1	pH-responsive biopolymer-based supramolecular architectures as biodegradable carriers for						
2	DOX delivery						
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#### 28 Abstract

29 Cellulose is the most abundant renewable biomaterial on earth and beta-cyclodextrin ( $\beta$ CD) is among 30 the most commonly used biocompatible drug encapsulation agents. Combining these bio-organic 31 materials is a very powerful approach to greatly enhance the bioavailability of many drugs. These 32 systems also allow for optimal selective drug release profiles, high biocompatibility, as well as "green 33 nanomedicine" approaches that are eco-friendly in their synthesis and have minimal ecological 34 toxicity. Herein, we designed a new type of green trifunctional biopolymer-based nanosponge drug 35 carriers polymerized by crosslinking beta-cyclodextrin ethylene diamine (BCD-EDA) with 36 bifunctional hairy nanocellulose (BHNC). In this system, hairy nanocellulose particles is considered 37 as the backbone, and the immobilized cyclodextrin cavities capture the DOX (molecules) via host 38 guest inclusion complexation. BHNC contains aldehyde and carboxyl groups and the latter can react 39 with amino groups in  $\beta$ CD-EDA. Firstly, the crosslinker  $\beta$ CD-EDA was obtained from a simple 40 nucleophilic substitution reaction between beta-cyclodextrin carbonyl imidazole (BCD-CI) and 41 ethylene diamine (EDA). Secondly, BHNC was crosslinked by βCD-EDA through a facile 42 nucleophilic substitution crosslinking reaction of the BHNC activated carboxyl groups by the amine 43 groups on  $\beta$ CD-EDA. We refer to the highly crosslinked polymerized BHNC- $\beta$ CD-EDA network as 44 BBE. Various ratios of  $\beta$ CD-EDA and BHNC were polymerized with the help of DMTMM as an 45 activator, which resulted in different morphological shapes of BBE, and thus different release profiles 46 and pH-responsiveness were obtained for the BBE polymers. Unlike other polymer-based  $\beta$ CDs and 47 nanosponges, these new types of crosslinked polymer were prepared in a green and safe solvent 48 (water) and with very short reaction times and at room temperatures. Finally, the BBE polymers were 49 tested as biocompatible nanocarriers for controllable doxorubicin (DOX) delivery. These hyper 50 crosslinked polymers show a high capacity for loading DOX with extended DOX release.

51	Furthermore, breast cancer cell cultures show lower cell viability when DOX was loaded in various										
52	BBEs compared to control samples or DOX alone, indicating that our DOX-BBE drug delivery										
53	systems are better anticancer agents than DOX alone.										
54											
55	Keywords: biodegradable and biocompatible carriers, liquid crystals, nanosponges, bifunctional										
56	hairy	nanocellulose,	beta-cyclodextrin,	hyper	crosslinked	polymer,	DOX	delivery,			
57	biomac	cromolecules.									
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#### 70 **1. Introduction**

71 In the evolving landscape of small molecule therapeutic use, there has been recent emphasis 72 highlighting the importance of small molecules for chemotherapy in combination with several contemporary drug immunotherapy<sup>1-3</sup> and hormonal therapy<sup>4-5</sup> regimens for treating a wide array of 73 74 cancers. On the grounds of this increased therapeutic use, overcoming the limited bioavailability of 75 the small molecules has become the most important role of drug delivery systems since the majority 76 of these drugs are insoluble in body fluids. The drug delivery system morphology and surface material 77 enable improved control of the drug delivery. Predictable drug release kinetics can be optimized by altering the carrying capacity<sup>6-8</sup>, limiting off-target toxicity<sup>9</sup>, improving cellular uptake<sup>10</sup>, and 78 adjusting pharmacokinetics<sup>11</sup> adapted to the unique physiochemical properties of various small 79 80 molecules. A major concern in the field of drug delivery systems is the biodegradability of these 81 systems as they raise many environmental toxicology problems including various non-sustainable 82 synthetic requirements, bioaccumulation, environmental persistence, and the unknown interactions of these molecules in water and air ecosystems<sup>12-18</sup>. Thus, new nanotechnologies rely primarily on 83 84 biodegradable polymers in order to avoid environmentally harsh drug delivery systems<sup>19-21</sup>. These 85 polymers can be synthesized by exploiting green engineering to fabricate innocuous renewable, 86 biodegradable, biocompatible compounds. The usage of green engineering in concordance to the 87 environmentally-friendly synthetic principles of green chemistry gave birth to the field of "green 88 nanomedicine" with eco-friendly drug delivery systems with optimal drug loading and release<sup>22</sup>. Furthermore, there are various potent anticancer drugs from plant sources<sup>23</sup> and the incorporation of 89 90 novel environmentally friendly drug delivery systems can potentially make the growing field of 91 anticancer small molecules fully eco-friendly.

92 Nanocellulose fibers have been promising candidates among the polymers used for developing environmentally-friendly drug delivery systems<sup>24</sup>. The desirable drug delivery properties of 93 94 nanocellulose including high availability of surface area, mechanical reinforcement, adjustable size 95 and versatile surface functionalization are crucial for nano-scale bioavailability. These properties for 96 enhanced anticancer drugs have been increasingly utilized, but are found to be limited with much room to grow<sup>25</sup>. Cellulose nanocrystals (CNC) are stable colloidal dispersions that are rod-shaped 97 98 and are extracted from nanocellulose fibers. Nanorod formulations have more desirable drug delivery properties compared to nanospheres including tumor accumulation<sup>26</sup>, prolonged half-life due to lower 99 uptake by macrophages<sup>27</sup>, and tumor homing due to unique rheology<sup>28</sup>. Hence, cellulose-based 100 101 nanorod formulations have been advantageously used for drug delivery<sup>29</sup>. In order to resolve the low 102 capacity of surface functionalization of CNC, hairy nanocelluloses (HNCs) have been produced by 103 periodate oxidation which solubilizes the amorphous regions, and, at the same time, allows for the 104 cleavage of cellulose chains in the crystalline regions leaving the soluble amorphous regions attached on both ends<sup>30-31</sup>. HNCs have dialdehyde-modified cellulose chains on both sides of the crystalline 105 106 region and by oxidizing some of the aldehyde groups on the amorphous regions of HNC, bifunctional 107 hairy nanocelluloses (BHNCs) can be synthesized, containing both aldehyde and carboxylic groups, further increasing the functionalization capacity of these nanocelluloses<sup>32</sup>. These cellulose derivatives 108 109 are optimal guest polymers for the threading of Beta-Cyclodextrins (BCDs) supramolecular inclusion 110 complexes and they have attracted great attention in recent years for developing advanced supramolecular materials with outstanding self-assembly and structural stability<sup>33-34</sup>. βCDs are cyclic 111 112 oligosaccharides that consist of 7 glucopyranose units that form hydrophobic cavities and externally 113 hydrophilic surfaces. The use of  $\beta$ CD nanoparticles for drug delivery has been extensive in research 114 as they have shown to increase the bioavailability of many guests primarily due to the advantage of 115 using the well-known guest-host chemistry of isolated CDs for unique complexation of different 116 guests. The supramolecular inclusion complex hides the hydrophobic host functionality in the interior 117 cavity while the external surface hydroxyl groups of CD remain exposed to the aqueous environment. 118 Functionalizing the alcohol groups of  $\beta$ CDs with several molecules (including ethylene diamine 119 (EDA)) is known to enhance their physiochemical properties and can enable their cross-linking<sup>35</sup>. 120 The unique supramolecular inclusion complexes of cyclodextrins makes them the stepping stone for 121 the development of  $\beta$ CD-based polymers that largely optimize drug delivery depending on the drug's physiochemical properties. For example,  $\beta$ CDs were functionalized on hydrogels<sup>36-38</sup> and 122 nanosponge<sup>38</sup> polymers that change the morphology of  $\beta$ CD making them less soluble and easily 123 124 extractable. BCD can be substituted onto various polymers and CD-CD linkers can crosslink the 125 polymers enabling hyper crosslinked BCD-based polymers. This has been shown to be more 126 beneficial than the use of linkers alone to crosslink polymers since the hyper crosslinked  $\beta$ CD-based polymers have a larger specific surface area with more framework nanoporosity<sup>39</sup>, allow better 127 128 adsorption due to the accessible binding sites on the template. Moreover, these  $\beta$ CD-based polymers 129 are observed to be thermally stable<sup>40</sup>. βCD-based polymers can host hydrophobic compounds in their 130 internal cavity as well as hydrophilic compounds in the interstitial pores formed by the cross-linkers 131 and the external CD walls. Importantly, most  $\beta$ CD polymers have predictable release schedules due 132 to the polyester being predictably biodegradable<sup>41</sup> which puts them at an advantage compared to 133 nanoparticles. The release is usually relatively very slow since the polymer network functionalized 134 around the BCD cavity hampers the diffusion of entrapped guest molecules, thus promoting slower release kinetics<sup>42</sup>. The biodegradability of  $\beta$ CD polymers makes them candidates for use in the 135 remediation of micropolluted water environments and ultrafast water treatment<sup>43-44</sup>. Combining  $\beta$ CDs 136 137 with nanocellulose is highly beneficial for drug delivery due to the observed synergy between the

138 two materials for excellent biocompatibility, low cytotoxicity, biodegradability, efficient polymerization, and versatile functionalization<sup>33</sup>. In the present work, we hypothesize that BHNC 139 140 can provide a colloidally stabilized structure and a template with a high carrying capacity for  $\beta$ CD-141 DOX supramolecular inclusion complexes forming green polymers with controllable drug delivery. 142 In summary, we have developed a new rapid and green route to synthesize water-soluble and water-143 insoluble nanosponges using biodegradable materials. Here, nanosponges are defined as large 144 nanoporous structures which are compressible and can adsorb water (or another solvent). 145 Bifunctional hairy nanocellulose and beta-cyclodextrin materials have been crosslinked to each other 146 with amide bonds. The extent of  $\beta$ CD grafting on BHNC will determine the solubility of the polymer 147 along with the drug loading and release profiles. In this system, hairy nanocellulose particles are 148 considered to be the backbone, and the immobilized cyclodextrin cavities capture the DOX 149 (molecules) via host guest inclusion complexation. Moreover, the extent of cross-linking can be used 150 for varying the morphology of the hyper-crosslinked polymer altering the release profiles and the 151 solubility of the delivery system. Herein, we report, to the best of our knowledge, the first study that 152 aims to use BHNC-BCD-EDA polymers for drug delivery aiming to expand the applications of 153 BHNC. The polymerization of  $\beta$ CD-EDA with BHNC is fully characterized. This study advances the 154 understanding of BHNC- $\beta$ CD enhanced release profiles and its solubilizing capacity.

155 **2. Experimental section** 

#### 156 **2.1. Materials and methods**

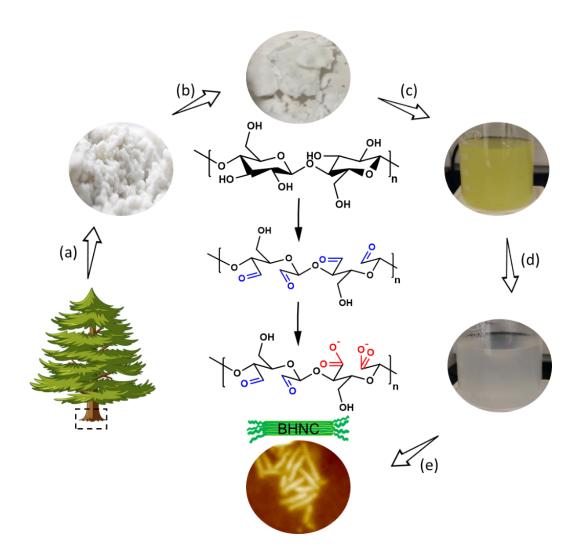
Milled softwood kraft pulp sheets (Domtar, Inc., Canada) were used as precursors to the synthesis of
BHNC. Sodium (meta) periodate (NaIO<sub>4</sub>), sodium chloride (NaCl), sodium chlorite (NaClO<sub>2</sub>, 80%),
hydrogen peroxide solution (H<sub>2</sub>O<sub>2</sub>, 30%), ethylene glycol for oxidizing softwood pulp and
hydroxylamine hydrochloride, sodium hydroxide (NaOH), and hydrochloric acid (HCl, 36.5-38.0%)

161 solutions for measuring the aldehyde content were purchased from Sigma-Aldrich. Beta-cyclodextrin 162 (C<sub>42</sub>H<sub>70</sub>O<sub>35</sub>, 99.9% (ELSD)) were purchased from AK Scientific. 1,1'-Carbonyldiimidazole 163  $(C_7H_6N_4O, 99.9\%)$ , Ethylenediamine  $(C_2H_4(NH_2)_2, 99.9\%)$  for the synthesis of  $\beta$ CD-EDA from  $\beta$ CD, 164 and DMTMM (C<sub>10</sub>H<sub>17</sub>ClN<sub>4</sub>O<sub>3</sub>, 96%) as a carboxyl activator, were obtained from Sigma-Aldrich. 165 Anhydrous dimethylformamide (HCON(CH<sub>3</sub>)<sub>2</sub>, 99.99%), anhydrous ethanol ( $C_2H_5OH$ , 99.5%), and 166 anhydrous acetone ((CH<sub>3</sub>)<sub>2</sub>CO, 99.8%) as solvents were purchased from Thermo Fisher Scientific. 167 All chemicals were used as received and deionized (DI) water (resistivity of ~10 M $\Omega$  cm at 25 °C) 168 was used in all experiments. Doxorubicin hydrochloride ( $C_{27}H_{29}NO_{11}$ ) and phosphate-buffered saline 169 (PBS, pH 7.4) for drug delivery experiments were supplied by Sigma-Aldrich were used in all 170 experiments except where specified otherwise. Fetal bovine serum and penicillin/streptomycin 171 solutions were purchased from Wisent bioproducts. Calcein-AM and Hoechst 33242 were purchased 172 from AAT Bioquest and TOCRIS Bioscience, respectively. The human breast cancer cell line MDA-173 MB-231 was purchased from ATCC.

#### 174 **2.2. Preparation of bifunctional hairy nanocellulose (BHNC)**

175 For the synthesis of BHNC, 12 g of milled softwood pulp was oxidized in 700 mL of a solution 176 containing 16 g of NaIO<sub>4</sub> and 46.8 g NaCl at room temperature for 42 h under continuous stirring. 177 The reaction beaker was wrapped with aluminum foil to prevent the deactivation of periodate, 178 according to the procedure of Heidari Nia *et al.* with applying little changes<sup>32</sup>. 10 mL of ethylene 179 glycol was added to this mixture. Ethylene glycol was used to quench the excess amount of periodate 180 in order to stop the oxidation reaction. The product of the oxidation reaction, which is called 181 dialdehyde modified cellulose (DAMC), was thoroughly washed with water using vacuum filtration. 182 The aldehyde content of DAMC was measured by the hydroxylamine hydrochloride titration method 183 as described in section 2.5.2. To convert half of the aldehyde groups to carboxylic groups, 6 g of as-

184 prepared never-dried DAMC was stirred at room temperature for 24 h with 7.5 g of NaClO<sub>2</sub>, 24 g of 185 NaCl, and 2.5 g of H<sub>2</sub>O<sub>2</sub> in 600 mL of water. During this reaction, the conditions of the reaction were 186 monitored by observing the color of the solution, with changed from dark fluorescent yellow to light 187 yellow. Dark fluorescent yellow signifies that 0.1 M NaOH needs to be added, light yellow 188 fluorescence means that the reaction is proceeding under optimal conditions, and a white color 189 indicates that the reaction is completed. The oxidized pulp solution was filtrated with centrifugation 190 at 8000 rpm for 10 min to remove the trace amounts of non-fibrillated fibers. The BHNC particles 191 appeared in the form of a white precipitate after adding isopropanol 1.5 times the weight of 192 supernatant. The precipitate was collected and purified through dialysis against DI water in 8-10 kDa 193 dialysis bags with daily water exchanges for 3 days. The resulting neutral BHNC suspension was 194 further dispersed by an ultrasonic processor (Hielscher UP200H, Germany) for 1 min. The synthesis 195 of BHNC, used as a co polymer in this study, is shown in Scheme 1.



**Scheme 1.** Schematic illustration of the synthesis of the bifunctional hairy nanocellulose (BHNC) from cellulose nanofibrils by oxidation. a) Preparation of milled softwood pulp from wood, b) dialdehyde modified cellulose (DAMC) was prepared trough periodate oxidation, c) second oxidation process with sodium chlorite, d) monitoring the reaction by observing the color of the solution, with changing from dark fluorescent yellow to light yellow, and e) preparation of hairy nanocellulose after centrifugation and dialyzing.

#### 196 **2.3. Synthesis of hyper crosslinker βCD-EDA**

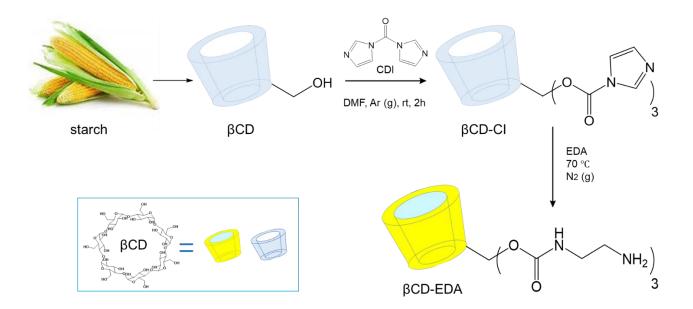
197  $\beta$ CD-CI. A simple nucleophilic substitution reaction method between 1,1'-carbonyldiimidazole 198 (CDI) and hydroxyl groups on the surface of  $\beta$ CD, resulted in the formation of  $\beta$ CD-CI. Briefly 1.64 g

199 of  $\beta$ CD and 1.96 g of CDI were dissolved in 30 ml DMF and then subjected to stirring with argon

200 protection at room temperature for 2 h. The white product obtained after the reaction was precipitated 201 in cold diethyl ether and extracted with centrifugation. The resulting precipitate  $\beta$ CD-CI was washed 202 several times with acetone. The white solid obtained by centrifugation was dried in a vacuum oven 203 for 8 h to obtain a white powdery solid (1.43 g, yield: 87.2%).

204  $\beta$ CD-EDA. Amine groups have been introduced on the surface of  $\beta$ CD through carbamate bonds 205 between EDA and  $\beta$ CD-CI. In this synthesis, 1 g  $\beta$ CD-CI was dissolved in a high amount of EDA 206 (40 mL) as a solvent in order to prevent the polymerization of  $\beta$ CD-EDA as EDA is the attacking 207 material (nucleophile) of the carbonyl bond of βCD-CI. The solution was stirred at room temperature 208 for 30 min under nitrogen protection, the final reaction mixture was heated to 40 °C under reflux 209 conditions and held for 24 h with nitrogen protection. The resultant vellow precipitate was purified 210 with the same method mentioned above and recrystallized with ethanol (0.92 g, yield: 92%). A 211 schematic representation of synthesis of  $\beta$ CD-EDA and  $\beta$ CD-CI are shown in Scheme 2.

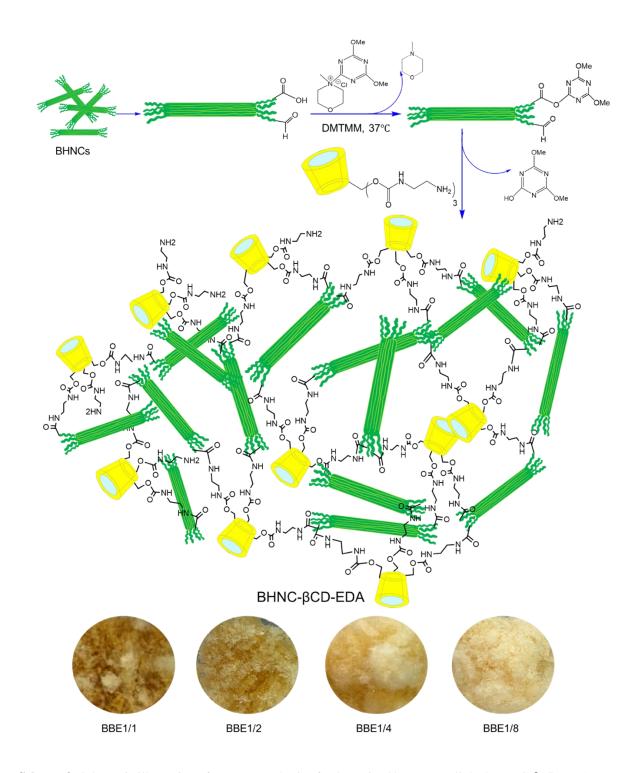
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Scheme 2: Schematic illustration of the preparation of the hyper crosslinker βCD-EDA.

# 213 2.4. Preparation of water soluble and water insoluble polymerized hyper crosslinked BHNC 214 βCD-EDA (BBE)

Cellulose and  $\beta$ CD as bio-organic materials are good candidates for biocompatibility and sustainable 215 216 drug delivery owing to the presence of a free amino groups per sugar unit in the hyper crosslinker 217 βCD-EDA and free carboxyl groups in BHNC. So, the design of a natural polymerized hyper 218 crosslinked BHNC-βCD-EDA (BBE) carrier is achieved through a facile crosslinking reaction with 219 nucleophilic substitution of activated carboxyl groups of BHNC by amine arms of BCD-EDA 220 (Scheme 3). BCD-EDA was wrapped around the BHNCs with amide bonds with different 221 morphological shapes forming pH-responsive polymers. The hyper crosslinker  $\beta$ CD-EDA was 222 polymerized with BHNC particles with the help of DMTMM as an activator reagent in water at low 223 temperature. DMTMM is considered a green activator that can be extracted after the activation of the 224 carboxylic content. Unlike other polymer-based  $\beta$ CDs and  $\beta$ CD-based nanosponges, these new types 225 of crosslinked polymer were prepared in a green and safe solvent (water) and with very short reaction 226 times and at low temperatures. The reaction was started by activating different ratios of the carboxyl 227 content of BHNC with DMTMM (100% (1/1; activated carboxyl content to DMTMM), 50% (1/2), 228 25% (1/4) and 12.5% (1/8)). Briefly, different amounts of DMTMM (37.2, 18.6, 9.3 and 4.65 mg) 229 were added to the 7.5 mL BHNC suspension (3.8 mg/mL with 4.5 mmol/g carboxyl content) and 230 stirred for 5 min at 37 °C and then, the desired amount of  $\beta$ CD-EDA (164, 82, 40 and 20 mg) was 231 dispersed in 20 mL water and added to the activated BHNC solutions. The resulting solutions were 232 stirred for 1 h at 37 °C. After less than 5 minutes. For BBE1/1 and BBE1/2, a yellow solid was 233 produced while for the other polymers (BBE1/4 and BBE1/8) the solution remained transparent 234 yellow. Afterwards, all the solutions were purified through dialysis against DI water in 8-10 kDa 235 dialysis bags with daily water exchanges for 3 days. The obtained solutions were lypholized and the 236  $\beta$ CD-EDA content was measured gravimetrically and with a phenolphthalein assay (see figure S1).



Scheme 3. Schematic illustration of one-pot synthesis of polymerized hyper crosslinked BHNC- $\beta$ CD-EDA (BBE).

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#### **2.5.**Characterization of BHNC, hyper crosslinker βCD-EDA, water soluble and water 239 insoluble polymerized hyper crosslinked BHNC-BCD-EDA (BBE)

#### 240 Morphological characterization

241 **Transmission electron microscopy (TEM).** The morphology and particle size of BHNC were 242 investigated using TEM. The images were obtained with a Philips Tecnai 12, operating at 120 kV, 243 equipped with a Gatan 792 Bioscan  $1000 \times 1000$  wide angle multiscan charge-coupled device (CCD) 244 camera. The TEM specimen were prepared from a suspension of the samples with a concentration of 245 0.1 % (w/w) in the corresponding particular solvent. A drop of the suspension was placed on a copper 246 grid coated by a thin carbon film and the drop was left to evaporate.

247 Field emission scanning electron microscope (FE-SEM) The morphology of the  $\beta$ CD-EDA and 248 BBE hyper crosslinked polymers were characterized using FE-SEM. The samples were prepared by 249 dropping the solution onto a coverslip, followed by evaporating the liquid in air and coating with a 3 250 nm thick layer of Pt by a high-vacuum coater (Leica EM ACE600). FE-SEM images were recorded 251 on a Q450 ESEM (FEI quanta 450 environmental scanning electron microscope). The images were 252 taken at an accelerating voltage of 7 kV.

253 Atomic force microscopy (AFM). The surface morphology and size of BHNC nanoparticles were 254 recorded by AFM with Nanoscope IIIa controller (Digital Instrument/Veeco, Santa Barbara, CA). To 255 prepare the AFM sample, a freshly cleaved mica was inclined to a magnetic disc by double-sided tape and coated with 0.1% poly-l-lysine (PLL). After 5 min, the excess PLL was rinsed off with Milli-256 257 Q water. A drop of BHNC suspension with concentration of 0.1 mg/mL was placed on a freshly 258 cleaved mica surface for 10 min, followed by washing off the excess liquid with Milli-Q water. 259 Before measurement, samples were air-dried overnight. The experiments were conducted in tapping

260 mode with a silicon probe (Nanoworld; type: SSS-NCH) with force constant of 42 N/m, resonance 261 frequency  $\sim$ 320 kHz, and nominal tip radius  $\sim$ 5 nm.

#### 262 Aldehyde content of DAMC and BHNC

263 The aldehyde content of DAMC and BHNC was determined using the hydroxylamine-hydrochloride titration method that was introduced in our previous work<sup>32</sup>. Shortly, the exact amount of oxidized 264 sample was suspended in water and the pH was adjusted to 3.5 using HCl. Afterwards, the pH of the 265 266 10 mL hydroxylamine-hydrochloride solution (5% w/w) was adjusted to 3.5 using NAOH solution 267 which was added to the sample suspension under vigorous stirring until the oxime and HCl form and 268 consequently, the pH drop was observed. The resulting HCl was titrated with NaOH solution (0.1 M) 269 until the pH becomes constant at 3.5. The aldehyde content was calculated using the following 270 formula:

271 Aldehyde content (mmol 
$$g^{-1}$$
) =  $V_{\text{NaOH}} \times N_{\text{NaOH}} / W_{(\text{DAMC or BHNC})}$ 

where  $V_{NaOH}$  is the volume of titrant (mL),  $N_{NaOH}$  is the normality of titrant (mol L<sup>-1</sup>), and W is the weight of dry samples (g) initially suspended.

#### 274 Carboxylic content of BHNC and amide content of βCD-CI

The carboxylic content of BHNC and the amide content of  $\beta$ CD-CI were measured with conductometric titration. Briefly, 5 mL of BHNC solution (2 mg/mL) or a certain amount of  $\beta$ CD-CI were added to a 100 mL milli-Q water and 1 mL NaCl solution (20 mM). The pH of the solution was adjusted to 3.5 by adding HCl (0.1 M) and the mixture was moderately stirred. Thereafter, using titrando titrator (Metrohm836, Switzerland), 10 mM NaOH solution were added to the solution with a rate of 0.1 mL/min until the pH reached to around 11 (see figure S2 and S3). The carboxylic content was determined from the following equation: 282 Carboxylic content (mmol  $g^{-1}$ ) =  $V_{NaOH} \times N_{NaOH} / W_i$ 

where  $V_{NaOH}$  is the volume of titrant (mL),  $N_{NaOH}$  is the normality of the titrant (mol L<sup>-1</sup>), and W<sub>i</sub> is W<sub>BHNC</sub> or W<sub>βCD-CI</sub>, being the weight of BHNC or βCD-CI used initially (g).

#### 285 Phenolphthalein (PHTH) assay for βCD grafting ratio of BBEs

286 In order to measure the amount of active  $\beta$ CD grafted on BHNC, phenolphthalein was used as a 287 quantitative alkaline calorimetric indicator that could insert into the active  $\beta$ CD cavities. Briefly, 288 increasing amounts of  $\beta$ CD were added to the PHTH solution containing 0.02M NaCO<sub>3</sub> buffer at pH 10.5 with freshly prepared  $0.5 \times 10^{-3}$  g.mL<sup>-1</sup> PHTH and the absorbance was measured by UV–Vis 289 290 spectrophotometry at 550 nm corresponding to the deprotonated PHTH in the  $\beta$ CD cavity. Thereafter, 291 a standard curve was plotted as absorbance vs  $\beta$ CD concentration in g/L (see Figure S1a). 10 mg of 292 each of the BBE polymers was added to 25 mL of the phenolphthalein solution, stirred, and 293 equilibrated overnight and the absorbance was measured and the BCD concentration in the polymers 294 was determined by referring to the standard curve. PHTH does not bind to the BHNC pristine and 295 the measured absorbance would hence only reflect PHTH in the  $\beta$ CD cavities<sup>45</sup>. The  $\beta$ CD content of 296 the BBE polymers was determined using the following equation:

297 
$$\beta \text{CD content (mmol } g^{-1}) = \frac{V C_{\beta \text{CD}}}{m M_{\beta \text{CD}}}$$

Here V is the volume of the solutions (mL),  $C_{\beta CD}$  is the concentration of  $\beta CD$  obtained from the calibration curve (g L<sup>-1</sup>), m is the mass of the BBE samples (g), and  $M_{\beta CD}$  is the molar mass of  $\beta CD$ (g mol<sup>-1</sup>).

#### **301 Amine content of βCD-EDA**

The amine content of βCD-EDA was measured using a titration. Firstly, 5 mg BHNC were dissolved in 100 mL milli-Q water. The pH of the solution was adjusted to 9.5 by adding NaOH (0.1 M) under vigorous stirring. Thereafter, using a titrando titrator (Metrohm836, Switzerland), 10 mM HCl solution was added at a rate of 0.1 mL/min until the pH reached about 3.5 (see figure S4). The amine content of βCD-EDA was calculated by the following equation:

307 Amine content (mmol 
$$g^{-1}$$
) = V<sub>HCl</sub> × N<sub>HCl</sub> / W<sub>βCD-EDA</sub>

308 where  $V_{HCl}$  is the volume of titrant base (mL),  $N_{HCl}$  is the normality of the titrant base (mol L<sup>-1</sup>), and 309  $W_{\beta CD-EDA}$  is the weight of  $\beta CD$ -EDA that used initially (g).

## 310 Particle size measurements using dynamic light scattering (DLS) and Zeta-potential analysis

#### 311 by electrophoretic light scattering (ELS)

312 The equivalent spherical hydrodynamic diameter and polydispersity of the BHNC and BBE samples 313 were determined by a Brookhaven light scattering instrument (BI9000 AT digital correlator). In all 314 DLS experiments, the scattered light intensity was measured at a 90° scattering angle and at 25 °C. 315 For preparing the samples, 100  $\mu$ L of 0.1% (w/w) BHNC suspensions and BBE samples were added 316 to 900 µL of DI water in low-volume disposable cuvettes. The magnitude of the surface charges of 317 the particles was monitored using a Zetasizer Nano ZS (Malvern instrument, Lt.d., U.K.). To measure 318 the zeta potential data, the suspensions of BHNC and BBEs were diluted to 0.1% (w/w) with DI 319 water.

#### 320 Fourier transform infrared (FTIR) spectroscopy

Functional groups in the structure network of BHNC and BBEs were identified using a FTIR spectrometer (PerkinElmer, Inc., Waltham, MA, U.S.A.) equipped with a single bounce diamond attenuated total reflectance (ATR) cell. The completely dried samples were placed directly on the ATR crystal. All FTIR spectra were averaged from 32 scans from wavelengths 400 to  $4000 \text{ cm}^{-1}$  with a resolution of 4 cm<sup>-1</sup>.

#### 326 Surface area and pore size analysis

 $N_2$ -sorption was measured on a Tristar 3000 from Micromeritics, and the data for the BET plot fitting, specific surface area and pore size distribution of the particles were analyzed by Tristar 3000 software package V6.07. Since the results are expressed in units of surface area per gram of sample, the true sample mass must be known and hence, solid samples were meticulously weighted (0.1 g <weight <1 g), degassed at 120 °C and purged with N<sub>2</sub> for 24 h. The pore size distributions of the different BBE bio-polymers were calculated using the Barrett–Joyner–Hollande (BJH) model based on the adsorption and desorption branches of the isotherms.

#### 334 Thermal gravimetric analysis (TGA)

335 The thermal stability of BHNC,  $\beta$ CD, and BBEs was investigated with a thermal gravimetric analyzer 336 (Discovery 5500, TA Instruments, USA). To prepare the samples for analysis, around 5-10 mg were 337 placed on a little trace and automatically reweighted with TRIOS instrument. Heating temperature 338 was chosen between 20 to 1200 °C with a controlled heating rate of 0.1-1500 °C min<sup>-1</sup> in a controlled 339 atmosphere. The flow rate of high purity N<sub>2</sub> gas was 50 mL min<sup>-1</sup>.

#### 340 **Powder X-ray diffraction (PXRD)**

The presence of the BHNC crystalline region after polymerization with  $\beta$ CD-EDA with examined with XRD. The measurements were performed on a Bruker Discover D8 Discover two-dimensional diffractometer with VANTEC 2D detector and CuK $\alpha$  radiation ( $\lambda$ =1.54 Å). The X-ray diffractograms were acquired at 40 kV and 40 mA with a 2 $\Theta$  (Bragg angle) range of 10-30° at a scan rate of 0.005° s<sup>-1</sup>.

18

#### 346 Solid-State carbon-13 NMR

347 Solid carbon-13 NMR data was acquired using a VNMRS 400 widebore spectrometer operating at

348 399.9 MHz for <sup>1</sup>H and 100.5 MHz for <sup>13</sup>C in a 4 mm Varian Chemagnetics double-resonance probe.

349 The recycle delay was 4 s. The sample was spun at 8 kHz. The CP contact time was 2 ms, and a total

of 12000 scans (17 hours) were collected.

#### 351 Liquid-State hydrogen and carbon-13 NMR

352 The 13C NMR spectrum (126 MHz, D2O) was acquired using a Bruker AVIIIHD spectrometer using

a BBFO and Smart Probe. The <sup>1</sup>H spectra (500 MHz, D2O) were acquired using a Varian VNMRS

using a switchable broadband probe. 3000 scans were used for the <sup>13</sup>C spectrum and 16 for the proton,

#### 355 2.6. Preparation and characterization of DOX-loaded BBEs

#### 356 DOX loading

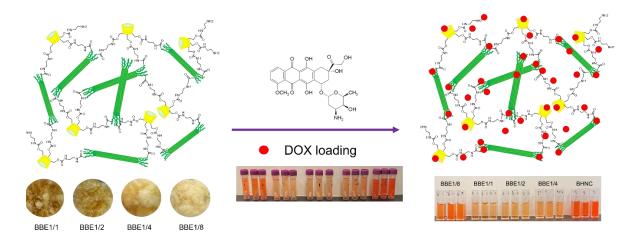
DOX loading of BHNC and BBEs was carried out by dispersing 10 mg samples in 1 mL ultrapure water and 500 µL DOX solution (10 g/mL). The final mixtures were kept in a dark area and incubated on a magnetic stirrer at 800 rpm overnight. Then, samples were collected using ultra-centrifugation (12000 rpm, 10 min) and after several washes, the supernatants were used to determine the DOXloading efficiency into the particles by UV–vis spectrophotometry at 480 nm. A calibration curve specific to DOX was prepared to convert the absorbance to DOX concentration. The amount of DOX loaded into the samples was calculated using the following equation (2):

$$q_e(\text{mg DOX/g samples}) = \frac{(C_0 - C_e)V}{m}$$
 (2)

- 364 where  $C_0$  is the initial concentration of DOX (mg/L),  $C_e$  is the concentration of DOX at equilibrium
- 365 (mg/L), V is the volume of DOX (L), and m is the mass of samples (g). The equilibrium DOX uptake
- $(q_e)$  is expressed in mg DOX per g samples.

#### 367 **Release profile**

*In vitro* release of DOX from the BHNC-DOX and BBEs-DOX was measured by UV-Vis. Briefly, a series of BHNC-DOX and BBEs-DOX were respectively dispersed in 1 mL of PBS (pH 5.5, 6.5 and 7.6) and vigorously stirred for 160 h at 120 rpm at 37 °C. After specified time periods, 1 mL release solution was taken out and replaced with an equal volume of fresh fluid. The release of DOX from PBS solution was monitored at 480 nm by UV–Vis. The procedure of drug loading and release is shown schematically in Scheme 5.



Scheme 5: Schematic illustration of DOX loading in pH-responsive BBE structures and BHNC as a control structure.

#### **Biocompatibility tests**

- 375 The toxicity of BBE and DOX loaded BBE was tested with MDA-MB-231 breast cancer cell lines.
- 376 MDA-MB-231 was cultured in DMEM medium with 10% fetal bovine serum, 100 U/mL penicillin,
- and 100  $\mu$ g/mL streptomycin at 37 °C and 5% CO<sub>2</sub> until 80% cellular confluency was attained. Cells

378 were harvested and 6,000 cells/well were seeded into a 96-well plate and cultured for 24 h. The 379 medium was then replaced by a fresh medium containing BBE-DOX prepared from ultrapure water, 380 containing a 10 mg/mL BBE1/4-DOX stock solution loaded with 835 µM doxorubicin and 15 mg/mL 381 or BBE1/1-DOX containing 732 µM loaded-doxorubicin. Our previous experiments show that 5.04 382  $\mu$ M DOX has a more remarkable toxic effect on MDA-MB-231 cells when exposed for 24 h<sup>29</sup>. A 383 final concentration of 5.04 µM BBE-DOX was used (based on BBE-DOX loading), and incubated at 384 37 °C and 5% of CO<sub>2</sub> for 24 h. As controls, we used MDA-MB-231 culture without BBE structures, 385 (BBE-Free (without doxorubicin), 10 mg/mL or 15 mg/mL stock solution for BBE1/4 and BBE1/1, 386 respectively), and doxorubicin (Dox, at 5.04 µM). Calcein-AM and Hoechst 33242 were used to 387 measure cell viability and nuclear staining, respectively. After incubation, the culture medium was 388 removed and the wells were washed three times with DPBS (Dulbecco's phosphate-buffered saline). 389 For viability, 100  $\mu$ L working solution composed of 1  $\mu$ L of Calcein-AM (4 mM) and 2  $\mu$ L of Hoechst 390 (18 mM) in 1 mL of DPBS was added per well and incubated for 45 min at 37 °C. Spinning-disk 391 confocal microscope Olympus IX83 was used to obtain the confocal microscopy images. By manual 392 counting the viability was determined using the following equation (3):

% of viability = 
$$\frac{\# \ live \ cells}{\# \ total \ cells} \times 100$$
 (3)

393 Ethidium homodimer I or propidium iodide were not used as dead cell markers since-doxorubicin394 interferes with the measurement. All experiments were performed in quadruplicates.

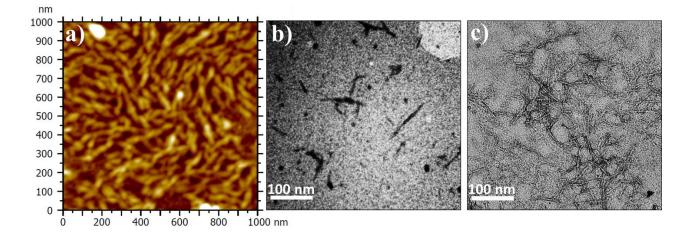
**395 3. Results and discussion** 

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# 397 3.1.Characterization of BHNC and bio-derived BHNC-βCD-EDA trifunctional polymerized 398 nanosponges (BBEs)

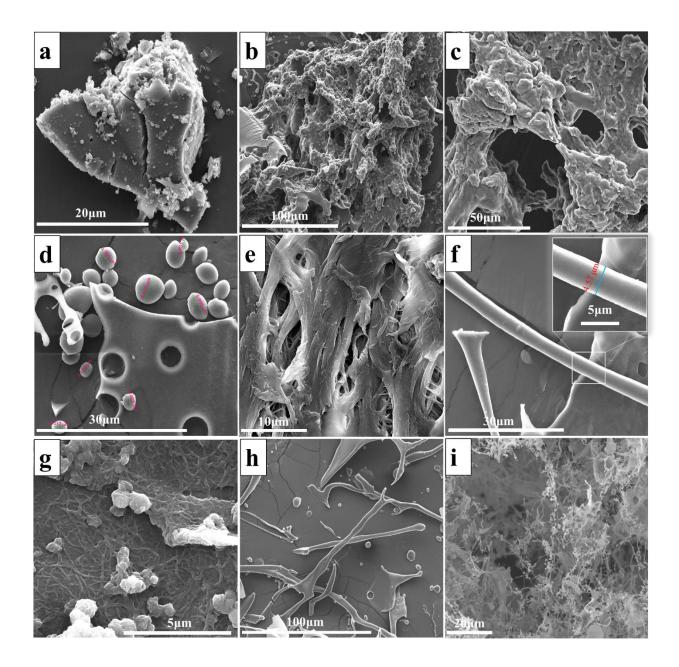
#### 399 Morphological, structural and dimensional analysis

400 The morphologies and the dimensions of rod-like BHNC particles and the different structural 401 morphologies of polymer-based BBEs in three different conditions (freeze-dried, non-freeze dried 402 and supramolecular inclusion complex with DOX) were evaluated with AFM, TEM, HRTEM and 403 FE-SEM imaging. BHNC nanorods were analyzed using both AFM (Figure 1a), TEM (Figure 1b) 404 and HRTEM (Figure 1c). TEM and HRTEM images of BHNC clearly show the typical rod-like 405 nanoparticles with a more or less uniform size and shape (Figure 1b and 1c). The BHNC polymer 406 template particles have, on average, a length of about 150 nm and width of about 5 nm. AFM images 407 of the BHNC particles shown in (Figure 1a) reconfirm the morphology and dimension properties of 408 BHNCs. DLS showed an average size of BHNC particles of 138 nm with a polydispersity of 0.21 409 (Table 1).

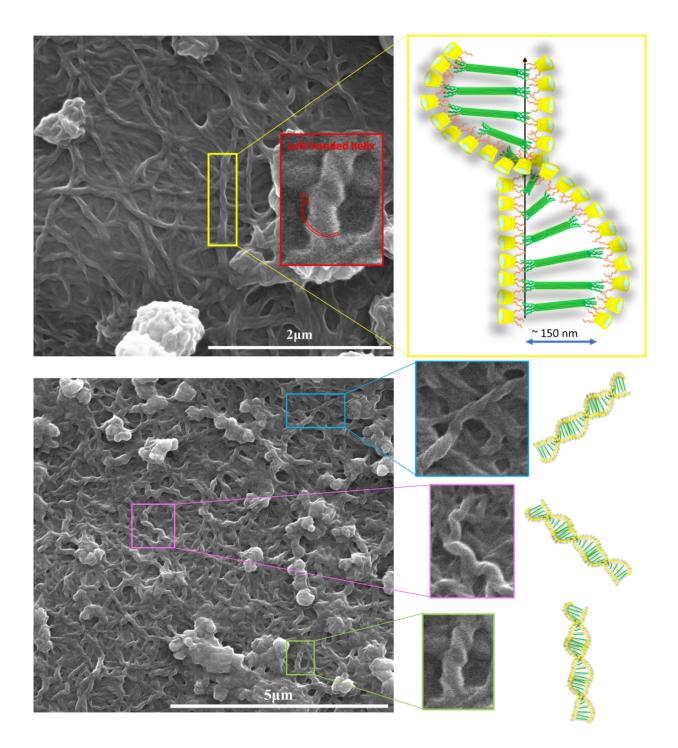


**Figure 1:** (a) AFM, (b) TEM, and (c) HRTEM images of BHNC rod-shape particles used as building blocks for preparation of polymer-based BBEs.

410 Biopolymer-based BBEs with different physical properties at two different conditions (freeze-dried 411 and non-freeze dried) were analyzed by FE-SEM (Figure 2a-i) and (Figure S5-7). These images 412 demonstrate that each of the 4 synthesized BBEs has a different structure before and after drying. 413 The SEM images of  $\beta$ -cyclodextrin (Figure S5) revealed that the material displayed a dense block-414 shaped morphology with uniform shapes and flat surfaces. After functionalization with EDA, the 415 white  $\beta$ CD turned yellow, which indicated that the crosslinking reaction had successfully occurred. 416 Besides, as shown in (Figure 2a), unlike the pristine  $\beta$ CD, the surface of the modified  $\beta$ CD, ( $\beta$ CD-417 EDA), became extremely loose and uneven, and displayed an evidently porous structure. Hence, it is 418 clear that the pristine  $\beta$ CD-EDA has an amorphous morphology and this morphological irregularity 419 was diminished by functionalization with the rod-like BHNC as seen from the uniformly ordered 420 morphologies in the different BBE SEM images. After polymerization of BHNC with  $\beta$ CD-EDA, it 421 is obvious that the BBEs presented a different morphology compared to BHNCs. For better 422 understanding, the different morphologies of BBEs with different  $\beta$ CD-EDA contents are presented 423 in Figure 2. FE-SEM images in Figure 2b and 2c are those of BBE1/1 with the highest amount of 424 βCD-EDA and both non-freeze dried and dry samples of BBE1/1 consist of water-insoluble nanosponges. Both images exhibit a spongy structure that makes the bio-polymers absorptive for 425 426 DOX. Surprisingly, the morphologies of BBE1/2, 1/4 and 1/8 were completely different from BBE1/1427 and furthermore, each of these 4 types of BBE biopolymers had different morphologies in the non-428 freeze dried and dry state. In BBE1/1 and BBE1/2 with increasing amounts of  $\beta$ CD-EDA, different 429 extents of agglutination between BHNC and  $\beta$ CD-EDA resulted in water-insoluble nanosponges with 430 different structures. As can be seen in Figure 2d and 2e, BBE1/2 nanosponges have totally different 431 structures: in non-freeze dried conditions BBE1/2 has the expected rod-shape polymerized fibrous 432 architectural structure that is due to the BHNC particle morphology, and the freeze-dried structure 433 consists of microspheres with an average size of 2-6  $\mu$ m. It is observed that with decreasing  $\beta$ CD-434 EDA content in the polymerization process, the presence of BHNC finger-print as a building template 435 in the polymerized product is more obvious, especially for the BBE1/4 and BBE1/8. Unlike the non-436 freeze dried condition, after freeze-drying the samples, BBE1/4 and BBE1/8 nanosponges had 437 decorated rod-like morphologies with an average thickness of 4-6 µm and they possesses micro-rod 438 sites and some micro-sheets (Figure 2f and 2h). An important morphological property observed in 439 the FE-SEM images of non-freeze dried BBE1/4 in Figure 2g is the unique agglutination between 440 BCD and BHNCs which resulted in remarkable helical morphologies. These helical structures are 441 less than 200 nm in diameter and have a left-handed twist. Cellulose nanocrystals are known to form 442 chiral nematic phases with helical nanorod arrangements; they also form in the presence of polymer. The FE-SEM image of these unique helical structures is shown enlarged in Figure 3 (left). A possible 443 444 explanation of how these structures are formed is shown in Figure 3 (right). They could be formed 445 by crosslinking the hairs of the BHNC nanorods by  $\beta$ CD-EDA, with subsequent near alignment of 446 the cellulose nanocrystals. Moreover, FE-SEM images of non-freeze dried BBE1/8 (with the lowest 447 amount of  $\beta$ CD-EDA) clearly show that this type of nanosponge has a fibrous structure with very 448 thin fibers (Figure 2i), which might have a similar structure as fibrils from BBE1/4. In summary, the 449 FE-SEM images of BBE1/2, BBE1/4, and BBE1/8 (Figure 2e, 2g, and 2i) show that BHNC nanorods 450 are trapped and polymerized with  $\beta$ CD-EDA in different morphologies. Hence, the polymer network 451 morphology is strongly dependent on  $\beta$ CD-EDA monomer content.

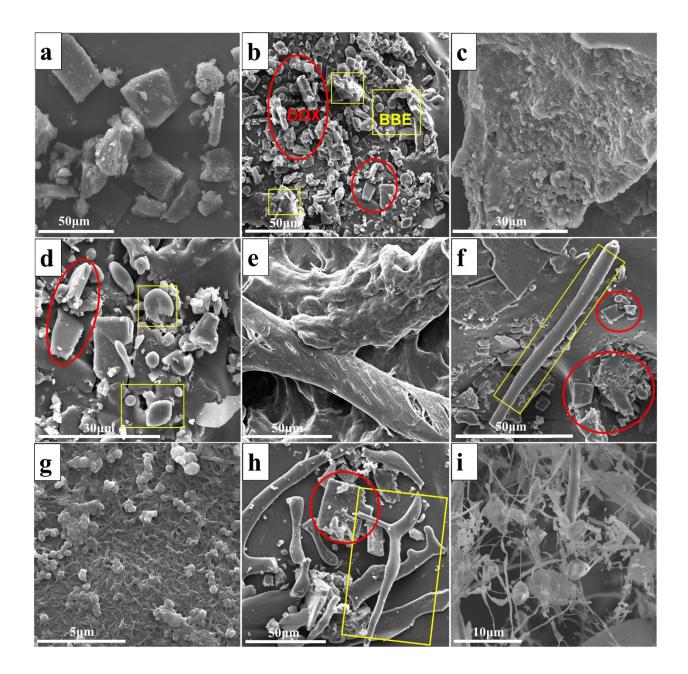


**Figure 2:** FE-SEM images of; (a) pristine  $\beta$ CD-EDA; (b, c) freeze dried and non-freeze dried BBE1/1 nanosponge; (d, e) freeze dried and non-freeze dried BBE1/2 nanosponges, (f, g) freeze dried and non-freeze dried BBE1/4; and (h, i) freeze dried and non-freeze dried BBE1/8 with different magnifications.



**Fgure 3:** (Left): FE-SEM image showing left-handed helical BBE1/4 fibrils; (Right): possible structure of helical fibrils, formed by hairy nanocellulose rods (green with hairs), crosslinked at their ends by  $\beta$ CD-EDA (yellow with red carbamate-EDA arms).

452 Further morphological investigation of different BBE biopolymers in physical mixtures with DOX 453 unveiled an outstanding morphology for these nanosponges and the supramolecular inclusion 454 complex of DOX-BBEs were observed under FE-SEM images (Figure 4a-i). As expected, the pristine 455 structure of DOX with clear columnar crystal edges and corners morphology are shown in (Figure 456 4a). The FE-SEM images represent the cubic crystalline sites of DOX in all of the mechanical mixing 457 structures with BBEs marked with red circles (Figure 4b, 4d, 4f and 4h), and in all of these images, 458 BBE morphologies remained unchanged (marked with yellow squares). The supramolecular 459 inclusion complexes in the non-freeze dried nanosponges can be seen in Figure 4c, 4e, 4g and 4i, 460 where they appear with different polymerized architectural morphologies. When compared, the 461 morphology of the non-freeze dried BBEs and their supramolecular inclusion complex structures are 462 very similar. Clearly, there is strong evidence of the presence of BHNC particles in the 463 supramolecular inclusion complexes. Also, the helical polymerizable liquid crystal-like assemblies 464 of BHNC particles were also seen in FE-SEM images of supramolecular inclusion complexes of 465 BBE1/4 with DOX (Figure 4g). In breast cancer cells, helical DOX-BBE1/4 showed the minimal cell 466 viability compared to the control or DOX cultures, suggesting that DOX-BBE1/4 has a greater 467 anticancer effect than free DOX. Compared to BBE1/1 (with higher DOX capacity), the effect on the 468 breast cancer was approximately the same.



**Figure 4:** FE-SEM images of; (a) DOX cubic crystalline particles; (b, c) physical mixture and supramolecular inclusion complex of DOX with BBE1/1 nanosponges. In red circles: DOX cube-shape morphology. In yellow boxes: BBE loaded DOX with different morphologies are shown; (d, e) physical mixture and supramolecular inclusion complex of DOX with BBE1/2 nanosponges; (f, g) physical mixture and supramolecular inclusion complex of DOX with BBE1/4 nanosponges; and (h, i) physical mixture and supramolecular inclusion complex of DOX with BBE1/8 nanosponges.

### 469 FTIR spectroscopy

470 To confirm the structure of BHNC particles as building blocks in polymerized products obtained 471 from soft wood pulp and DAMC and to confirm the surface modification of the precursor, FTIR 472 spectra of the different nanocelluloses,  $\beta$ CD,  $\beta$ CD-CI, and  $\beta$ CD-EDA were collected (Figure 5a-d). 473 The results displayed in Figure 5a suggest that along with the characteristic cellulose peaks at 3310 474 cm<sup>-1</sup> (stretching vibration modes of -OH groups), 1306 cm<sup>-1</sup> peak (-OH bending vibration), 2909 cm<sup>-1</sup> peak (C–H stretching vibration), 1426 cm<sup>-1</sup> peak (–CH<sub>2</sub> scissoring), 1034 cm<sup>-1</sup> peak (CH<sub>2</sub>–O– 475 476  $CH_2$  stretching), and a very small peak at 1630 cm<sup>-1</sup> (cellulose carboxyl vibration in the sodium form), wet DAMC had two characteristic peaks at 1730 cm<sup>-1</sup> related to non-protected aldehydes and 477 478 880 cm<sup>-1</sup> related to the dialdehyde group hemiacetal linkages. Compared to the FTIR spectrum of DAMC, the BHNC spectrum had a sharp increase in the 1603 cm<sup>-1</sup> peak related to carboxyl vibration 479 480 in the sodium form, indicative of successful oxidation of surface aldehyde groups to carboxylic acid. 481 However, the oxidation was not complete since the aldehyde peaks persist in the BHNC spectrum with the 880 cm<sup>-1</sup> peak (unprotected dialdehyde group hemiacetal linkages in BHNC particles) and 482 483 the 1730 cm<sup>-1</sup> protected aldehyde group peak which is expected to overlap with the strong carboxylic 484 acid 1603 cm<sup>-1</sup> peak. That  $\beta$ CD functionalization with EDA for preparing the EDA arms on the  $\beta$ CD 485 body had occurred, was confirmed by FTIR, as can be concluded from Figure 5b. Firstly, the 486 nucleophilic hydroxyl groups located on the  $C_6$  of  $\beta$ CD attacks the carbonyl of CDI and one of the 487 imidazoles that acts as a leaving group is replaced with the nucleophile. As a result, new unique 488 carbonyl peaks for carbamate bonds were observed in the FTIR spectrum of  $\beta$ CD-EDA at 1741 cm<sup>-</sup> <sup>1</sup>. On the other hand, as can be seen in Figure 5b, comparing  $\beta$ CD-CI to CDI shows that the unique 489 hydroxyl groups attributed to  $\beta$ CD at 1010 cm<sup>-1</sup> and the aromatic amine peaks for imidazole in CDI 490 491 at 1267 cm<sup>-1</sup> are only present together in the  $\beta$ CD-CI spectrum. Furthermore, the imine stretching peak at 1640 cm<sup>-1</sup> is observed in  $\beta$ CD-CI and CDI confirming successful functionalization of 492

493 carbonyl-imidazole on  $\beta$ CD. This suggests that the resonance electrons are only shared between the 494 carbonyl and imidazole in  $\beta$ CD-CI confirming the site of the nucleophile attack to be the carbonyl of CDI. Secondly, the EDA nucleophilic attack on the carbonyl of  $\beta$ CD-CI is confirmed with the 495 introduction of the new EDA band at 1637 cm<sup>-1</sup> corresponding to N-H bending vibration. The 496 497 successful preparation of bio-derived BHNC-BCD-EDA trifunctional polymerized nanosponges was 498 observed with FTIR in Figure 5c. The successful novel bio-polymeration of the BBE nanosponges 499 can be confirmed through the amide bond formation between the activated carboxyl content in BHNC 500 and the primary amines of BCD-EDA. This characteristic peak overlaps with two other broad characteristic double peaks at 1722 cm<sup>-1</sup> and 1736 cm<sup>-1</sup> attributed to the carbonyl vibration of 501 502 aldehydes on the BHNC and carbonyl vibration band of carbamate related to  $\beta$ CD-EDA respectively. The disappearance of the peaks at  $1603 \text{ cm}^{-1}$  that corresponds to carboxyl stretching vibrations on 503 504 the BHNC is good evidence for proving that the polymerization of  $\beta$ CD-EDA with BHNC building 505 blocks has occurred. Finally, to verify that there are no differences between different types of bio-506 derived BHNC-βCD-EDA trifunctional polymerized nanosponges, the FTIR spectra were compared 507 in Figure 5d. It is evident that the trifunctional polymerization fabrication of BHNC particles with 508 βCD-EDA via a facile and green one-pot crosslinking reaction method using EDA as a cross-linker 509 (BHNC- $\beta$ CD-EDA). There are no structural changes observed between them. The only difference 510 was for the EBB1/8 structure that had a sharper carboxyl vibration band at 1603 cm<sup>-1</sup> which confirms 511 that the BBE1/8 has the highest carboxyl content compared to the other BBE polymers.

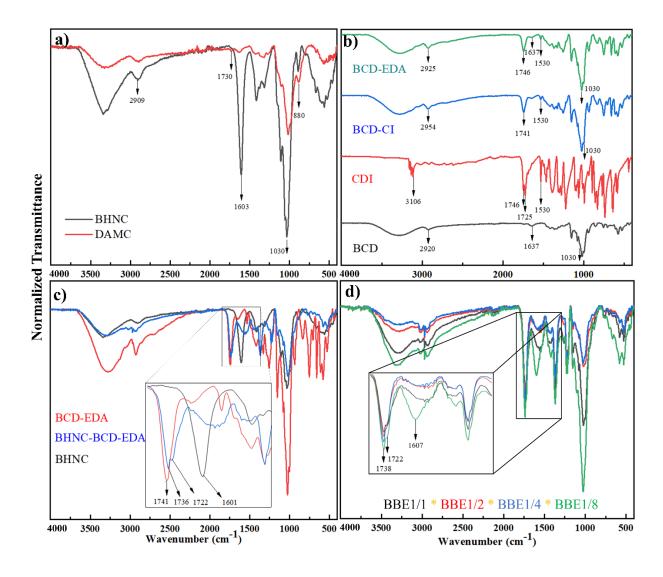


Figure 5: FTIR analysis of different control materials in each preparation steps and trifunctional BBE biopolymers.

#### 512 **XRD**

513 The crystallinity of the BBEs was investigated by XRD (Figure S8). The results indicated that the 514 high index of crystallinity of BBE corresponds mainly to BHNCs which are crosslinked with  $\beta$ CD 515 and shows a peak around 20.5°, characteristic for cellulose crystals with Miller index (021).

516 **TGA** 

517 The TGA curves of BHNC and trifunctional BBE biopolymers are shown in Figure 6. The thermal 518 behavior of the biopolymers involves only a three-step process with two isothermal conditioning 519 periods. At the first step all samples showed a decrease in weight below 150 °C owing to the 520 evaporation of physically absorbed water on the surface of biopolymers (10%). The second and third 521 steps account for most of the weight and results in the formation of the residue of the  $\beta$ CD polymer 522 and cellulose. For dried bifunctional hairy nanocellulose, the disintegration at the second stage (the 523 first pyrolysis stage) occurs at 200-300 °C and was mainly caused by the degradation of cellulose 524 chains, showing a drastic weight loss of 45%. The third stage (second pyrolysis stage) at 300-600 °C 525 was caused by the slow carbonization of cellulose and was accompanied by a 20% additional weight 526 loss. For BBE, the second degradation stage of biopolymers takes place at a higher temperature (200-527 350 °C) than for BHNC, with high weight loss (50%), and the third degradation stage of biopolymers 528 takes place at a higher temperature (350-600 °C) with lower weight loss (40%). These results indicate 529 that BBE biopolymers are more stable than the non-crosslinked BHNC. This shows that the stability 530 of BHNC increases after modification with  $\beta$ CD. Among all TGA curves, that of BBE1/1 is the most 531 stable, indicating that the highest content of  $\beta$ CD has the largest effect on the hydrothermal stability 532 of the biopolymers. This enhancement in thermal stability of BBE biopolymers could be due to the 533 strong crosslinking interactions between the BHNC and  $\beta$ CD-EDA and serves as a thermal protective 534 layer surrounding the BHNC particles, which enables this material to be used in high-temperature 535 applications. Basically, differences could only be found in the thermal stability of the biopolymers. 536 When comparing TGA curves of BBEs to that of BHNC, it can be concluded from the weight loss 537 that  $\beta$ CD was successfully grafted on the surface of BHNCs.

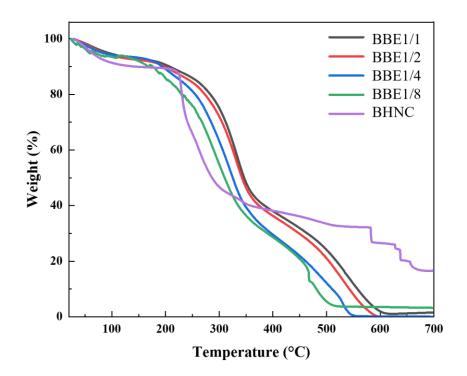
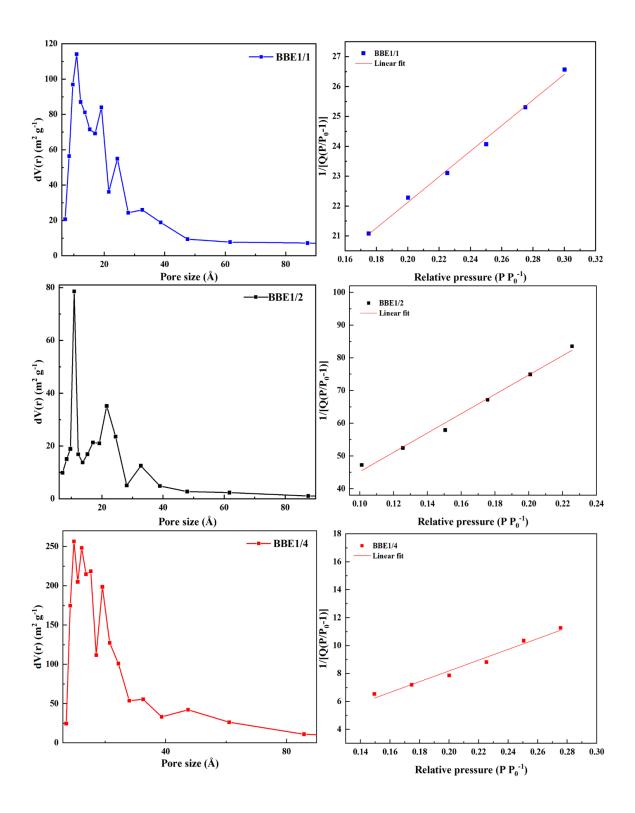


Figure 6. TGA of BHNC and trifunctional BBE biopolymers.

#### 538 Characterization by BET analysis

539 In order to characterize the drug loading and release potential of the different BBE morphologies it 540 is important to measure the surface area, pore size and, pore volume as these structures have unique 541 porosity. N<sub>2</sub> adsorption-desorption isotherms were obtained to determine the pore size distribution 542 and the surface of the different BBE polymers area, the results of which are shown in Figure 7. The 543 surface area, pore size, and pore volume results are summarized in Table 1. From the BET plots, the 544 BET equation yielded a pronounced decline of the surface area (determined from slope and intercept of linear BET plots in Fig.7) from 533 m<sup>2</sup>/g in BHNC to 39-300 m<sup>2</sup>/g in BBE, confirming the 545 546 functionalization of  $\beta$ CD-EDA on BHNC. The decrease shows that the BHNC particles are not 547 present as individual particles within the BBE matrix, but must be in a dense aggregated form. The 548 various BBE polymers have 1-3 nm pores present in all the structures, presumably related to the  $\beta$ CD 549 cavity that has a width of approximately 0.8 nm.



**Figure 7.** Pore size distribution of various BBEs (left) and corresponding linear BET plots (right), from which the specific surface area can be determent from the slope and intercept.

Sample	Charge content (mmol/g)	Average size (nm)	Polydispersity	Surface area (m <sup>2</sup> /g)	Pore volume (cm <sup>3</sup> /g)	Average pore size (Å)	Zeta potential (in PBS solution) (mV)		
							рН 5.5	рН 6.5	рН 7.6
βCD-EDA	6	-	-	-	-	-	-	-	2.31
BHNC	4.5	138	0. 21	533*	-	-	-14	-18	-20
<b>BBE</b> 1/1	0	1583	0.30	116	0.17	22	-4#	-7#	-9#
BBE1/2	2.25	1200	0.28	39	0.05	29	-9	-11	-12
BBE1/4	3.48	515	0.22	271	0.29	38	-10	-12	-16
BBE1/8	3.94	202	0.13	-	-	-	-13	-16	-20

Table 1. Physiochemical properties of BHNC and the different BBE polymers.

Notes: \*Calculated assuming cylindrical nanoparticles of 5 nm diameter and density of 1.5 cm<sup>3</sup>/g.

<sup>#</sup>Negative zeta potentials for BBE1/1 are likely due to the adsorption of anionic impurities, which is almost unavoidable for neutral colloids. The same is observed for SNCC (sterically stabilized nanocrystalline cellulose)<sup>47</sup>.

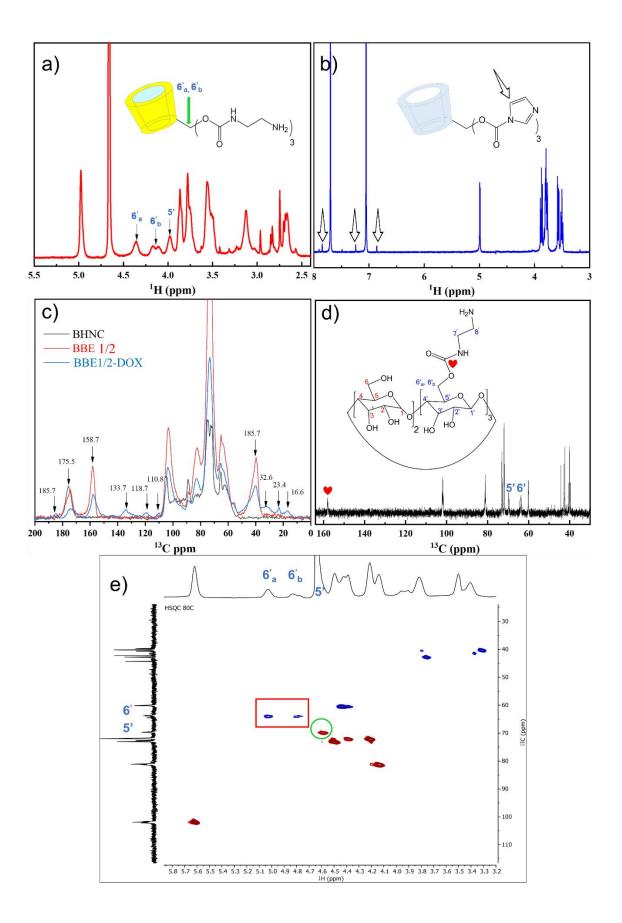
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#### 552 NMR spectroscopy

553 Structural characterization of all compounds was performed via NMR spectroscopy and the spectra 554 obtained are shown in figure 8 and figure S9-11. The liquid <sup>1</sup>H NMR spectrum of pure βCD shows 555 the characteristic peaks at 4.99, 3.89, 3.77, 3.79, 3.57, and 3.51 ppm corresponding to the H<sub>1</sub>, H<sub>3</sub>, 556  $H_5$ ,  $H_6$ ,  $H_2$ , and  $H_4$  respectively (figure S9). Upon functionalization of  $\beta$ CD with CDI through 557 hydroxyl groups of  $\beta$ CD, the peaks of  $\beta$ CD were observed and the peaks of carbonyl imidazole 558 (CI) attached to  $\beta$ CD was observed with the characteristic peaks at 7.85, 7.23, and 6.86 ppm for 559 hydrogen atoms of imidazole and two sharp peaks at 7.74 and 7.04 ppm related to the released 560 imidazole from the nucleophilic reaction between CDI and  $\beta$ CD which was trapped in to the  $\beta$ CD 561 cavity (Figure 8b). After that, the imidazole leaving groups in  $\beta$ CD-CI, were substituted by EDA 562 with the carbamate bond formation. The successful preparation of  $\beta$ CD-EDA was structurally 563 validated with liquid <sup>1</sup>H and <sup>13</sup>C NMR, HSQC and HMBC spectra shown in figure 8a, d, e, and 564 figure S10, respectively. Compared to the <sup>1</sup>H NMR spectrum of  $\beta$ CD,  $\beta$ CD-EDA had new peaks 565 at 2.6 and 3.1 for hydrogens of ethylene in EDA (figure 8a). On the other hand, these characteristic 566 peaks were observed at 40 and 42.76 ppm ( $C_8$  and  $C_9$ ) in the <sup>13</sup>C NMR of  $\beta$ CD-EDA (figure 8d). Even more evidence for the successful grafting of  $\beta$ CD on EDA comes from <sup>1</sup>H, <sup>13</sup>C NMR, HSOC 567 568 and HMBC. After attaching the EDA to  $\beta$ CD through the  $\beta$ CD-CI of the carbamate bond, peak 569 shifts could be detected due to the functionalization of one  $C_6$  of  $\beta$ CD, namely, a downfield shift 570 of  $C_6$  from 60.1 to 63.73 ppm, as well as upfield shifts of the adjacent  $C_5$  from 73 to 70 ppm and  $C_1$  from 102.1 to 101.9 ppm. These values are in accordance with the literature<sup>46</sup> which shows that 571 572 the preparation of carbamate bonds with participation of hydroxyl groups on  $C_6$  leads to a 573 downfield shift of the carbon carrying the hydroxyl ( $C_6$ ), a small upfield shift of the carbon  $C_5$  as 574 well as an even smaller upfield shift of the carbon  $C_1$ . The carbonyl of carbamate located at 158

ppm also shows that decorating  $\beta$ CD with EDA has clearly occurred. Moreover, the <sup>1</sup>H NMR 575 576 spectrum of  $\beta$ CD-EDA (Figure 8a) showed new characteristic peaks of diastereotopic hydrogens 577 on C<sub>6'</sub> (H<sub>6'a</sub> and H<sub>6'b</sub>) at 4.14 and 4.38 ppm, and on H<sub>5'</sub> at 3.97 ppm. For accurate evaluation of 578 these new peaks, the two-dimensional (2D) HSQC <sup>13</sup>C-<sup>1</sup>H plot has been taken at 80 °C depicted in 579 Figure 8e. One of the great advantages of HSQC compared to <sup>13</sup>C NMR and <sup>1</sup>H NMR is the 580 determination of proton-carbon single bond correlations. From this diagram, a detailed 581 interpretation of these protons is provided: firstly, both hydrogens at 4.1 and 4.4 ppm were 582 correlated with the same carbon ( $C_{6}$  at 133 ppm), therefore, the two diastereotopic protons belong 583 to a methylene group which is attached to the carbamate bond. Secondly, another new hydrogen 584 peak at 4.0 ppm was correlated with  $C_{5'}$  at 70 ppm implying that EDA was grafted on  $\beta$ CD. The 585 H<sub>6'a,6'b</sub> chemical shift changes further prove that the reaction position was 6-OH. This result 586 indicated that  $\beta$ CD reacted with EDA by forming the carbamate bond. Comparing the area of 587  $H_{6'a,6'b}$  to the area of  $H_1$  allows us to determine that the degree of substitution of  $\beta$ CD with EDA, 588 which was found to be 3.5. This value is in consistent with the amide content of  $\beta$ CD-CI (3.2 589 mmoL/g obtained by titration (Figure S3)). Finally, the successful preparation of bio-derived 590 BHNC-βCD-EDA (BBEs) trifunctional polymerized nanosponges and supramolecular inclusion 591 complexes with DOX were observed in <sup>13</sup>C CP/MAS NMR spectra, illustrated in Figure 8c. In our 592 previous work<sup>32</sup>, the solid <sup>13</sup>C NMR spectrum revealed the expected pulp cellulose peaks of 105 593 ppm for (C<sub>1</sub>), 85 and 90 ppm for (C<sub>4</sub>), 80-70 ppm for (C<sub>2</sub>, C<sub>3</sub>, C<sub>5</sub>), and 70-60 ppm for (C<sub>6</sub>). 594 Oxidation of pulp to DAMC fibers was confirmed by a broad peak at 85-105 ppm without any 170 595 ppm (carbonyl) peak suggesting the formation of hemiacetal groups. After oxidation of DAMC to 596 BHNC the appearance of a new peak at 170-180 ppm (carboxylic acid) confirms the formation of 597 BHNC and shows that the aldehyde peaks from DAMC were not completely lost. The region

598 between 50-110 ppm closely resembles that of cellulose, indicating that any remaining hemiacetal 599 linkages were no longer observable after oxidation of DAMC. Compared to the BHNC spectrum, 600 the presence of a new peak at 158 ppm, representing the carbamate groups and a new peak at 40 601 ppm due to the ethylene carbons in EDA are strong evidence that the BHNC building block 602 particles were polymerized with  $\beta$ CD-EDA. Amide bonds which resulted from attaching amines 603 from  $\beta$ CD-EDA to the activated carboxylic acid groups of BHNCs with the DMTMM activator 604 overlap with the remaining carboxyl carbons and aldehyde carbonyls peaks in the BHNC 605 spectrum. This broad peak is located at 170-178 ppm. The region 50-110 ppm closely resembles 606 that of cellulose and  $\beta$ CD indicating that both bio-precursors are present. The inclusion complex 607 of DOX-BBE has also been verified by comparing the solid-state <sup>13</sup>C CP/MAS NMR spectra of 608 DOX-BBE and BBE (Figure 8c). Three sharp DOX characteristic peaks were present in the 609 spectrum of DOX-BBE due to the carbon signal of aliphatic carbons in the DOX structure (17, 23, 610 and 33 ppm). Moreover, multiple wide peaks around 50–110 ppm were observed corresponding 611 to carbon signals of BBE. The peaks at 110, 119, and 134 ppm are ascribed to aromatic carbons 612 corresponding to the DOX structure in the supramolecular inclusion complex and the new sharp 613 peak at 187 ppm is related to the carbonyl of ketone in DOX, confirming the formation of 614 supramolecular inclusion complex in  $\beta$ CD of BBE.

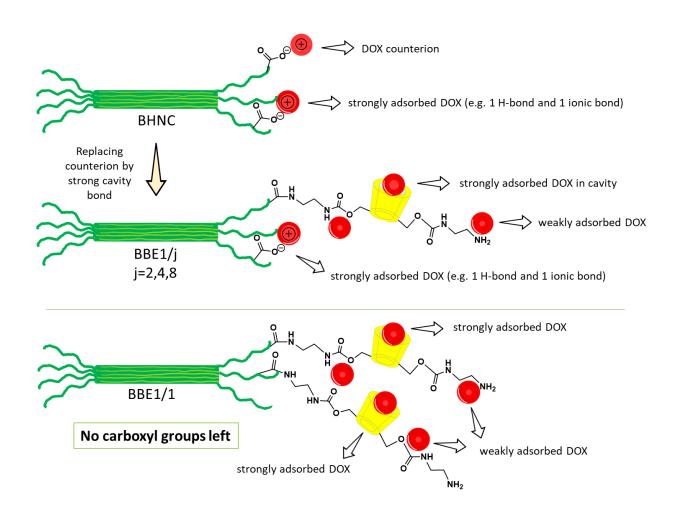


**Figure 8.** a) <sup>1</sup>HNMR spectrum of  $\beta$ CD-EDA; b) <sup>1</sup>HNMR spectrum of  $\beta$ CD-CI; c) solid-state <sup>13</sup>C CP/MAS NMR of BHNC, BBE1/2 and BBE1/2-DOX; d) <sup>13</sup>C NMR spectra of  $\beta$ CD-EDA; (e) HSQC spectrum of  $\beta$ CD-EDA at 80 °C.

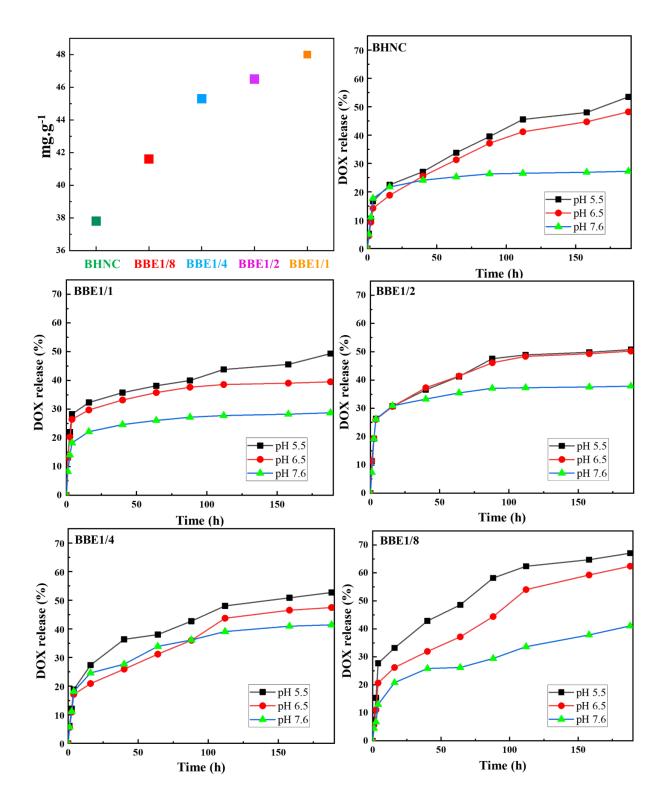
## 615 **3.2. pH sensitive DOX loading and** *in vitro* release of various BBEs

616 For the application of BBEs as biopolymer-based carriers, various BBEs were loaded with DOX 617 to evaluate the effect of the polymer morphology, the ratio of bio-precursors, and the content of 618 both BHNC and βCD on drug loading and release. UV spectrophotometry was used to characterize 619 the loading and release of the doxorubicin from BHNC and the BBE polymers (Figure 9). The 620 loading and release were examined at body temperature (37°C) under slightly acidic conditions 621 (pH 5.5 and 6.5) that resemble tumor endosomal and lysosomal pHs, as well as at neutral blood 622 pH 7.6, in order to estimate the effect of the tumor microenvironment on the release profile of 623 DOX from the BBE polymers. The release behavior of DOX from BBEs directly influences the 624 anticancer effect on the cancer cells. It is clear from Figure 9 that the loading capacity was higher 625 for all BBE polymers compared to the BHNC control structure. Moreover, the absence of the cubic 626 DOX morphology in the SEM images of the DOX-loaded biopolymers as well as the <sup>13</sup>C NMR 627 results, proof that DOX is loaded in the supramolecular inclusion complexes of  $\beta$ CD and hence, 628 we expect to see an increase in DOX loading capacity with increased  $\beta$ CD content, as observed. It 629 appears that  $\beta$ CD enhances the loading by allowing DOX to be loaded in the  $\beta$ CD cavity and by 630 allowing physisorption of DOX on the nanocellulose polymer matrix to occur. The release profiles 631 of DOX-loaded BBEs (designated as BBE-DOX) with various morphologies were evaluated and 632 the amount of DOX released is plotted in Figure 9 together with the release of DOX from BHNC. 633 The release profile of DOX from BHNC suggests that there are two types of adsorbed molecules: 634 those released very fast and those released very slowly. This is likely due to two different

635 adsorption mechanisms: possibly one due to ion-exchange, in which protonated DOX molecules 636 act as counterions for the carboxyl groups in BHNC, and one due to physisorption. The counterions 637 are released fast in the presence of a PBS buffer, whereas the physisorbed DOX molecules are 638 released slowly. The effect of pH on DOX release from BHNC is counterintuitive. One would 639 expect that more protonation of DOX at lower pH would increase the electrostatic attraction 640 between DOX and carboxyl groups on BHNC, thus slowing down the release kinetics. However, 641 the opposite is observed, implying that the physisorption of DOX on BHNC is not governed by 642 electrostatics (scheme 6). The release profiles of BBEs, in which a certain amount of carboxyl 643 groups is replaced by  $\beta$ CD-EDA cavities, are remarkably similar to those of BHNC. This is 644 puzzling, as few or no counterions should be present in BBE1/1, implying the absence of ion-645 exchange. Taken in isolation, these profiles would imply the presence of weakly adsorbed DOX 646 molecules and the adsorption of strongly adsorbed molecules, presumably in the  $\beta$ CD-EDA 647 cavities. The comparison with BHNC suggests that DOX molecules adsorbed in  $\beta$ CD-EDA 648 cavities are released at a similar rate as DOX molecules physiosorbed on BHNC. The pH 649 dependence of the slow release of DOX from BBE can be due to the fact that at higher pH the 650 DOX molecules are not protonated and from a stronger bond with the hydrophobic interior of the 651 cavities. The weakly adsorbed DOX molecules are likely adsorbed on  $\beta$ CD-EDA, possibly due to 652 hydrogen bonding between secondary amines in EDA and oxygen atoms in DOX. We confirmed 653 that complex formation occurs between EDA and DOX by UV-vis (see Figure S12), which proves 654 that DOX associates with EDA. With a decrease in  $\beta$ CD-EDA content, and an accompanied 655 increase in carboxyl groups, the number of weakly adsorbed ions decreases and that of counterions 656 increases, keeping the number of fast desorbing DOX molecules roughly equal (about 25%). The 657 schematics of a plausible mechanism of DOX loading in BBEs and in BHNC as control, is shown in scheme 6. From the mechanism one would expect that BBE1/1 has a much larger adsorption
capacity than BHNC. Instead, the increase is only about 25%. This is like due to the extensive
washing of the loaded sample, prior to measuring the DOX content, in which process most of the
weakly adsorbing DOX molecules were removed.



Scheme 6. Schematic illustration of strongly and weakly adsorbed DOX in BBE structures and BHNC.



**Figure 9.** DOX loading capacity of BBE structures and BHNC and cumulative release profiles of DOX loaded-BBEs in pH 5.5, 6.5 and 7.4 PBS at 37 °C.

## 662 **3.3. Evaluation of the cytotoxicity of BBE-DOXs on human breast cancer cells**

663 Biocompatibility of the BBE polymers was assessed using the metastatic chemo-resistant MDA-664 MB-231 breast cancer cell line that is widely used as a cancer cell model for drug screening due 665 to its metastatic aggressiveness and chemoresistance. BBE1/1 and BBE1/4 were selected to be 666 used as drug delivery bio-polymers in biological model MDA-MB-231 cells (Figure 10) since they 667 represent soluble and insoluble BBE bio-polymers, respectively, and they did not have exaggerated 668 DOX bursts especially, BBE1/4 with the fascinating helix structures. The reported IC50 for doxorubicin for MDA-MB-231 is 1.25 µM for 48 h incubation<sup>48</sup>. In our previous studies, we found 669 670 that 5.04  $\mu$ M doxorubicin was toxic for MDA-MB-231 cells at 24 h, we used this concentration 671 for the toxicological assay. The cell viability was not significantly altered compared to control 672 when DOX-free BBE bio-polymers were added, suggesting that they are biocompatible in vitro 673 (Figure 10 b, c, and, Figure 11). After 24 h of exposure, a significant decrement in cell viability 674 was found in BBE-DOX and DOX samples compared with control. BBE1/4-DOX show more 675 anticancer effect than free doxorubicin. The anticancer effect of BBE1/1-DOX with more loaded 676 DOX was similar to BBE1/4-DOX. This is probably related to the helical structural that formed 677 by capturing the liquid crystals of BHNC particles and water-soluble properties of this novel bio-678 polymer. Moreover, BBE1/4 has higher surface area, pore size and pore volume with compared to 679 BBE1/1 which has a dense nanosponge structure, and hence  $\beta$ CD cavities in BBE1/4 are more 680 accessible and more effective in releasing DOX. As a result, on average, BBE-DOX samples had 681 a higher cytotoxicity than doxorubicin cultures, suggesting more anticancer effect than free 682 doxorubicin. These BBE trifunctional bio-polymers as novel drug carriers can be used as 683 remarkable structures for drug delivery with adjustable loading, release, and solubility.

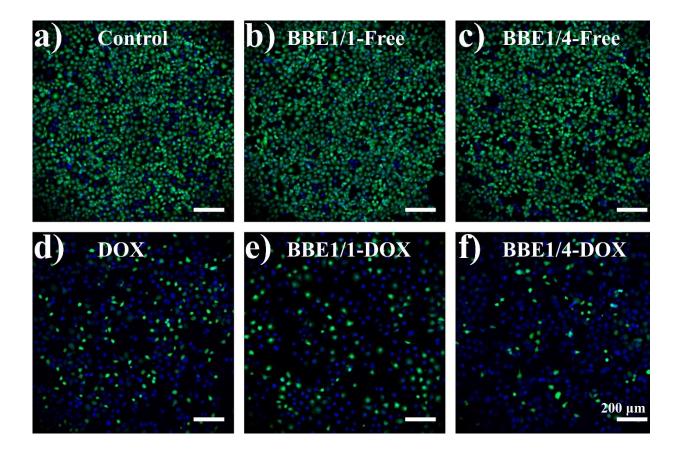


Figure 10. Confocal gallery of MDA-MB-231 exposed to BBE-DOX loaded. Breast cancer cells were incubated with BBE-Frees (b,c), BBE-DOXs (e,f), or DOX (d) for 24 h 5.04  $\mu$ M (DOX-based concentration). Magnification ×10, scale bar 200  $\mu$ m. Calcein-AM stains live cells as green; Hoechst stains nucleus of live and dead cells. MDA-MB-231 without nanomaterials were used as control (a).

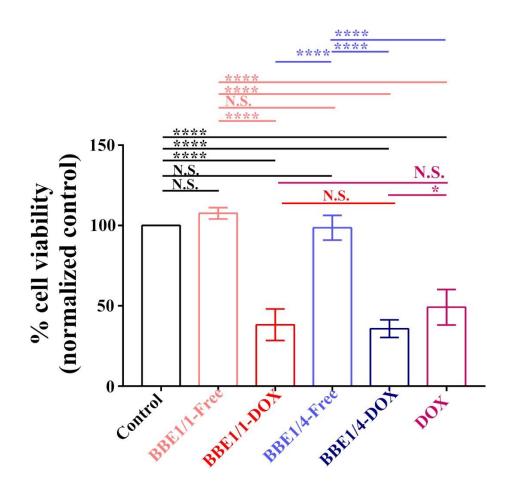


Figure 11. Viability of MDA-MB-231 exposed to BBE-Free, BBE-DOX, and DOX at 5.04  $\mu$ M concentration of biopolymers loaded-doxorubicin or doxorubicin; data are presented as mean  $\pm$  SD, n  $\geq$  3, \*P < 0.05, \*\* P < 0.01, \*\*\*\* P < 0.001, \*\*\*\* P < 0.0001.

## 684 **4. Conclusions**

Novel and environmentally friendly bio-derived cellulose-beta cyclodextrin trifunctional anticancer carriers, BBEs, were fabricated via a facile one-pot and green synthesis by using  $\beta$ CD-EDA groups as cross-linkers. The objective was to determine the effectiveness of doxorubicin-loaded BBE against breast cancer cells. This system, based on the preparation of biopolymers with different morphology and novel DOX-loading and DOX-release behavior and characterization results, provides evidence that each component of BBEs has a crucial role in its functioning:

691 BHNC nanorods function as the backbone of this novel polymer; importantly, the  $\beta$ CD-EDA 692 groups play a role not only as cross-linkers but also a substrate for DOX molecules weakly 693 physisorbed on the tether (EDA) which links the cavity to the nanocellulose particles, and acting 694 as counterions for unreacted carboxyl groups. On the other hand, the DOX-loading and DOX-695 release behavior mechanism of BBE is, after a burst of weakly adsorbed DOX, primarily due to 696 the host-guest inclusion complexation of immobilized BCD cavities. Moreover, these bio-based 697 hyper crosslinked polymers had a significant impact of DOX loading and DOX molecules could 698 be released effectively. The BBEs are biocompatible with cells, but the addition of doxorubicin 699 into their structure generates an attractive new drug delivery system with better outcomes than 700 doxorubicin alone enhancing the anticancer effects of DOX. These polymers have shown pH 701 responsiveness with enhanced release in acidic conditions which is very beneficial for passive 702 tumor targeting. Varying the grafting ratio of  $\beta$ CD-EDA on BHNC resulted in different polymers 703 with some different physiochemical properties, different morphologies, and different loading and 704 release profiles. Among the different polymers, the 1:4 crosslinking ratio (BBE1/4) resulted in 705 fascinating helical morphologies possibly due to the grafted BCD-EDA being optimum for 706 polymerizing liquid crystals of BHNCs stabilized in a polymer network. As a result, they had 707 enhanced anticancer effects in the breast cancer cell lines after they displayed high solubility in 708 water and high drug loading. More in-depth studies should be done to validate the effectiveness of 709 these new BBEs as drug carrier systems, including but not limited to: time-dose response, time-710 dose drug releasing, applications on different cancer cell types, and more in vivo studies. This 711 study may lead to advancing practical applications of BBEs in drug delivery systems as they are 712 green, biocompatible, and versatile nanostructures with huge potential. Additionally, this work 713 adds a new insight to design and preparation of a trifunctional cyclodextrin-based polymer carriers,

714	which can be used for other drug delivery systems as well. It is believed that this green and
715	ecofriendly cross-linking technology can be extended to prepare a wide variety of trifunctional
716	materials for various applications.
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