

Design of supramolecular hybrid nanomaterials comprising peptide-based supramolecular nanofibers and *in situ* generated DNA nanoflowers through rolling circle amplification

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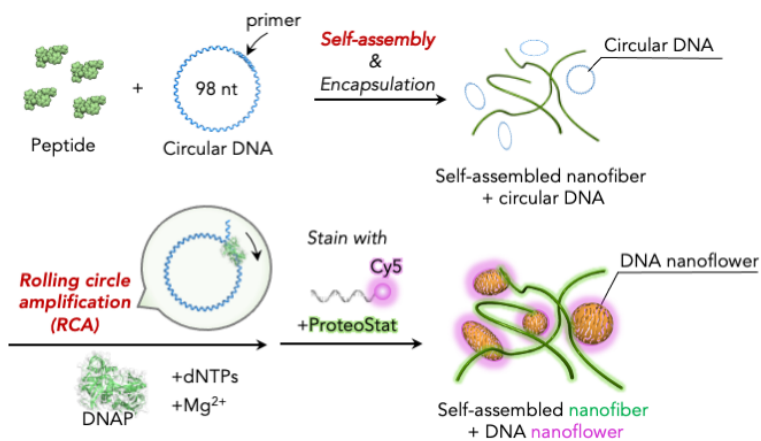
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1 **Abstract**

2 The artificial construction of multicomponent supramolecular materials comprising plural
3 supramolecular architectures that are assembled orthogonally from their constituent molecules has attracted
4 growing attention. Here, we describe the design and development of multicomponent supramolecular materials
5 by combining peptide-based self-assembled fibrous nanostructures with globular DNA nanoflowers constructed
6 by the rolling circle amplification reaction. The orthogonally constructed architectures were dissected by
7 fluorescence imaging using the selective fluorescence staining procedures adapted to this study. The present,
8 unique hybrid materials developed by taking advantage of each supramolecular architecture based on their
9 peptide and DNA functions may offer distinct opportunities to explore their bioapplications as a soft matrix.

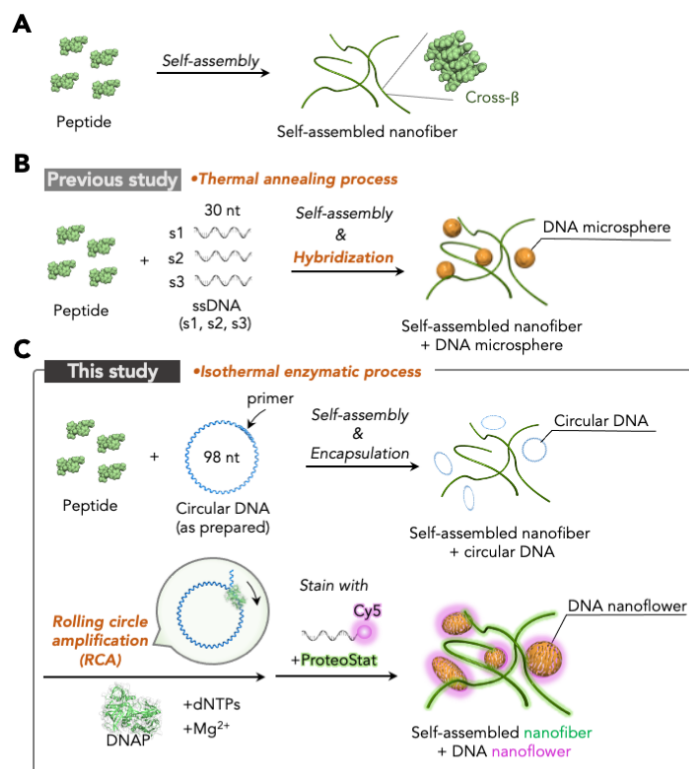
11 **Introduction**

12 Molecular assembly based on noncovalent synthesis has allowed the construction of various
13 supramolecular nanostructures,[1,2,3] thereby helping in advancing our understanding of elaborate biological
14 systems.[4] However, the artificial construction of multicomponent supramolecular materials comprising plural
15 supramolecular architectures assembled exclusively and precisely from their constituent molecules has remained
16 largely unexplored. In this context, because orthogonal self-assembly is of paramount importance, efforts to
17 expand the repertoire of supramolecular architectures to meet the requirements for orthogonal self-assemblies
18 have recently and actively been investigated.[5,6,7,8,9] Examples of aqueous multicomponent supramolecular
19 materials include biomolecules like peptides and nucleic acids, which may be superlative candidates due to their
20 potentially and tunable orthogonal molecular self-assembling propensities, biocompatibility, sustainability for
21 future bioapplications, and availability.[10,11,12,13]

22 Rapidly expanding DNA nanotechnology has offered a bottom-up approach to access various
23 supramolecular architectures with structural precision at nanometer and submicrometer scales, thereby eliciting
24 controlled functions at the respective levels.[14,15,16] Accordingly, our previous study explored aqueous
25 multicomponent supramolecular materials comprising semi-artificial glycopeptide-based self-assembled
26 nanostructures, DNA microspheres,[17] and DNA tile nanotubes.[18] These DNA architectures were
27 constructed through thermal annealing-induced hybridizations of multiple and sequence-programmed nucleic-
28 acid chains (typically, three DNA-microsphere strands [19,20] and five DNA tile nanotube strands [21]).
29 Meanwhile, rolling circle amplification (RCA) has attracted growing attention as an isothermal enzymatic process
30 to obtain DNA-based nanostructures and microstructures.[22,23] Notably, for example, under typical RCA
31 conditions in the presence of divalent cations such as the magnesium ion, unique globular and flower-like
32 morphology (referred to as DNA nanoflowers) emerge spontaneously as RCA products. These products most
33 presumably arise through the complexation of the as-synthesized long single-stranded (ss) DNA molecules with
34 divalent cationic pyrophosphate (e.g., Mg_2PPi).[24] However, to our knowledge, the orthogonal coexistence of

1 DNA nanoflowers as RCA products and peptide-based supramolecular nanostructures had not been investigated
2 yet.

3 We herein describe the construction of multicomponent supramolecular materials by combining
4 peptide-based self-assembled fibrous nanostructures and DNA nanoflowers constructed by the RCA reaction.
5 As depicted in **Fig. 1**, circular DNA molecules could be entrapped inside a network of peptide-based self-
6 assembled fibrous nanostructures. Then, subsequent isothermal RCA reactions catalyzed by DNA polymerase
7 (DNAP) could give rise to the *in situ* formation of DNA nanoflowers in the presence of peptide-based self-
8 assembled fibrous nanostructures. In this study, we focused on investigating self-assembled nanostructures
9 constructed from newly designed and synthesized anionic peptide derivatives and constructing hybrid materials
10 by combining them with DNA nanoflowers. We expect that the developed unique hybrid materials could offer
11 distinct opportunities to explore their bioapplications as soft matrices by taking advantage of the functions of
12 each supramolecular architecture.



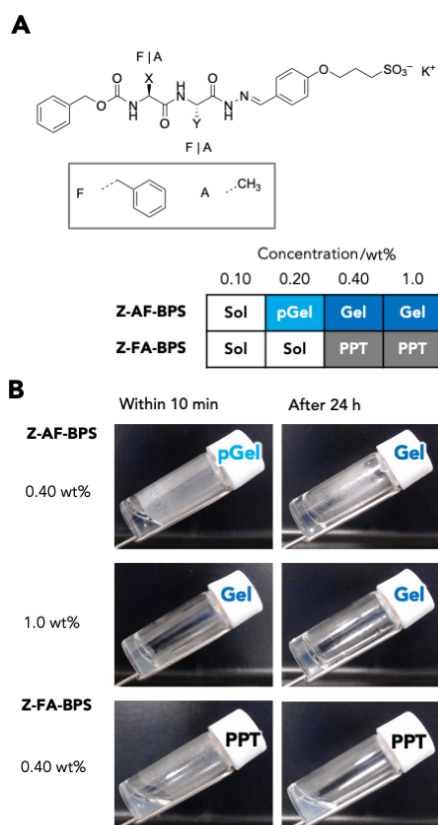
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14 **Fig. 1** Schematic (not to scale) showing (A) self-assembly of peptides to form fibrous supramolecular (self-
15 assembled) nanostructures, (B) an orthogonal construction of hybrid materials comprising peptide-based self-
16 assembled nanofibers and DNA microspheres through the thermal annealing process [17], and (C) the isothermal
17 enzymatic construction of DNA nanoflowers through RCA in the presence of peptide-based self-assembled
18 nanofibers, giving rise to hybrid materials.

19

20 Results and discussion

21 Synthesis and hydrogel formation abilities of the anionic peptide derivatives

1 In this study, we newly designed two anionic peptide derivatives (**Z-AF-BPS** and **Z-FA-BPS**, **Fig. 2A**),
 2 comprising phenylalanine (F) and alanine (A). Their syntheses were carried out according to **Scheme S1** modified
 3 slightly from our previous report on the similar compound.[25] As shown in **Fig. 2B**, **Z-AF-BPS** showed
 4 hydrogel formation abilities above 0.40 wt% (6.4 mM) whereas no hydrogel formation was observed with its
 5 inversed sequence (**Z-FA-BPS**), even at a higher concentration such as 1.0 wt% (15 mM) and after 24-h
 6 incubation. In the following studies, we investigated the properties of **Z-AF-BPS** hydrogel because the
 7 formation of a self-assembled nanostructure network was reasonably anticipated for the **Z-AF-BPS** hydrogel.



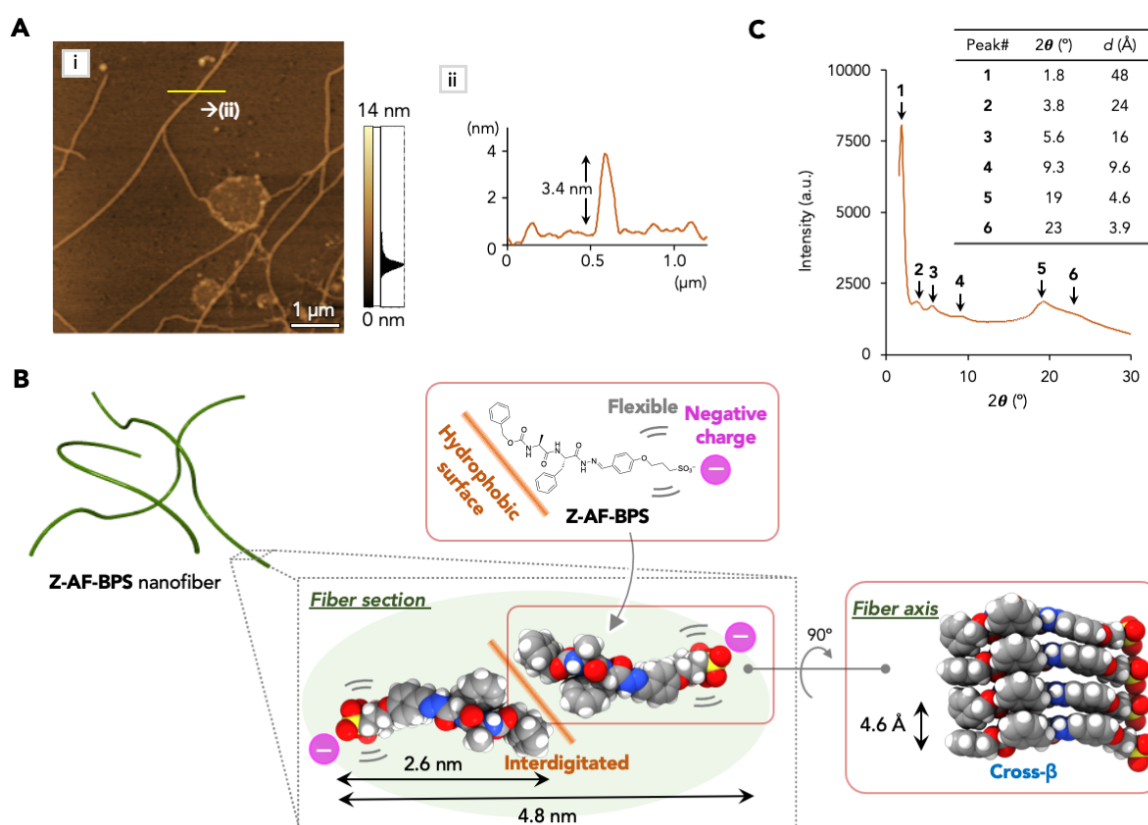
8
 9 **Fig. 2 (A)** Chemical structures of the self-assembling peptides (**Z-AF-BPS** and **Z-FA-BPS**) designed and
 10 synthesized in this study, showing their gel formation abilities based on their concentrations [Sol: solution, pGel:
 11 partial gel, Gel: gel, and PPT: precipitation (or suspension)]. **(B)** Photographs showing **Z-AF-BPS** hydrogels and
 12 **Z-FA-BPS** suspensions obtained 10 min and 24 h after dissolution by heating [**Fig. S9A** shows photographs of
 13 the other conditions in panel **(A)**]. *Conditions*: 50 mM MES-NaOH (pH 7.0) containing DMSO (2.0 vol%).

15 Characterization of the **Z-AF-BPS** hydrogel

16 Viscoelastic properties of the **Z-AF-BPS** hydrogel was evaluated by rheology measurements. Although
 17 an almost linear viscoelastic region was uncovered by strain sweep oscillatory rheology, the mechanical
 18 weaknesses of the hydrogel were evident from its relatively low storage (G') and loss modulus (G'') compared
 19 with similar peptide-based hydrogels (**Fig. S10**).[25] Microscopic observations were subsequently conducted to
 20 gain insight into the self-assembled structures of **Z-AF-BPS**. As shown in **Fig. 3A**, the atomic force microscopy

(AFM) images (under ambient air conditions) revealed long straight nanofibers with a length over several μm and an average height of 3.4 nm. The formation of such extended one-dimensional self-assembled architectures would be attributed to the hydrogel formation ability of **Z-AF-BPS**. Nevertheless, the weak entanglement of the nanofibers, most probably due to electrostatic repulsion, could be correlated with the mechanical weakness of the hydrogel. On the other hand, the smooth fibrous structures with the height (3.4 nm), which is less than the double of the molecular length (2.6 nm) of **Z-AF-BPS**, suggest the interdigitated bimolecular structure as a unit for the one-dimensional self-assembled structure. A plausible model for the self-assembled **Z-AF-BPS** structure displayed as **Fig. 3B** may be explained, at least in part, as a weak entanglement and bundling of the negatively charged nanofibers through electrostatic repulsion, which would be desirable for the orthogonal coexistence with other supramolecular architectures in aqueous media, owing to mitigated nonspecific interactions that produce undesired and less-controlled aggregations. Transmission electron microscopy (TEM) image (*vide infra*) results disclose the comparable thin long nanofibers.

Subsequently, we gained further insight into the molecular assembly mode of **Z-AF-BPS** to form fibrous architectures by X-ray diffraction (XRD) measurements. To this end, freeze-dried xerogel samples were prepared from hydrogels using deionized water instead of the aqueous buffer [50 mM MES-NaOH (pH 7.0)] because the aqueous buffer contains salts that potentially interfere with the XRD data and was unremovable. Almost comparable hydrogel formation abilities were validated under both conditions (**Fig. S9B**). As shown in **Fig. 3C**, we observed that a broad peak centered as the Bragg spacing at $d = 4.6 \text{ \AA}$ (peak #5), which is assignable to the inter-strand distance of the β -sheet structure. We also observed a shoulder peak at $d = 3.9 \text{ \AA}$ (peak #6), probably assignable to the π - π stacking of the aromatic groups in phenylalanine (F) as well as the Z and BPS moieties in **Z-AF-BPS**. These findings were consistent with the cross- β structure [26] for **Z-AF-BPS** nanofibers. Additionally, a weak but distinguishable peak at $d = 9.6 \text{ \AA}$ (peak #4), most probably ascribed as the inter-sheet stacked distance of the β -sheet structure for a typical cross- β structure, was observed. Furthermore, a peak was observed at a small angle region, $d = 48 \text{ \AA}$ (peak #1), which was presumably accompanied by higher-order reflections [$d = 24 \text{ \AA}$ (peak #2) and 16 \AA (peak #3)]. These diffractions suggest that **Z-AF-BPS** self-assembled to form a layer structure with a spacing of 4.8 nm in the xerogel (dried) state, meaning probably a lateral bundling of **Z-AF-BPS** nanofibers. Moreover, the spacing of 4.8 nm, most likely ascribed to the diameter of **Z-AF-BPS** nanofibers, was shorter than double the molecular length (2.6 nm) and the height (3.4 nm) of the fibrous structures observed in the AFM images, in which a tip- and/or surface-induced deformation should be considered.[27,28] Collectively, these XRD results indicate the interdigitated bimolecular and cross- β structure of **Z-AF-BPS** nanofibers (**Fig. 3B**) while tilting the β -strand against the long axis of the fibers was also conceivable.[29]

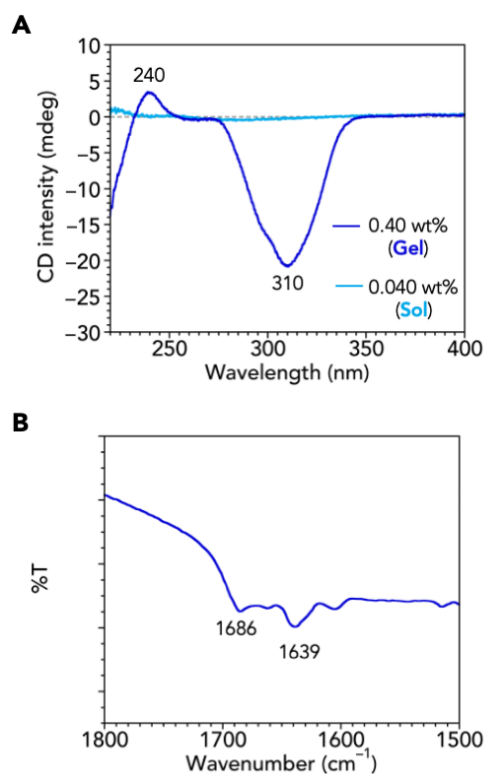


1
2 **Fig. 3** (A) Representative AFM (tapping mode) images showing the height image (i) of **Z-AF-BPS** nanofibers
3 on freshly cleaved mica. The panel (ii) represents a cross-sectional profile along the white line in the image (i).
4 (B) Plausible models for the self-assembled structure of **Z-AF-BPS**, giving rise to one-dimensional (cross-β)
5 bimolecular structures in the hydrogel state. (C) XRD pattern from a freeze-dried sample of **Z-AF-BPS** hydrogel
6 (1.0 wt%).

8 Spectroscopic studies on the **Z-AF-BPS** hydrogel

9 **Fig. 4A** shows a circular dichroism (CD) spectrum of the **Z-AF-BPS** hydrogel (0.40 wt%, 6.4 mM)
10 exhibited a negative CD signal at 310 nm, assignable to a chiral arrangement of BPS moiety in the self-assembled
11 fibrous structures that can be originated from chiral information transfer from peptide moiety to BPS moiety
12 enhanced by self-assembly.[25] Furthermore, a positive CD signal at 240 nm and a negative CD signal shorter
13 than 235 nm were observed. In this wavelength region (200–250 nm), the presence of phenylalanine (F) as well
14 as the Z and BPS moieties in **Z-AF-BPS** could lead to a mixed spectra ascribable to the electronic transitions of
15 its peptide backbone and aromatic moieties.[25,30] The CD signals became almost silent at a lower concentration
16 (0.040 wt%, 0.64 mM), indicating the absence of self-assembled structures at the lower concentration. Indeed,
17 the thioflavin T (ThT) assay conducted under similar conditions revealed that critical aggregation concentration
18 of **Z-AF-BPS** was 0.73 ± 0.03 mM ($n = 3$; **Fig. S11**). Fourier-transform infrared (FTIR) spectra of **Z-AF-BPS**
19 hydrogel exhibited two major bands centered at 1639 and 1686 cm^{-1} with shoulder peaks as displayed in **Fig. 4B**,
20 suggesting that hydrogen bonding of an amide backbone and (Z-related) carbonate moieties at the N-terminal

1 via β -sheet formation, consistent with the proposed self-assembled structure depicted in **Fig. 3B**.



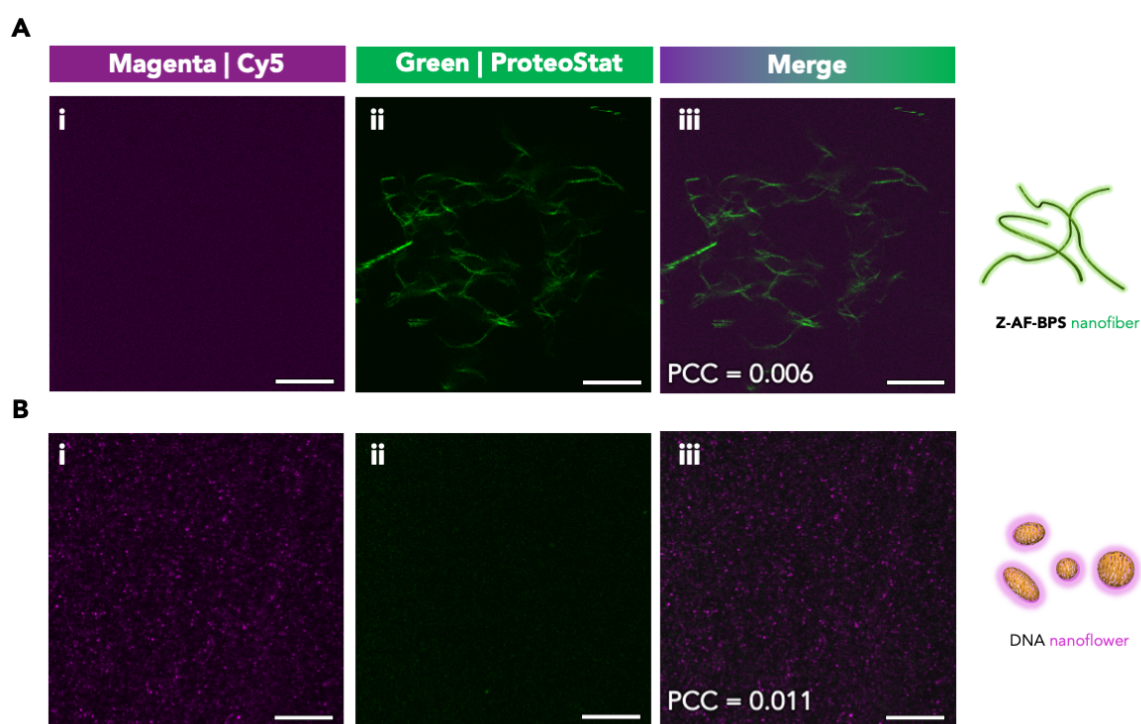
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3 **Fig. 4 (A)** CD spectra of the **Z-AF-BPS** hydrogel (0.40 wt%, 0.1-mm cell) and sol (0.040 wt%, 1.0-mm cell).
4 *Conditions:* 50 mM MES-NaOH (pH 7.0) containing DMSO (2.0 vol%). **(B)** IR spectrum of the **Z-AF-BPS**
5 hydrogel [5.0 wt%, prepared with D₂O (**Fig. S9B**) to detect the bands in the amide I region].

6 7 **Isothermal enzymatic construction of DNA nanoflowers through RCA in the presence of peptide-** 8 **based self-assembled nanofibers**

9 Next, we envisioned the isothermal enzymatic construction of aqueous hybrid materials by mixing the
10 newly developed, negatively charged **Z-AF-BPS** nanofibers with the DNA nanoflowers that were obtained as
11 RCA products. Notably, selective fluorescent staining dyes for each supramolecular architecture are indispensable
12 to visualizing **Z-AF-BPS** nanofibers and DNA nanoflowers individually and in their mixed states through *in situ*
13 confocal laser scanning microscopy (CLSM) observations.[31] In this study, ProteoStat [32], a commercially
14 available fluorescent molecular rotor dye (lacking an open chemical structure) for staining amyloid plaques, was
15 used to stain **Z-AF-BPS** nanofibers because of its robust selectivity demonstrated during the intracellular
16 fluorescence imaging of peptide-based aggregates, as reported by other groups.[33,34] Then, a fluorescent-dye-
17 labeled oligonucleotide (Cy5-ON, 12 nt), having a complementary sequence to an ssDNA part (which repeatedly
18 appears) and constructed by an RCA reaction, was employed to visualize DNA nanoflowers according to
19 previous reports.[35,36]

20 As shown in **Fig. 5Aii**, a fibrous morphology was observed using the ProteoStat green channel for **Z-**

1 **AF-BPS** nanofibers while **Fig. 5Bi** showed the successful observation of a particulate morphology under a Cy5
 2 magenta channel for DNA nanoflowers obtained by the RCA, according to the standard procedure reported
 3 previously (incubation was conducted at room temperature (rt, ~25 °C) for 4 h).[37,38] As the RCA template, a
 4 circular DNA (98 nt) was prepared, the formation of which was verified by PAGE (**Fig. S12**) and directly applied
 5 to the RCA reaction according to the previous report.[38] TEM observations of the RCA reaction products (*vide*
 6 *infra*) disclosed the formation of globular DNA nanoflowers, consistent with previous reports.[37,38] Most
 7 importantly, we found that the nonspecific staining was insignificant (**Fig. 5Ai**: Cy5-ON against **Z-AF-BPS**
 8 nanofibers, **5Bii**: ProteoStat against DNA nanoflowers) and supported by the low Pearson's correlation
 9 coefficient (PCC) values displayed in the merged channels (**Fig. 5Aiii** and **5Biii**).

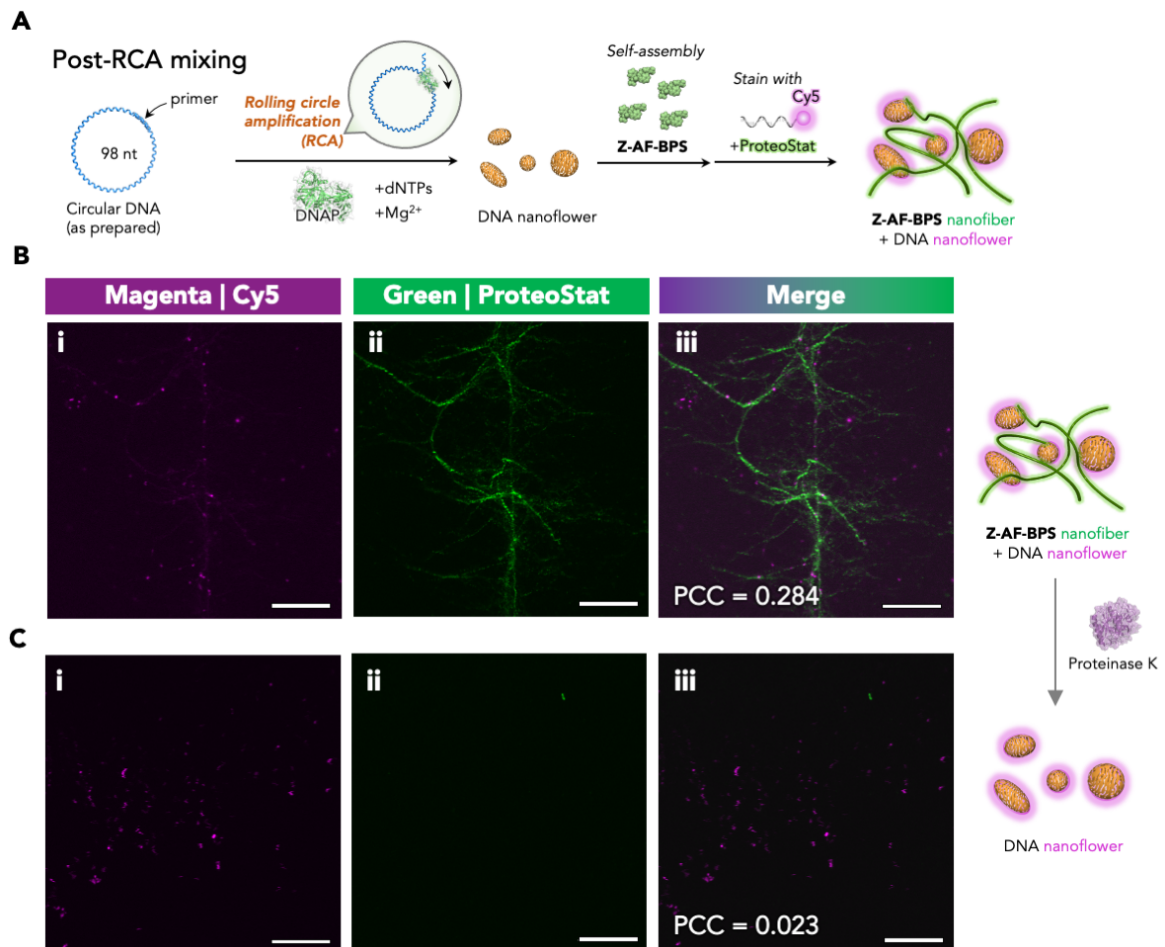


10
 11 **Fig. 5** Representative CLSM images showing (A) **Z-AF-BPS** nanofibers and (B) DNA nanoflowers obtained by
 12 the RCA and subsequently stained with ProteoStat and Cy5-ON. Details on sample preparation and CLSM
 13 observation procedures are described in the supplementary information. Scale bar: 10 μm . *Conditions:* (A) [**Z-AF-**
 14 **BPS**] = 8.2 mM, (B) [dNTPs] = 1.0 mM and [DNAP] = 1000 U/mL, staining with ProteoStat (1/500 dilution
 15 of the ProteoStat staining solution) and Cy5-ON (16 μM).
 16

17 Next, two protocols were designed to establish procedures for constructing new hybrid materials
 18 comprising **Z-AF-BPS** nanofibers and DNA nanoflowers. As outlined in **Fig. 6A**, pregrown DNA nanoflowers
 19 (typically at rt for 4 h) were mixed with a small amount of DMSO stock solution containing monomeric **Z-AF-**
 20 **BPS** to induce the formation of the **Z-AF-BPS** nanofibers for (i) the post-RCA-mixing protocol. In contrast,
 21 the RCA reaction (conducted at rt for 4 h) was performed to grow DNA nanoflowers in the presence of **Z-AF-**
 22 **BPS** nanofibers for (ii) the pre-RCA-mixing protocol (**Fig. 7A**). First, the hybrid materials prepared according

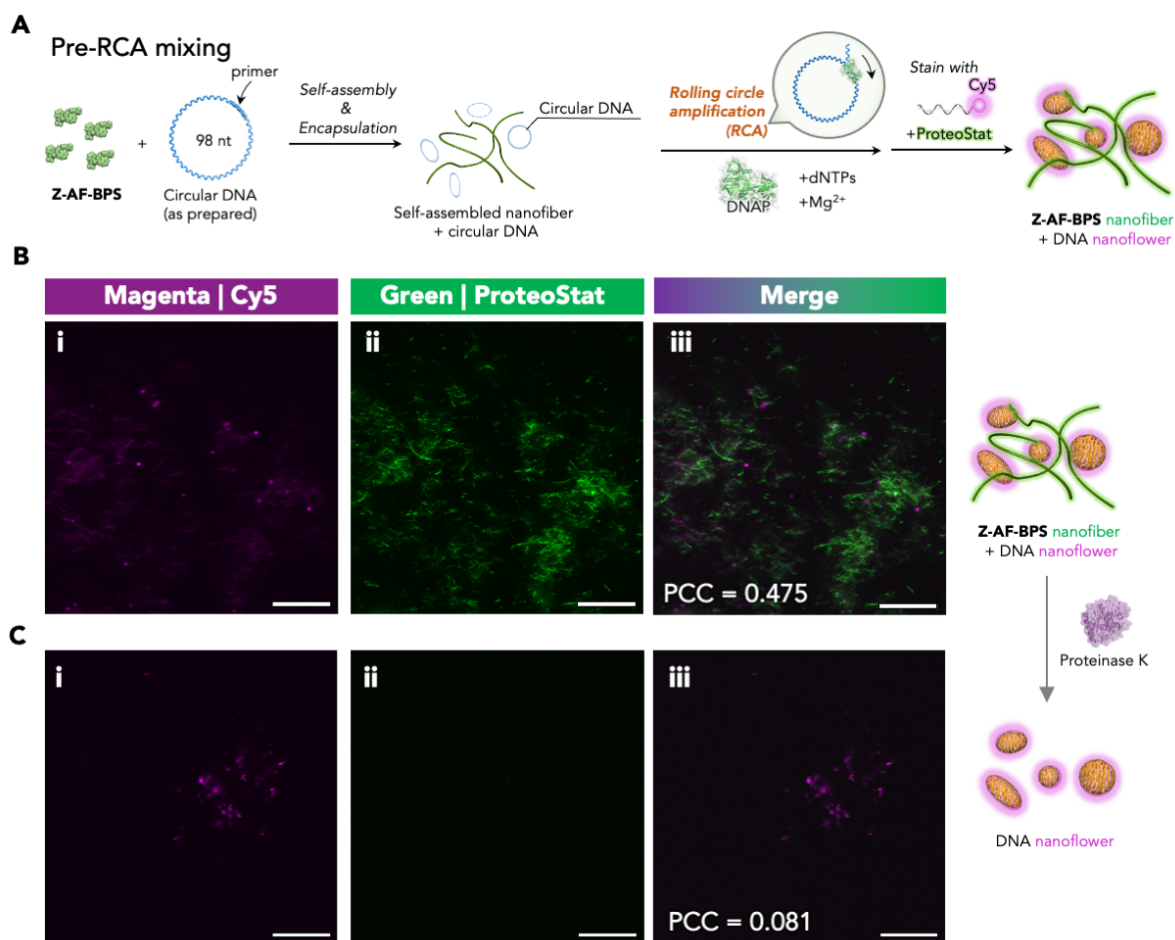
1 to (i) the post-RCA-mixing protocol were subjected to CLSM observations through the above-described
2 fluorescence staining method. As shown in **Fig. 6B**, fibrous morphology in the green channel and particulate
3 morphology in the magenta channel were individually observed, indicating the orthogonal coexistence of **Z-AF-**
4 **BPS** nanofibers and DNA nanoflowers obtained from the RCA reaction before the mixing. The PCC value was
5 evaluated to be 0.284, indicating an appreciably low correlation between the green and magenta channels.
6 Encouraged by this selective fluorescence staining results, even under the mixed state, we subsequently prepared
7 hybrid materials according to (ii) the pre-RCA-mixing protocol, which was, in fact, our original purpose as
8 depicted in **Fig. 1C**. As displayed in **Fig. 7B**, CLSM images comparable to those in (ii) the pre-RCA-mixing
9 protocol were obtained, the PCC value (0.475) was larger, a possible reason of which will be discussed later.
10 Nonetheless, this finding indicates that RCA reactions in the presence of **Z-AF-BPS** nanofibers orthogonally
11 proceeded to allow for the *in situ* production of DNA nanoflowers, which manifests the successful, isothermal
12 enzymatic construction of multicomponent supramolecular hybrid materials containing two distinct
13 supramolecular architectures (nanofibers and nanoflowers) assembled from their component peptides and
14 nucleic acids, respectively.

15 Subsequently, proteinase K [39] was added to the hybrid materials constructed according to the two
16 distinct protocols to evaluate biostimuli responsiveness and further dissect the orthogonal coexistence of the
17 DNA nanoflowers and **Z-AF-BPS** nanofibers.[17,18] We observed that although the fibrous morphology
18 visualized by the ProteoStat green channel disappeared almost entirely after 16-h incubation at 40 °C in the
19 presence of proteinase K, the particulate morphology in the Cy5 magenta channel remained as shown in **Figs.**
20 **6C** and **7C**. As anticipated, the selective degradation of the **Z-AF-BPS** nanofibers by proteinase K was evident.
21 Furthermore, this protease-selective degradation supports our view that fibrous architectures visualized by CLSM
22 observations can be constructed orthogonally from peptide derivatives (**Z-AF-BPS**) against the DNA-based
23 globular architectures.



1
 2 **Fig. 6** (A) The post-RCA-mixing protocol to construct hybrid materials comprising **Z-AF-BPS** nanofibers and
 3 DNA nanoflowers. Representative CLSM images of the **Z-AF-BPS** nanofibers and DNA nanoflowers were
 4 shown by staining with ProteoStat and Cy5-ON, respectively, (B) before and (C) after adding proteinase K (11
 5 mg/mL). Details on the sample preparation and CLSM observation procedures are described in the
 6 supplementary information. Scale bar: 10 μ m. *Conditions:* (B) [**Z-AF-BPS**] = 8.2 mM, [dNTPs] = 1.0 mM, and
 7 [DNAP] = 1000 U/mL, staining with ProteoStat (1/500 dilution of the ProteoStat staining solution) and Cy5-
 8 ON (16 μ M).

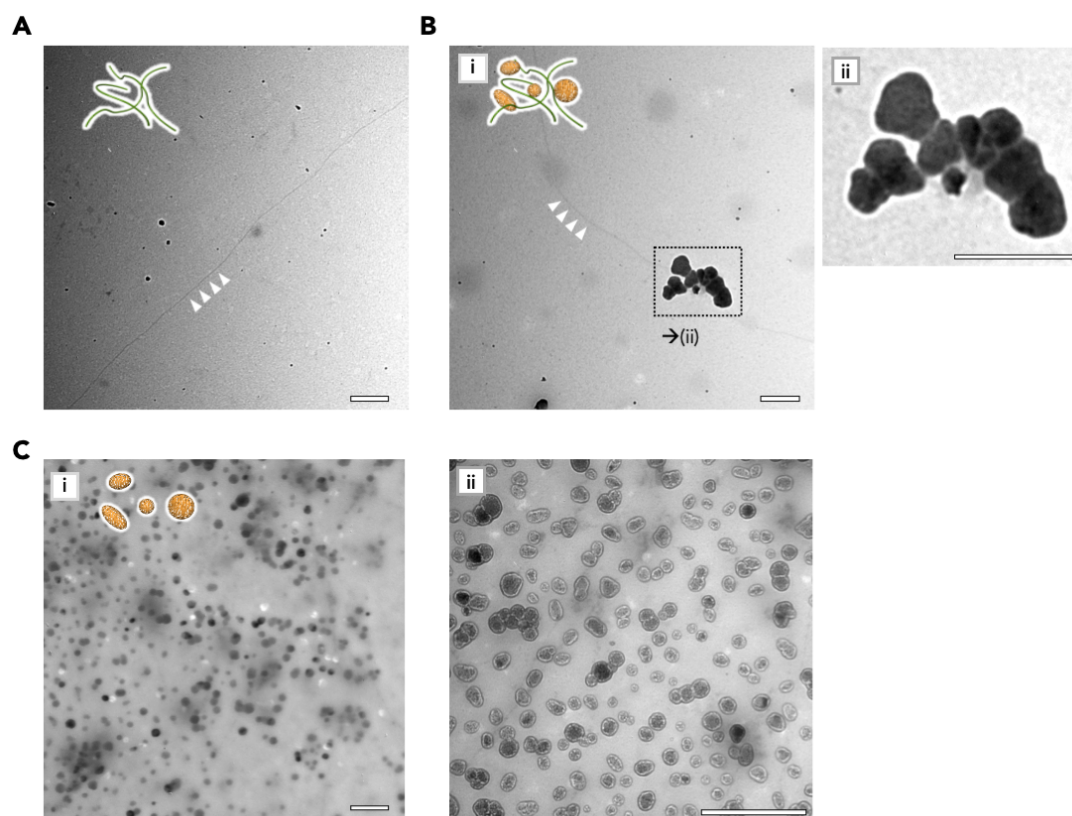
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1
2 **Fig. 7** (A) The pre-RCA-mixing protocol to construct hybrid materials comprising the **Z-AF-BPS** nanofibers
3 and DNA nanoflowers. Representative CLSM images of the **Z-AF-BPS** nanofibers and DNA nanoflowers were
4 shown by staining with ProteoStat and Cy5-ON, respectively, (B) before and (C) after adding proteinase K (11
5 mg/mL). Details on the sample preparation and CLSM observation procedures are described in the
6 supplementary information. Scale bar: 10 μ m. *Conditions:* (B) [**Z-AF-BPS**] = 8.2 mM, [dNTPs] = 1.0 mM, and
7 [DNAP] = 1000 U/mL, staining with ProteoStat (1/500 dilution of the ProteoStat staining solution) and Cy5-
8 ON (16 μ M).

9
10 Finally, TEM observations were conducted to obtain further insight into the orthogonal coexistence of
11 **Z-AF-BPS** nanofibers and DNA nanoflowers at a nanoscale. As shown in **Fig. 8A**, long thin nanofibers with a
12 length of several μ m and a diameter of several nm were discovered. Notably, the **Z-AF-BPS** nanofibers found
13 in these TEM images were significantly thin and scarcely entangled compared with those of the supramolecular
14 nanofibers constructed from the similar peptide derivatives reported previously by our group [25]. Hence, the
15 fibrous morphology coincided with the AFM images (**Fig. 3A**). More importantly, aggregated DNA nanoflowers
16 were often found along with **Z-AF-BPS** nanofibers (**Fig. 8B**). This structural nanoscale feature could be
17 correlated with marginally larger PCC values for the CLSM images (**Fig. 7B**). Moreover, although the sizes of
18 the individual DNA nanoflowers (ca. 50–200 nm) were comparable to those reported previously,[37,38,40,41]

1 they were more polydisperse. In contrast, as shown in **Fig. 8C**, less aggregated DNA nanoflowers, with the size
2 of 72 ± 18 nm ($n = 190$), were found in the absence of **Z-AF-BPS** nanofibers. Therefore, we presume that the
3 aggregation of DNA nanoflowers found frequently along with the **Z-AF-BPS** nanofibers could be facilitated by
4 attractive interactions between DNA nanoflowers and the **Z-AF-BPS** nanofibers and/or the surface of **Z-AF-**
5 **BPS** nanofibers could act as a template for the growth of DNA nanoflowers,[18] which could be mediated by a
6 divalent cation (Mg^{2+} under this conditions).



7
8 **Fig. 8** Representative TEM images showing (A) **Z-AF-BPS** nanofibers and DNA nanoflowers constructed by
9 the RCA reaction in (B) the presence (during the pre-RCA-mixing protocol, corresponding to **Fig. 7B**) and (C)
10 absence of **Z-AF-BPS** nanofibers. Magnified images for (B) and (C) are shown in panels (ii). Fibrous
11 architectures are highlighted using white arrows in panels (A) and (B). Schematics showing each supramolecular
12 architecture are also presented in the top left corner of the images. Scale bar: 500 nm.

13

14 **Conclusions**

15 We have successfully constructed multicomponent hybrid supramolecular materials comprising fibrous
16 nanostructures through the self-assembly of peptide derivatives and DNA nanoflowers using the RCA reaction
17 under isothermal conditions. To the best of our knowledge, such orthogonal coexistence of peptide-based
18 supramolecular nanostructures and DNA nanoflowers has not been reported. Furthermore, this study
19 demonstrated the protease-responsive, selective degradation of a peptide-based supramolecular nanofibers

1 embedded in hybrid supramolecular materials, which additionally supports the view that each supramolecular
2 architecture was constructed through the orthogonal assembly process from the component bio-related
3 molecules. We believe that this unique supramolecular hybrid (nano)materials could offer a distinct opportunity
4 of exploring bioapplications like cell-culturing or drug-releasing matrices. Research along such lines is in progress
5 in our laboratory.

6 7 **Conflicts of interest**

8 There are no conflicts to declare.

9 10 **Acknowledgments**

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