

Three-Dimensional Covalent Organic Frameworks with scu Topology

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ABSTRACT: Three-dimensional (3D) covalent organic frameworks (COFs) exemplify a new generation of crystalline extended solids with intriguing structures and unprecedented porosity. Notwithstanding substantial scope, the reticular synthesis of 3D COFs from pre-designed building units leading to new network topologies yet remains a demanding task owing to the shortage of 3D building units and inadequate reversibility of the linkages between the building units. In this work, by linking a tetragonal prism (8-connected) node with a square planar (4-connected) node, we report the first 3D COF with **scu** topology. The new COF, namely TUS-84, features a two-fold interpenetrated structure with well-defined porosity and a Brunauer–Emmett–Teller surface area of 679 m² g⁻¹. In drug delivery applications, TUS-84 shows efficient drug loading and sustained release profile.

An emerging class of porous organic materials developed from linking molecular building blocks with strong covalent bonds into crystalline, extended (two-dimensional) 2D and (three-dimensional) 3D structures called covalent organic frameworks (COFs)¹⁻¹⁴ have recently aroused great interest in catalysis^{15,16}, sensing^{17,18}, separation,¹⁹ semiconduction,²⁰ proton conduction,²¹ biomedicine^{22,23}, among others. COFs emerged in 2005²⁴ as the second series of reticular materials, the first one being metal-organic frameworks (MOFs)^{25,26}. ‘Reticular’ means ‘anything that has the structure of a net’. By reticular synthesis, we refer to the extended structure regime that combines (i) molecular-level control over matter and (ii) robustness.²⁷⁻²⁹ A top-down reticular synthesis scheme starts with a desired net topology followed by disassembling it into vertices and edges, finding secondary building units with the right connectivities and aligning them with the vertices, obtaining an augmented net by replacing the vertices of an *n*-connected net by a group of *n*-vertices, and finally linking the molecular building blocks by robust bonds into crystalline extended structures.^{1,30-34} Alternatively, the bottom-up scheme of reticular synthesis proceeds from pre-designed building units leading to unprecedented network topologies.³⁵ COFs feature one of the highest open-pore scaffolds. The COF scaffold is built out of organic units and it imparts tunable chemical environments for encapsulating a wide array of guest molecules.

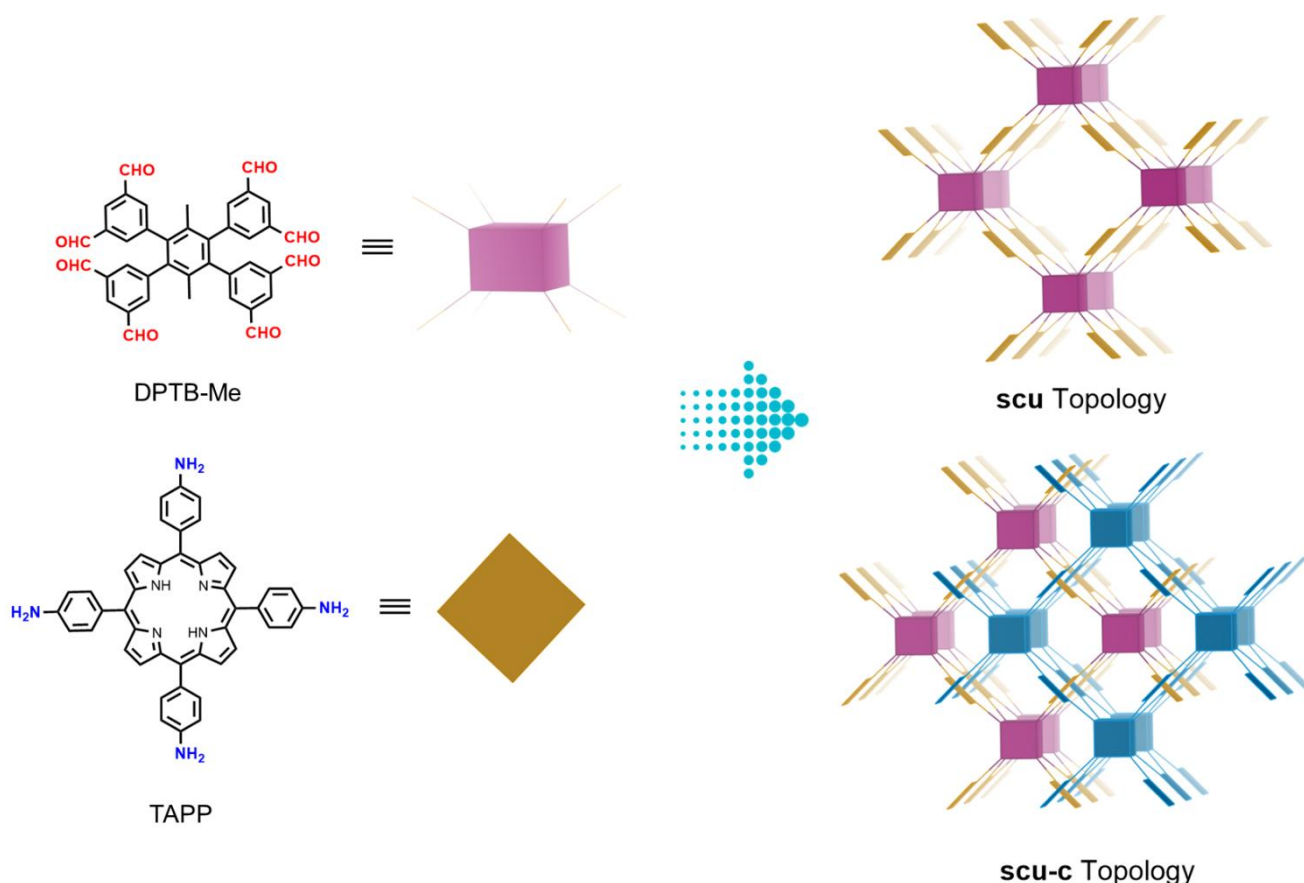
Based on the extension of their covalent connectivity, COFs can be categorized into 2D and 3D COFs.³⁶ With covalent

connectivity extending only in 2D, 2D COFs crystallize as layered structures in which the layers are stacked through non-covalent interactions (π - π stack, Van der Waals interactions, hydrogen bonds), giving rise to 1D straight channels.^{14,37} On the other hand, with covalent connectivity extending along the entire 3D scaffold, 3D COFs often have the upper hand over 2D COFs, attributed to their interconnected channels and readily accessible active sites.³⁸

Topological consideration is crucial to 3D extended structures considering that it largely dictates their pore architecture, active site formation and mass transport behavior.³⁸ Albeit highly sought after, discovery of new 3D COF topologies yet remains a herculean task because the highly-connected 3D organic building blocks are hard to come by and it is very difficult to solve the crystal structures. Thus far, the type of 3D topologies of COFs is limited to about 20.³⁸ The tetratopic (T_d)-based 3D COF nets are **bor**,³⁹ **ctn**,³⁹ **dia**,⁴⁰ **pts**,⁴¹ **rra**,⁴² **lon**,⁴³ and **ljh**.⁴⁴ Fang et al. and He et al. prepared several hexatopic (D_{3h})-based 3D COFs e.g. **stp**,⁴⁵ **acs**,^{46,47} **ceq**,^{47,48} and **hea**.^{49,50} Different from triangular prismatic (D_{3h}) nodes, Mateo-Alonso et al. utilized triangular antiprismatic (D_{3d}) nodes to construct **pcu** topology 3D COFs.⁵¹ Other unprecedented 3D COF topologies reported are **ffc**,⁵² **srs**,⁵³ **fjh**,⁵⁴ **tbo**,⁵⁵ and **nbo**.⁵⁶ Recently, octatopic nodes have been used to prepare **pcb**⁵⁷ and **bcu**⁵⁸ topology 3D COFs.

Herein, we report for the first time a novel 3D COF, namely TUS-84, with **scu** topology formed through the combination of a tetragonal prism (8-connected) node with a square

Scheme 1. Strategy for Constructing 3D COF with the scu Topology^a



^aThe condensation reaction of a D_{2h} -symmetric linker, DPTB-Me, and a C_4 -symmetric linker, TAPP, yielding 3D COF with a **scu-c** net, which belongs to **scu** topology.

planar (4-connected) node. TUS-84 exhibits a two-fold interpenetrated **scu** net, denoted as **scu-c** (c for catenated), with permanent porosity and a Brunauer–Emmett–Teller (BET) surface area of $679 \text{ m}^2 \text{ g}^{-1}$. Structural elucidation of the COF was carried out thoroughly through different characterization techniques. Interestingly, the COF shows efficient drug loading and extended-release profile in simulated physiological media. Scheme 1 depicts the strategic approach to construct 3D COFs with **scu-c** net via the [8+4] imine condensation reaction of a D_{2h} -symmetric linker, 4,4'-bis(3,5-diformylphenyl)-3,3',5,5'-dimethyl-[1,1':2',1''-terphenyl]-3,3'',5,5''-tetracarbaldehyde (DPTB-Me), and a C_4 -symmetric linker, 5,10,15,20-tetrakis(4-aminophenyl)porphyrin (TAPP).

TUS-84 was synthesized by the solvothermal reaction of DPTB-Me (19.0 mg, 0.03 mmol) and TAPP (40.48 mg, 0.06 mmol) in a 5:5:2 (v/v/v) mixture of mesitylene, 1,4-dioxane, and 6 M aqueous acetic acid under 120°C for 3 days. The acid-catalyzed Schiff-base condensation reaction yielded the COF as a dark purple crystalline solid at a yield of 76%. The solid-state ^{13}C cross-polarization magic angle spinning (CP/MAS) NMR and Fourier-transform infra-red (FT-IR) spectroscopies provided definitive evidence for atomic-level connectivity of the imine linkage in TUS-84. The ^{13}C CP/MAS NMR spectrum displayed a characteristic peak at 158 ppm for the imine carbon of TUS-84 (Figure

S1). In the FT-IR spectrum of TUS-84, the C=N vibration peak at 1625 cm^{-1} was observed. Significant attenuation of the N-H ($3433, 3465 \text{ cm}^{-1}$ for TAPP) and C=O (1703 cm^{-1} for DPTB-Me) stretching vibration bands in the FT-IR spectrum of TUS-84 implies high degree of polymerization for the imine COF (Figure S2). Isometric microcrystals of TUS-84 were observed from scanning electron micrographs (Figure S3). High-resolution transmission electron microscopy (HRTEM) imaging (Figure 1b,c, S4,5) showed the ordered structure of TUS-84, comprising rhombus pores viewed along the z-direction in the simulated structure (Figure 2a). Thermogravimetric analysis (TGA) curve indicates high thermal stability for TUS-84 retaining 95% of its weight up to 500°C (Figure S6). Chemical stability of the COF was substantiated from its preservation of crystallinity and imine linkage after treatment with organic solvents, water, and aqueous HCl and NaOH solutions, as can be seen from the PXRD profiles (Figure S7) and FT-IR spectra (Figure S8).

The crystal structure of TUS-84 was unraveled by powder X-ray diffraction (PXRD) analysis combined with structural modeling and simulation (Figure 1a). Geometry optimization (energy minimization) was performed in *Materials Studio 7.0*⁵⁹ Forcite program that afforded the unit cell parameters of TUS-84 with a **scu-c** net and Pm space group as $a = 39.9205 \text{ \AA}$, $b = 18.7162 \text{ \AA}$, $c = 23.6564 \text{ \AA}$, $\alpha = \beta = \gamma =$

90°. The simulated PXRD pattern (Figure 1a, green curve) showed great alignment with the observed diffraction pattern (Figure 1a, red dots). Sharp Bragg peaks observed at 4.38 and 6.44° correspond to the (200) and (111) facets, respectively and relatively weak peaks at 9.10, 10.23, 11.15, 12.83, 14.38 and 16.19° correspond to the (112), (020), (221), (222) (130) and (132) facets, respectively (Figure 1a). Pawley refinement was applied against the experimental PXRD data using Reflex that resulted in a space group of *Pm* with unit cell parameters $a = 39.9179 \text{ \AA}$, $b = 18.7054 \text{ \AA}$, $c = 23.6772 \text{ \AA}$, $\alpha = \beta = \gamma = 90^\circ$ and good agreement factors $R_p = 4.37\%$, $R_{wp} = 3.19\%$. The Pawley refined PXRD pattern (black curve, Figure 1a) shows good consistency with the experimental PXRD pattern (red dots), as indicated by the minor difference plot (blue curve). Furthermore, we also explored alternative topologies for TUS-84, including the non-interpenetrated **scu** net (Figure S16, Table S2), and **csq** topology (Figure S17, Table S3). However, the simulated PXRD patterns did not accord with the experimental PXRD pattern. All things considered, we propose the **scu-c** net for TUS-84.

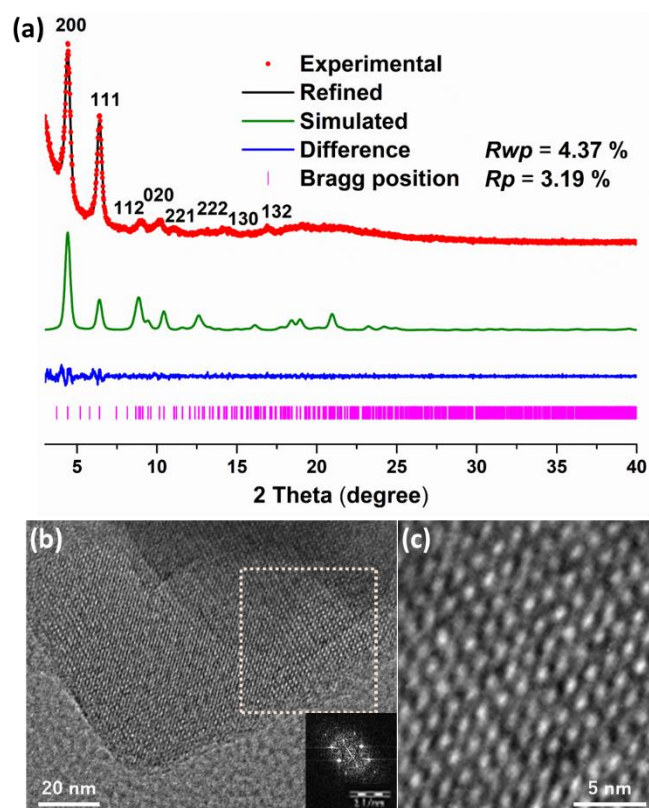


Figure 1. (a) PXRD patterns of TUS-84: experimental pattern (red dots), Pawley refined (black curve), simulated (green curve) pattern from the **scu-c** modeled structure, and the difference plot (blue curve) between the experimental and refined patterns. The Bragg positions are denoted by magenta ticks. (b,c) HRTEM images of TUS-84. Inset in Figure b shows the fast Fourier transform (FFT) pattern acquired from the area enclosed by the white box.

The permanent porosity of TUS-84 was ascertained by N_2 sorption measurements on activated COF sample at 77 K. As can be observed in Figure 3a, TUS-84 displayed a reversible type-I isotherm with a sharp uptake at low pres

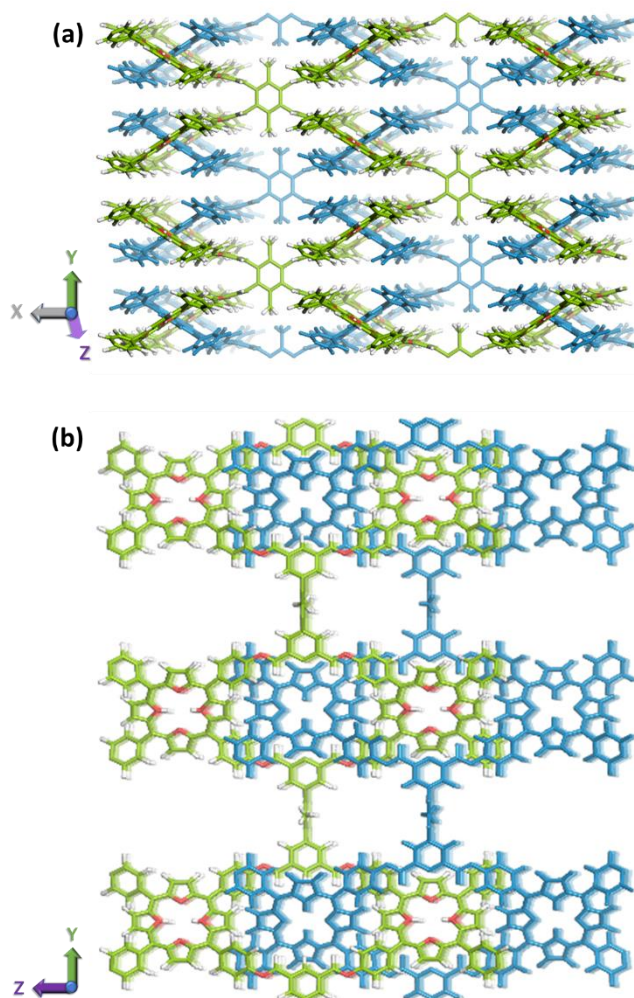


Figure 2. Extended structures of TUS-84.

sure ($P/P_0 < 0.1$), indicative of its microporous character. The BET specific surface area of TUS-84 was evaluated as $679 \text{ m}^2 \text{ g}^{-1}$ (Figure S9). Applying the nonlocal density functional theory (NLDFT) method, the pore volume of TUS-84 was derived as $0.7613 \text{ cm}^3 \text{ g}^{-1}$ and its pore size distribution was calculated as 0.97 nm (Figure 3b), consistent with the pore size predicted from the simulated structure (1.05 nm). We also evaluated the H_2 , CO_2 and CH_4 gas adsorption capacities of TUS-84 to reinforce its prospects in carbon capture and clean energy applications. As illustrated in Figure S10, the H_2 uptake capacities at 77 and 87 K under 1 bar are $131 \text{ cm}^3 \text{ g}^{-1}$ and $88 \text{ cm}^3 \text{ g}^{-1}$, respectively. The isosteric enthalpy of adsorption (Q_{st}) of H_2 was calculated to be 6.8 kJ mol^{-1} (Figure S11). TUS-84 shows a CO_2 uptake capacity of $55 \text{ cm}^3 \text{ g}^{-1}$ and $31 \text{ cm}^3 \text{ g}^{-1}$ at 273 and 298 K, respectively, under 1 bar (Figure S12). The Q_{st} of CO_2 adsorption was evaluated as 24.9 kJ mol^{-1} (Figure S13). The CH_4 sorption isotherms shown in Figure S14 reveal an uptake capacity of $14 \text{ cm}^3 \text{ g}^{-1}$ and $10 \text{ cm}^3 \text{ g}^{-1}$ at 273 and 298 K under 1 bar, respectively, and the value of Q_{st} was obtained as 6.5 kJ mol^{-1} (Figure S15).

Intrigued by the 3D functional scaffold with permanent porosity and high chemical stability, we utilized TUS-84 in *in vitro* drug delivery studies. Ibuprofen is one of the most common nonsteroidal anti-inflammatory drugs (NSAIDs)

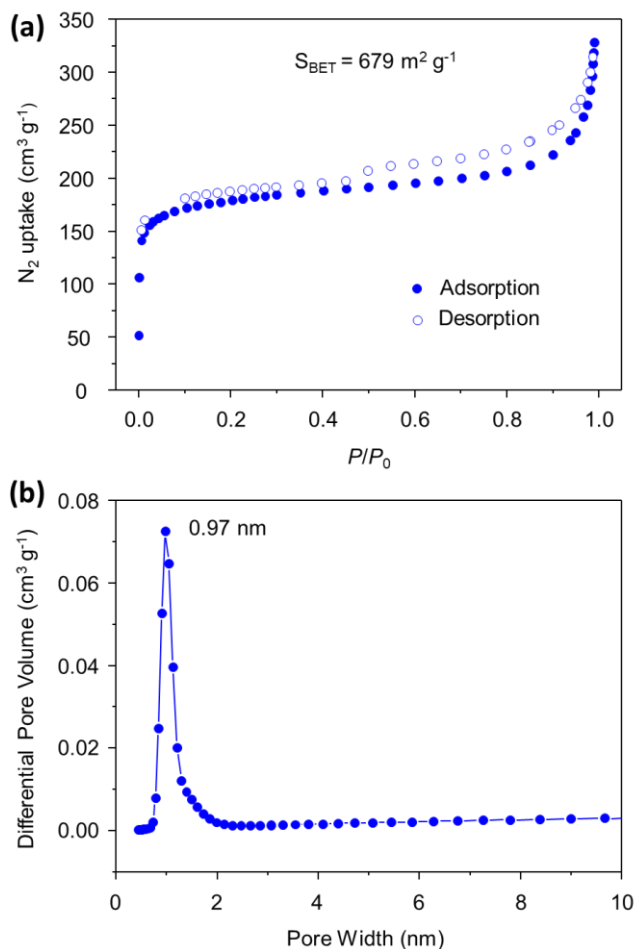


Figure 3. (a) Nitrogen sorption isotherms and (b) pore size distribution profile of TUS-84.

used for the treatment of rheumatoid arthritis, osteoarthritis, mild-moderate pain and primary dysmenorrhea.⁶⁰⁻⁶² The selection of ibuprofen as the drug in this study is based on: (a) ibuprofen has short half-life (1.8-2.0 h) that calls for extended-release formulations,¹ and (b) pore dimensions of TUS-84 (1.05 nm) is befitting for encapsulation of ibuprofen with molecular size of $0.5 \times 1 nm^2$.^{63,64}

For drug loading, 50 mg of TUS-84 was suspended in 30 mL of 0.1 M hexane solution of ibuprofen under magnetic stirring for 5 h. The drug-loaded COF sample was isolated from suspension via vacuum filtration, washed with hexane, and subsequently dried at room temperature. 1.0 mL of the filtrate was collected and 50 times diluted to evaluate the loading amount of ibuprofen using a UV-Vis spectrophotometer by measuring the absorbance at 261 nm of ibuprofen in hexane and supernatant (see SI for details). UV-Vis absorption data showed 11.05 wt% loading of ibuprofen in TUS-84 (Figure S23). As can be seen from Figure S21, the value of drug loading amount is in good agreement with that obtained from the TGA (11 wt%). PXRD analysis (Figure S18) and scanning electron micrograph (Figure S19) of ibuprofen-loaded TUS-84 revealed that the COF crystalline structure was retained after drug loading. The drug release study was performed by placing 40 mg of the drug-loaded TUS-84 sample inside a semipermeable

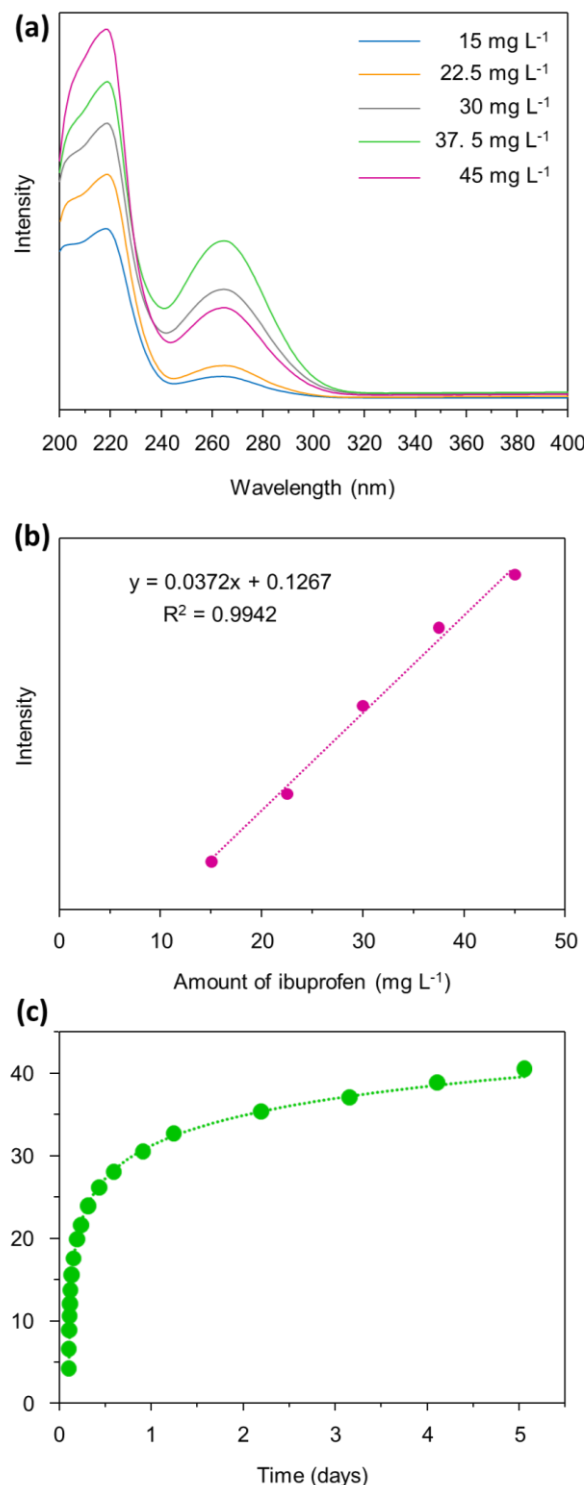


Figure 4. (a) UV-Vis spectra of ibuprofen in simulated body fluid (pH 7.4, phosphate buffer solution) at different concentrations. (b) Calibration curve of ibuprofen. (c) Release profile of ibuprofen from ibuprofen-loaded TUS-84 and corresponding fitting curve.

bag followed by immersing in 10 mL of phosphate buffer solution (simulated body fluid, pH 7.4) at a constant temperature of 37 °C. The dissolution solvent was taken out at specified time intervals for evaluation of the ibuprofen concentration and replenished with 10 mL of fresh buffer

solution. The ibuprofen concentration was determined UV-Vis spectrophotometrically using calibration curve (Figure 4a,b). TUS-84 showed an extended drug release performance of about 40% after 5 days (Figure 4c). This long-acting ibuprofen formulation could deliver sustained concentrations of drug over a prolonged period of time, thereby reducing dosing frequency and ensuring more consistent control of long-lasting pains.^{65,66}

To conclude, a 3D COF with a novel **scu-c** topology was designed and synthesized, utilizing a D_{2h} -symmetric linker, DPTB-Me, and a C_4 -symmetric linker, TAPP. The resultant TUS-84 COF displays an ordered microporous structure with high crystallinity and excellent stability. Furthermore, TUS-84 shows great promise as drug delivery vehicle owing to its efficient drug loading and controlled release behavior. This study may not only expand the library of 3D COF topologies but also facilitate the design of new 3D COF structures for biomedical applications.

ASSOCIATED CONTENT

Supporting Information.

Materials and characterization; Synthetic procedures; Solid-state ^{13}C CP/MAS Nuclear Magnetic Resonance (NMR) spectroscopy; Fourier-Transform Infrared (FT-IR) spectroscopy; Scanning electron microscopy (SEM) and transmission electron microscopy (TEM) images; Thermogravimetric analysis (TGA); Chemical stability test; N_2 adsorption; Structure simulations and X-ray diffraction analyses; Crystallographic information.

This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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