

1 DNA Crosslinked Alginate Hydrogels: Characterization,
2 Microparticle Development and Applications in Forensic Science.

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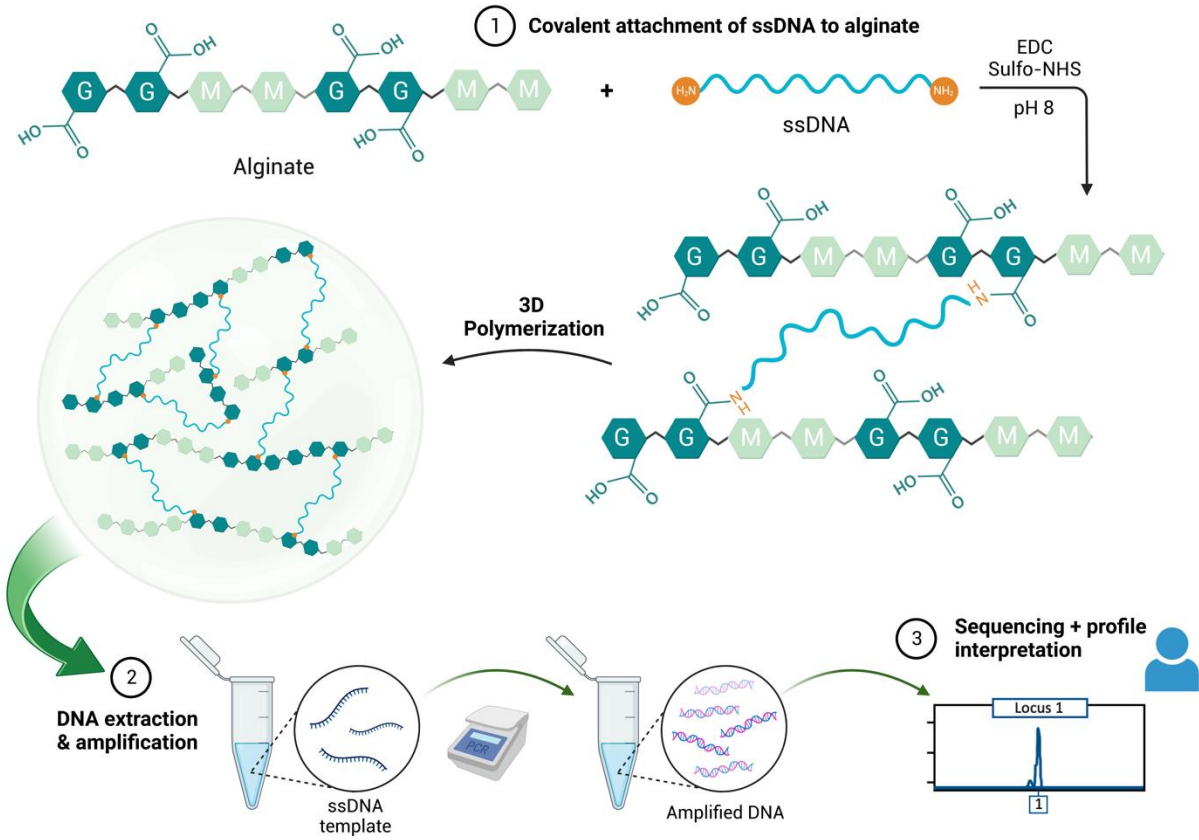
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16 **KEYWORDS** Forensic biomaterials; covalent crosslinking; tissue simulant; carbodiimide chemistry;
17 chemical hydrogel; standard reference material; functional DNA

18

1 **GRAPHICAL ABSTRACT** Covalent crosslinking of alginate and ssDNA and general processing steps
2 of hydrogel for forensic applications. Created with BioRender.com



3

1 **ABSTRACT**

2 DNA-based hydrogels are attractive materials due to the integration of highly specific DNA
3 sequences that can perform targeted functions for multiple fields. In this work we present a suite
4 of materials with covalently bound ssDNA to an alginate-based hydrogel for targeted forensic
5 applications. These crosslinked materials not only promote a more stable 3D polymeric network,
6 but also achieve localization of functional ssDNA, a more desirable feature compared to previous
7 DNA encapsulated versions. Specifically, dual amine terminated ssDNA (N-DNA-N) of three
8 different concentrations was bound to alginate using carbodiimide chemistry. Rheological
9 characterization showed that each DNA-crosslinked material forms similar structures, but the
10 higher DNA concentration behaved like a dynamic viscoelastic material. FTIR analysis confirmed
11 the formation of amide bonds, indicative of successful crosslinking between the N-DNA-N and
12 alginate. SEM visualization also showed that each material had distinct topographies, where the
13 covalent crosslinked alginate-DNA materials had more ordered particles and networked
14 structures. We also investigated ssDNA with three different amine functionalities to understand
15 the amine reactivity, which revealed that the N-DNA-N attaches primarily from the terminal
16 primary amines on the DNA strands. From this, microparticles (MPs) using the DNA-crosslinked
17 materials were developed, and the particle morphology and sizes were measured. It was
18 determined that MPs made using DNA-crosslinked materials had larger particle diameters
19 compared to the non-DNA controls, which is ideal for the generation of white blood cell (WBC)
20 mimetics in forensic materials. In addition, these MPs could be successfully processed in a
21 relevant forensic scenario through extraction, amplification, and genotyping, demonstrating the
22 functionality of these materials to forensic blood simulants.

1 1. Introduction:

2 Alginate-based hydrogels have continually been developed in the areas of bioengineering,
3 biomedicine, and the food industry owing to the many benefits of alginate such as
4 biodegradability, printability, and low toxicity¹⁻⁵. Not only is alginate a desirable polymer for these
5 applications given its biocompatibility and mechanical similarities to the extracellular matrix
6 (ECM)⁶, but also due to its versatile crosslinking potential. Covalent crosslinking of alginate can be
7 used to design chemically modified hydrogels (chemical gels) with targeted functions, mechanical,
8 and/or chemical capabilities. Alginate is comprised of repeating blocks of β -D-mannuronate (M)
9 and α -L-guluronate (G), each containing carboxyl (COOH) functional groups. The COOH moieties
10 are one of the main sources of covalent crosslinking of alginate⁷, where typical crosslinking can be
11 achieved through incorporation of photoionizable groups⁸⁻¹⁰, click chemistry¹¹, carbodiimide
12 chemistry^{12,13}, or through esterification reactions with alcohol- or alkyl halide-functionalized
13 groups¹⁴. Other types of crosslinking include ionic and pH-induced crosslinking, but are often not
14 used in biomaterial development due to properties such as gel instability and lack of stiffness when
15 compared to covalently crosslinked materials^{15,16}. Recent work has even demonstrated the
16 potential of hydrogels to be developed without the use of traditional crosslinkers, where hydrogel
17 polymerization is controlled via physical forces such as hydrogen bonding and electrostatic
18 interactions while retaining similar properties to crosslinked hydrogels¹⁷.

19
20 Hydrogels can incorporate a variety of crosslinkers including deoxyribonucleic acids (DNA). DNA-
21 based hydrogels are an emerging and attractive option for polymer materials primarily due to the
22 predictable and base-specific binding of DNA¹⁸. Further, DNA integration into hydrogels procures
23 additional benefits such as controlled biodegradation, spatial control, and stimuli responsive
24 behaviour^{18,19}. DNA-containing hydrogels are extremely versatile and programmable materials
25 that make these hydrogels useful in host recognition or controlled transfection of DNA¹⁵,
26 biosensing²⁰, and drug delivery applications^{19,21}. Commonly, these materials incorporate
27 aptamers in the hydrogel matrix; which are short single stranded ligands with high specificity for
28 a given target molecule(s)²². The aptamers can be used as either single stranded crosslinkers, or
29 can be hybridized with its complementary strand to form crosslinked hydrogels^{18,22}. Most often,
30 the aptamers are constructed through functionalization or decoration of polymers²³ that primarily
31 provide functional properties within a hydrogel¹⁹. However, the DNA can be included for both
32 mechanical and functional purposes, where the three-dimensional (3D) crosslinked networks are
33 formed through hybridization of two complementary immobilized single stranded DNA (ssDNA).
34 One example of a crosslinked 3D hydrogel using alginate and DNA was developed for controlled
35 release of fluorescein isothiocyanate-dextran (FITC-dextran) using an egg white lysozyme
36 aptamer²⁴. By including the aptamer hydrogel, targeted degradation occurred in the presence of
37 the specific lysozyme, thus releasing the encapsulated drug simultaneously. Beyond aptamer-
38 based DNA hydrogels, various forms of DNA can be integrated into the polymer network, including

1 X- or Y-DNA/nanostars, i-Motifs, DNAzymes, and Guanine-quadruplexes using a variety of
2 modification strategies^{19,25,26}. For example, using X- and Y-shaped DNA, hydrogels can be easily
3 tuned to possess properties such as self-healing²⁷, controlled stiffness and degradation²⁸, all while
4 produced under physiological and non-toxic conditions. G-quadruplexes and i-motifs can be used
5 to achieve hydrogels with photo-responsiveness and shape-memory functionalities^{29,30}, and
6 DNAzymes can be effective in the colorimetric detection of hydrogen peroxide (H₂O₂) down to 1.0
7 μm³¹.

8
9 While DNA-crosslinked hydrogels have been investigated previously, the included sequence is
10 usually some form of an aptamer, and these materials typically focus on other polymer bases such
11 as polyacrylamide^{18,20,32,33}, polyethylene glycol^{18,34}, or even DNA itself^{35,36}, but infrequently
12 alginate. In addition, the design of these materials is typically tailored for therapeutic or biosensing
13 applications in biomedicine and bioengineering. While these applications are advantageous,
14 biofunctional materials – such as DNA-based hydrogels, have clear relevance to other fields such
15 as forensic science. In forensic science, DNA-based hydrogels can be used for the detection of a
16 variety of analytes drugs, metabolites, and identification of biological evidence³⁷; and for imaging
17 of latent fingerprints³⁸. Beyond these applications, and the focus of this research, is the
18 development of functional tissue mimetics or standard reference materials with controlled
19 hydrogel properties. These materials can assist in training and understanding the complex
20 degradation mechanisms experienced by biological tissues in crime scene scenarios.

21
22 The present work focused on the 3D crosslinking of alginate and ssDNA that can be amplified with
23 human CODIS loci primers^{39,40}. The purpose of this research was to characterize the covalent
24 crosslinking between alginate and dual amine ssDNA to expand on our previous research with
25 ionically crosslinked DNA-encapsulated alginate hydrogels (CATGs)⁴¹. Here, we explore covalent
26 crosslinking with DNA strands of similar size; where the dual amine-terminated 98-base ssDNA
27 was covalently crosslinked to alginate using carbodiimide chemistry. Covalently crosslinked
28 alginate-DNA hydrogels are desirable to localize DNA binding in the alginate and develop more
29 stable hydrogels, which have significant forensic applications

30

1 **2. Experimental Section**

2 2.1 *Materials*

3 1% w/v alginate (low viscosity, Sigma, Oakville, ON) was dissolved in MES buffer (0.05 M, pH 6.0)
4 and allowed to mix overnight at room temperature ($22^{\circ}\text{C} \pm 2^{\circ}\text{C}$). 1-ethyl-3-(3-
5 dimethylaminopropyl) carbodiimide•HCl (EDC, BioShop, Burlington, ON) and *N*-hydroxysuccinimide
6 (Sulfo-NHS, ThermoFisher, Mississauga, ON) were used as received. The ssDNA crosslinker was
7 purchased from Integrated DNA Technologies (Toronto, ON) and reconstituted in UltraPure™
8 water to create a 100 μM stock solution. The crosslinker was a 98-base ssDNA⁴¹ with primary
9 amine groups ($\text{NH}_2 - \text{R}$) on both the 5' and 3' ends (N-DNA-N). Various amine modified ssDNA were
10 also investigated, including: i) primary amines on both the 5' and 3' ends – “N-DNA-N” (AXD2N),
11 ii) primary amine on the 5' end – “N-DNA” (AXD1N) and iii) no terminal amines – “DNA” (AXDON)
12 (Table 1).

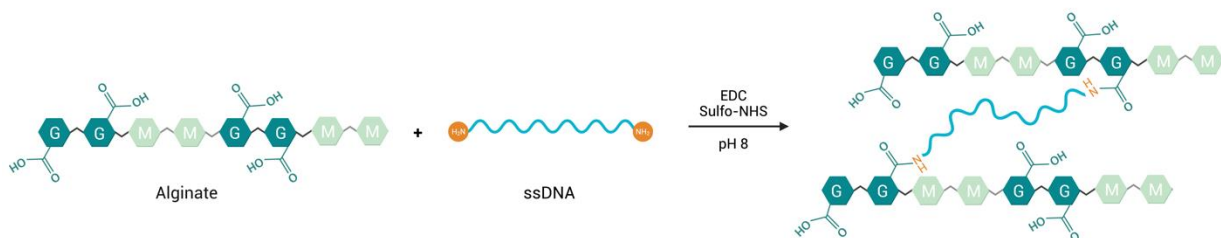
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14 2.2 *Crosslinking Scheme*

15 Each crosslinking reaction had a total volume of 5 mL. First, 1% w/v alginate (MES) was added to
16 individual vials on a stir plate, and EDC (261 μmol) and Sulfo-NHS (65 μmol) were added and
17 allowed to react for 20 minutes at room temperature ($22^{\circ}\text{C} \pm 2^{\circ}\text{C}$). 100 μL of NaOH (1 M) was
18 added to raise the pH to 8 before the addition of the ssDNA (N-DNA-N, N-DNA or DNA). Various
19 concentrations of the DNA crosslinker were added (0.01, 0.025 and 0.05 μmol = 0.02, 0.05 and 0.1
20 μmol amines, respectively) to assess the effects of crosslinking on the materials. The general
21 reaction scheme is shown below (Scheme 1). The samples were allowed to react for 24 hours and
22 EDC/Sulfo-NHS removal was completed using Amicon Ultra 30K MWCO filtration units. Three
23 filtration/wash steps were performed: i) filtration step (1000 x g, 30 minutes), ii) wash step 1 (1000
24 x g, 15 minutes), and iii) wash step 2 (1000 x g, 15 minutes). 1 mL of MilliQ® water was added to
25 each sample before each of the wash steps. In total, nine materials were formed: seven different
26 materials with the carbodiimide crosslinkers present, as well as two controls without crosslinkers
27 (Table 1).

28

29 **Scheme 1.** Covalent attachment of dual amine-terminated ssDNA to alginate polymer. Created
30 with BioRender.com



31

1 **Table 1:** Outline of crosslinking reactions. The presence of “X” indicates that the covalent crosslinkers EDC
 2 and Sulfo-NHS were added to reaction, whereas “P” indicates the DNA was mixed with alginate without
 3 EDC/Sulfo-NHS. Identification in square brackets represents the classification with the addition of Ca²⁺.

Main Crosslinking [+ Ca ²⁺]	Material	EDC (μmol)	Sulfo-NHS (μmol)	ssDNA Structure	ssDNA Concentration (nmol)	pH
Covalent [Dual]	AX ₀	261	65	--	--	
	AXD10	261	65	N-DNA-N	10	8
	AXD25	261	65	N-DNA-N	25	
	AXD50	261	65	N-DNA-N	50	
	AXD0N	261	65	DNA	50	
	AXD1N	261	65	N-DNA	50	8
	AXD2N	261	65	N-DNA-N	50	
Non-Covalent [Ionic]	A ₀	--	--		--	8
	APD50	--	--		50	

4

5 *2.3 Materials Characterization*

6 *2.3.1 Rheology*

7 For the rheological characterization, samples were made to be dual crosslinked (4:1 volume ratio
 8 alginate:calcium (50 mM CaCl₂); where “alginate” in this case refers to the final material – or
 9 “material” – from Table 1). Rheological assessments were conducted using a Discovery HR20
 10 stress-controlled rheometer with an advanced Peltier plate and solvent trap system (TA
 11 Instruments). All rheological tests were conducted at 22°C using a 40 mm parallel plate geometry
 12 with Millipore water in the solvent trap and a 500 μm gap. Oscillation amplitude sweeps and
 13 frequency sweeps were performed between strains of 0.01-150% and frequencies of 0.05-200
 14 rad/s, respectively. The frequency sweeps were conducted at a strain of 2%.

15

16 *2.3.2 Fourier Transform Infrared Spectroscopy (FTIR)*

17 FTIR and SEM analyses were performed using both the DNA crosslinked materials and controls,
 18 both with and without the addition of Ca²⁺. For the FTIR characterization, 1 mL disposable transfer
 19 pipettes were used to place three drops of each material into individual Petri dishes and were
 20 allowed to dry. The dried films were placed directly on the FTIR crystal for analysis. FTIR
 21 measurements were conducted using a Thermo Fisher Nicolet 380 FTIR spectrometer in
 22 absorbance mode (ATR) between wavenumbers of 4000-500 cm⁻¹ with a total of 32 scans.

23

24 *2.3.3 Scanning Electron Microscopy (SEM)*

25 SEM analyses were used for morphological characterization of the materials, using a Hitachi's
 26 FlexSEM 1000 Scanning Electron Microscope. The materials were allowed to air dry on stainless

1 steel stubs, and analyzed at 5 kV. Images were magnified at 500 (100 μm scale), 1K (50 μm scale),
2 and 2K (20 μm scale) SE, and certain images were analyzed at 4K SE (10 μm scale).

3 4 *2.3.4 Thermogravimetric Analysis (TGA)*

5 TGA was used for materials characterization of the different amine variation ssDNA materials,
6 where only dual crosslinked materials were investigated. For this, a TA Instrument SDT Q600
7 thermal analyzer was used. The dual crosslinked materials were freeze-dried for 24 hours prior to
8 TGA analysis and the freeze-dried materials (8 – 12 mg) were analyzed between 25°C to 800°C,
9 with a heating rate of 10°C/min, and under Argon gas (50 mL/min).

10 11 *2.4 Electrospray Ionization Optimization & Microparticle Generation*

12 Before microparticle (MP) development with the crosslinked materials, an optimization
13 experiment was first conducted using alginate dissolved in water, Alg(H₂O), to understand the
14 effects of concentration, voltage (V), spraying distance (*d*), and flow rate (Q) on MP formation. For
15 the purposes of this research, the optimized conditions were found to be: 2.5% w/v alginate +
16 0.5% v/v Tween 20 at a flow rate of 2 $\mu\text{L}/\text{min}$, with 10 kV applied to the needle tip that was 4 cm
17 from the CaCl₂ collection bath (7.35 wt/v%). Following optimization, MPs were created using the
18 2.5% w/v alginate dissolved in MES buffer, Alg(MES), a crosslinking control (AX₀) and two DNA
19 crosslinked materials – a low concentration of DNA (AXD10) and a high concentration (AXD50).
20 Each crosslinked material was made using the same conditions as Section 2.2, with the exception
21 that 2.5% w/v alginate was used instead of 1% w/v. The DNA used in the crosslinking reactions was
22 the dual amine terminated 98-base strand (N-DNA-N).

23
24 The electrosprayed MPs were washed three times with MilliQ® water before visualizing on an
25 EVOS FL Auto microscope (Life Technologies) under 20x and 40x magnification in brightfield mode.
26 Once imaged, the images were imported into Fiji (Image J, 2.3.0/1.53q), scaled, and the particle
27 diameters were manually measured using the “Measure” tool. RStudio (2022.07.1) was used to
28 statistically compare the respective particle diameters using a student’s t-test, and generate the
29 particle size distributions.

30 31 *2.5 DNA extraction and amplification of the MPs*

32 Finally, the washed MPs underwent an extraction and amplification process, similar to a method
33 used in our previous work with CATGs⁴¹. Specifically, the MP solutions were first inverted to
34 ensure homogenous distribution of MPs, and 100 μL was added to 450 μL of gel dissolving buffer
35 (Monarch® DNA Gel Extraction Kit, New England Biolabs). The MPs were incubated at 52°C and
36 500 rpm for 20 minutes (Eppendorf ThermoMixer F1.5), and the manufacturer’s protocol was
37 followed as suggested hereafter. The DNA was eluted with 20 μL of Invitrogen™ UltraPure water
38 (ThermoFisher Scientific). 2 μL of the extracted DNA was added to a 10 μL PCR reaction (95°C for

- 1 10 minutes; 94°C for 30 seconds; 53°C for 1 minute; 72°C for 1 min; repeated for a total of 30
- 2 times with a final step of 65°C for 15 minutes). A 1:50 dilution of the amplified product was
- 3 genotyped using an ABI 3730 DNA Analyzer with a GeneScan 500 ROX size standard.

3. Results and Discussion

3.1 Rheology

The rheological analyses show that overall, the ionically crosslinked samples have much higher storage (G') and loss moduli (G'') compared to the dual crosslinked counterparts (Figure 1). This is unlike other studies with dual crosslinking^{8,10,42}; however, there are two likely explanations for the lower moduli values in the dual crosslinked samples. First, the egg box model of Ca^{2+} -Alg hydrogels leads to tighter junctions, and therefore stiffer gels compared to the hydrogels crosslinked with elastic and larger ssDNA. Further, if alginate is bound to either N-DNA-N, or in the case of the Ca-AX_0 , likely bound to Sulfo-NHS (see Section 3.2), there are less available COO^- groups to participate in Ca^{2+} coordination, leading to softer materials. Additionally, the DNA used in this research is typically larger than the length of DNA incorporated into other hydrogels through 3D crosslinking or functionalization. For example, aptamers can typically range between 20-60 nucleotides⁴³, but a majority of works that include functionalized ssDNA is less than 40 nucleotides. Further, the attachment of ssDNA is usually bound by a single end, where the 3D hydrogel formation is achieved through either hybridization of complementary sequences of bound strands²⁴ or through hybridization of another unbound complementary strand⁴⁴. Using similar attachment strategies to the one used in our research (ssDNA immobilized by both the 5' and 3' ends), Murakami et al²⁰ used 29-base amine-terminated ssDNA to investigate the ability of the DNA-crosslinked hydrogel to shrink or swell when bound to complementary strands, but did not investigate the rheological profile of these materials. Pan et al⁴⁵ investigated the effect of rigid and flexible DNA strands on the hydrogel rheology and found that flexible chains can lead to better stability and stretchability, although Jiang and coauthors have shown that longer DNA strands can lead to softer gels⁴⁶. In our hydrogels, we observed the latter, where the long ssDNA incorporated into our hydrogels leads to very flexible, and soft gel characteristics. Because of this, we do not see linear trends in our moduli values, where the G' should be much larger than G'' as the DNA concentration (and theoretically number of crosslinks) is increased.

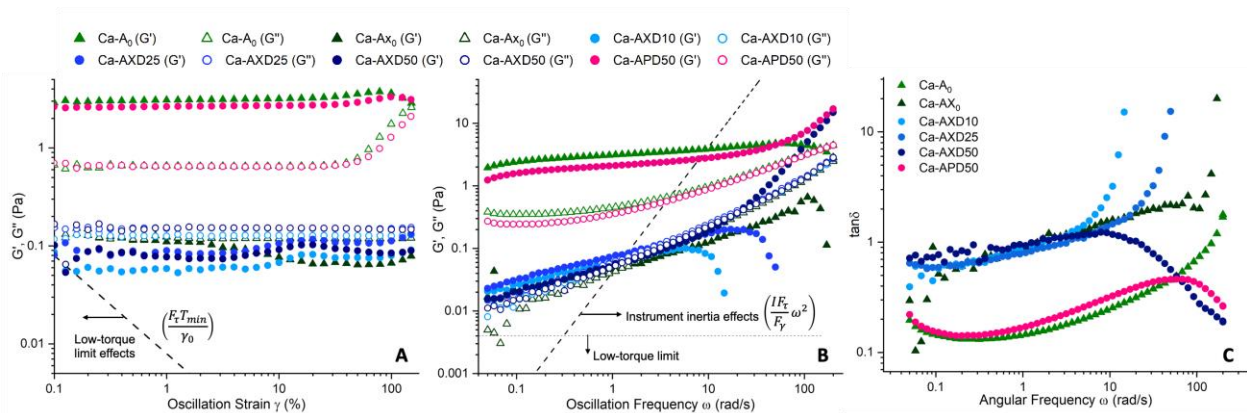


Figure 1. Amplitude sweeps (A), frequency sweeps (B) and $\tan\delta$ (C) for 6 different ionic and dual crosslinked materials. The filled in symbols represent the Storage Modulus (G') and the corresponding non-filled symbols

1 are the Loss Modulus (G'') for a given sample. Overlaid lines represent potential inertial limits based on low torque
2 and instrument inertial effects as outlined by Ewoldt⁴⁷. Limits are based on geometry-specific stress constant (F_τ),
3 minimum torque of instrument (T_{min}), minimum strain (γ_0), instrument inertia (I), geometry-specific strain constant
4 (F_γ) and frequency (ω).

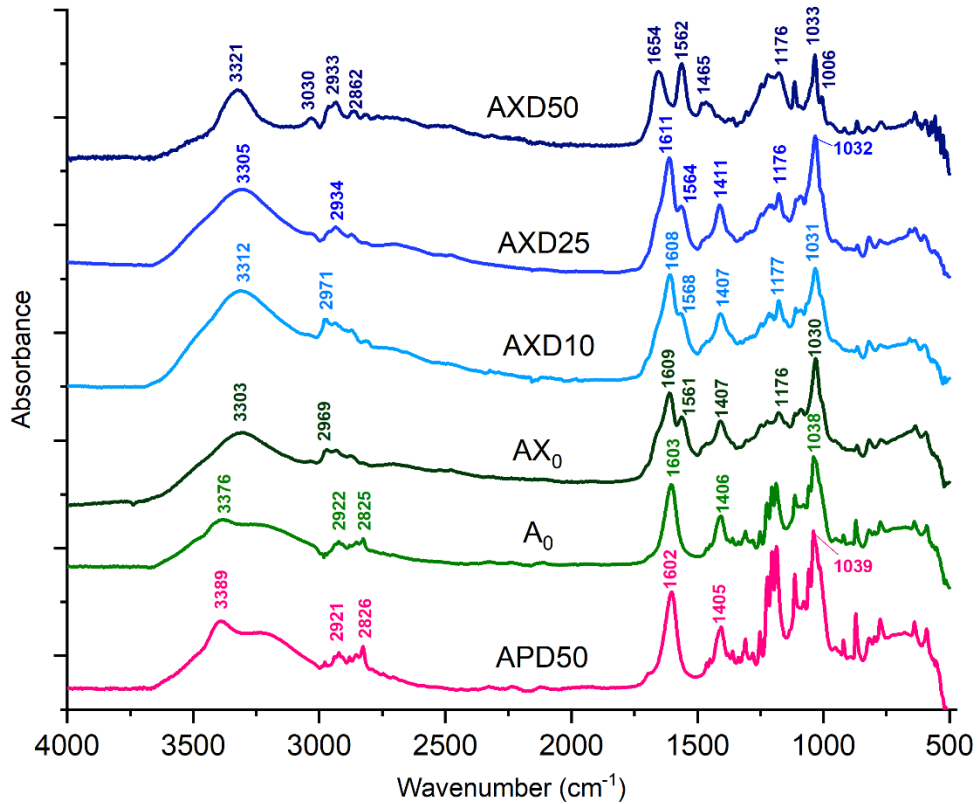
5
6 In the amplitude sweeps, the dual crosslinked materials each had $G'' > G'$ indicating the formation
7 of viscous materials instead of elastic (Figure 1A). Each DNA crosslinked material showed similar
8 curve characteristics, demonstrating similar structural formations, and had distinct behaviour in
9 the amplitude sweeps as demonstrated in another study⁴⁸. The DNA exhibits strain stiffening at
10 the higher oscillation strains, due to its elastic nature. This behaviour is different from the Ca-AX₀
11 sample as the material has a smaller linear viscoelastic region, and exhibits more of a structural
12 breakdown in the same strain range (Figure 1A).

13
14 In the frequency sweeps, the Ca-AXD50 was the only material of the dual crosslinked materials
15 that did not exhibit structural breakdown at higher frequencies (Figure 1B). This breakdown was
16 concentration-dependant, where Ca-AXD10 deforms first (7.9 rad/s), followed by Ca-AXD25 (23.2
17 rad/s), and then Ca-AX₀ (107.7 rad/s), where the Ca-AXD50 does not show structural breakdown
18 (Figure 1B). In the high frequency region (>100 rad/s), Ca-AXD50 behaves like a viscoelastic
19 material⁴⁹, where $G' > G''$, and the material shows high frequency dependence. Frequency
20 dependence can indicate dynamic gels⁵⁰, which is likely the case. Besiri et al⁵¹ have shown that
21 higher frequencies can lead to increased distribution of Ca^{2+} during *in situ* assessments of Ca-
22 alginate crosslinking. These higher frequencies can theoretically disrupt the coordinated Ca^{2+} ions
23 in the dual crosslinked materials, causing structural deformation. Consequently, the Ca-AXD50,
24 which has more covalent crosslinks, can then withstand the higher frequencies despite the
25 reorganization of the Ca^{2+} ions. The Ca-AXD50 also showed less frequency dependence at lower
26 frequencies in the $\tan\delta$ (Figure 1C), indicating a more relatively stiff material compared to the
27 other dual crosslinked counterparts. It should be noted that other work has suggested the
28 influence of inertial effects in the high frequency range of low viscosity materials which can be
29 misinterpreted as structural response⁴⁷. When the inertial limits are overlaid over the rheological
30 data (Figure 1) the high frequency region could be influenced by inertial effects, although replicate
31 analyses suggest this is a true material behaviour (Figure 1, Figure 4).

32
33 Interestingly, even though we see a distinct structural difference in the Ca-AXD50 in the frequency
34 sweep, there was not a large difference between the Ca-AXD50 and Ca-AXD25 in the amplitude
35 sweep. In fact, the Ca-AXD25 shows a higher G' compared to the Ca-AXD50. This observation could
36 be due to the Ca^{2+} interactions offsetting the extra DNA crosslinking occurring in the Ca-AXD50;
37 where less DNA crosslinks in the Ca-AXD25 means that more Ca^{2+} can interact with the COO⁻
38 groups, leading to higher G' overall.

1 3.2 FTIR Spectroscopy

2 The FTIR spectra showed similar trends to the rheology analyses, especially regarding the AXD50
3 sample. The main comparison is shown between the DNA-crosslinked samples (AXD10, AXD25,
4 and AXD50) and controls (AX₀, A₀, APD50) (Figure 2). The control materials were included to
5 directly assess the effects of the covalent crosslinkers (AX₀) as well as the presence of ssDNA on
6 the hydrogel structure (APD50).



7
8 **Figure 2.** FTIR spectra for six different alginate materials. Materials AX₀, AXD10, AXD25, and AXD50 were
9 covalently crosslinked using EDC/Sulfo-NHS only (no Ca²⁺); and samples A₀ and APD50 were mixed without the
10 addition of EDC/NHS or Ca²⁺.

11
12 Alginate was observed to have characteristic peaks at 1603 and 1406 cm⁻¹, which corresponds to
13 the COO⁻ moieties of the M and G blocks, specifically the symmetrical stretching of the carbonyl
14 groups (C=O) at 1603 cm⁻¹ and asymmetrical stretching (C-OH) at 1406 cm⁻¹⁵². In the presence of
15 EDC/Sulfo-NHS, the 1603 cm⁻¹ peak is split into two peaks at 1609 and 1561 cm⁻¹, which is likely
16 due to residual C=O in solution from Sulfo-NHS. This is also supported given that S=O stretching
17 can be seen only in the covalent materials at 1176 cm⁻¹.

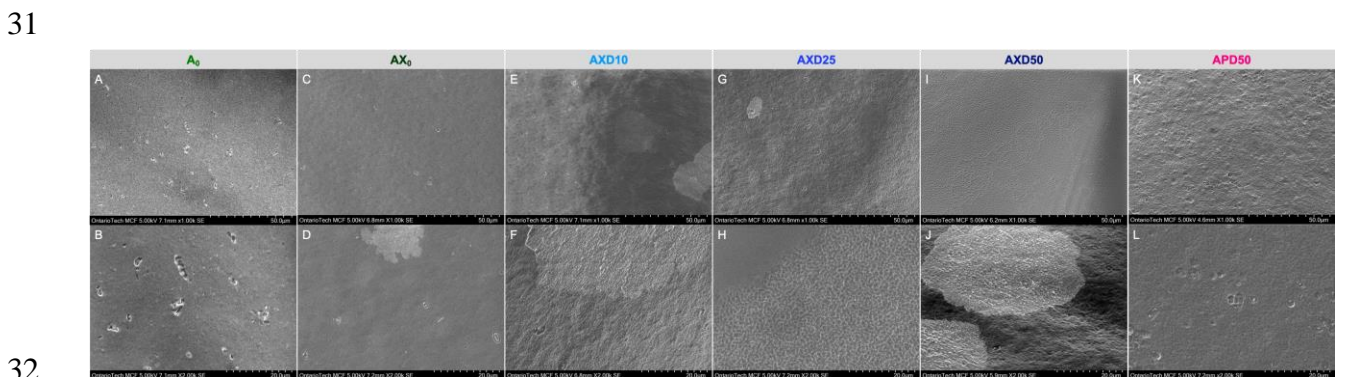
18
19 In the AXD50 material, there is an even further red shift in the split peaks to 1654 cm⁻¹, attributed
20 to the amide I bond formation. There is also an additional simultaneous loss of the 1407 cm⁻¹ peak.
21 Both changes indicate the loss of COO⁻ groups, which occurs during amide bond formation from

1 N-DNA-N binding to the alginate (Scheme 1). Other indicators of amide bond formation are
2 observed in the increased absorbance of the peak at 1562 cm^{-1} , likely due to the N-H deformation
3 of amide II⁵³. There is also a more distinct peak at 3321 cm^{-1} , which is reasoned to be caused by
4 N-H stretching of the amide A bond⁵³. The addition of N-DNA-N in the alginate material without
5 crosslinkers (APD50) shows relatively no change to the COO^- groups. This indicates that the
6 presence of DNA alone is not leading to the amide bond detection observed in the DNA-crosslinked
7 materials, which naturally exist in three of the four DNA bases. An interesting note is the shift of
8 the asymmetrical stretch of the $-\text{COO}^-$ to a higher wavenumber in the in the APD50 sample from
9 1407 cm^{-1} to 1410 cm^{-1} . This could be due to the interaction of the N-DNA-N ions with the
10 negatively charged alginate, although another study showed a shift to higher numbers without
11 Ca^{2+} interaction⁵⁴.

12
13 In both the FTIR and SEM assessments, the calcium-crosslinked samples (ionic, dual) showed less
14 distinct differences overall (Supporting Information), especially between covalently crosslinked
15 samples. This is likely the result of the calcium ionic crosslinks masking any underlying covalent
16 linkages between DNA and alginate, as mentioned previously. It should be noted that even though
17 materials made with the addition of calcium showed less distinct differences in the SEM and IR
18 assessments, the rheological assessments did observe distinct differences, largely as the technique
19 can directly probe the viscoelastic properties of materials on a more sensitive molecular level.

20 21 3.3 SEM

22 Analyses with SEM showed that the DNA-crosslinked materials had more order and structured
23 surfaces compared to encapsulated N-DNA-N (APD50) and the non-DNA containing materials (AX_0 ,
24 A_0) (Figure 3). Both the A_0 (Figure 3, A and B) and AX_0 (Figure 3, C and D) materials had relatively
25 smooth and pitted surface morphologies compared to any of the DNA-containing materials. The
26 DNA-crosslinked materials exhibited varying surface morphologies, signifying more structured and
27 complex materials (Figure 3, E-J). Compared to the DNA-crosslinked materials, the APD50 sample
28 displayed a more porous structure, with a less structured fibrous surface (Figure 3, K and L). This
29 contrasts the covalently crosslinked counterpart, where more circular aggregations were observed
30 with more apparent order (Figure 3, J) and intricate networks (Figure 3, I).



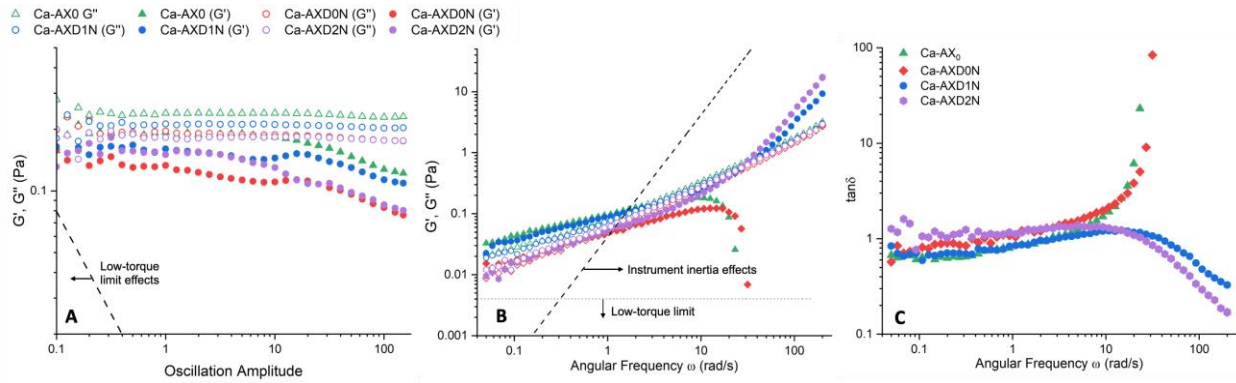
1 **Figure 3:** SEM images of six different materials with/without crosslinkers and with/without N-DNA-N. Samples
2 from left to right are: A₀, AX₀, AXD10, AXD25, AXD50, and APD50. The first row shows magnification at
3 approximately x1000 (50 μm scale intervals), and the second row is x2000 (20 μm intervals).
4

5 As mentioned previously, the calcium crosslinked samples (dual or ionic) demonstrated less
6 distinct surface morphologies between samples, but still showed general material differences
7 including more bulbous surfaces in the non-crosslinked controls and particle shape differences
8 between the DNA-crosslinked materials compared to the alginate controls (see Supporting
9 Information).
10

11 *3.4 Influence of ssDNA Structure*

12 The different ssDNA structures used in the crosslinking reaction (N-DNA-N, N-DNA, DNA) did
13 appear to influence the overall material properties. When assessed with rheology, materials
14 crosslinked with either dual ended amine ssDNA (Ca-AXD2N) or single amine ssDNA (Ca-AXD1N)
15 did not exhibit any structural breakdown at high frequencies, similar to Ca-AXD50, and was unlike
16 the alginate material covalently crosslinked with DNA without any terminal amines (Ca-AXD0N)
17 (Figure 4). The replicate dual amine DNA crosslinked material (Ca-AXD2N) demonstrated more of
18 a linear viscoelastic region (LVR) in the amplitude sweep (Figure 4A) compared to the first reaction
19 (Ca-AXD50, Figure 1A). When ssDNA without any terminal amines was reacted in a covalent
20 crosslinking reaction (Ca-AXD0N), the material has a lower G' compared to the materials
21 crosslinked with DNA that contain terminal amines (Figure 4A). Interestingly, the Ca-AXD0N
22 material behaves differently than both encapsulated ssDNA (Ca-APD50, Figure 1A) and the
23 alginate covalent controls (Ca-AX₀, Figure 1A, Figure 2A), indicating a unique material influenced
24 by the lack of terminal amines. Further, the FTIR analyses detected no amide bonds the AXD0N
25 material (Figure 5), suggesting that the DNA without terminal amines is not covalently attaching
26 to the alginate as the AXD2N and AXD1N do. Without chemically attaching to the alginate
27 backbone directly, the EDC can still react with the phosphate backbone of the DNA⁵⁵, which could
28 be causing nonspecific reactions between DNA molecules and therefore a structurally weak
29 material hydrogel. This is also supported through the SEM, where the AXD0N material appears
30 more porous similar to the AX₀ material, but is unlike the AXD1N and AXD2N have more smooth
31 and structured surfaces (Figure 5 A-H).

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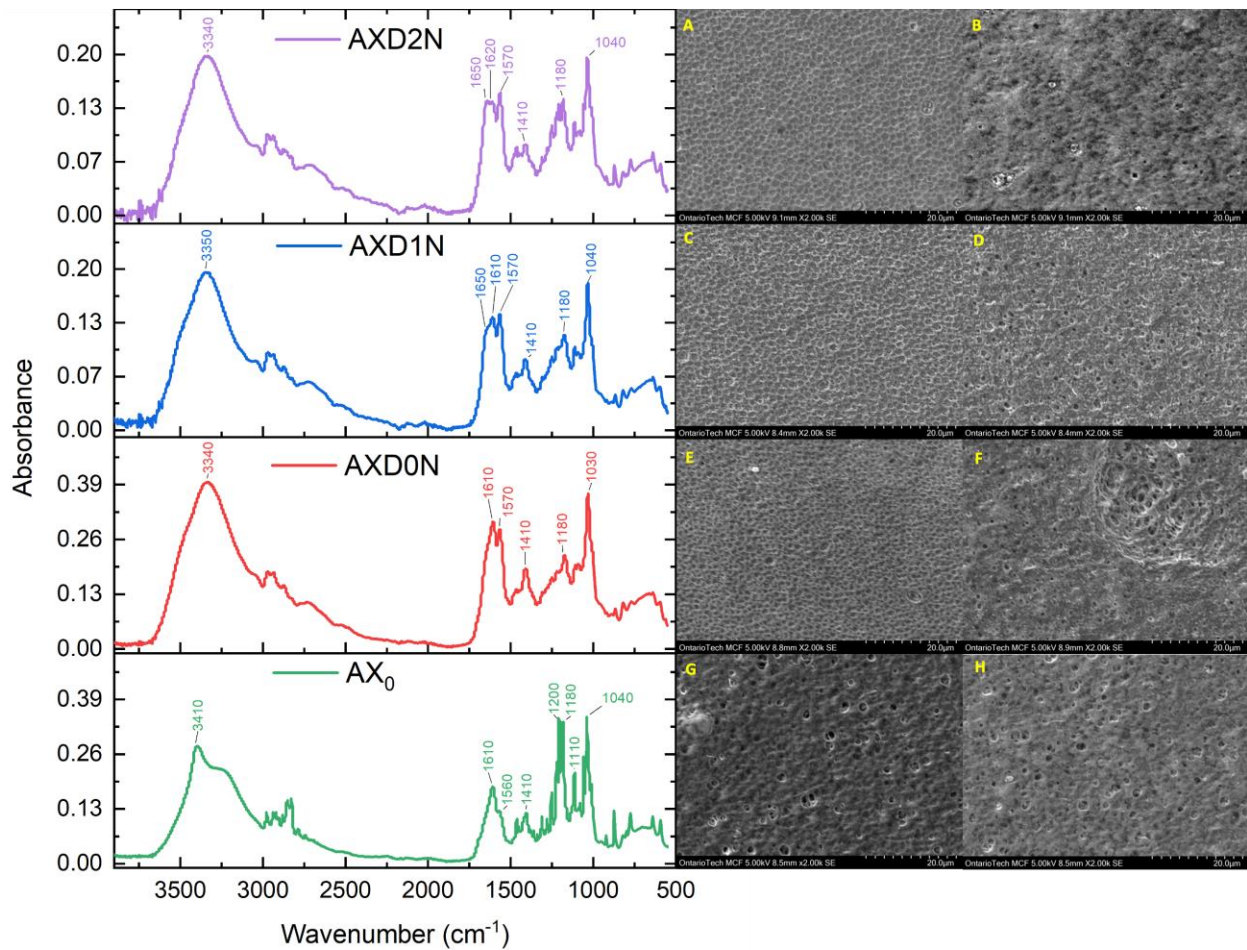
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Figure 4. Amplitude sweeps (A) and frequency sweeps (B) and $\tan\delta$ (C) for 4 ssDNA with varying amine reactivities: no DNA (Ca-AX₀), ssDNA with no terminal amines (Ca-AXD0N), one terminal amine (Ca-AXD1N), and two terminal amines (Ca-AXD2N) covalently crosslinked with alginate. The samples were made to be dual crosslinked with Ca²⁺. The filled in symbols represent the Storage Modulus (G') and the corresponding non-filled symbols are the Loss Modulus (G'') for a given sample.



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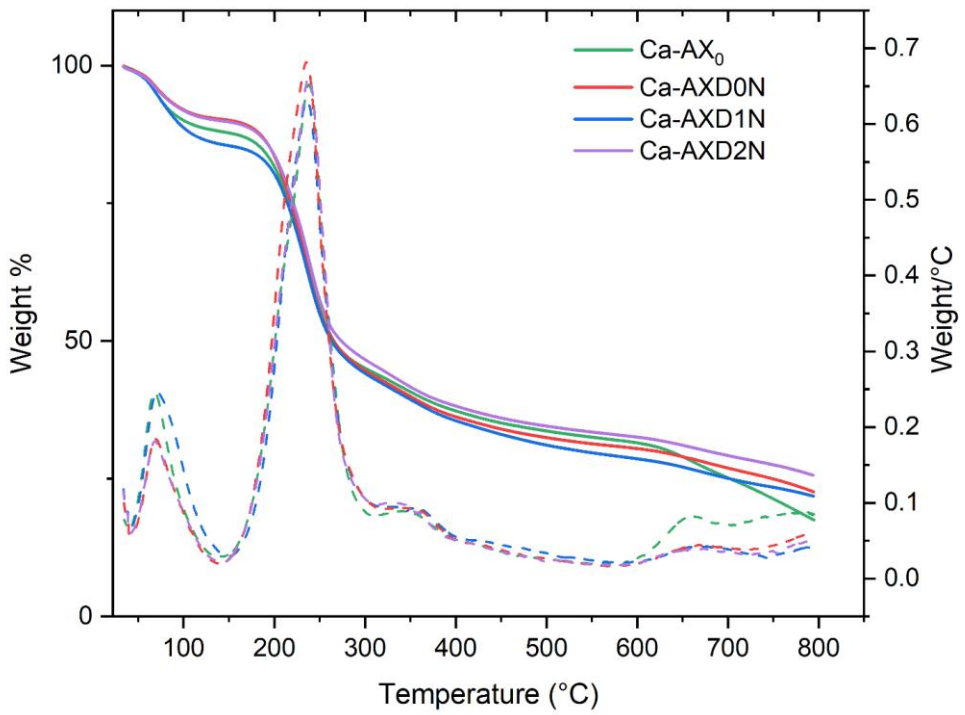
Figure 5. FTIR spectra (left) 4 different alginate materials crosslinked with no DNA (AX₀), ssDNA with no terminal amines (AXD0N), ssDNA with one terminal amine (AXD1N), and ssDNA with two terminal amines (AXD2N).

1 Images on the right show the corresponding SEM images of each material. Two SEM images are included
2 to show the variation in surface morphologies.

3

4 Finally, TGA denoted overall similar degradation profiles (Figure 6), with slight variation during
5 each degradation event. Ayouch et al⁵⁶ indicate that alginate-based materials can undergo three
6 degradation events; between 30-150°C (water desorption), 200-300°C (decomposition of carbon
7 in alginate backbone and formation of sodium carbonate – NaCO₃), and 600-700°C (two-stage
8 breakdown of NaCO₃). Our TGA analyses showed similar degradation characteristics in the
9 materials, with more distinct material differences shown at higher temperatures. The Ca-AXD2N
10 material had the least relative amount of weight loss (74.37%) compared to Ca-AXD1N (78.16%),
11 Ca-AXD0N (77.38%) and Ca-AX₀ (82.49%). At the higher temperatures (over 600°C), the alginate
12 control (Ca-AX₀) is more easily degraded compared to the other materials, which is likely due to
13 the strong molecular networks of the DNA-crosslinked materials⁵⁷, and explains why the Ca-AXD2N
14 has more residual material.

15



16
17 **Figure 6:** TGA curves of four different dual crosslinked alginate materials covalently crosslinked with no DNA
18 (Ca-AX₀), ssDNA with no terminal amines (Ca-AXD0N), ssDNA with one terminal amine (Ca-AXD1N), and ssDNA
19 with two terminal amines (Ca-AXD2N). Samples were made to be dual crosslinked with Ca²⁺. The dashed
20 curves represent Weight/°C, while the solid curves are the Weight % at each temperature

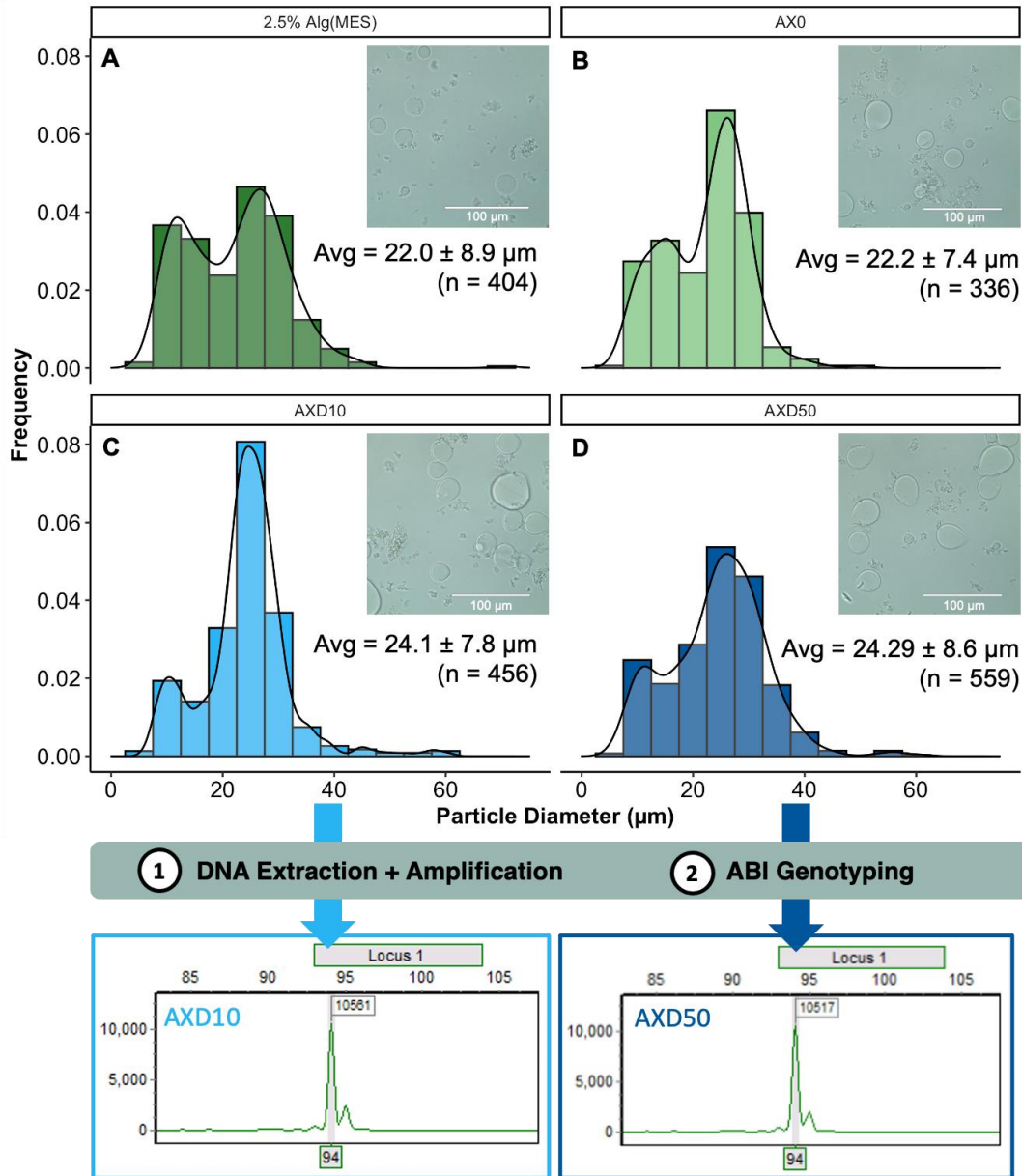
21
22 Together, the analyses signify that the amine modification of ssDNA at either one or both
23 terminal ends leads to the immobilization of DNA to alginate. The ability to covalently attach the

1 DNA by either one end or both ends is particularly beneficial as it allows for many applications of
2 the hydrogel materials. Specifically, the development of microparticles (MPs) using DNA-
3 crosslinked alginate is of particular interest and is discussed in the next section.

4 5 *3.5 Microparticle Generation & Applications*

6 Human reference standards with the hydrogels can be developed in many forms, but one
7 valuable application is the development of forensic blood simulants that have both relevant fluid
8 properties and genetic capabilities similar to human blood^{58,59}. To accomplish this, the covalently
9 crosslinked alginate and ssDNA hydrogels were used to develop microparticles (MPs) of relevant
10 white blood cell (WBC) size(s). MPs can be effectively generated through electrospray ionization
11 (ESI), where the covalent crosslinked alginate-DNA is extruded through an electrified needle into
12 a calcium bath with a counter electrode. The overall MP size and morphology is influenced by
13 many factors⁶⁰, but the covalent materials must be able to coordinate effectively with Ca^{2+} to
14 achieve relevant MPs, especially for our applications.

15
16 The MPs assessed in this work included those derived from Alg(MES), AX₀, AXD10 and AXD50, all
17 with 2.5% w/v alginate. The 2.5% Alg(MES) was used as a control to compare to materials that
18 had undergone the conditions of a crosslinking reaction (AX₀) and a crosslinking reaction with
19 both low (AXD10) and high (AXD50) ssDNA concentrations. Tween 20 was added to the materials
20 as a non-ionic co-solvent to lower the surface tension of the materials^{61,62}, which allowed for a
21 more stable Taylor cone and more homogenous particle distributions. The two non-DNA
22 materials (2.5% Alg(MES) and AX₀) had similar average particle diameters ($p = 0.78$), and were
23 smaller than the DNA-crosslinked MPs (Figure 7). The AXD10 material had MPs that were
24 statistically different from both 2.5% Alg(MES) ($p = 0.0003$) and AX₀ ($p = 0.0004$). Similarly, the
25 AXD50 was also statistically different from both 2.5% Alg(MES) ($p = 6.81 \times 10^{-5}$) and AX₀ ($p = 9.91$
26 $\times 10^{-5}$). Interestingly, it appears that by crosslinking alginate with DNA, the MPs are larger than
27 MPs electrosprayed without DNA. This is true irrespective of the concentration of DNA attached
28 to alginate, as both AXD10 and AXD50 had similar average particle sizes ($p = 0.68$). This finding is
29 appropriate for the development of human blood simulants, since WBCs are typically larger than
30 red blood cells (RBCs) naturally.



1
2 **Figure 7:** Microparticle (MP) size distribution of four different materials: 2.5% w/v alginate in MES buffer
3 (A), AX0 (B), AXD10 (C) and AXD50 (D) electrospayed with 0.5% v/v Tween 20 into a CaCl_2 bath. Inlet
4 inset images show microscopic images of MPs visualized in brightfield mode at 40x magnification.
5 Corresponding genotype of ssDNA extracted and amplified from AXD10 (E) and AXD50 (F) microparticles.
6
7 Therefore, by simply crosslinking alginate with ssDNA, the particles formed (our “WBCs”) would
8 be larger than those electrospayed without DNA (our “RBCs”) under the same conditions.
9 Having two different cell mimetic types is not only important for the overall functionality of the
10 blood material, but also the incorporation of viscoelastic RBCs is a critical feature to accurately
11 simulate blood drop dynamics of spreading and splashing⁶³ in crime scene scenarios. Another
12 important consideration in forensic blood simulants is the ability for materials to be processed in

1 a similar manner to blood evidence – specifically, the ability to extract, amplify and genotype the
2 contained DNA. We investigated the overall functionality of the candidate WBCs and determined
3 that following a typical blood processing procedure with an additional modification to break
4 down the alginate material⁴¹), the crosslinked DNA could be extracted, amplified and produced
5 an interpretable genotype similar to non-crosslinked controls (Figure 7). This is promising
6 because it indicates that the crosslinked DNA still remains both functional and intact through the
7 electrospraying process, generating viable WBC mimetics for forensic blood simulants.

1 **4. Conclusions**

2 This work explored the effect of covalent crosslinking between amine-terminated ssDNA and
3 alginate and its role in generating MPs for forensic blood simulants. The covalent attachment of
4 ssDNA to alginate provides more stable linkages and immobilization of the DNA within the
5 hydrogels. The characterization of the alginate-DNA hydrogels demonstrated covalent
6 crosslinking between alginate and ssDNA based on the detection of amide bonds in the highest
7 DNA concentration material. The rheological assessments also showed that the AXD50 material
8 had less frequency dependence in the $\tan\delta$, as well as dynamic gel characteristics in the high
9 frequency range, unlike the other covalent crosslinked materials. This unique behaviour can be
10 attributed to the higher number of crosslinks formed in the high DNA concentration material,
11 where the flexible ssDNA allows for soft gel characteristics. Similarly, SEM revealed that the DNA
12 crosslinked alginate hydrogels were more structured, with interconnected networks distinct
13 from the alginate and DNA control materials. Finally, investigation into the molecular dynamics
14 of the alginate and DNA crosslinking suggest that the binding of ssDNA primarily occurs from the
15 primary amines at 5' and 3' ends, further supporting the development of 3D crosslinked
16 networks in the reaction.

17
18 The attachment of DNA to the alginate backbone while maintaining functionality in MP form
19 lends the material to such roles as mimetic WBCs in forensic blood simulants. The low G'
20 characteristics of the covalent crosslinked Alg-DNA hydrogel is favourable for the development
21 of MPs containing immobilized DNA, much like traditional WBCs in human blood. This work
22 provides the baseline for the development of our alginate-based forensic blood simulant. Future
23 works include characterizing these materials in the complex biological and physical processes
24 that occur during evidence formation and degradation at the crime scene.

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16

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19

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25

26 **ABBREVIATIONS**

27 **DNA**, deoxyribonucleic acid; **ssDNA**, single stranded DNA; **RBCs**, red blood cells; **WBCs**, white
28 blood cells; **MP**, microparticle; **N-DNA-N**, dual amine terminated ssDNA; **N-DNA**, single amine
29 terminated ssDNA. **FTIR**, Fourier Transform infrared spectroscopy; **SEM**, scanning electron
30 microscopy; **TGA**, thermogravimetric analysis. **CODIS**, Combined DNA Index System; **CATG**,
31 calcium alginate tissue gels; **MES**, 2-(N-morpholino)ethanesulfonic acid; **EDC**, 1-ethyl-3-(3-

1 dimethylaminopropyl) carbodiimide•HCl; Sulfo-NHS, N-hydroxysuccinimide; G' , storage modulus;
2 G'' loss modulus.

3

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