

**Table 1:** Composition and textural properties of catalysts prepared via IWI (I) and in one step via the AASG process (A).

[a] Total pore volume, estimated from the adsorption branch of the isotherm at  $p/p0=0.98$ . [b] Microporous volume, estimated from the t-plot. [c] Average pore diameter, estimated from the BJH model applied on the adsorption isotherms.

The specific surface area and pore volume were generally higher for A-catalysts (prepared via the one-pot AASG process), reaching as high as  $505 \text{ m}^2/\text{g}$  for A-Cu2Si. Upon increase in copper loading, both impregnated and aerogel catalysts surface area and pore volume tend to decrease. Interestingly, the micropore volume appears to be more important for one pot catalysts than for bare silica or impregnated materials. A possible explanation is related to the direct introduction of copper nitrate in the formation of A-CuX catalysts, trapped in the porous structure during the rapid droplet's condensation and drying step. Since  $Cu^{2+}$  ionic radius is -larger than that of  $Si^{4+}$ , it is possible that the silica network rearranges to host copper species, creating roughness at the surface. Alternatively, nitrates may behave as a "foaming agent" during calcination creating a family of smaller pores upon the departure of  $NO<sub>x</sub>$  species.



**Figure 1:** N<sub>2</sub> adsorption/desorption isotherms (insets A and C) and pore size distributions (B and D) of investigated catalysts.

The diffraction patterns were recorded for all fresh catalysts (Figure 2A and 2B). For both I-CuXSi and A-CuXSi series of catalysts, a broad diffraction peak at about  $2\Theta = 22^{\circ}$  is attributed to the amorphous silica support  $^{41}$ . In all catalysts having a Cu content equal or higher than 5 wt.%, the CuO phase (tenorite, mS8; PCD ref. 1243879)<sup>38</sup>, is observed. Expectedly, the CuO diffraction peaks intensity increases with the increasing copper content in the catalyst. The Scherrer equation, applied on the most intense CuO peak at  $2\Theta = 35.6^{\circ}$ , indicates an average crystallite size of 32-33 nm for both I-Cu7.4Si and I-Cu9.1Si samples. In the XRD patterns of samples obtained by AASG synthetic procedure (Fig. 2 B), the peaks intensity of CuO phase is lower; in fact, only the two main peaks, at 2ϴ=35.8° and 38.8°, are visible and broader than those observed for I-CuXSi series. For A-Cu7.4Si and A-Cu9.1Si, the crystal size evaluation with Scherrer equation leads to an average size of about 10 nm. For low loading catalysts, it is only possible to hypothesize the presence of nanoparticles with an average size equal or lower than 10 nm, hypothesis that is consistent with TEM images (see below and ESI).



**Figure 2:** XRD patterns of the fresh I-samples (inset A), and A-samples (inset B). Please note that broad and weak peaks around  $2\Theta = 63^\circ$  and  $79^\circ$  are attributable to the fixing agent used in the sample preparation for the analysis.

FE-SEM images of bare silica support shows spherical silica particles, with diameters between  $0.1 \div$ 5 μm, characterised by a rough surface (Figure 1S(A)). The obtained spherical shape is a reminiscence of the synthetic procedure that relies on a drying step of spherical aerosol droplets. In low loading impregnated catalysts i.e., I-Cu2Si and I-Cu5Si, no bright crystalline aggregates have been detected (Figure 1S(B) and (C)). At higher loading i.e., I-Cu7.4Si and I-Cu9.1Si, bright crystals are clearly distinguishable and might be ascribed to CuO (Fig. 3). Indeed, STEM energy dispersive X-ray

spectroscopy (STEM-EDXS) confirms the presence of Cu-rich aggregates that are correlated with well-visible crystallites. These particles come out from the silica matrix, often sandwiches between spherical silica particles (Fig. 3A and 3C). The morphology of the latter remains unaffected upon impregnation (Fig. 1S). In fact, focusing on the smooth silica microspheres (away from the Cu aggregates), EDX spectroscopy reveals a fixed copper amount (Fig. S1(D)), equal to  $\approx 3.0$  wt.%, for I-Cu5Si, I-Cu7.4Si, and I-Cu9.1Si. This agrees with XRD results, where the intensity of CuO peaks is very high for I-Cu7.4Si and I-Cu9.1Si specimens, due to the presence of isolated CuO crystals outside of the silica spheres. To sum up, starting from I-Cu5Si, only a fraction of the copper is introduced into the silica microspheres by means of the impregnation step and a significant fraction accumulates outside of the support particles, in the form of CuO crystals.



**Figure 3:** FE-SEM micrographs obtained on I-Cu7.4Si (A, B, C, using the BSE signals for A and C and the SE signal for B). Insets D (dark field), E (Cu mapping), F (bright field) and G (O, Si, Cu mapping) are referred to STEM images of I-Cu5Si.

For aerosol-made catalysts (see the example of A-Cu7.4Si in Fig. 4 and other images for A-Cu5Si in ESI, Fig. S2), FE-SEM microphotographs do not reveal the presence of copper-containing aggregates. This is in full agreement with XRD analyses. Moreover, by looking at EDX analysis for A-catalysts, the copper amount into the spheres agrees with the total Cu amount in the specimen (Fig. S2, F-G). Thus, in agreement with FE-SEM microphotographs, it is confirmed that all the copper is incorporated into the silica microspheres. Elemental mapping by means of STEM-EDXS unambiguously confirms the homogeneous distribution of copper throughout the silica microspheres (Fig. 4).



**Figure 4:** FE-SEM micrographs of A-Cu7.4Si using SE (left, top) and BSE (left, bottom), STEM images of A-Cu7.4Si with elemental mapping (right).

Insight on surface copper chemical state has been obtained by X-ray photoelectron spectroscopy (XPS) (Table S1 and Fig. 5). In agreement with the literature <sup>42</sup>, Cu 2p<sub>3/2</sub>, Cu 2p<sub>1/2</sub> peaks are found at ∼933.3 eV and ∼952.5 eV, accompanied by the shake-up satellite peaks (∼943 eV and ∼963 eV), indicating the presence of CuO (Cu<sup>2+</sup> with a  $d^9$  electron configuration), in agreement with the adopted synthetic route (calcination in air) and with XRD results. As already reported in the literature  $43$ , two contributions can be seen for Cu  $2p_{3/2}$ , indicating the existence of two Cu (II) species with different chemical environment at the catalyst surface. The first one at ∼933 eV is typically due to bulk-like CuO species while the second feature at higher binding energy (~936 eV) is indicative of well dispersed Cu (II) interacting with the silica network. This may suggest the presence of -Cu-O-Sispecies (i.e. well dispersed CuO and or copper phyllosilicate-like species). While this feature at 936 eV is also seen in impregnated catalysts, as shown in table 1S its intensity is much smaller than in aerosol-made catalysts, pointing again to a higher dispersion of copper via the AASG synthesis procedure. In fact, for A-catalysts upon increasing of the loading we linearly increase the surface Cu concentration (ESI, Fig. S4) while over impregnated catalysts there is no increment upon Cu loading. According to microscopy characterization and XRD patterns, this means that via impregnation only a certain amount of Cu is introduced into the pores of the silica microspheres and well dispersed at the surface. After that, Cu start to heavily aggregate without further increasing Cu superficial concentration.



Figure 5: High-resolution XPS spectra of Cu 2p<sub>3/2</sub> species on fresh I-catalysts (A) and A-catalysts (B).

FT-IR spectra featured the characteristic features of amorphous silica absorption bands for both series of catalysts. The IR spectrum of silica is characterized by the bands assigned to the Si-O-Si asymmetric stretching ( $v_1 = 1091$  cm<sup>-1</sup> with the corresponding pronounced shoulder at 1230 cm<sup>-1</sup>), the Si-O stretching of SiOH group  $(968 \text{ cm}^{-1})$ , the coupled Si-O-Si symmetric stretching/in plane bending ( $v_2 = 803$  cm<sup>-1</sup>), and the corresponding out of plane bending (rocking) mode ( $v_3 = 467$  cm<sup>-1</sup> <sup>1</sup>)<sup>44</sup>. The spectrum obtained for the bare aerosol-made silica support is almost unchanged after impregnation of copper, showing that in this case the bulk silica is not significantly perturbed. In contrast, in the case of A-CuXSi catalysts, the position of the  $v_1$  mode is shifted to 1085 cm<sup>-1</sup>, indicating that in this case the bulk is perturbed, probably because of the formation of Cu species in strong interaction with the silica network <sup>45,46</sup>.



**Figure 6:** FTIR skeletal spectra (KBr pressed disks) of the fresh I- (A) and A- (B) catalysts.

UV-vis spectroscopy (Fig. S3) indicated – for both catalysts' series – that the absorption in the far UV region (band in the range 200-250 nm) increases when the Cu loading increases, owing to the charge transfer (CT) transition  $O^{2-}(2p) \to Cu^{2+}(3d)$ . Also, an additional broad absorption is detected at the opposite limit of the visible range, due to the  $d \rightarrow d$  transitions in dispersed Cu<sup>2+</sup>. In the case of A-CuXSi samples the intensity of both absorptions increases as a function of copper loading, with a linear trend (see function in Figure S3). In parallel, the maximum of the  $d \rightarrow d$  transition tends to shift to lower wavelengths. This suggests that the nature of copper species is essentially always of the same nature with increasing loading, the shift of the  $d \rightarrow d$  transition being likely due to the decreasing distance between the dispersed copper centres. In the spectra of I-CuXSi samples, at the highest loading, the shape of the absorption band due to the  $d \rightarrow d$  transitions of dispersed Cu<sup>2+</sup> ions are modified, suggesting that an additional absorption grows due to the  $d \rightarrow d$  transition in bulk CuO. This is fully consistent with the appearance of CuO diffraction peaks in the XRD pattern of these samples. The linear trend between the specific absorption intensities and the copper loading (as described above for A-catalysts) is verified also for I-Cu2Si and I-Cu5Si but not for I-Cu7.4Si and I-Cu9.1Si, suggesting a more accentuated heterogeneity of these catalysts in terms of Cu speciation.

#### *3.2 Ethanol dehydrogenation*

Before being tested in the ethanol dehydrogenation reaction, the catalysts were pre-reduced in situ in the reactor. For the sake of completeness, XRD data of pre-reduced catalysts are introduced in Figure 5S, showing effective catalyst reduction by production of metallic copper particles for all catalysts (above 2 wt% Cu). Crystal sizes evaluation, carried out on the most intense peak ( $2\theta = 35.6^{\circ}$ ) by means of the Scherrer equation, indicated the presence of Cu particles around 35 nm for the impregnated catalysts. For aerosol-made catalysts, on the other hand, crystal size evaluation is not possible because the peaks are broad and weak. These pattern features suggest that upon pre-reduction Cu remains highly dispersed throughout the silica spheres at a nanometric scale.

In Figure 7, catalytic performances achieved over investigated materials are summarised in terms of ethanol conversion and acetaldehyde selectivity. Over impregnated catalysts, the lowest conversion values are found for I-Cu2Si, achieving almost 20% at 573 K. The highest conversion performances are reached by I-Cu5Si, showing already 54% ethanol conversion at 573 K. However, further increasing the Cu loading leads to a marked drop in performance, both in terms of conversion and in terms of selectivity. The drastic drop in selectivity is particularly marked for the catalysts with the highest Cu loading. This appears in line with the characterization survey that indicated a marked aggregation of large CuO (Cu, after reduction) in these high-loading samples prepared by impregnation. Full selectivity is obtained at 473 K and high selectivity (97%) is maintained at higher temperatures for the catalysts with the lowest loading. However, the two catalysts with the highest Cu loading show a marked drop in acetaldehyde selectivity when the temperature was increased to 523 K and higher (down to 86% at 673 K). Selectivities to diethyl ether and ethylene are reported in Fig. S6 A.



**Figure 7:** Ethanol conversion (A, C) and acetaldehyde selectivity (B, D) obtained with I-catalysts and A-catalysts, as a function of reaction temperature. Performances are measured by maintaining each reaction temperature constant for 1 hour, from 473 to 673 K.

In comparison with impregnated catalysts, A-CuXSi catalysts reached higher conversion. Performance increases with the Cu loading, up to A-Cu7.4Si, then were slightly lower over A-Cu9.1Si. In particular, at low temperatures aerosol-made catalysts feature almost double catalytic activity when compared to the most performant I-sample, and increased acetaldehyde selectivity with respect to I-catalysts (ESI, Fig. S6, A). These materials reached a plateau in terms of conversion above 573 K (Fig. 7C). Only a slight decrease in the selectivity (from full selectivity to  $\sim$ 97%) to acetaldehyde is observed when increasing the reaction temperature, due to the occurrence of ethanol dehydration to ethylene (ESI, Fig. S6, B).

A recent review on the topic allows comparing these catalysts with the state of the art  $^{24}$ , even if it must be stressed that operating conditions often diverge significantly from one study to another. Apart from one Cu/SiO<sub>2</sub> catalyst reported by Zhang et al.<sup>45</sup> using a relatively high copper amount and a much lower GHSV, the catalysts shown in the present work clearly outcompete all other Cu-silica formulations  $^{24}$ . Beyond silica, looking at Cu-ZnAl<sub>2</sub>O<sub>4</sub> as one the most intensively studied catalysts, we can compare with our recent results, obtained in very similar reaction conditions and with similar copper loading <sup>47</sup>. A-Cu7.4Si shows comparable, or slightly higher, ethanol conversion values in the entire temperature range 573 K-673 K. Furthermore, it seems to be more active at lower temperature (i.e. 49% vs 34% at 523 K).

Looking at the catalytic performances shown in Figure 7, for many of the prepared materials i.e. I-Cu7.4Si, I-Cu9.1Si, A-Cu5Si and A-Cu2Si a drop in the ethanol conversion values can be seen at the highest temperatures, suggesting that catalyst deactivation is occurring. This is also confirmed looking at conversion values versus time plots (ESI-Fig. S6, C, D) that systematically show a decreasing trend during each temperature dwelling. Previous studies pointed out that both copper sintering and carbon deposition can slow down catalytic ethanol dehydrogenation <sup>48–50</sup>. Thus, to shade some light on the possible deactivation causes, spent catalysts were characterized, as discussed in the next paragraph.

# *3.3 Spent catalysts characterization*

XRD patterns of the catalysts after ethanol conversion experiments are reported in Figure 8. In almost all cases, the peaks of metallic copper (cF4-Cu, PCD ref. 1146504) are detected as the main crystalline phase, together with the appearance of characteristic peaks of Cu<sub>2</sub>O (cP6, PCD ref. 1831225) phase. The presence of oxidized Cu species might be due to copper oxidation at the expenses of water arising from ethanol dehydration to ethylene <sup>51</sup> or as well to slight oxidation occurring in the presence of air during the handling of the powders. I-catalysts generally reveal a clear bigger crystal size (i.e. 48 and 10 nm for I-Cu9.1Si and A-Cu9.1Si) respect to the aerosol made ones.



**Figure 8:** XRD patterns of spent catalysts after catalytic measurements (inset A: I-catalysts; inset B: A-catalysts).

FE-SEM characterization of spent catalysts is reported in Figure 9. In impregnated catalysts, coppercontaining particles are visible and appear bigger than in the fresh catalysts. This shows that sintering occurred upon exposure to reaction conditions. Impregnated catalysts show a remarkable high number of copper rich particles that apparently grew in the intraparticle spaces and around small silica particles (Figure 9 A, B). Remarkably, FE-SEM characterization of the A-catalysts (Fig.9 C, D) reveal that upon the same catalytic test conditions, no bright aggregates are detected. The absence of heavy sintering can be interpreted as a sign of a stronger intimacy between the Cu species (nanoparticles) and the silica network, owing to of the aerosol-based method.



**Figure 9:** FE-SEM micrographs of spent ICu5Si (A, B) and Cu5Si (C, D) catalysts after the catalytic test (473-673 K). A and C are SE signals. B and D are BSE signals.

TGA analyses for all investigated catalysts are reported in Figure S7 A, B. Figure 10, shows the MS signal for  $m/z=44$  (CO<sub>2</sub> evolving from carbon oxidation). For all catalytic materials two peaks are found at 530-560 K and 760 K accounting for two different carbon species deposed over catalysts during reaction. The first mass loss can be due to the presence of C-O/C=O containing carbon residue  $52$  or amorphous carbon species  $53$ . It is systematically more important for A-catalysts (that are more active) as compared to I-catalysts (Fig. S7, C). The second  $CO<sub>2</sub>$  peak ( $\sim$ 760 K) is attributable to the presence of graphitic-like carbon. It is again more important for A-CuXSi, especially at low Cu loading (Fig. S7, C).



**Figure 10:** MS monitoring at m/z = 44 during the TGA-MS experiment on spent I-catalysts (A) and A-catalysts (B) after catalytic measures (473-673 K).

XPS measurements (Fig. S8, A, B) of the spent catalysts reveal the presence of  $Cu^+$  and/or  $Cu^0$  species (consistent with XRD and FE-SEM characterizations) with clear peaks at 932-933 eV (Cu  $2p_{3/2}$ ) orbitals) and 954 eV (Cu 2 $p_{1/2}$  orbitals) <sup>54</sup>. It should be noted that partial re-oxidation can occur during handling which complicates the analysis of the Cu oxidation state in spent catalysts. However, it appears clearly that the Cu/Si ratio tends to decrease after catalytic reaction (Table S2). Causes may be both copper sintering and/or selective coke deposition on copper particles. At this stage of the study is impossible to determine which cause of deactivation is predominant but certainly the consequence, as suggested in literature  $54,55$ , is a reduction of active phase sites at the surface. To further investigate the coke deposition phenomenon (Fig. S8, C, D and Tables S3 and S4), XPS peaks for C1s region was investigated for both fresh and spent catalysts. The surface carbon concentration was found to increase in all spent catalysts, as compared to the fresh counterparts, confirming carbonaceous species deposition. The C1s peak was further decomposed into four contributions: O=C-O (~ 289 eV), C=O/O-C-O (~288-289 eV), C-O (~286 eV), C-(C, H) (~284.8-285 eV) <sup>56</sup>.

Comparing fresh and exhausts analyses, the C=O/O-C-O and C-O components remain stable among the different catalysts. Probably underlining the adventitious nature of these species. Components C- (C, H), referred to aliphatic carbon, is more subjected to a variation among the spent catalysts. Considering the components of the C1s peak, it seems that aliphatic carbon species are predominant. This probably confirms the presence of coke with a main  $C-C-C_n$  pattern, probably deriving from oligomerized carbonaceous species, in fully agreement with exhaust TGA-MS analyses on spent catalysts. Finally, no satellite peaks due to  $\pi$ - $\pi$ <sup>\*</sup> bond (BE > 290 eV) are found, so extended delocalised electrons derived structures (i.e. aromatic species)  $57$  can be excluded. Thus, we suggest that the carbonaceous deposit that is formed during reaction is mainly composed of amorphous/polymeric coke. Finally, globally looking at the surface carbon concentration over the spent catalysts (ESI, Fig. S9), it seems to be independent of the copper loading for A-catalysts. Instead, over the impregnated ones there is a clear negative trend upon Cu increasing loading. Being that the most active sites are at the border between the support and the nanoparticle, it seems to be present a correlation between Cu crystallites dimensions, catalytic activity and coke deposition. In fact, over impregnated catalysts higher is the loading, and then, proportionally, the crystal size, lower is the coke deposition. Instead, over A-catalysts, where dimensional variation of nanoparticle is less marked, coke deposition seems to be constant.

#### *3.4 Effect of time on stream, deactivation, and rejuvenation*

A-Cu7.4Si and I-Cu7.4Si catalysts were tested at 573 K for 500 min on stream to study their deactivation trend (Fig. 11 A). I-Cu7.4Si starts at a conversion of 39%. A-Cu7.4Si reaches a conversion of 55 % (the test was done with 40 mg of catalyst instead of 50 mg, so that the initial conversion is in the same range for both catalysts). Both catalysts showed total selectivity to acetaldehyde. Then, conversion slowly declines with time on stream, tending to a dwelling after 400 minutes. Meanwhile, however, selectivity remains at 100%. In the course of this stability test, the

yield of acetaldehyde is divided by ~4 with I-Cu7.4Si (down to ~11%), while it is only divided by a factor 2 (down to  $\sim$ 24%) with A-Cu7.4Si.



**Figure 11. (**A) Stability tests over A-Cu7.4Si and I-Cu7.4Si with ethanol conversion and acetaldehyde selectivity values. (B) stability test and  $2<sup>nd</sup>$  stability test, on A-Cu7.4Si, after rejuvenation step (mild oxidative treatment) in terms of ethanol conversion.

FE-SEM microphotographs of I-catalyst post stability test (Fig. 12 A, B) reveal a large number of bright particles (Cu) that aggregate outside of the silica spheres. Thus, it is clear that, although the stability test has been carried out only at 573 K (i.e.  $T < T_{Tamman}$ ), segregation of active phase and progressive sintering occurred. As already reported <sup>58</sup> atomic migration and crystallite migration (diffusion-controlled processes that occur along the surface, followed by coalescence of atoms/crystallites) are well-known deactivation causes for supported metal nanoparticles. Moreover, pronounced sintering may also occur after prolonged time on stream or/and at high temperatures, as reported by others <sup>18</sup>, generating irreversible deactivation. This effect is very visible on the I-catalyst. On the contrary, after the same test on the A-catalyst (Fig. 12 C, D) only a small number of Cu crystallites is detected. Moreover, looking together at fresh and spent A-catalysts micrographs, no variations in Cu distribution over the support is found. This confirms a slightly reduced mobility of Cu nanoparticles in the catalyst made by aerosol. In fact, it seems reasonable to affirm that A-catalyst

shows a higher resistance to coalescence because the Cu nanoparticles are truly embedded in the silica matrix (yet accessible for catalysis, so at least partially pointing towards the external surface of the pores). The fact that Cu species tend to aggregate more on the I-catalyst than on the A-catalyst is unambiguously confirmed by elemental mapping in STEM-EDXS (Fig. 12 E, F) and by XRD (Fig. S10, E, F).



**Figure 12:** FE-SEM micrographs obtained on spent catalysts after the stability tests: A (SE signals), B (BSE signals) refer to ICu7.4Si while C (SE signals) and D (BSE) refer to A-Cu7.4Si. STEM images obtained on I-Cu7.4Si (E, dark field, bright field and Cu/Si mapping from left to right) and on A-Cu7.4Si (F, dark field, bright field, and Cu mapping from left to right).

All in all, both fouling by carbon deposition (ESI Fig. S10, A, B, C, D for TGA-MS), occurring on both catalysts, and more markedly on A-catalysts and Cu sintering, occurring mostly on I-catalysts and only marginally on A-catalysts, can be the main cause of the progressive deactivation. To understand whether the carbon deposition was the major source for the deactivation or not, a rejuvenation was attempted by applying an oxidative treatment (up to 788 K) on the best catalyst (i.e. A-Cu7.4Si) after the stability test. Surprisingly, the 2<sup>nd</sup> stability test (after the rejuvenation) reached conversion values slightly higher than in the first test (Fig. 11 B) and total selectivity to acetaldehyde

was maintained. The origin of this slight increase in activity would require further investigations. Yet, this experiment reveals that the main limiting factor for catalyst stability is the progressive coke deposition <sup>59</sup>. However, the carbonaceous deposit can be easily removed using a mild oxidative treatment.

#### **4. Conclusions**

Simple and direct synthesis method relying on the aerosol-assisted sol-gel (AASG) process can be utilized to prepare efficient Cu-based catalysts for the non-oxidative dehydrogenation of (bio)ethanol to acetaldehyde. The process leads to spherical and mesostructured silica-based microparticles with highly dispersed Cu species. From a detailed characterization survey, it comes up that preparation methods strongly influence the catalytic properties. In fact, our experimental data suggest that respect to catalysts fabricated by impregnation with one pot made ones is possible to obtain i) smaller nanoparticles, ii) a higher dispersion of Cu at the surface iii) a stronger intimacy between the active phase and the support. A-catalysts present better dehydrogenation performance respect to impregnated ones, in particular at 573 K. Characterization of spent catalysts reveals that deactivation occurs by copper segregation and coke deposition. The main cause of catalyst deactivation seems to be the second one. However, over A-catalysts, where it has been proved that sintering has a limited effect, carbon deposition can be easily reversed with a mild oxidative treatment rejuvenating the catalytic activity.

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