

1 **Ultra-stable Transformer Oil nanofluids**  
2 **with significant AC Breakdown Voltage**  
3 **enhancement at ultra-low filler fraction of**  
4 **functionalized boron nitride nanosheets**

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**33 ABSTRACT**

34 This paper reports the purposeful fluorination of hexagonal boron nitride (h-BN) nanofillers  
35 and its impact on reinforcement of AC breakdown strength and stability of transformer oil (TO)  
36 nanofluid. Fluorine functionalized boron nitride nanosheets (f-BNNs) of 5 nm thickness was  
37 synthesized in house via wet synthetic exfoliation route of pristine h-BN utilizing Ammonium  
38 Fluoride (NH<sub>4</sub>F) as the shedding agent. This promotes attachment of some highly electro-  
39 negative fluorine atoms to boron. This tailored f-BNNs exhibit a diminished band gap and  
40 induced electrical conductivity which helps in elevating the AC breakdown Voltage to 26 %-  
41 20% and a surge in resistivity at appreciably low nanofiller fraction of 0.005-0.01wt. %. These  
42 noteworthy improvement of electrical insulation properties compared to the state-of-the-art  
43 Boron nitride nanoparticles or nanosheets is explained by the parallel role played by fluorine  
44 in charge trapping as well as the role played 2D morphology of the nanofillers. Here, fluorine  
45 facilitated extrinsic energy bands in the oil-nanofiller interface acted as efficient charge  
46 trapping sites and helped to accumulate large quantities of streamer charges, more than h-BN  
47 nanosheets or BN nanoparticle for a longer time and improved the electric insulation properties  
48 to a large extent. The ultra-high steadiness of the nanofluid is also observed at these lower filler  
49 concentrations. 2D morphology, lipophilicity and electro-negativity induced electrostatic  
50 repulsion between the f-BNNs nanosheets are attributed to achieve this alluring property of the  
51 nanofluid. Such significant improvements at very low filler fractions justifies the fluorination  
52 of hexagonal boron nitride as a novel idea and a better alternative among all the reported BN  
53 brothers for high voltage applications of nano engineered liquid insulation.

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64 *Keywords:* Fluorine functionalization, f-BNNs, AC Breakdown, Electronegativity, stability

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## 66 1. Introduction

67 Since their advent by Choi in 1995 [1], nanofluids have played extensive role in advanced  
68 cooling systems such as electronic industry, automobile sector to nuclear power plants [2] and  
69 etc. In the electrical power industry, the concept of nanofluids was introduced primarily to  
70 improve the electrical properties (dielectric breakdown strength, impulse breakdown voltage,  
71 PDIV, etc.) of insulating fluids (mineral oil, transformer oil & alternatives) so that the rising  
72 problems of liquid insulation failures could be taken good care of. The registration of the  
73 research on nanodielectric fluids can be dated back to the work of Segal *et. al.* [3] where  
74 magnetic nanoparticles (spherical,  $\text{Fe}_3\text{O}_4$ ) were used for the aim of improvement of dielectric  
75 breakdown (DB) properties of mineral oil. It was seen that the dielectric breakdown strength  
76 (DBS) of mineral oil was dependent on external magnetic field direction [3]. Later, several  
77 non-magnetic nanoparticles such as  $\text{TiO}_2$ ,  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$ , BN & etc gained their due importance  
78 in this field [4-7]. In the last decade, along with electrical insulation, the idea of dielectric  
79 nanofluids was extended towards improving the cooling performance of insulating fluids [8].  
80 Parallely, research using some alternative oils as base oil such as ester/ vegetable oils were  
81 conducted due to poor biodegradability of mineral oil [9].

82 The mechanism of electrical breakdown is extremely complicated. It depends upon various  
83 parameters of base fluid (moisture content, viscosity, temperature) [10], measurement  
84 protocols [11], purity [12] etc. Electrical breakdown in nanofluid is a surface phenomenon [4].  
85 The oil-nanomaterial interface plays a huge role in enhancing BDV by acting as a scavenger  
86 of streamer charges [4]. Hence, in nanofluids, apart from the previously mentioned factors,  
87 properties of the nanofiller such as conductivity, permittivity, morphology, size of nanofiller  
88 own their respective influences. For example, Mohammad *et al.* [13] compared the AC BDV  
89 and viscosity of natural ester oil based three types of nanofluids. The ester oil-based nanofluids  
90 were prepared by mixing palm fatty acid ester (PFAE) oil with three types of nanoparticles  
91 (conducting  $\text{Fe}_3\text{O}_4$ , semi-conducting  $\text{TiO}_2$  and insulating  $\text{Al}_2\text{O}_3$ ) of size, 15-20 nm at a  
92 concentration of 0.01 g/L The ferrofluid was the best to show a 43 % improvement in AC BDV.  
93 Similarly, Wenxia Sima *et al.* [14] studied the relative performances of conducting  $\text{Fe}_3\text{O}_4$ ,  
94 semiconducting  $\text{TiO}_2$ , and insulating  $\text{Al}_2\text{O}_3$  nanoparticles with size,  $10 (\pm 0.1)$  nm, modified by  
95 Oleic acid, Stearic acid and Span 80 in TO. Due to the differences in the conductivity and the  
96 dielectric constant, the rate of electron scavenging at the interface was found to be different for  
97 different material. This led to different degree of improvement in impulse BDV.  $\text{Fe}_3\text{O}_4$ ,  $\text{TiO}_2$  &  
98  $\text{Al}_2\text{O}_3$  showed almost 44, 33 & 35 % increase in impulse BDV at 0.003, 0.006 & 0.006 Wt. %

99 respectively. Hence, in the context of conductivity and permittivity, high conductive & low  
100 relaxation time of the conducting nano fillers enhance the breakdown properties to a large  
101 extent at ultra-low concentrations [15]. In line with this, some of the recent studies with ultra-  
102 high conducting fillers such as CNT, graphene [15] have been reported. On the other hand,  
103 semi-conducting [16, 17] and insulating [18] fillers require moderate to high filler fraction to  
104 achieve the same performances. At relatively higher filler fraction, due to large attractive van  
105 der Waals forces among nanomaterials, the agglomeration accelerates and the long-term  
106 stability of the NFs are questionable. Many researchers have demonstrated that stability of NFs  
107 using of surfactants like oleic acid have been enhanced due to static hindrance [19]. But long-  
108 term functionality of these long chain molecule are doubtful, as loosely surrounds over the  
109 fillers [19, 14]. Moreover, these surfactants are corrosive in nature and not suitable candidate  
110 for transformer core.

111 On the other hand, morphology of the nanofillers also play a significant role to improve the  
112 charge trapping capacity and other thermo physical properties [17]. Hence, search for  
113 customized nanofillers is the need of the hour to address the issues of insulating and thermal  
114 properties along with long-term stability of the NFs. Recent development of two-dimensional (2D)  
115 layered materials presents a unique opportunity to address several of the aforementioned problems.  
116 Properties of these 2D materials range from metallic ( $\text{NbS}_2$ ,  $\text{VSe}_2$ ), conducting (graphene), semi-  
117 metallic ( $\text{WTe}_2$ ,  $\text{TiSe}_2$ ), semiconducting ( $\text{MoS}_2$ ,  $\text{WS}_2$ ) to insulating (h-BN,  $\text{HfS}_2$ ); and their  
118 properties can even be tailored to suit specific base fluids [20]. Owing to their 2D nature, these  
119 nanofillers have extremely high surface to volume ratio and thus can be operated at much lower  
120 filler fractions to obtain targeted outcomes. The layered structure gives ultra-thinness to these  
121 materials so that they are easily stable in the base fluids [20, 21]. Hence, a proper tailor made 2D  
122 nanomaterial would lead to design an application specific nanofluid. Among many 2D  
123 nanomaterials, h-BN is highly insulating with outstanding thermal conductivity, first-rate  
124 mechanical steadiness, and prominent chemical dullness. Du *et al.* [7] demonstrated that BN  
125 nanoparticle improves thermal responses, electrical insulation and BDV of TO. However, BN  
126 particles tend to agglomerate easily [7]. Later, Bhunia *et al* [8] &Tijerina *et al* [21] pointed out that  
127 chemically exfoliated h-BN nanosheets, owing to their 2D structure and ultra-thinness with large  
128 surface exposal of (002) planes can be a better choice for designing next-generation TO NFs. As  
129 an extension to the previous study on h-BN nanosheets as well as recalling he extraordinary  
130 performances of conducting nanofillers at ultralow concentrations [15], the present work is a novel  
131 approach to explore the effect of a tailor made, 2D conducting nanofiller in TO. Conducting and

132 electro-negative fluorine functionalized h-BN nanosheets called fluorinated boron nitride  
133 nanosheets (f-BNNs) has been picked up for this purpose. Novelty of the investigation lies on the  
134 intended fluorination of h-BN Nanosheets for band engineering to achieve elevated AC breakdown  
135 voltage at very low filler fraction as compared to h-BN. Eventually the diluted filler fraction and  
136 electro-negativity of the functionalised material has improved the stability of the TO NF to a great  
137 extent.

138 The following sections discusses the methods employed and the experiments performed. The  
139 outcomes of the results are discussed elaborately with suitable references. Schematics are used  
140 to discuss the mechanisms involved.

## 141 **2. Materials and Method**

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### 143 *2.1 Synthesis*

144 All the required chemicals were bought from Sigma Aldrich and were used without further  
145 purification. Ammonium Fluoride ( $\text{NH}_4\text{F}$ ) and bulk hexagonal boron nitride (h-BN) powder  
146 ( $\sim 1\ \mu\text{m}$ ) were taken in certain proportions in a mortar and pestle (in 5:1 ratio) and grounded to  
147 homogeneous form. Next, 20 mL distilled water (DI) was added to it and mixed thoroughly.  
148 The solution was afterwards transferred to Teflon lined stainless steel autoclave and then heated  
149 at  $180^\circ\text{C}$  for 24 h. After natural cooling down, the resultant product was washed with DI until  
150 the pH reduced to neutral. Thereafter, the sample was dried in a petri dish at  $80^\circ\text{C}$  for 24 h in a  
151 preheated oven. Finally, white powder of f-BNNs was collected and stored for further  
152 characterization.

153 The formation of f-BNNs from commercial BN is achieved by fluorinated cum exfoliated  
154 route. The consumed mediator for exfoliation, i.e.,  $\text{NH}_4\text{F}$  breaks up into  $\text{F}^-$  and  $\text{NH}_4^+$  ions.  
155 Now,  $\text{F}^-$  being highly electro-negative and reactive, causes some B atoms of BN to come out  
156 of its basal planes disturbing the regular  $\text{SP}^2$  configuration of BN. This causes edge buckling  
157 in bulk h-BN and results in successful scrolling of individual sheets to reduce surface tension.  
158 The edge scrolling enables the diffusion and intercalation of  $\text{NH}_4^+$  ions.  $\text{NH}_4^+$  ions having  
159 relatively larger size than inter planer space can increase the distance between h-BN stacks  
160 with a motion perpendicular to the BN plane with high mobility. The resultant mechanism  
161 results in exfoliation and fluorination of few layers h-BN sheets. The mechanism is  
162 schematically expressed in **Fig. 1a**.

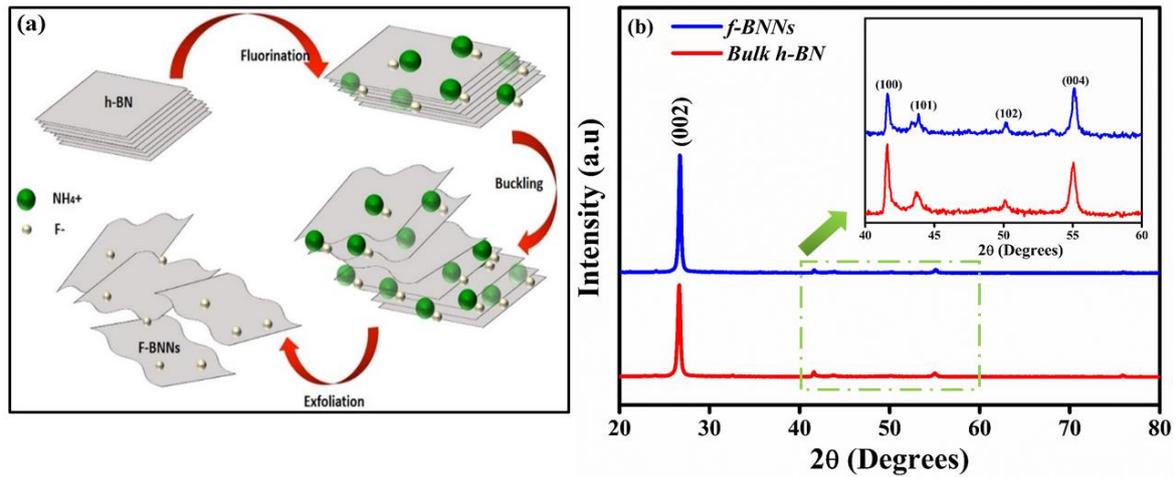


Fig. 1 (a) Step by step exfoliation route & (b) XRD of f-BNNs

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## 165 2.2 Characterization and Experimentation

166 The phase purity of the synthesized samples was studied by recording X-ray diffraction (XRD)  
 167 pattern (via *Bruker D-8 Advance Diffractometer, operating at 40 kV, 40 mA*) using Cu-K $\alpha$   
 168 radiation ( $\lambda=1.54$  A). Characterization and identification of the functional groups were  
 169 performed by *Shimadzu IR PRESTIGE 21* spectrometer within the range of 500-4000  $\text{cm}^{-1}$ .  
 170 Raman spectra of the synthesized samples were recorded by *WI Tec Raman spectrometer, Nd:*  
 171 *YAG* laser with an excitation wavelength of 532 nm. The samples were further characterized  
 172 by high resolution transmission electron microscope (HRTEM), *JEOL 2100 (Operating*  
 173 *voltage: 200 kV)* and Field emission Scanning Electron Microscope (FESEM), *ZEISS* for  
 174 determining its nanostructures. The Energy Dispersive X-Ray Analysis (EDAX) was carried  
 175 out (in the FESEM machine) for compositional study of the material. X-Ray Photo Electron  
 176 Spectroscopy (XPS) analysis (using monochromatic Al K $\alpha$  ( $h\nu = 1486.6$  eV) as X-ray source  
 177 and a hemispherical analyser (SPECS, HSA 3500)) was conducted to identify the chemical  
 178 states. Tests of AC breakdown voltages of the f-BNNs-Transformer oil nanofluids were carried  
 179 out using Neo-Tele-Tronix made 120 kV, 50 Hz automatic breakdown voltage testing  
 180 equipment. The Tan Delta and Resistivity of the nanofluids were measured by *ADTR 2K Plus*  
 181 instrument, supplied by ELTEL industries. The temperature dependent thermal conductivity of  
 182 nanofluids at various filler fractions were carried out by KD2 probe thermal Property analyser  
 183 (*Decagon Device Inc., model KD2 Pro, KSI* probe for liquids) following a lab set up similar  
 184 to [22]. The surface temperature of the nanofluids were recorded by infrared thermal imager,  
 185 Ray Cam 1886. The stability of the nanofluids over time were observed by a red laser torch.

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### 187 3. Results and Discussions

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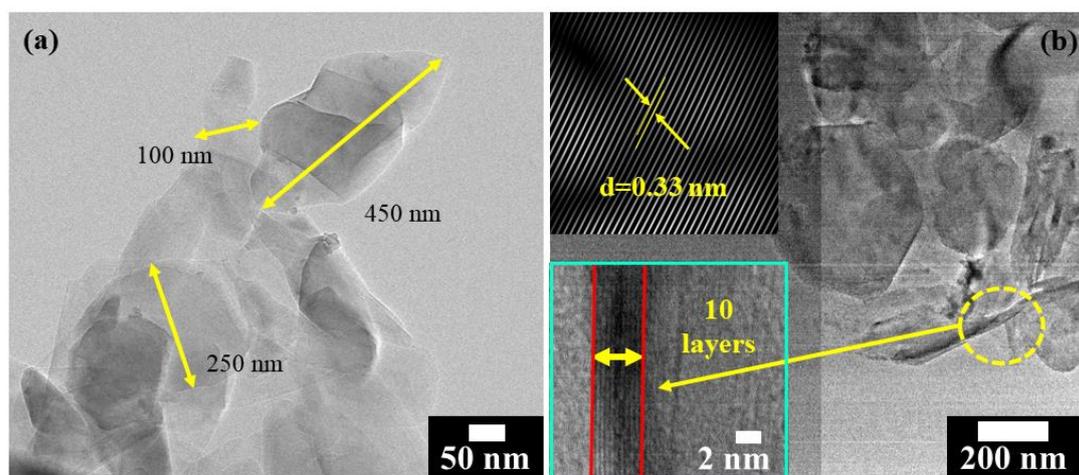
#### 189 3.1 XRD and TEM Analysis

190

191 The XRD pattern of f-BNNs is presented in **Fig. 1b** along with that of bulk BN powder. Peaks  
 192 observed at  $2\theta$  values of  $26.2^\circ$  and  $42.8^\circ$  were assigned to diffraction originating from (002)  
 193 and (100) planes of h-BN. From relative intensity ratio it can be concluded that the sample is  
 194 having (002) preferred orientation agreeing well with JCPDS Card number 340421. The  
 195 exposed (002) crystal surface of h-BN resembles the (002) plane of graphite [8, 20] -. In case  
 196 of f-BNNs, presence of no other peak indicated its phase purity. An estimation of the intensity  
 197 ratio of peaks corresponding to (004) & (100), ( $I_{004}/I_{100}$ ) plane (inset of **Fig. 1b**) shows a  
 198 higher ratio for f-BNNs. Hence exfoliation have worked well on (004) plane [23].

199

200 The TEM images as shown in **Fig. 2** reveals the morphology. It indicates the presence of  
 201 puckered sheet like structures similar to graphene with an average size approximately of 100-  
 202 450 nm.



203

204

**Fig. 2** TEM images of f-BNNs

205 Further, the crystalline nature of the nanosheets is clearly observed in the lattice imaging (inset,  
 206 **Fig. 2b.**) obtained by HRTEM. The  $d$  spacing (top inset of **Fig.2b**) as calculated by *Image J*  
 207 was 0.33 nm, which corresponds to (002) plane of XRD plot. The thickness of the exfoliated  
 208 sheets were around 5 nm, consisting of 10-12 layers (bottom inset of **Fig. 2b**).

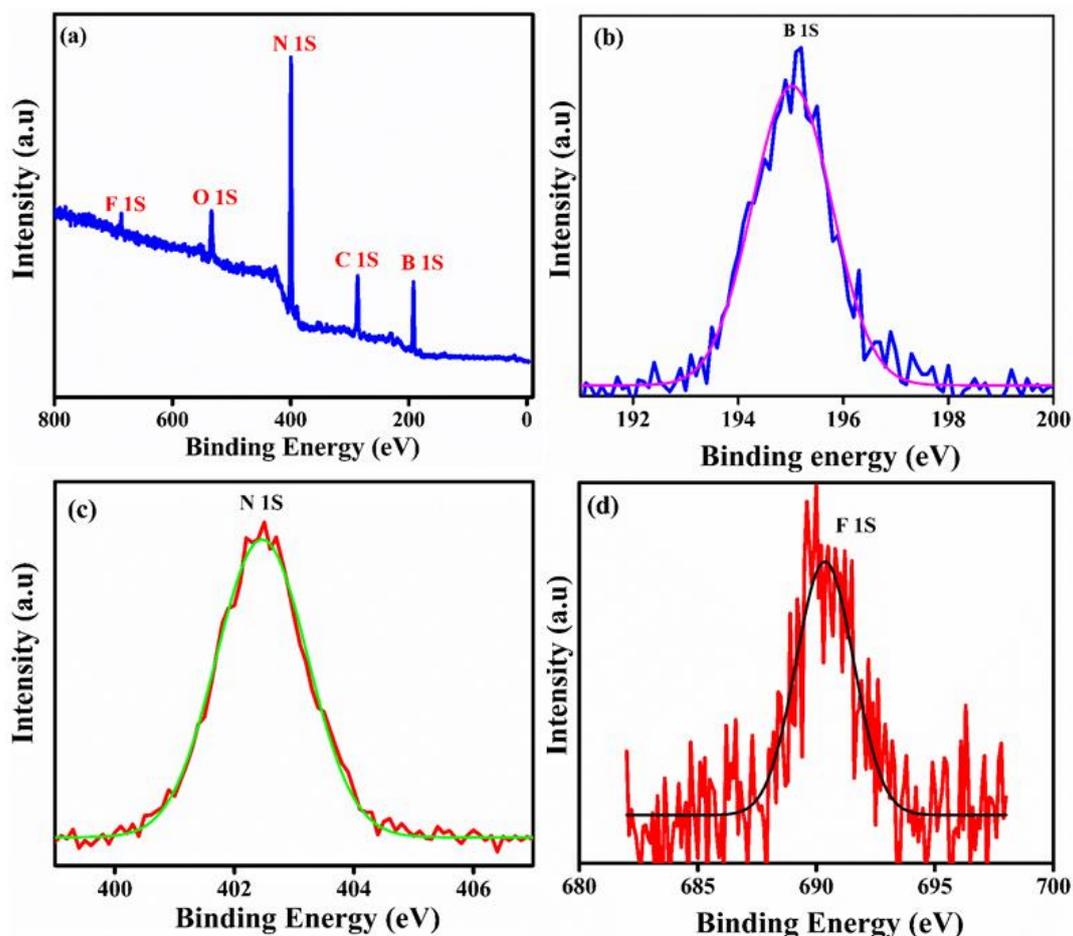
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#### 210 3.2 XPS, RAMAN & FTIR Analysis:

211

212 To identify the chemical states of f-BNNs, characterization of f-BNNs by XPS was performed  
 213 (**Fig. 3**). The XPS survey plot is depicted in **Fig. 3a** whereas the high resolution (HR) scan for

214 B 1s, N 1s, and F 1s are presented in **Fig.3b-3d**, after charge correction considering the C1s  
 215 peak at 284.6 eV. Oxygen and carbon peaks shown in survey plot most likely resulted from the  
 216 exposure of f-BNNSs to air or during the XPS measurement groundwork. The detected binding  
 217 energy of B 1s, N 1s, and F 1s are at 195, 402.4, and 690.33 eV. The measured values are in  
 218 good agreement with the previous reports [23]. The presence of fluorine is further supported  
 219 by EDAX analysis provided in the *SI* and FTIR analysis described in the next section.



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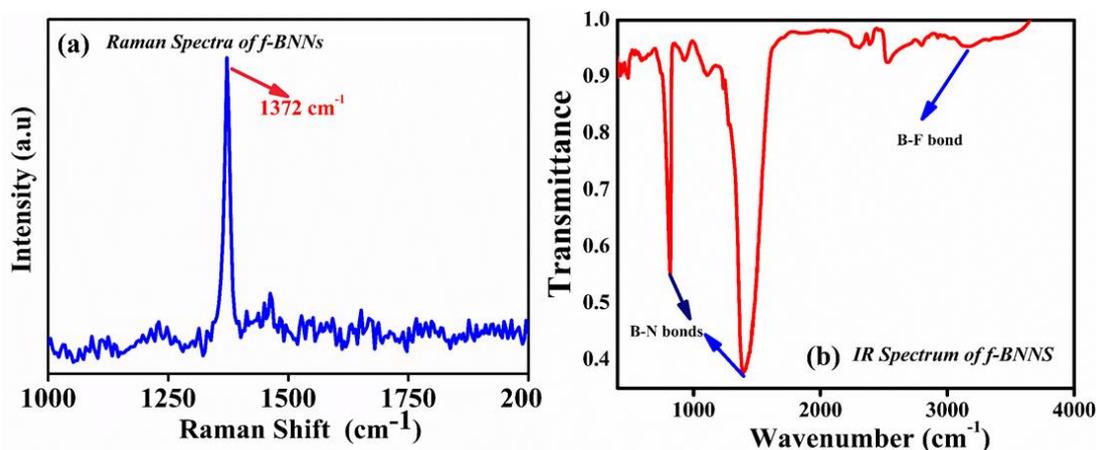
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222

**Fig. 3** XPS Analysis of F-BNNS

223 The RAMAN spectra of the sample as depicted in **Fig.4a**. shows the presence of sharp peak at  
 224  $1372\text{ cm}^{-1}$ . This peak corresponds to the  $E_{2g}$  mode of vibration and appears as a red shift from  
 225 bulk BN ( $1374\text{ cm}^{-1}$ ) [8]. The red shift occurs due to the exfoliation procedure where number  
 226 of layer decreases [24]. In FTIR spectra (**Fig. 4b**), two noticeable bands at  $814$  and  $1389\text{ cm}^{-1}$   
 227 is noticed which rises due to in-plane stretching and out-of-plane bending vibrations of B-N  
 228 bond. In addition to these two, an additional broad band at about  $3000\text{ cm}^{-1}$  is observed which  
 229 indicates B-F bond [24] which proves the bonding of fluorine with Boron. It is also to be noted

230 that generally a broad -OH bond occurs at around  $3000\text{ cm}^{-1}$  due to moisture adsorption on the  
 231 nanomaterial surface. Hence it is concluded that here, B-F bond and -OH bond have overlapped  
 232 in this region [24]. This is in line with the results of XPS analysis as discussed in the previous  
 233 paragraph.



234  
 235 **Fig. 4** (a) RAMAN & (b) FTIR Analysis

236

### 237 3.3 Nanofluid preparation

238

239 Nanofluids of f-BNNs in transformer oil (TO) were prepared via conventional two-step process  
 240 [22]. Two different grades of TO similar to our previous work were used as base fluid (BO)  
 241 [22]. The properties of base fluids are provided briefly in **Table S1** in SI. Nanofluids at three  
 242 different weight percentages (wt. %), 0.0025, 0.005 & 0.01 in both oils were prepared.

243

### 244 3.4 Impact on Electrical insulation

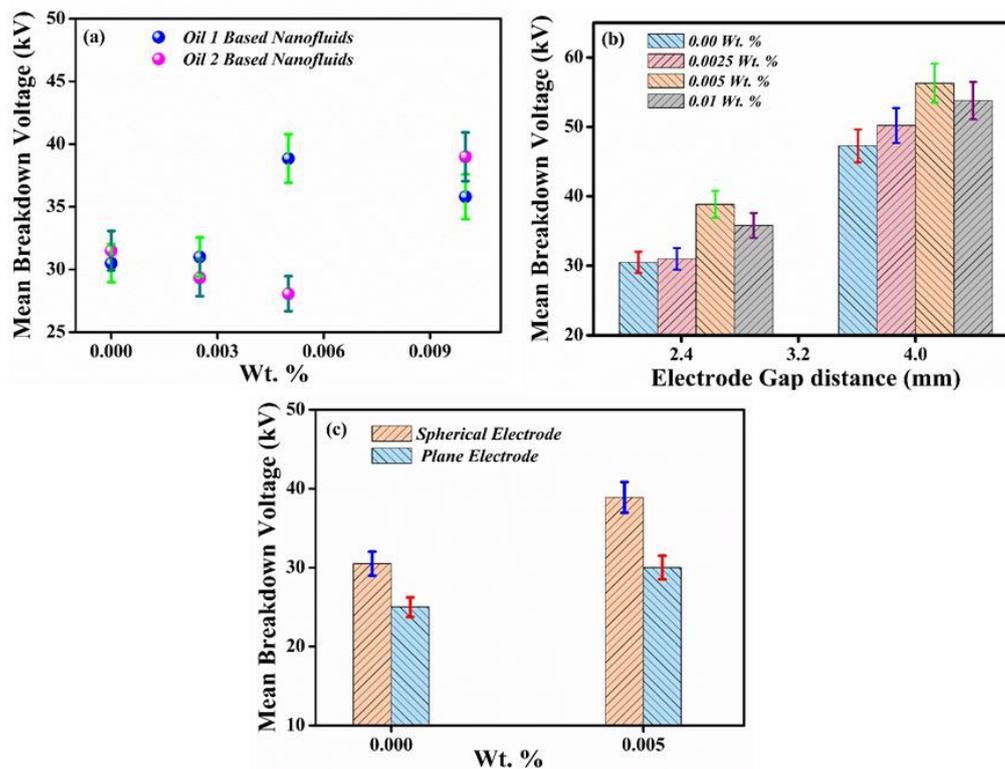
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246 TO acts as the electrically insulating medium between the high voltage windings. Hence, the  
 247 impact of the nano-agents on various insulating properties such as, dielectric breakdown (DB)  
 248 strength, resistivity, dielectric loss, etc. of the liquid insulation system are very important.  
 249 Following sections evaluate these important electrical properties of the TO nanofluids under  
 250 study.

#### 251 3.4.1 AC BDV of Nanofluids

252 The AC voltage tolerance level of nanofluids were tested on Neo-Teletronix made automatic  
 253 breakdown voltage tester (120 kV), following IEC 156 standard with a ramp voltage of  $2\text{ kV/s}$   
 254 using copper electrodes. For all DB measurements conducted here, an average of 18

255 independent measurements was carried out to calculate the mean BDV. AC BDV depends on  
 256 various parameters of nanofiller as well as oil [25]. In order to investigate the role of several  
 257 parameters in monitoring the BD mechanism, different type of measurements was carried out  
 258 by altering various parameters. This helps to identify and analyse clearly the dominant role  
 259 played in BDV enhancement and give better clarity of the mechanism. Here, the AC BDV was  
 260 measured at different experimental conditions such as: (i) *Filler fraction dependent BDV*  
 261 *measurement*: For this, spherical copper electrodes (Diameter: 13 nm) at a gap distance of 2.5  
 262 mm were kept fixed for both set of nanofluids (**Fig. 5a**). A maximum increase of around 26 %  
 263 and 20 % in AC BDV for nanofluids with 0.005 & 0.01 wt. % in oil 1 and oil 2 respectively  
 264 was observable. For oil 1 based nanofluids, detrimental nature in BDV is observed after 0.005  
 265 wt. % which was called as optimum concentration.



266

267 **Fig. 5** (a) Filler fraction dependent BDV plot of nanofluids, Variation of BDV with (b) electrode distance & (c)  
 268 electrode geometry

269 (ii) *Electrode Gap dependent BDV study*: For this, the set of oil-1 based nanofluids was chosen  
 270 and the BDV was measured with spherical copper electrodes with an increased electrode  
 271 distance of 4 mm (**Fig. 5b**). With increased electrode gap (4 mm), the AC BDV of both the  
 272 base fluid and nanofluid increases, however the increment in nanofluid was greater by almost  
 273 17% (iii) *Electrode Geometry dependent BDV study*: For this, spherical electrodes were

274 replaced by plane electrodes (gap distance: 2.5 mm, diameter: 30 mm). Optimum concentration  
275 of oil-1 based nanofluids, i.e., @ 0.005 wt. % was used for the study here (**Fig. 5c**). With plane  
276 electrode configuration, the BDV in both nanofluid and base fluid decreases compared to  
277 spherical electrode system but here too, the nanofluid maintained a higher BDV; almost 16 %  
278 higher. Hence, either with similar electrode geometry or with similar electrode-gap, nanofluids  
279 tolerate more AC voltages than base fluid.

280 The ability of the utilized nanofillers to enhance BD voltages of transformer oil to a huge extent  
281 is explained according to the mostly accepted conjecture behind ‘breakdown in insulating  
282 fluids’; that is the phenomena of ‘*Charge trapping*’ and the associated ‘*changing streamer*  
283 *dynamics*’; as explained clearly for better clarification below.

284 The Maxwell-Wagner-Sillar (MWS) theory of polarization for solid-liquid suspensions (such  
285 as nanofluids) states that, several surface energy states get induced in the interface of base fluid  
286 and nanofiller due to their energy level (*valance, conduction & Fermi, etc...*) differences [25].  
287 The oil-nanofiller interface is a perfect contact; devoid of any abrupt discontinuities in physical  
288 features [25] (so that critical theories of grain-boundary or grain resistance is neglected). Now,  
289 when the suspension is subjected to high electrical stresses, a copious number of charges  
290 (streamer charges) gets injected in the medium. The interfacial states act as sinks of those  
291 charges and holds them. The sinks can be of deep or shallow type, depending on the  
292 conductivity ( $\sigma$ ) and permittivity ( $\epsilon$ ) differences of both oil and the nanofiller. A shallow trap  
293 can accommodate a streamer charge for a short span and de-trap it easily. While a deep trap  
294 stores the charges for a longer time and can create permanent electrets [26]. Presence of these  
295 traps and accumulation of charges in the interface states (also known as MWS region [25], trap  
296 sites [22]) gives rise to a layer wise charge distribution -surface layer, stern layer & diffuse  
297 layer, referred as Electrical double layer (EDL) having thickness in the range of 2 to 9 nm [27].  
298 Due to these charge arrangements in EDLs, the natural streamer orientation and propagation is  
299 hindered. A net dipole moment ( $p$ ) is induced and there is an overall surge in BDV of nanofluid.

300 The degree of streamer hindrance depends on the density of charge traps [28]. It is generally  
301 seen that a conducting nanofiller offer shallow traps with high trap density and can improve  
302 BDV at ultralow concentrations [15]. Semi-conducting and insulating materials generally  
303 provide shallow to deep trap density at higher filler concentrations [5, 6]. Again, the quantity  
304 of charge trapped also depends upon the area of oil-material interface. In this context, choice  
305 of appropriate morphology is required [13]. This is where 2D materials prove to be eminent

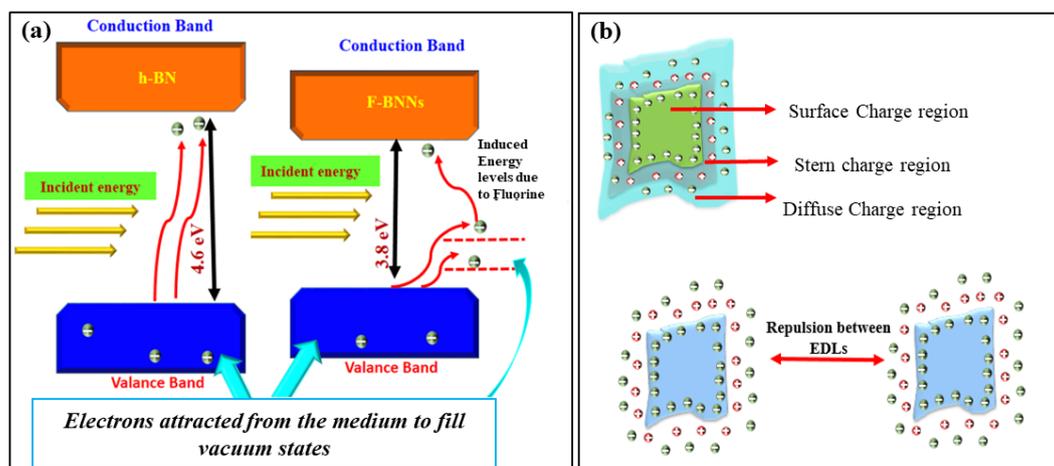
306 participators. They voluntarily provide large oil-nanomaterial interface with the largest number  
307 of surface-active atoms to participate in a phenomenon among all its morphological  
308 brotherhoods. Recalling the role of conducting nanofillers at ultra-low concentrations, choice  
309 of 2D nanofiller with a finite conductivity would be a viable approach.

310 In the current experiment, f-BNNs shows same relative permittivity as of pristine BN, but a  
311 finite conductivity (**Fig. 6a**) is induced due to fluorine incorporation. Hence, the trap type and  
312 the BD performance trend are anticipated to be similar to previous studies by BN particles or  
313 sheets [7, 8, 20]. Conventional BN nanomaterial undergoes polarization and provides traps for  
314 streamers under external electric field [7, 8]. What provides additional attention here is the role  
315 of *2D morphology* and *incorporated fluorine* as explained below. (While discussing, it is  
316 assumed that the other parameters that monitor the insulation of the medium such as purity,  
317 moisture content, and acidity are unaltered in this experiment).

- 318 (i) The specific surface area (SSA) of the interfacial zone or the potential barrier region (oil-  
319 nanofiller interface) is proportional to the aspect ratio of the nanofiller. The quantity of  
320 trapped charges depends on this SSA. This defends the usage of 2D nanofiller as it will  
321 help to trap more charges than other morphologies such as spherical/ cylindrical/ micro  
322 particles at the same concentrations. I.e, the spatial distribution of traps increases with  
323 the rise in SSA of the barrier.
- 324 (ii) Fluorine is highly electro-negative in nature. The introduction of fluorine has helped to  
325 induce a finite (though small) conductivity (**Fig.S2**) via decrease of band gap (Kubelka  
326 Munk plot, **Fig.S3**). The change in bandgap indicates there is induction of extrinsic  
327 energy bands between the valence and conduction bands (VB & CB) of f-BNNs. Hence,  
328 the oil-nanofiller interface will be enriched with more trap centres than that of pristine  
329 BN (**Fig. 6a**). Which implies higher electro-negativity of f-BNNs than pristine BN.  
330 Though a precise determination of electronegativity (via surface charge measurements  
331 techniques) are not easily accessible for oil-based solutions, an estimation of the resultant  
332 electro-negativity can be done by taking the mean of electro-negativities of the  
333 constituent elements of a compound [29]. Therefore, there is trapping of copious charges  
334 in the Layers of EDL of f-BNNs-TO nanofluid. An electron from oil is captured in VB  
335 as well as other extrinsic states (near VB) of f-BNNs (**Fig. 6a**). This brings a -ve charge  
336 accumulation in layer 1 (surface layer) with increased density of charges. The next layer  
337 (stern layer); layer of counter-particles (+ve) has a gradient of decreasing intensity from  
338 the material towards the oil. Due to high charge density of surface layer, stern and diffuse

339 layer will also contain more charges (**Fig. 6b**). Hence, the temporal distribution of traps  
 340 increases with the induction of fluorine in pristine BN. As a result, more BDS is observed  
 341 at lower filler fraction compared to BN sheets / particles (**Fig. 7a**).

342 The spatio-temporal charge accumulation with time continues in a manner analogous to  
 343 capacitor charging phenomena and prevents the streamers to bridge the electrodes. However,  
 344 the process cannot continue for infinite time as there would be a saturation. The dispersed  
 345 nanosheets gets charge saturated and can't accommodate incoming charges further. The  
 346 migration of charges to CB via hopping takes place through the induced states as shown  
 347 schematically in **Fig. 6a**. This leads to transient generation of Schottky current and breakdown  
 348 [26].



349

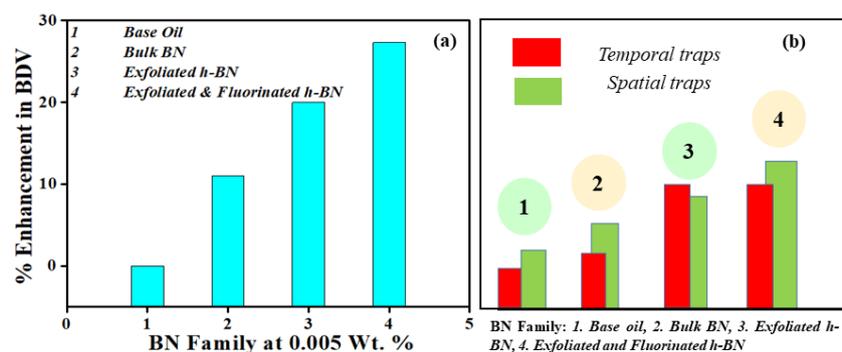
350 **Fig. 6** (a) Trapping of streamer charges in pristine BN and f-BNNs & (b) EDL arrangement in f-BNNs/oil  
 351 interface

352 With increasing wt. %, the number of traps increases resulting in further increase of BDV (**Fig.**  
 353 **5a**). But, after a critical concentration, (here 0.005 wt. % in oil 1), the inter-sheet distance in  
 354 the medium decreases which causes an overlap of the interfacial zones /EDL regions of two or  
 355 more sheets. This causes hindrance in electrostatic stabilization and upsurges Van der Waals  
 356 forces and creates a scattering-free and resistance free escape of streamers through the EDL  
 357 region [22]. The apparent time required to trap a free charge becomes larger than the time  
 358 required for a free charge to traverse the gap. Hence accumulation of carriers occurs resulting  
 359 in numerous arc-discharges and ultimately the oil become conducting.

360 Further, the augmentation in BDV in oil 1 based nanofluids are higher than that observed in oil  
 361 2 based nanofluids. This is due to the fact of oil 2 possessing 10 times lower resistivity than oil  
 362 1. Hence oil 2 intrinsically have low tolerance level to electrical shocks. This requires a large

363 number of nanosheets in oil 2 to enhance the electrical BDV to the level of oil-1 based  
 364 nanofluids. With increasing electrode distance, the distance to be travelled by the streamers to  
 365 cause breakdown increases. Hence a delay in streamer propagation is reflected in the higher  
 366 BDV with increased electrode distance (**Fig. 5b**). The dependency of BDV on electrode  
 367 geometry is explained on the context of the differences of electric field intensity between the  
 368 electrodes. With spherical electrode configuration, the streamers orientation is divergent while  
 369 with plane electrode configuration, a linear orientation of streamer is obvious between the  
 370 electrodes [30]. Hence the field intensity in the former case is smaller than the latter. Which  
 371 implies that the probability of trapping streamer charges will be higher for spherical electrode  
 372 system. With plane electrode system, due to high density of lines of force, streamers would  
 373 successfully short the electrode at relatively shorter time. This explains the results obtained  
 374 with electrode variation conditions (**Fig. 5c**).

375 Finally, a comparative experimental study was carried out to support the claims made. For this,  
 376 BD study with bulk BN, exfoliated BN and f-BNNs at 0.005 wt. % (**Fig. 7a**) was carried out.  
 377 From the relative performances, it is concluded that, the currently developed material is the  
 378 most reliable one among the rising opponent in the field of liquid insulation.

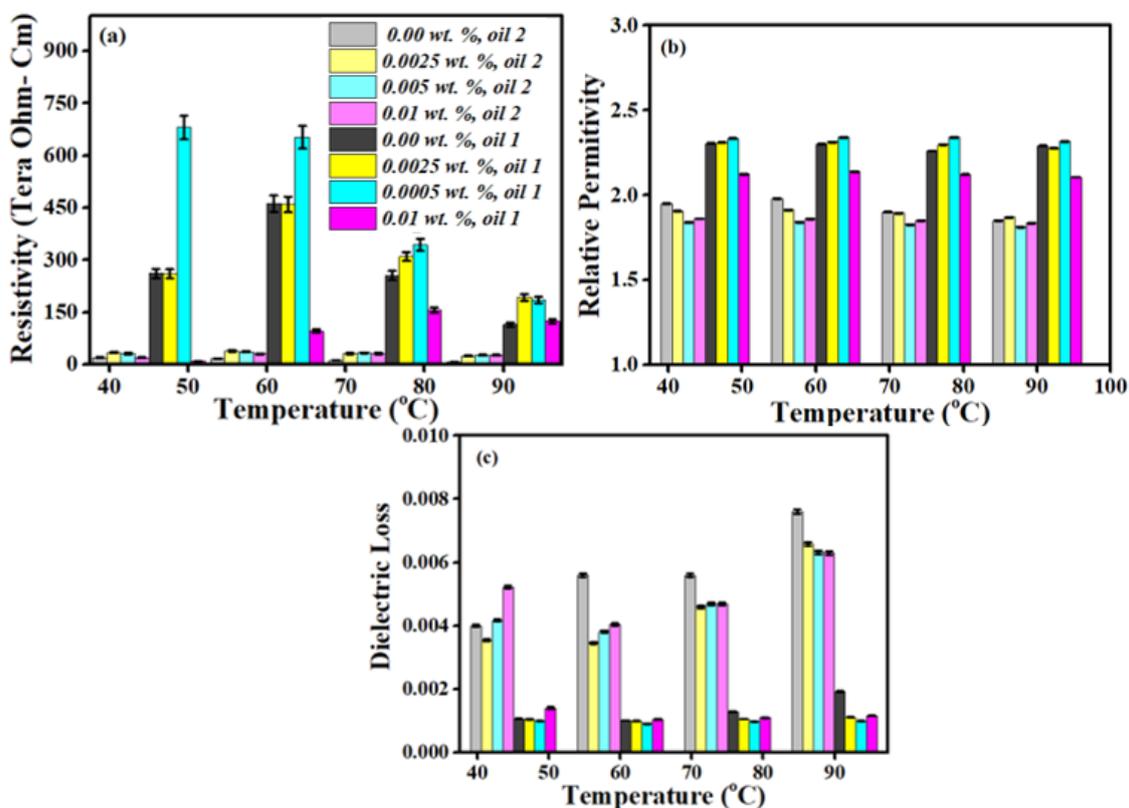


379  
380 **Fig. 7** Comparative BDV Study of BN family

### 381 3.4.2 Resistivity, Relative permittivity and Dielectric Loss Analysis:

382 These are some other electrical properties, necessary to assess the quality of oil. It is well  
 383 known that every dielectric consists of capacitive and resistive component. The capacitive  
 384 component gives idea about the ability of the dielectric to store charges whereas the resistive  
 385 component gives idea about how the dielectric allows the flow of current through it. The loss  
 386 tangent describes the resistive to capacitive loss. Under AC voltage, the ability of the dielectric  
 387 to bound charges is defined by relative permittivity [9]. Under DC voltage, the ability of the  
 388 dielectric to block the flow of current is defined by resistivity [9]. All these are essential to  
 389 check the quality of the oil. These measurements were done using ADTR 2K PLUS instrument

390 for both set of nanofluids at four different temperatures (45, 60, 75, 90°C). The results are  
 391 plotted in **Fig. 8a-8c** below.



392

393 **Fig. 8** (a) Resistivity, (b) Relative permittivity & (c) Dielectric Loss of nanofluids in two different oils

394 **Fig. 8a** depicts the variation of resistivity of nanofluids in two different base fluids. In both  
 395 cases, nanofluids (upto certain concentration) exhibits higher resistivity than base fluids. With  
 396 the increase in temperature, the resistivity falls due to increase in thermal agitation. These  
 397 results are consistent with AC breakdown results. The variation in relative permittivity ( $\epsilon$ ) as  
 398 depicted in **Fig. 8b** shows that there is a rise in  $\epsilon$  value for nanofluids in oil 1, compared to oil  
 399 2. The dielectric loss is seen to decrease in nanofluids in both oils (**Fig. 8c**). It is interesting to  
 400 note that loss at higher temperature is more for base oil, with nanofluids, a drastic drop in loss  
 401 is observed.

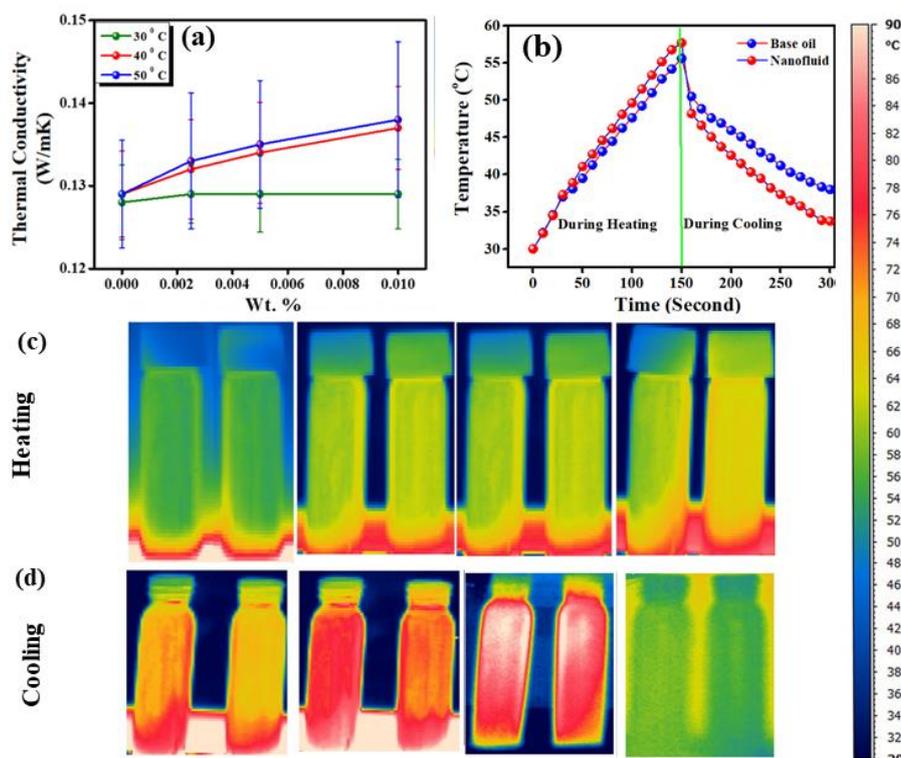
402 The reason for the variation of all three properties described above is due to the similar  
 403 phenomena of charge trapping in extrinsic energy states as discussed in the AC BDV section.  
 404 Also, the loss in oil 2 is higher than oil 1 due to the resistivity differences between the two  
 405 original oil.

406

### 407 3.5 Impact on Thermo-physical Properties

#### 408 3.5.1 Thermal conductivity, Thermal response & Thermal Images

409 The temperature dependent thermal conductivities ( $k$ ) of base fluid and nanofluids (in oil 1)  
 410 were measured and represented in **Fig. 9a**. There exists a linear relationship between thermal  
 411 conductivity and wt. % & temperature. A maximum enhancement of ~8% in  $k$  is obtained for  
 412 0.01 wt. % nanofluid at 50° C. The results follow the trend predicted by Maxwell [20] and  
 413 remain within limits predicted by HC hypothesis [31]. Further, to support the results, thermal  
 414 response characteristics (heating and cooling) and infrared (IR) images during different  
 415 instances of heating and cooling of base oil and nanofluid (0.01 wt. %) were recorded under  
 416 same physical conditions; as presented in **Fig. 9b-Fig.9d** respectively.



417

418

419 **Fig. 9** (a) Temperature dependent Thermal conductivity, (b) Thermal characteristics & (c) Infrared Images of  
 420 the base fluid (left) and nanofluid (0.01 wt. %, right) during heating and cooling

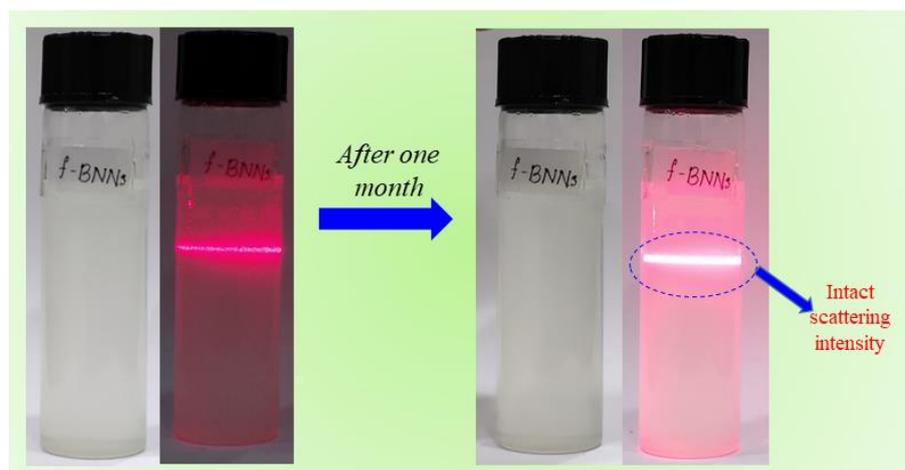
421 It follows that the nanofluid gets heated faster than the base fluid as well as cool down easily;  
 422 i.e the specific heat decreases in nanofluids. Analysing the IR images from the IR temperature  
 423 scale, it follows that the temperature profile is higher in every interval of heating in nanofluid.

424 During cooling, the nanofluid gets cooled much easily. Hence nanofluids endure more heat  
425 dissipation capability than base oil.

426 The rise in thermal conductivity is attributed to the rise in the rate of phonon transport in the  
427 medium. For nano-suspensions, the molecular liquid layering at the interface acts as the bridge  
428 of ballistic heat transfer. The nature (smoothness) of the interface is hence of immense  
429 importance. In this experiment, 2D f-BNNs, owing to its higher aspect ratio, thinness (5nm,  
430 10-12 layers) and lipophilic nature provides a large smooth interface; which enables restriction  
431 free propagation of acoustic phonons facing low Kapitza resistance [32]. With rising  
432 temperature, the molecules in the nanostructures tremble about their mean site and the  
433 contribution of Brownian motion in heat transfer gains merit [33]. Apart from the morphology,  
434 hexagonal phase of the dispersed material has played its role in providing consent to long range  
435 phonon transfer along the c-axis of hexagonal unit cell [8, 20].

### 436 3.5.2 Stability

437 The most important factor that decides the industrial applicability of a nanofluid is its stability.  
438 The stability of the nanofluid was checked by Tyndall scattering (room temperature) as shown  
439 in **Fig. 10**. It is seen that nanofluid is stable after one month of preparation.



440

441 **Fig. 10** Tyndall scattering image of nanofluid (0.01 wt. %) after just preparation & after 1 month of preparation.

442 The reason for the stability is embedded in the physics of fluorine incorporation. It is well  
443 known that the electro-static repulsion between nanofiller defends the stability against the  
444 attractive van der Waals forces among the nanofillers in nanofluids [34]. Inoculation in BN by  
445 highly electro-negative fluorine and the comparatively low wt. % of the nanofluid provided  
446 natural electrostatic repulsion (**Fig. 6d**) and helped to stabilize the medium. The nanofluid was  
447 seen to be resistant against gravity for a couple of months without any agglomeration. Hence,

448 it is obvious that in real time operating conditions in electrical transformers, it will provide  
449 great solutions in reducing the burden of modern power system challenges.

450 Finally, a comparative table showing the relative performance of all BN based materials in  
451 transformer oil were surveyed and presented in **Table 1**.

452 **Table 1.** Comparative Table

<b>Morphology, Size (nm) &amp; Base Fluid</b>	<b>AC BDV increment (%)</b>	<b>Other Electrical Properties</b>	<b>Stability</b>	<b>Reference</b>
Nanosphere, 50 nm, Transformer oil	11 % (@ 0.05 wt. %)	Improved Resistivity, DC BDV, relative permittivity & dielectric Loss	Not studied	[7]
Nanosheets, 100-450 nm, Transformer Oil	-	-	3 months	[8]
h-BN Nanosheets, Cottonseed oil	12 % (@0.01 Wt. %)	Improved impulse BDV & resistivity, reduced dielectric loss	Stable (studied for a week)	[9]
h-BN Nanosheets, few layers, Mineral Oil	-	Improved resistivity & reduced dielectric loss	Surfactant-free Ultra-stable	[21]
BN nanoparticles. 50 nm, Vegetable oil	-	Improved resistivity, relative permittivity & trap characteristic, reduced dielectric loss & electrical conductivity	Not studied	[35], year
<i>Present work:</i> f-BN nanosheets (100-450nm, 5 nm thickness)	20-26 % (@0.01 -0.005 wt %)	Improved resistivity, relative permittivity & reduced dielectric loss	Ultra-stable for several months	-

453

#### 454 **4. Conclusions:**

455 A simple chemical method of fluorination is adopted here for improvement of twin goals of  
456 TO NFs – AC Breakdown Voltage and long-term stability. The attachment of a small quantity  
457 of conductive fluorine on highly insulating h-BN nanosheets has played a vital role in reducing  
458 the band gap. It offered plenty of charge trapping sites and resulted in huge surge of AC BDV  
459 as compared to the state-of-the-art BN family. Whereas the lipophilic nature, 2D morphology

460 and high electronegativity enabled electrostatic repulsion among the EDLs of the nanosheets  
461 provided long term stability of the nanofluid without usage of any long chain surfactant. This  
462 path breaking idea opens a new horizon in designing future tailor made functionalised  
463 nanofillers for next generation NFs to meet the multiple objectives.

464

#### 465 **Declaration of competing interest**

466 The authors declare that they have no known competing financial interests or personal  
467 relationships that could have appeared to influence the work reported in this paper.

#### 468 **Authorship contribution statement**

469 **Mississippi M. Bhunia:** Conduction of experiments (synthesis & application part), analysis &  
470 plotting of results, drafting, synchronizing, editing, images formatting. **Kalyan Kumar**  
471 **Chattopadhyay & Paramita Chattopadhyay:** Idea behind the work, Material selection,  
472 discussions and manuscript correction.

473

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