

Correlating topography and elastic properties of Elastin-Like Polypeptide scaffolds probed at the nanoscale: Intermodulation Atomic Force Microscopy experiments and Molecular Dynamic simulations

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Abstract

The synthesis and property characterization of soft biomaterials has taken precedence in recent years. Although bulk physical-chemical properties are well known for these bio-materials, nanoscale properties still need to be probed and evaluated to fine tune the bio-compatibility (structural as well as functional) with natural tissues for regenerative medicine, prosthetics and other biological applications. In this study, we focus on a popular soft bio-material, ELastin-like polypeptide (ELP) which has been prepared under different pH conditions. We explore the topographical features of the ELP at the nanoscale using Atomic Force Microscopy (AFM). Additionally, we employ a non linear mode of AFM called Intermodulation-AFM (ImAFM) to correlate the elastic properties (Young's modulus) of ELP probed at the nanoscale with the topographical features which gives us a deep insight into the mechanical properties offered by ELP when the structural features are

altered by change in the ELP synthesis conditions. The noteworthy point is that we measure these properties at a spatial resolution of 0.9 nm. Finally, we explain the change in the structural features of ELP with varying pH through atomistic Molecular Dynamics Simulations. We follow the interaction mechanisms of the amino acid sequences and crosslinkers with proteins as they form the backbone and sidechain of the ELP at different pH.

Keywords: Elastin-like Polypeptide, Intermodulation Atomic Force Microscopy, Mechanical Properties, Young Modulus, Nanoscale regime, Topography, Soft biomaterials

1. Introduction

The science of synthesizing biomaterials to enhance, support or replace damaged tissue or body functions is at least five decades old. They can either be natural or engineered to gear their function to specific needs. The development of biomaterial is a vital area in the fabrication of scaffolds. These engineered substrates are studied predominantly for their surface structural features correlating with cellular responses. The engineered substrates tend to be stiff and lack cell adhesive properties as compared to native extracellular matrix (ECM). Also, they are found to have angular and repetitive surface structures as compared to ECM that have undulating surfaces [1]. To overcome these differences in properties from native ECM, elastin-based material have been introduced into these engineered substrates since it is one of the most abundant protein found in native ECM [2]. Elastin is a protein that possesses inherent chemical resistance and high durability in entire body owing to the presence of more hydrophobic amino acids [3]. Elastin

16 acts as a main component to induce rubber-like elastic nature to the tis-
 17 sues, that support them by sustaining infinite deformation/relaxation cycles
 18 without getting ruptured [4]. This property of providing elastic recoil to stiff
 19 substrates makes elastin a better choice of material as a component in bioma-
 20 terial fabrication [5]. Additionally, elastin facilitates accelerated skin wound
 21 repair, enhances the proliferation of endothelial cells, vascular smooth muscle
 22 cells and skin fibroblast since it has the ability to interact with large number
 23 of cell surface receptors [6]. However, the extensive cross-linked structure
 24 and hydrophobic nature of elastin hinders the processing and fabrication of
 25 elastin into biomaterials [7]. Therefore, recombinant, synthetic and solu-
 26 ble forms of elastin like alpha-elastin, elastin-like polypeptides (ELPs) and
 27 tropoelastin are used to fabricate composites scaffolds [8].

28 Synthetic elastins such as ELPs belong to artificial peptide polymers class.
 29 The general composition of ELPs especially Val-Pro-Gly-X-Gly is a repeated
 30 unit of pentapeptide, where X can be any amino acid except proline. The
 31 Val-Pro-Gly-X-Gly domain is derived from tropoelastins hydrophobic domain
 32 [9]. ELPs are biodegradable and biocompatible. These ELPs exhibit lower
 33 critical solution temperature (LCST) phase behavior which results in insol-
 34 uble coacervate phase formation similar to tropoelastin [10]. The alteration
 35 of stimuli such as light, pH, concentration of salts and proteins can tune the
 36 LCST of ELPs. The stimuli-triggered self-assembly of ELPs can be achieved
 37 by their reversible phase behavior. The ELPs can also self assemble to form
 38 hydrogels [11] and nanoparticles [12] that act as potential candidates to use
 39 in drug delivery and tissue engineering applications [13, 14]. As an added
 40 advantage, the purification of ELPs without chromatography is easier owing

41 to its ability to undergo reversible phase separation when it exceeds tunable
42 transition temperature [15].

43 The formation of hydrogel can be performed with proper peptide sequence
44 and under specific aqueous conditions (*e.g.*, ionic strength, pH or tempera-
45 ture) via a self-assembly technique [16]. These polypeptides adapt to form
46 secondary structural α -helices and β -sheets. This happens within peptide
47 chains through intramolecular hydrogen bonding that renders specific con-
48 formation based self-assembling nature and bioactivity. The polypeptide-
49 based hydrogels have grabbed huge potential in biomedical applications ow-
50 ing to structural similarity to ECM, enzyme biodegradability, cell adhesive
51 properties and remarkable biocompatibility [17]. Moreover, the hydrogels
52 can enhance cells adhesion, proliferation and differentiation and result in
53 good bioactivity. The change in peptide length or replacement of amino
54 acid residues is useful to alter the biological and physical properties of these
55 hydrogels [18].

56 The common natural polymers (gelatin, collagen, pectin, gellan gum,
57 chitosan and alginate) derived hydrogels also exhibit structural similarity to
58 tissues, biocompatibility and good degradability. However, drawbacks con-
59 cerning the suboptimal mechanical stability, immunogenic or inflammatory
60 response still persists [19]. The application of hydrogel is mainly based on
61 the mechanical properties of hydrogels. Indeed, the success and fate of the
62 implant is purely based on the hydrogel mechanical properties that are com-
63 patible with host tissue and is an important factor for consideration during
64 design and fabrication of tissue regenerative systems. This should avoid the
65 induced host inflammatory response [20]. The cellular responses are directed

by the extent of matrix stiffness [21]. The existence of mismatch in stiffness between native tissue and hydrogel often results in local stress that concentrates at the interfaces. Also, the stress can be induced due to the lack of proper adhesion of the gel to the adjacent tissue that disturb relative mobility. The immune cell response by generation of pro-inflammatory signals can cause fibrosis and lead to shear stress [22]. Hence, hydrogels that usually exhibit less stiffness than the neighboring tissue could minimize the stress accumulation, and reduce formation of fibrous capsule. The peptide-based hydrogen biomaterials are appealing in areas of regenerative medicine and tissue engineering applications as they can adapt to the stiffness of tissue microenvironments *in vivo* [19].

The ELPs mechanical and topographical surface analysis plays a vital role in their development stage. In recent years, dynamic AFM technique has been extended to be driven with multiple frequencies as compared to traditional AFM where the cantilever is driven with a single frequency. Here, excitation of probe and measurement of response occurs at two or more frequencies which allow us to obtain more details on tip-surface interaction [23]. One of the multi-frequency dynamic AFM method is Intermodulation AFM (ImAFM) that employs for capturing of several frequency mixing products by two driving frequencies closer to single frequency cantilever resonance instead of higher eigenmodes or higher harmonics excitation of the cantilever [24]. In ImAFM, same thermal noise measurement is applied to calibrate both the cantilever and deflection sensor. ImAFM is capable of measuring the tip-surface force accurately as a function of the cantilever deflection using this calibration. The route to map the materials elastic response and local

viscous nature arise from the possibility to measure how both the dissipative and conservative components of the interaction depends on the amplitude of oscillatory motion [25].

In this study, we validate the ImAFM technique by measuring the effective elastic properties i.e. Young's modulus of ELP at the nanoscale in spacings of 1 nm and correlate them with topographical features. The ELP biomaterial were synthesized at different pH conditions to observe the change in their nanoscopic structural features and their relative elastic behavior. We also explain the underlying mechanisms for different structural properties of ELP at different pH through Molecular Dynamic simulations.

2. Experimental section

2.1. *Elastin-like Polypeptide (ELP) production and purification*

The construction of the plasmid, the expression of ELP in escherichia coli and the following purification steps have been described in detail elsewhere[1]. Shortly, the pET15b plasmid containing the ELP sequence was transformed with a heatshock at 42°C into competent BL21-PLyS e.coli (Invitrogen). The ELP contained either a fibronectin cell binding site (RGD) or a scrambled version of that amino sequence (RDG). The transformed bacteria were cultured in terrific broth containing 100 $\mu\text{g}/\text{ml}$ ampicillin until an optical density of 0.8 at 600 nm has been reached, at which point protein expression was induced with 1 mM isopropyl β -D-1-thiogalactopyranoside. After 5-7 h of expression, the cells were lysed by freeze-thaw cycles with the addition of DNase and 1 mM phenylmethylsulfonyl fluoride. From there the protein was recovered by iterative inverse temperature cycling. As a final step, the

115 protein is dialyzed three times through a 3.5 kDa membrane at 4°C and then
116 lyophilized.

117 2.2. Intermodulation Atomic Force Microscopy

118 The surface morphology of the films produced was imaged by an Atomic
119 Force Microscope (AFM) by NT-MDT working in tapping mode. Images
120 of the sample surface were acquired on a $5 \times 5 \mu\text{m}^2$ area using silicon can-
121 tilever from Nanosensors (type NCH-50), working at the resonance frequency
122 $f=362\text{KHz}$, tip radius $r < 10 \text{ nm}$, quality factor of $Q=490$, and force con-
123 stant 34.4 N/m . During the acquisition of the samples surface topography
124 their elastic properties were acquired using the intermodulation technique.
125 In a standard AFM measurement, the tip is driven at its resonance frequency
126 where the tip oscillation amplitude and phase gather information of the topo-
127 graphical features and nature of the investigated material. A linear response
128 of the tip at the driving frequency is assumed. However, the behaviour of
129 the tip motion is highly non linear when the tip interacts with the sample
130 surface.

131 Measurements were performed with pixel dimensions of 512×512 on
132 areas of $5 \times 5 \mu\text{m}^2$ giving us a spatial resolution of about 0.9 nm . Within
133 the same intermodulation measurement both the surface topography and the
134 force curves were obtained for each scanned point. The value of the elastic
135 modulus was then obtained by performing the fitting procedure with the
136 modified DMT model to the force curve as analyzed from the inversion of
137 the intermodulation signals. The fitting procedure is performed within the
138 IMP software suite at each point of the scanned surface.

139 2.3. MD simulation details

140 We run all-atom molecular dynamics (MD) simulations of elastin-like
 141 polypeptide (ELP) system. Our system contains cell-binding (TVYAVT-
 142 GRGDSPASSAA) and structural domains ((VPGIG)2VPGKG(VPGIG)2)3,
 143 QK peptides (KKLTWQELYQL[K(Ac)]Y[K(Ac)]GI), and THPC, TrHP and
 144 TrHPO ligands as cross-linkers. Here we use GROMACS 2016.4 package to
 145 run our MD simulations, using CHARMM36 force field. We run three simu-
 146 lations to mimic the system at three different pH values: pH 5, 7, and 9. The
 147 only amino acid in the range that could be affected by the change of pH in
 148 this range are HIS and CYS[30]. Thus, based on the amino acids in our ELP,
 149 the pH range that is introduced to our system does not affect the charge of
 150 the proteins. However, the change in the pH affects the properties of cross-
 151 linkers in the system. At lower pH values (pH 5), the only cross-linkers in
 152 the system are THPC, while at higher pH values THPC changes into TrHP
 153 and TrHPO ligands (Fig.1)[31, 32, 33, 34]. The content of each system could
 154 be found in Table 1. Each simulation box contains 9 cell binding and struc-
 155 tural domain peptides, 16 QK peptides and 200 cross-linkers. At pH 5, all
 156 of our cross-linkers are THPC molecules, at pH 7 THPC, TrHP and TrHPO
 157 ligands co-exist (1:2:1 ratio) as our cross-linkers and in pH 9, only TrHP and
 158 TrHPO ligands (3:2 ratio) are present. Cl counterions are used to neutralize
 159 our system and TIP3P water molecules are used to solvate the simulation
 160 box (Table 1).

161 The same approach is used to set up and run our simulations for all
 162 three systems. After energy minimization, under NVT conditions (NVT: a
 163 canonical ensemble, where the number of particles (N), the volume of the

Table 1: System content for each simulation at different pH values

Simulation box content	pH5	pH7	pH9
Cell and structural binding	9	9	9
QK-peptide	16	16	16
THPC	200	50	-
TrHP	-	100	120
TrHPO	-	50	80
Cl ⁻	243	93	43
TIP3P	45270	45661	45824

164 system (V) and the temperature of the system (T) remains constant) we
 165 run brief MD simulation at T=300 K using Nose-Hoover thermostat for 2
 166 ns with time steps of 1 fs. Protein positions are restrained in this step to
 167 prevent any dramatic structural changes before the production run. Next,
 168 we run a brief simulation under NPT condition (NPT: an isothermalisobaric
 169 ensemble, where the number of particles (N), the pressure of the system (P)
 170 and the temperature of the system (T) remains constant) at T=300 K, and
 171 pressure=1 atm, using Parrinello-Rahman and isotropic pressure coupling,
 172 with $\tau_p=5$ ps and compressibility = $4.5 \times 10^{-5} \text{bar}^{-1}$ for 100ps with time steps
 173 of 2fs, keeping the protein positions restrained. After this short simulation,
 174 we start the production run under NPT conditions and using the same con-
 175 ditions as the previous step, without position restraints for 400 ns. Visual
 176 Molecular Dynamics (VMD) package and customized C++ codes are used
 177 to calculate our results. VMD is also used to visualize our structures.

178 3. Results and Discussion

179 3.1. Elastic Properties Modeling

180 The elastic properties of the ELP surface were obtained by modelling
 181 the contact mechanism of the AFM probe with the samples surface using
 182 the Derjaguin-Muller-Toropov (DMT) model [26], a model suitable in cases
 183 where there is low adhesion force and small tip radii. The reconstructed force
 184 curves $F(z)$, where z is the tip to surface distance, obtained from the non
 185 linear analysis of the tip motion [27, 28] can be modeled by the following
 186 relations:

$$F(z) := \begin{cases} -F_{min} \frac{a_0^2}{(z+a_0)^2} - k_{long}d & z > -h \\ -F_{min} + \frac{4}{3}E^* \sqrt{-Rz^3} - k_{long}d & z < -h \end{cases} \quad (1)$$

187 where h is the tip equilibrium position above the surface, F_{min} is the minimum
 188 of the force, a_0 is the intermolecular distance, R is tip radius and k_{long} is an
 189 additional parameter introduced in the DMT model which takes into account
 190 long distance interaction force between the tip and the sample surface. As an
 191 example, in Fig.2 we report the results of the fitting procedure with the DMT
 192 model to the experimental curve obtained at a given point on the surface of
 193 the ELP sample synthesized at pH3. As it can be seen the different regimes
 194 *e.g.* the attractive and repulsive regions of the force-distance curve, *e.g.* when
 195 the tip is not in contact and after the contact are well reproduced. The elastic
 196 modulus E^* obtained by the fit in this case was of 0.98 GPa.

197 3.2. Atomic Force Microscopy

198 The morphology represented by the ELP at microscopic scales are 'bead
 199 and string' like features [29]. When measuring these features with AFM,

200 we observe the beads to be bigger than their probable size while the strings
 201 are not visualized at all. In Fig.3 are presented the surface topography of
 202 the samples grown at pH3, pH6 and pH9 respectively. This is due to the
 203 resolution of the AFM which is presented by the tip while it scans the sample.
 204 As it can be clearly seen, the surface morphology presents marked differences
 205 as a function of the pH value. The surface of the pH3 sample is very smooth
 206 when observed at short distances, however it shows strong variations of the
 207 surface height along distances of the μm order. In fact, a height variation of
 208 about $2 \mu m$ is observed along a surface distance of $5 \mu m$. As the pH values
 209 increases, a different behaviour is observed: at long distances the surface
 210 topography seems to become more flat but at the same time height variation
 211 at short distances increases. In order to highlight this behaviour in Fig.4 we
 212 report the surface profiles collected along a line on the surface of the samples.
 213 As it can be clearly seen the bead size of ELP decreases as we move to higher
 214 pH hence the image for pH9 shows a smoother surface compared to the image
 215 for pH3.

216 3.3. *Elastic modulus*

217 In Fig.5 are shown the map of the elastic modulus obtained by fitting
 218 the force curves resulting from the intermodulation measurements at each
 219 point of the samples surface. The value of E^* is within the 0-1.2 GPa range
 220 for all the samples. Nevertheless null or very low E^* values usually corre-
 221 spond to surface points where the force curve could not be properly fitted by
 222 the DMT model, probably the consequence of a bad tip-surface interaction.
 223 Looking at the maps in Fig.5 the details of the surface topography can be
 224 clearly observed, pointing out how the surface morphology, and hence the

225 pH value plays a role in determining the elastic properties of the samples.
 226 However, a clear picture of the E^* dependence on pH does not emerge from
 227 the maps. To this purpose we report in Fig.6 the histogram of distribu-
 228 tion of the elastic modulus in the investigated areas for ELP synthesized at
 229 different pH. Looking at the E^* distribution function for the pH3 prepared
 230 sample, a continuous distribution of values between 0 and 1.2 GPa is ob-
 231 served. Nevertheless three maxima are present: the first peaked at $E^*=0$,
 232 whose probable origin has been discussed earlier, the second at about 0.21
 233 GPa and the third at about 0.6-0.7 GPa. The second and third region can
 234 be regarded as *soft* and relatively *hard* components respectively. It worth
 235 noticing that a continuous tail extends till the maximum observed value .
 236 The elastic modulus distribution of the samples prepared at pH6 and pH9
 237 show a similar behaviour, with the exception of the drastic decrease of the
 238 values around 0 GPa, pointing out how the different sample morphologies
 239 lead to a better tip-surface interaction as long as the surface become more
 240 *flat*. Also for these samples two maxima are present in the E^* frequency
 241 distribution peaked at 0.3 and 0.7 GPa, confirming the observation of a *soft*
 242 and the relatively *hard* phases. Thus, a well defined two-phase structure,
 243 *soft* and *hard*, emerges in the elastic properties of the ELP as long as the
 244 pH value is increased above pH3. The origin and the molecular structure
 245 of these two phases cannot be easily individuated solely on the basis of the
 246 AFM and ImAFM measurements. In order to understand the evolution of
 247 the ELP structure at a microscopic level as a function of the pH values, we
 248 performed molecular dynamic simulations of the ELP system to understand
 249 the underlying mechanisms for change in structural behavior at varying pH.

250 *3.4. Role of pH on the micro-structural properties of the ELP explained by*
251 *MD simulations*

252 *3.4.1. General Interaction with Proteins*

253 Figure 7 indicates the general initial configuration of our system at pH
254 7 and final configuration for pH 5, pH 7, and pH 9. We can see that in
255 all systems, the proteins self-assemble and interact with the cross-linkers.
256 Moreover, in each system, we have cross-linkers floating in the water, which
257 indicates the possibility of these molecules to solubilize in water.

258 Our results indicate that the number of contacts between proteins and
259 the cross-linkers in the system is pH dependent (Fig.8A). At pH 5 we have
260 the least amount of interactions between the cross-linkers (THPC) and the
261 protein units, while at pH 7 this value is increased by more than twice be-
262 tween the mixture of the cross-linkers (THPC, TrHP, and TrHPO) and the
263 proteins. At pH 9 the number of interactions is slightly higher than pH 7.
264 At this pH, the mixture of cross-linkers contains TrHP and TrHPO. Based
265 on the results obtained in our simulations, we do not see any major change
266 in the interactions during the last 200 ns of the simulations. Thus, we have
267 used the last 200 ns of the trajectory for calculating the rest of our results
268 in this paper unless stated otherwise.

269 *3.4.2. Detailed Interactions with amino acids*

270 **pH5:** At pH 5, we can see that 30% of THPC cross-linkers interact with
271 the protein (Fig.8B). Similarly, at pH 7, around 30% of THPC is interacting
272 with proteins, while more than 50% of TrHPO and 80% of TrHP is interact-
273 ing with protein (Fig.8C). At pH 9 TrHPO and TrHP are interacting with
274 proteins at almost the same ratio as the pH 7 system (Fig.8D). Different

275 cross-linkers interact with different residues on the proteins. At pH 5, THPC
276 is mainly interacting with ASP and GLU, which are both negatively charged
277 (Fig.9A). This is compatible with the general view that THPC has a positive
278 charge, and thus interacts with negatively charged residues, based on the
279 electrostatic interactions. Here, we can see that THPC is mainly interacting
280 with side chains of the ASP and GLU. More generally, THPC is mainly in-
281 teracting with the sidechain of negatively charged or polar residues, while for
282 positively charged or non-polar residues, such as ILE, THPC interacts with
283 the backbone of the residue.

284 **pH7:** At pH 7, we can see that THPC is mainly interacting with GLU,
285 ASP (similar to the pH 5 system), which are both negatively charged, GLN,
286 which is polar with partial negative charge and ARG, which is positively
287 charged (Fig. 10). More detailed studies indicate that in the case of ARG,
288 THPC is interacting with the backbone part of the amino acid. In this pH,
289 THPC is mainly interacting with the sidechain of the negatively charged
290 residues and polar residues, similar to the pH 5 system (Fig.10B). The two
291 exceptions are TYR, and ALY (acetylated LYS). First, THPC is mainly in-
292 teracting with the backbone of TYR at pH 7, unlike pH 5. Second, THPC
293 is mainly interacting with the side chain of ALY, where the amino acid has
294 partial negative charge. In the case of TrHP, the molecule interacts with a
295 variety of charged, polar and non-polar residues, without any main prefer-
296 ence. However, the interactions with charged or polar residues are mainly
297 through sidechains, while for non-polar residues the interactions are mainly
298 through the backbone (Fig.10C). We can see that TrHP has significant inter-
299 action with the sidechain of positively charged residues compared to THPC.

TrHPO has generally much higher interactions with charged residues compared to the other cross-linkers, especially in the case of interacting with LYS (Fig. 10A). Similar to THPC and TrHP, TrHPO interacts with the polar and charged amino acids mainly through the sidechain, and for non-polar residues, the TrHPO mainly interacts with the backbone region (Fig. 10D).

pH9: At pH 9, TrHP and TrHPO have similar interaction with the amino acids in the system (Fig. 11). These patterns are also similar to the patterns of TrHP and TrHPO at pH 7 (Fig. 10). Here, we can see that both cross-linkers have higher interactions with charged and polar residues, and these interactions mainly occur with the side chain of the amino-acids (Fig. 11B and C).

3.4.3. Interaction with the backbone

We can see in Fig. 12 that in the case of interaction with the backbone, all cross-linkers mainly interact with the CO group of the backbone, which has the highest electronegativity. NH group has much less interaction compared to CO group, while the C_α has the lowest interaction with cross-linkers in the backbone.

3.4.4. Interaction with proteins

Figure 13 indicates the normalized share of interaction between cross-linkers and the protein. We can see that the cross linkers have more normalized interactions with QK peptides in the system. Figure 14 indicates the higher number of interactions between cross-linkers and proteins in higher pH values. Moreover, these results indicate higher interaction between QK peptides and the cross-linkers in all pH values.

324 Based on our simulations, proteins do aggregate, while the cross-linkers
325 covers the outer area of the protein chunks and produce some domain-like
326 configurations (Figure 15). At lower pH values, a lower number of cross-
327 linkers cover protein surface. Thus, the domains could diffuse and produce
328 larger domains with similar properties. However, at higher pH values, the
329 higher surface density of the crosslinkers prevents the protein domains to
330 diffuse and hence the bead size reduces.

331 4. Conclusion

332 We were able to successfully analyze the elastic properties at the nanoscale
333 of soft biomaterials, ELP in our study, probed by AFM tip using the ImAFM.
334 ImAFM has proven its capabilities to measure the visco-elastic properties of
335 soft materials as a new mode of AFM technique. In particular, we measure
336 the elastic properties of the proteins and not of crosslinkers. The spatial
337 resolution offered by ImAFM to evaluate the elastic properties is about 0.9
338 nm. Hence, we see that the elastic values at different pH values remain
339 the same. The domain size dependent on pH does not change the Youngs
340 modulus value as can be seen in the scale bar (range: 0 - 1.8 GPa) of the
341 figures with different pH.

342 Our simulations reach energetically feasible configuration after 200ns of
343 simulations. These simulations indicate direct relation of pH dependency
344 in ELP systems. Since THPC is positively charged, in all simulations it
345 mainly interacts with negatively charged residues, through electrostatic in-
346 teractions. TrHPO mainly interacts with polar and charged residues (both
347 positive and negative). This is due to the neutral charge and the P=O bond,

348 which provides strong H-bond interactions with these residues. TrHP has
349 the highest normalized number of interactions with ELP, which is due to the
350 non-polar region in the center of the particle and O-H groups on the chains
351 which provides both polar and non-polar interaction with ELP. In terms of
352 interaction with LYS, TrHPO has the highest interaction, while TrHP has
353 lower interaction and THPC does not interact.

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364 6. References

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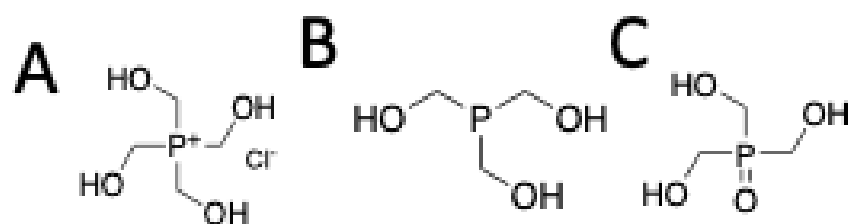


Figure 1: The structures of A) tetra-hydroxymethyl phosphonium chloride (THPC), B) tri-hydroxymethyl phosphine (TrHP), C) tri-hydroxymethyl phosphine oxide (TrHPO).

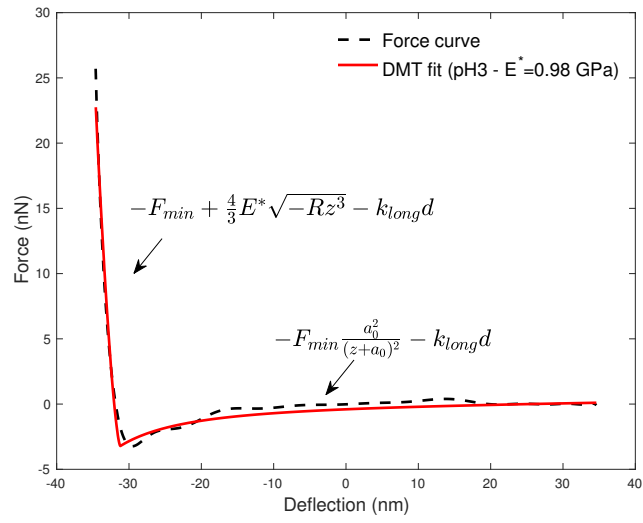


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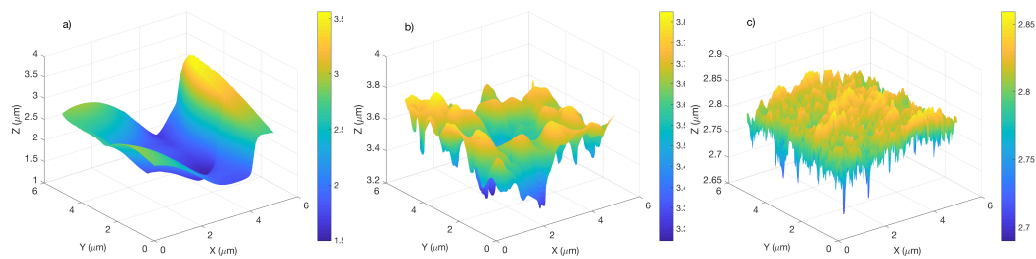


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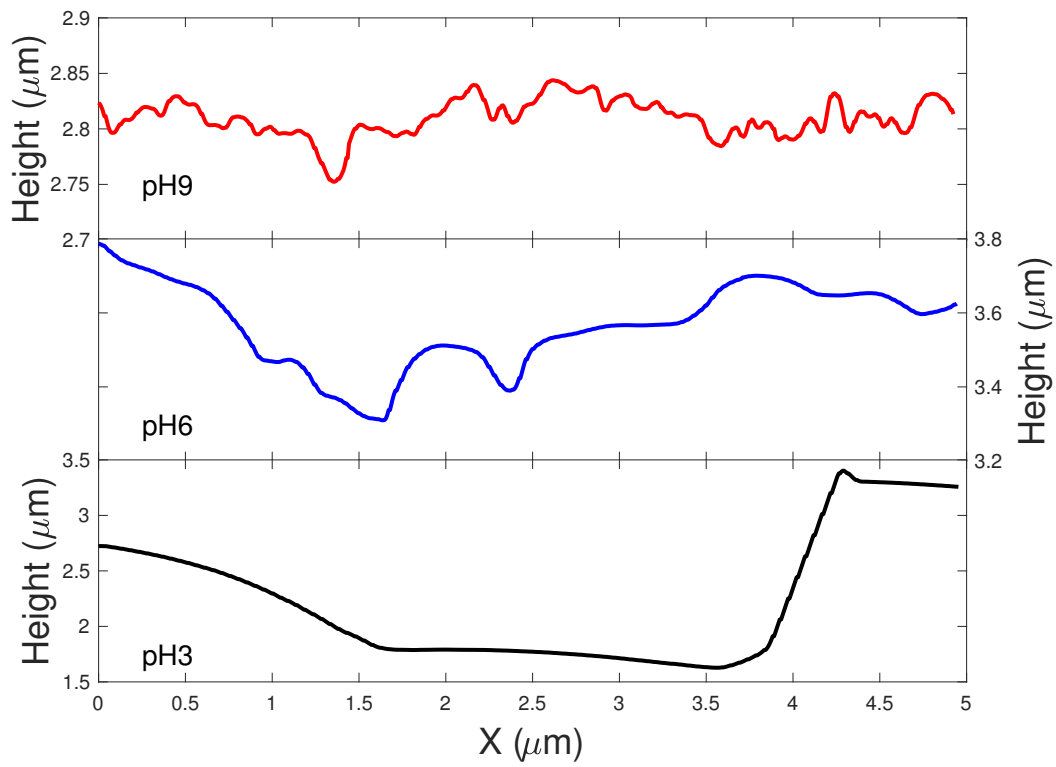


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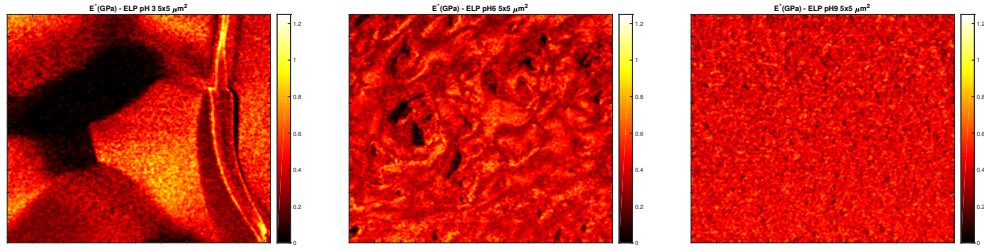


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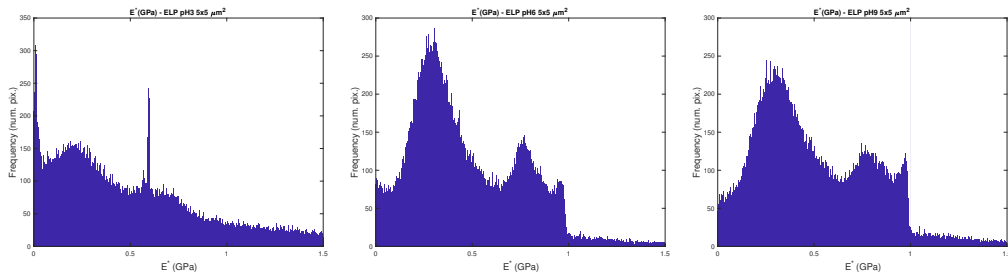


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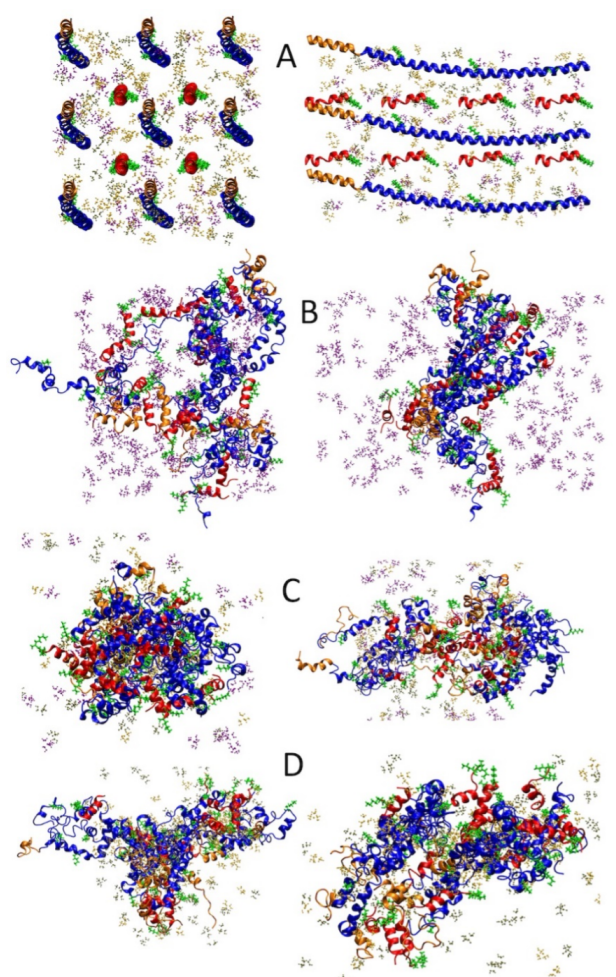


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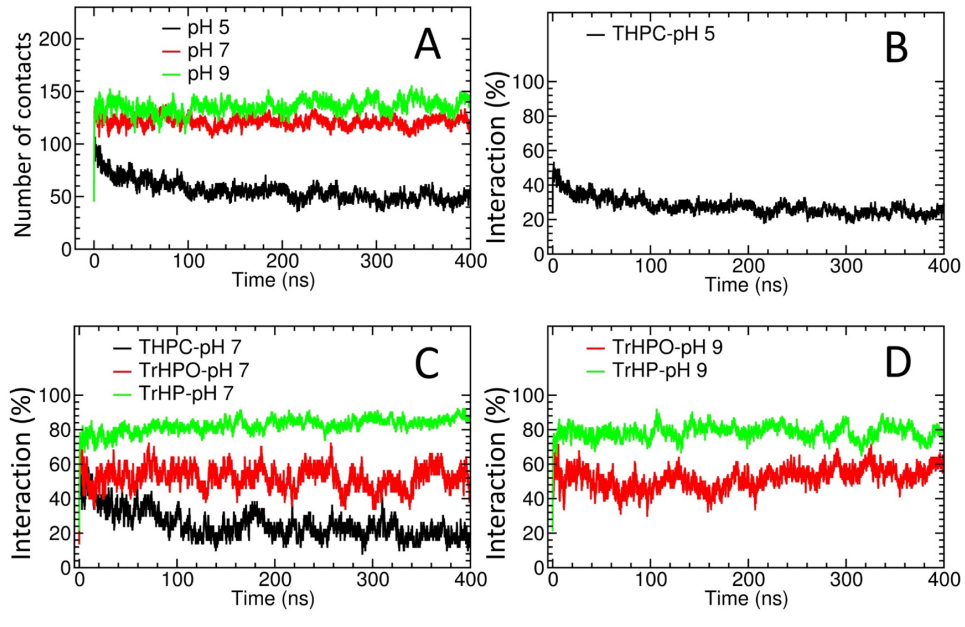


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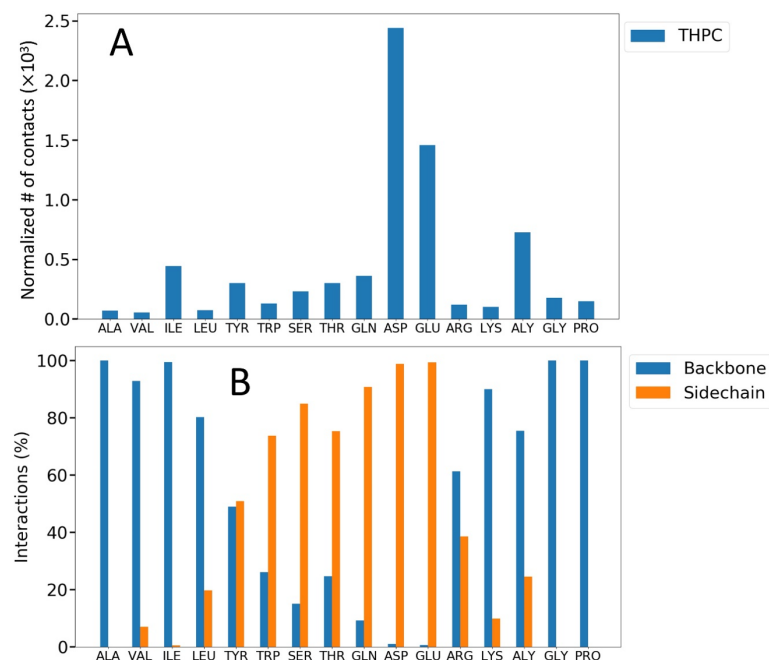


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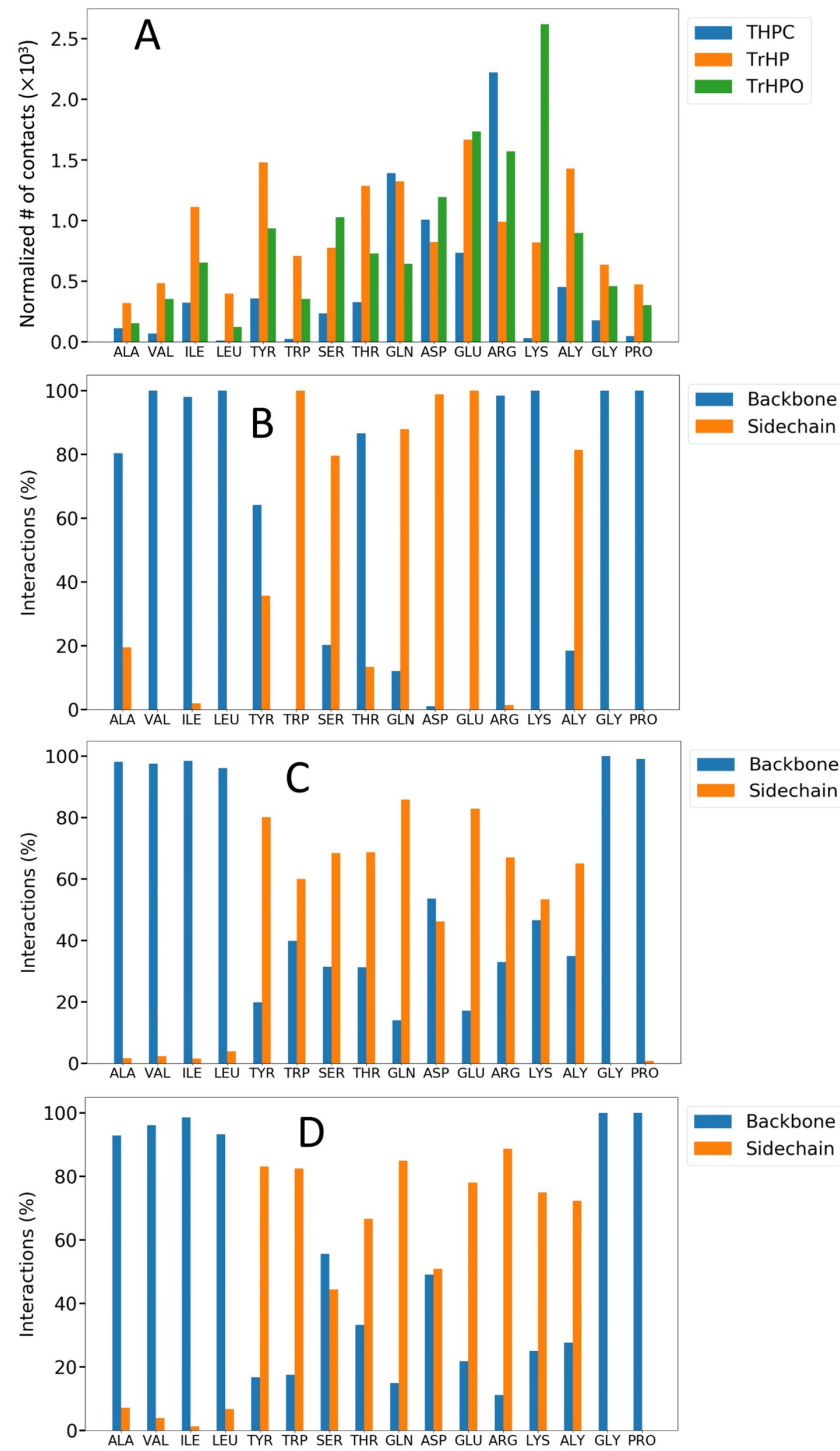


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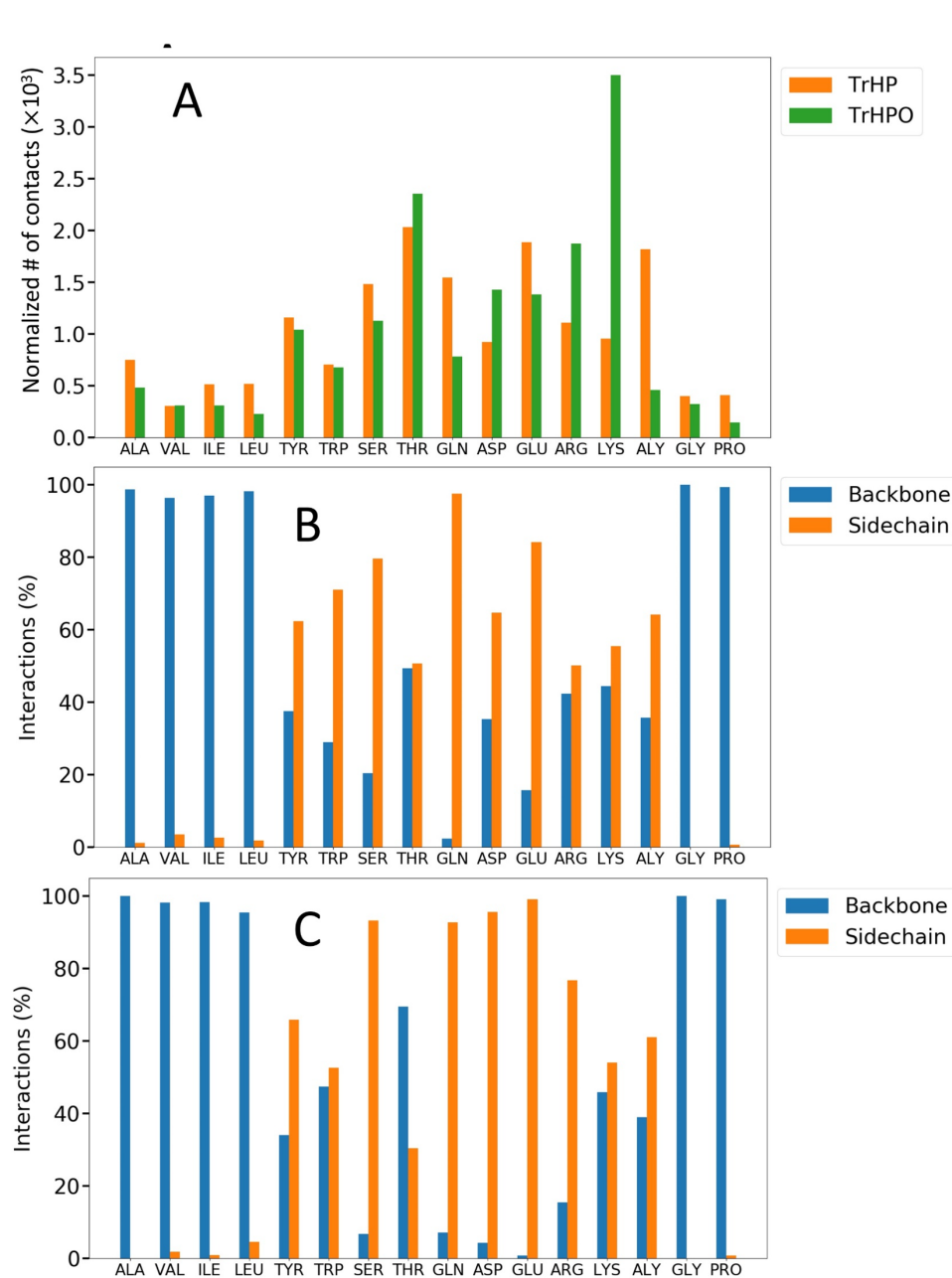


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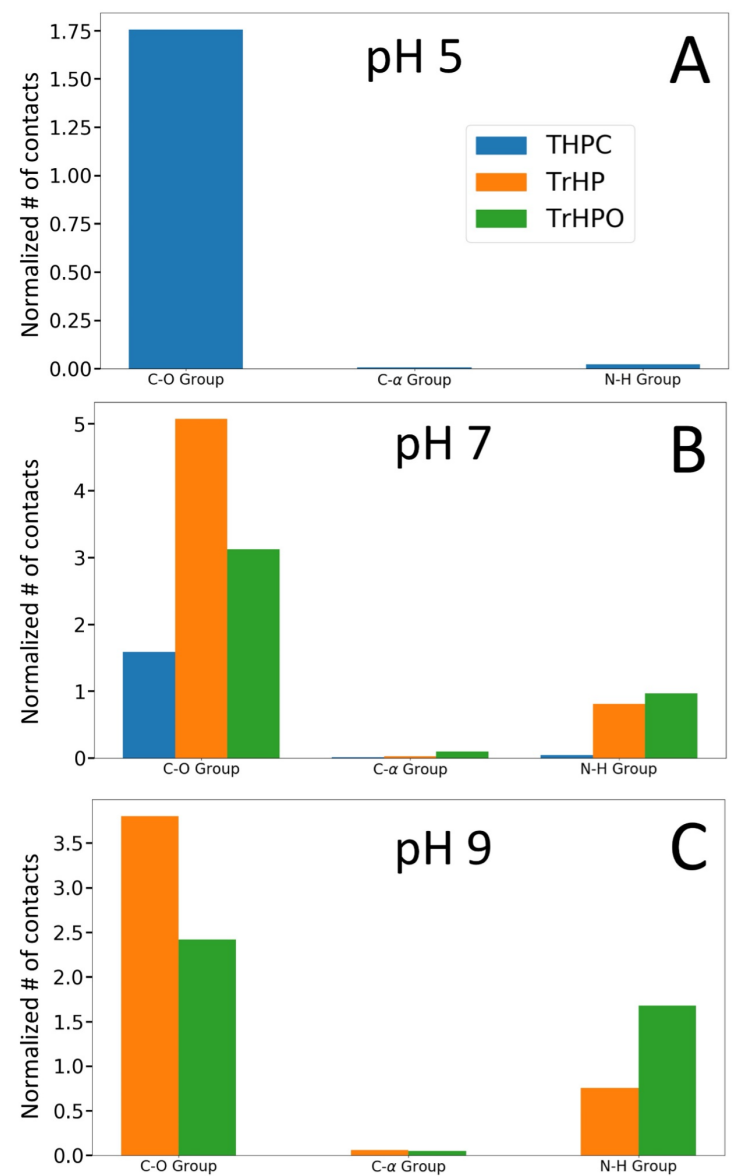


Figure 12: Number of contacts between different cross-linkers with different parts of protein backbone in different pH values. A) pH5, B) pH7 and C) pH9. Number of contacts are normalized by total number of the cross-linker.

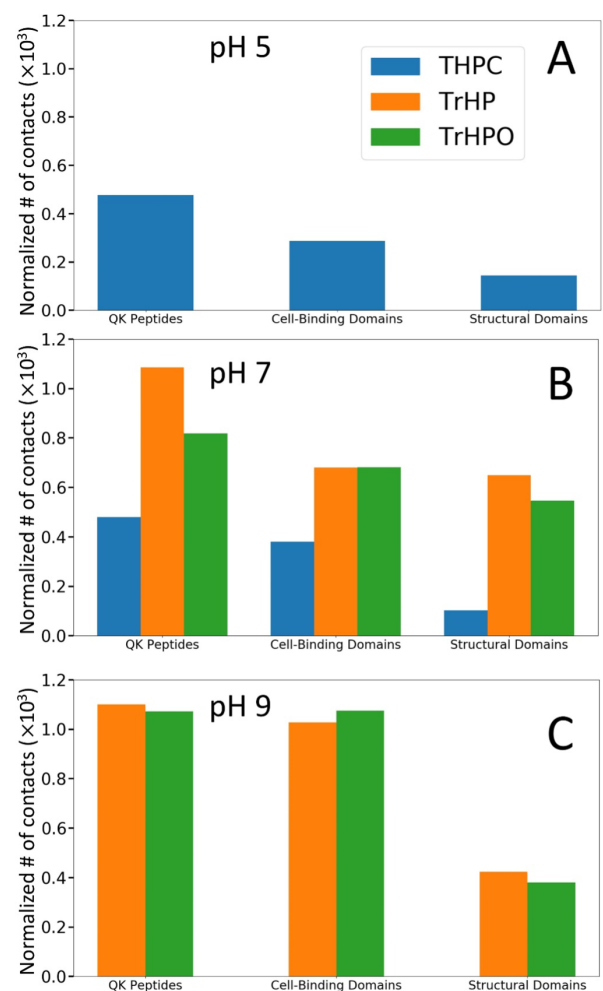


Figure 13: Number of contacts between different cross-linkers and different proteins in the system at different pH values. A) pH5, B) pH7 and C) pH9. Each cross-linker interaction is normalized by total number of the cross-linker. For each peptide, the number of interactions is normalized by number of the same type peptides, as well as the number of amino acids on each peptide.

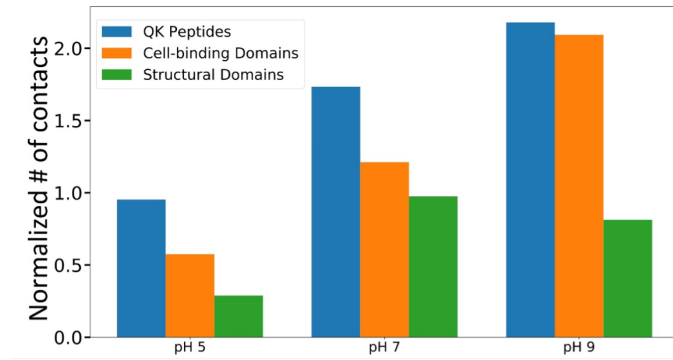


Figure 14: Number of contacts between all cross-linkers and different proteins in the system at different pH values. The values are normalized by total number of the cross-linkers and total the number of amino acids on each peptide.

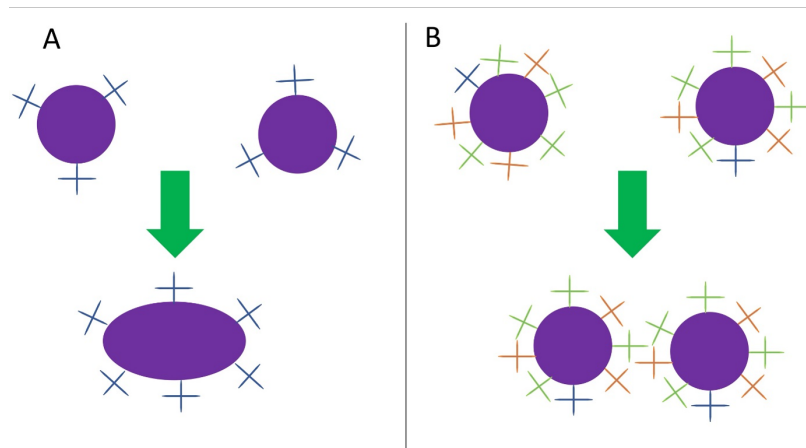


Figure 15: Schematics on the role of cross-linkers and proteins in producing the ELP configurations in different pH values. At low pH values (A), lower number of cross-linkers have covered the protein, while at higher pH values, (B) a greater number of cross-linkers have covered the protein. The aggregated proteins are shown as purple circles. The cross-linkers are shown as cross sign in different colors..