

1 **The Oxygen Reduction Reaction on Pt: Why Particle Size *and* Inter-particle**

2 **Distance Matter**

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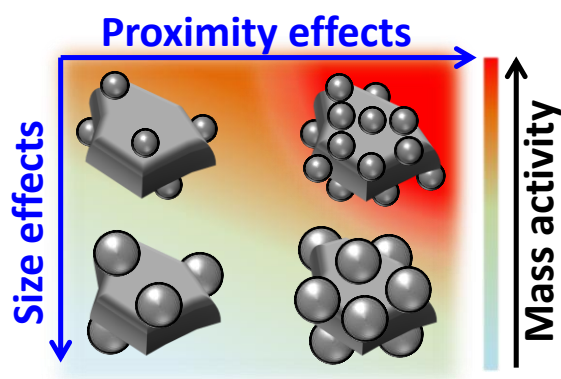
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27 **Keywords:** Platinum; Oxygen Reduction Reaction (ORR); Particle Size Effect; Inter-

28 particle Distance; *operando* electrochemical X-ray absorption spectroscopy (XAS);

29

30 **Abstract**

31 Carbon supported Pt based nanoparticles are important electrocatalysts for energy
32 conversion reactions such as the oxygen reduction reaction (ORR). Although this reaction
33 has been extensively studied, the influence of factors such as the particle size and inter-
34 particle distance of the nanoparticle-based or nano-sized electrocatalysts on the ORR
35 activity and durability are not yet fully understood and often intertwined. This lack of
36 understanding is mostly based on the limitation in the synthetic approaches of the
37 electrocatalysts which usually do not allow an independent variation of particle size and
38 inter-particle distance. In the presented work, we succeeded to disentangle both factors
39 using a “colloidal toolbox” approach and have demonstrated an effect of the inter-particle
40 distance on the electronic properties of the nanoparticle via *operando* electrochemical X-
41 ray absorption spectroscopy (XAS).

42

43 **1. Introduction**

44 In the last decade, the interest in electrocatalyst design has been boosted by the increasing
45 exploitation of regenerative energy in the form of electricity, in particular wind energy
46 and photovoltaic power¹. The intermittent nature of these energy sources requires suitable
47 energy storage and conversion technologies to be able to align the “*production*” with the

48 “demand” of energy. While battery technology shows continuous progress, the storage
49 and re-conversion of electricity in the form of energy carriers such as hydrogen gas still
50 faces substantial challenges².

51 For these conversion technologies a fundamental understanding and rational design of
52 electrocatalysts, on which the surface reactions of interest take place, are essential³. In
53 state-of-the-art academic electrocatalyst design, typically the focus is on optimizing the
54 surface structure of the catalyst on a nanometer or even atomic scale^{4–10}. This approach
55 led to the development of several highly performing electrocatalyst concepts in research,
56 nevertheless, their implementation into realistic conversion devices is not straight-
57 forward¹¹. In the presented work, we demonstrate that *mesoscopic properties*^{12–14} such as
58 the inter-particle distance of the catalytically active phase on the support material are of
59 equal importance for a rational catalyst design. Investigating the oxygen reduction
60 reaction (ORR) – the cathode reaction in a fuel cell – on different Pt nanoparticle based
61 electrocatalysts, we elucidate the influence of size (and thus structure) of the catalytically
62 active phase (Pt nanoparticles) as well as the mesoscopic property of the average inter-
63 particle distance between these nanoparticles on the carbon support. The influence of the
64 first factor, i.e., the particle or crystallite size of Pt nanoparticles on the ORR is
65 extensively and sometimes controversially discussed in literature^{15–23}; investigations date

66 back to the work of Stonehart, Watanabe, Boudart, Kinoshita, Ross, and others²⁴⁻²⁹ and
67 cannot be covered here in full length. Interestingly, the metal loading on the support and
68 thus the effective inter-particle distance of the nanoparticles is typically not addressed.
69 An exception is the discussion of reactant diffusion zones around the particles introducing
70 a “territory theory” by Watanabe et al.²⁸. Similar models exist in heterogeneous catalysis
71 discussion spill-over and reversed spill-over of reactants between active phase and
72 support material^{30,31}. The limited number of studies addressing the influence of the inter-
73 particle distance on catalytic reactions can at least partially be explained by the lack of
74 suitable synthesis strategies to control size and inter-particle distance independently. Only
75 advanced preparation strategies for metal clusters and nanoparticles³²⁻³⁵ allow such
76 investigations. In previous work, the authors investigated the ORR on size selected Pt
77 clusters prepared by a laser ablation source³⁴. It was proposed that at small inter-particle
78 distances an overlap in the electric double layer (EDL) structure at the catalyst-electrolyte
79 interface leads to an increase in ORR activity, i.e. the particle proximity effect of the
80 ORR¹⁴. Follow up work has demonstrated an improvement in ORR activity at small inter-
81 particle distance also for carbon supported high surface area electrocatalysts³⁶. Very
82 recently, it has been shown that the inter-particle distance of Pt nanoparticles deposited
83 on two dimensional as well as three dimensional carbon supports influences the amount

84 of dissolved Pt detected in scanning flow cell (SFC) measurements coupled to online ICP-
85 MS¹². Density functional theory modeling indicates that proton accumulation caused by
86 the overlap of the EDL (at high particle density) competes with proton depletion caused
87 by the proximity of the nanoparticles to the carbon support³⁷. On the other hand, it was
88 suggested in non-catalytic studies that the inter-particle distance (in addition to the
89 particle size) also influences the intrinsic electronic properties of metal nanoparticles. For
90 example, for alkane thiol stabilized colloidal gold nanoparticles a substantial cross-
91 coupling between the particles has been reported when depositing them onto a planar
92 substrate at small separation distances³⁸. These observations were interpreted as a result
93 of changes in the screening of charge transfer barriers in the ensembles of nanoparticle
94 due to electron tunneling. It was stated that films of closely separated nanoparticles
95 behave effectively as a collective metal film changing its electronic character³⁸. Other
96 reports have even claimed that at very low inter-particle distances, exchange interactions
97 cause otherwise insulating assemblies to become semiconducting, metallic, or
98 superconducting³⁹.

99 To test the hypothesis that in addition to the particle size, the inter-particle distance
100 influences the intrinsic electronic and thereby the catalytic properties of materials, these
101 parameters need to be adjusted independently. As mentioned above, in order to do so

102 suitable synthesis strategies are required. While vacuum based laser ablation or
103 magnetron sputtering sources combined with mass selection filters^{34,40} offer ultimate
104 control on the particle size, these methods can typically only produce minute sample
105 amounts and require enormous financial and technical investments. By comparison,
106 colloidal approaches^{33,40,41} offer similar control on the individual catalyst parameters but
107 can produce more material and are much broader available. In the presented work, we
108 synthesized carbon supported high surface area catalysts (Pt/Vulcan) with various Pt
109 particle sizes and Pt loadings via a toolbox approach that has been developed for this
110 purpose^{33,36,42}. The approach builds on the surfactant-free synthesis of colloidal
111 nanoparticles using alkaline ethylene glycol as solvent and reducing agent that was
112 introduced by Wang et al.⁴³. The size of the Pt nanoparticles is controlled by varying the
113 NaOH/Pt molar ratio in the colloidal synthesis⁴⁴.

114

115 **2. Experimental section**

116 *Chemicals, materials, and gases.*

117 The following chemicals were used in the catalyst synthesis and characterization:
118 ethylene glycol (EG, 99.8 %, Sigma-Aldrich), sodium hydroxide (NaOH, 98.9 %, Fisher
119 Chemical), hexachloroplatinic(IV) acid hexahydrate ($\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$, 99.9 %, Alfa Aesar),

120 30 % hydrochloric acid (HCl, Suprapur, Merck), 60 % nitric acid (HNO₃, Suprapur,
121 Merck), acetone (99.5+ %, Alfa Aesar), tin(II) chloride dehydrate (> 98 %, Sigma-
122 Aldrich), and platinum standard solution (1000 ± 4 mg L⁻¹, Sigma-Aldrich). Commercial
123 carbon black (Vulcan XC72R, Cabot Corporation, BET area: 235 m² g⁻¹) was employed
124 as carbon support in the catalyst synthesis. Deionized (DI) water (resistivity > 18.2
125 MΩ·cm at room temperature, total organic carbon (TOC) < 5 ppb) from a Milli-Q system
126 (Millipore) was used for acid dilutions, catalyst ink formulation, and the electrochemical
127 cell cleaning. 2-propanol (IPA, 99.7+ %, Alfa Aesar), 70 % perchloric acid (HClO₄,
128 Suprapur, Merck), and potassium hydroxide hydrate (KOH·H₂O, Suprapur, Merck) were
129 used for catalyst ink formulation and electrolyte preparation. The following gases from
130 Air Liquide were used in electrochemical measurements: Ar (99.999 %), O₂ (99.999 %),
131 CO (99.97 %), and (H₂ 99.999 %).

132 ***Synthesis of Pt/C catalysts.***

133 50 wt. % Pt/Vulcan electrocatalysts were synthesized via the tool-box approach we
134 previously reported⁴⁵. The synthesis method of the electrocatalyst consists of two main
135 steps: preparation of a suspension of colloidal Pt nanoparticles via an alkaline ethylene
136 glycol (EG) route and a subsequent immobilization of the nanoparticles onto carbon
137 support. A colloidal suspension of Pt nanoparticles (ca. 2 nm in diameter) was synthesized

138 by mixing 4 mL of a solution of 0.4 M NaOH in EG with 4 mL of a solution of
139 $\text{H}_2\text{PtCl}_6 \cdot 6\text{H}_2\text{O}$ at 40 mM in EG (NaOH/Pt molar ratio = 10) in a microwave reaction
140 vessel. The mixture was heated for 3 minutes at 160 °C with a microwave reactor (CEM
141 Discover SP, 100 W heating power). Colloidal suspensions of Pt nanoparticles with larger
142 diameters were synthesized by changing the molar ratio between NaOH and H_2PtCl_6 as
143 we recently reported⁴⁴. Pt nanoparticles with diameter of ca. 3 nm and Pt nanoparticles
144 with diameter of ca. 4 nm were synthesized by using a NaOH solution at 0.22 M
145 (NaOH/Pt molar ratio = 5.5) and a NaOH solution at 0.20 M (NaOH/Pt molar ratio = 5.0)
146 instead of the NaOH solution at 0.4 M (NaOH/Pt molar ratio = 10), respectively.

147 In order to support the Pt nanoparticles onto a carbon support, 30 mL of 1 M HCl solution
148 was added to 7.3 mL of the colloidal Pt nanoparticles suspension for precipitation. The
149 mixture was centrifuged at 2400 relative centrifugal force (4000 rotations per minute
150 (rpm), Sigma 2-5) for 5 minutes and the supernatant solvent discarded. This
151 washing/centrifugation with 1 M HCl was repeated twice. Then the Pt nanoparticles were
152 re-dispersed in 7 mL of acetone, and 27.5 mg of carbon black (Vulcan XC72R) was added
153 to the suspension. By sonicating the mixture in an ultrasonic bath (Ultrasonic cleaner,
154 VWR) for 1 hour, the acetone was evaporated, and a dried powder of Pt nanoparticles
155 supported on carbon (Pt/Vulcan) was obtained. The dried powder was re-dispersed into

156 water and sonicated for 10 minutes. The Pt/C catalyst powder was filtered and washed
157 with 200 mL of DI water, and then dried at 100 °C in air. Pt/C catalysts with different Pt
158 nominal loadings (10, 20, 30, 40, 60, and 70 wt. %) were prepared by changing the
159 amount of the carbon support mixed with the Pt nanoparticles in the immobilization step.

160 ***Characterization of Pt/C catalysts.***

161 The particle size distribution of Pt nanoparticles immobilized on the carbon support was
162 determined by transmission electron microscopy (TEM, Tecnai Spirit, FEI, 80 kV or Jeol
163 2100; Jeol 3000F for HR-TEM). The size analysis was performed using ImageJ software
164 (without automated processing of the image) by measuring the size of more than 80
165 nanoparticles by taking micrographs in 5 different areas of the TEM grid. The mean
166 number diameter (d_{MN}) and the mean area diameter (d_{MA}) of the Pt nanoparticles were
167 calculated using the following equations:

168
$$d_{MN} = \frac{\sum d_i}{N}$$

169
$$d_{MA} = \frac{\sum(A_i d_i)}{\sum A_i} = \frac{\sum(\pi d_i^3 / 4)}{\sum(\pi d_i^2 / 4)}$$

170 where d_i , N , A_i are the diameters of the individual particles, the number of the particles,
171 and the surface area of the individual particle, respectively.

172 The actual Pt loadings of the catalysts were evaluated by using UV-Vis spectroscopy⁴⁶.
173 For sample preparation, 2.0 mg of catalyst powder was weighed in a ceramic crucible and

174 heated in a muffle furnace (air, 900 °C, 30 min) to burn off the carbon support. Afterward
175 4 mL of aqua regia (mixture of 30 % HCl and 65 % HNO₃ in a volume ratio of 1:3) was
176 added in the crucible and heated for 90 minutes on a heating plate at ca. 80 °C to digest
177 platinum. After the digestion the final volume of the aqua regia sample was adjusted to
178 100 mL with DI water. 1 mL of the aqua regia sample, 0.25 mL of 1 M SnCl₂ in 4 M HCl,
179 and 0.75 mL of 2 M HCl were mixed in a quartz cuvette and a UV-Vis spectrum of the
180 sample mixture was measured. Subsequently 5 μL of Pt standard solution (1000 ppm)
181 was added to the sample solution and a UV-Vis spectrum of the sample mixture was
182 measured again. The addition of the platinum standard solution and the measurements
183 were repeated 4 times. The absorbance at 402 nm was plotted against the concentration
184 of the added platinum to obtain a calibration curve. Finally, the concentration of platinum
185 in the aqua regia sample was determined from the x-intercept of the calibration curve.

186 Inter-particle distance (edge-to-edge distance) of Pt nanoparticles was estimated using the
187 following equation³⁶:

$$188 \quad d_{ip} = \sqrt{A/N} - d_{NP}$$

189 where d_{ip} , A , N , and d_{NP} are inter-particle distance, BET surface area of the carbon support,
190 number of Pt nanoparticles, and diameter of Pt nanoparticles, respectively. The mean area
191 diameters that were obtained by analyzing the TEM images of 30 wt.% Pt/Vulcan

192 catalysts (see Figure 1) were used as d_{NP} for each respective nanoparticle size. The Pt
193 particle size, nominal Pt loading, actual Pt loading, and inter-particle distance based on
194 actual loading of each Pt/C catalyst are summarized in Table S1.

195 ***RDE measurements.***

196 6.27 mg of the homemade 50 wt. % Pt/Vulcan catalyst powder was mixed with 8 mL of
197 IPA:DI water (1:3, v:v) mixed solvent to formulate a catalyst ink. The Pt concentration in
198 the catalyst ink was always kept at $0.392 \text{ mg}_{Pt} \text{ mL}^{-1}$ for all Pt/Vulcan catalysts with
199 different Pt loadings. A small amount of 1 M KOH solution ($\sim 8 \text{ } \mu\text{L}$) was added to adjust
200 the pH of the ink to be around 10⁴⁷. The glass vial containing the mixture was placed in
201 an ultrasonic bath filled with cold water ($< 5 \text{ } ^\circ\text{C}$) and sonicated for 15 minutes.

202 A glassy carbon (GC) disc electrode (5 mm in diameter, 0.196 cm^2 in geometric area) was
203 polished to mirror finish using alumina oxide paste (0.3 and $0.05 \text{ } \mu\text{m}$ AP-D suspension,
204 Struers), and cleaned ultrasonically in DI water. A $5 \text{ } \mu\text{L}$ aliquot of the catalyst ink was
205 pipetted onto the GC electrode leading to a Pt loading of $10 \text{ } \mu\text{g}_{Pt} \text{ cm}^{-2}$. Subsequently the
206 ink on the GC electrode was dried under Ar gas flow humidified with mixture of IPA and
207 DI water (17:3, v:v) in a bubbler⁴⁶. After the drying, the surface of the GC electrode was
208 checked with the help of a CCD camera to confirm that the electrode surface was
209 uniformly covered with the catalyst thin film.

210 An in-house Teflon cell based on a three-compartment configuration was employed in all
211 electrochemical measurements⁴⁸. A platinum mesh was used as a counter electrode. A
212 saturated calomel electrode (B3510+, Schott) was employed as a reference electrode and
213 placed in a sub-compartment separated by a Nafion membrane to avoid the contamination
214 of chloride ions into the main compartment. All potentials in this study are referred to the
215 reversible hydrogen electrode (RHE) potential, which was experimentally determined for
216 each measurement series. The Teflon cell and components were soaked in mixed acid
217 ($\text{H}_2\text{SO}_4:\text{HNO}_3 = 1:1, \text{v:v}$) overnight. Subsequently the Teflon cell and other components
218 were rinsed thoroughly by DI water, and boiled in DI water twice.

219 All electrochemical measurements were performed using a computer controlled
220 potentiostat (ECi 200, NordicElectrochemistry) and the Teflon cell with an RDE rotator
221 (EDI101, Radiometer Analytical). The measurements were performed in 0.1 M HClO_4
222 solution at room temperature. Prior to the measurements, the electrolyte was de-aerated
223 by purging with Ar gas, and the catalysts were cleaned by potential cycles between 0.05
224 and 1.20 V at a scan rate of 500 mV s^{-1} until a stable cyclic voltammogram was observed
225 (typically 30–50 cycles). The electrochemical surface area (ECSA) of the catalyst was
226 determined by conducting CO stripping voltammetry²¹. The working electrode was held
227 at 0.05 V during purging CO through the electrolyte followed by saturating with Ar. The

228 ECSA was calculated from the CO oxidation charge recorded at a scan rate of 50 mV s^{-1}
229 using conversion coefficient of $390 \mu\text{C cm}^{-2}_{\text{Pt}}$. To determine the ORR activity, linear
230 sweep voltammetry (LSV, positive scan) was conducted in O_2 -saturated electrolyte at a
231 scan rate of 50 mV s^{-1} and a rotation speed of 1600 rpm. The polarization curves were
232 corrected for the nonfaradaic background by subtracting the cyclic voltammograms (CVs)
233 recorded in Ar-purged electrolyte at the identical scan rate. Furthermore, the solution
234 resistance (R_{sol}) between the working electrode and the Luggin capillary ($\sim 23 \Omega$) was
235 determined using an AC signal (5 kHz, 5 mV) and thereafter compensated for using the
236 potentiostat's analog positive feedback scheme. The resulting effective solution
237 resistance was less than 3Ω for each experiment. The ORR kinetic current density j_k was
238 extracted from the measured current density j using Koutecky-Levich equation described
239 as: $1/j = 1/j_k + 1/j_l$
240 where j_l is diffusion limiting current density. In this study, the ORR activity of the
241 catalysts was always evaluated at 0.9 V vs. RHE.

242 ***XAS measurements.***

243 XAS measurements were carried out at the SuperXAS (X10DA) beamline, Swiss Light
244 Source (SLS), Switzerland (energy of storage ring 2.4 GeV; beam current 400 mA) and
245 at the B18 beamline of the Diamond Light Source, UK (energy of storage ring 3 GeV;

246 beam current 300 mA). N₂ was used to fill both ionization chambers for the detection of
247 incident (I₀) and transmitted (I₁ and I₂) X-ray radiation. For *ex situ* XAS measurements,
248 a mixture of Pt/Vulcan powders and cellulose as a binder was pressed to form a
249 homogeneous pellet with a uniform thickness. The *operando* electrochemical XAS
250 experiments were performed in a home-made spectro-electrochemical flow cell with a
251 three-electrode configuration. The basic cell design has been reported elsewhere⁴⁹. Some
252 modifications like electrolyte thickness, diameter of channels, compartment of the
253 reference electrode, etc. were made to optimize the signal-to-noise ratio for the XAS
254 measurements at the Pt L₃ edge in transmission mode. A dispersion of Pt/Vulcan in IPA
255 and water was dropped onto a carbon paper. After drying, the resulting catalyst film was
256 used as working electrode. A graphite foil was applied as counter electrode, a non-leakage
257 Ag/AgCl (3 M KCl, EDAQ) serves as a reference electrode. The electrolyte solution (0.1
258 M HClO₄) at a constant rate flew from a reservoir to the cell and then to the waste
259 container. XAS data were analyzed by FEFFIT software suite⁵⁰. The data processing
260 included the background subtraction, edge step normalization and conversion of the
261 energy units (eV) to photoelectron wave vector k units (Å⁻¹) by assigning the
262 photoelectron energy origin, E₀, corresponding to k = 0, to the first inflection point of the
263 absorption edge. Extended X-ray absorption fine structure (EXAFS) spectra were

264 Fourier-transformed to obtain pseudo radial structure functions (RSFs). The amplitude
265 reduction factor (S_0^2) was obtained from the fit of the EXAFS spectrum of a Pt foil to be
266 0.82.

267 *XPS measurements.*

268 X-ray photoelectron spectra (XPS) were measured using an ESCALAB 250 Xi
269 instrument (Thermo Fisher, East Grinstead, UK) with monochromatized Al K α ($h\nu =$
270 1486.6 eV) radiation and an in-lens electron source was used for charge compensation.
271 The high-resolution spectrum for Pt 4f was measured with a pass energy of 10 eV, 50 ms
272 dwell time and 0.02 eV steps referenced to graphitic carbon at 284.3 eV. Peak fitting was
273 performed with the XPS Peak41 software using a Gaussian Lorential (G/L) mixed
274 function after a Shirley background subtraction.

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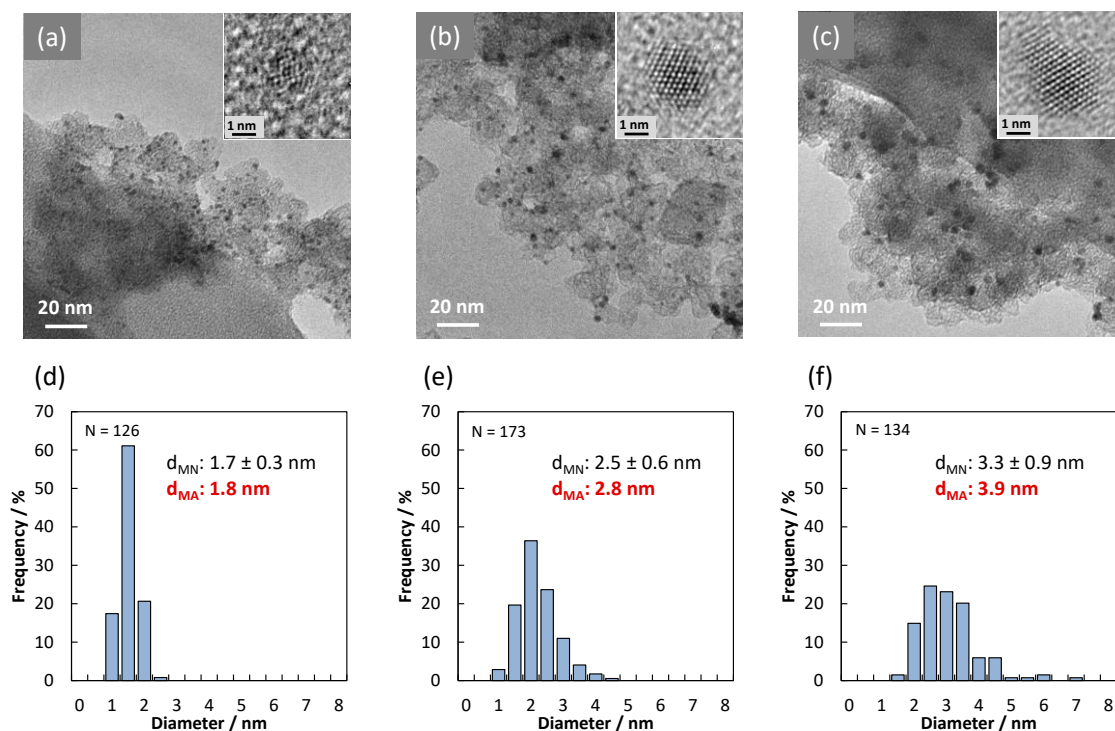
276 **3. Results and discussions**

277 As an example, Figure 1 displays TEM micrographs of 30 wt. % Pt/Vulcan catalysts
278 where the NaOH/Pt molar ratio for the Pt nanoparticle preparation was 10, 5.5, and 5.0,
279 respectively. High resolution (HR) TEM signifies that all nanoparticles are highly
280 crystalline (see insets of HR-TEM of the unsupported face centered cubic (*fcc*) Pt
281 nanoparticles) and vary in particle size. The mean number diameters (d_{MN}) increase from

282 1.7 to 2.5, and 3.3 nm, respectively, and the mean area diameters (d_{MA}) from 1.8 to 2.8,

283 and 3.9 nm, respectively.

284



285

286 Figure 1. TEM micrographs of 30 wt. % Pt/Vulcan catalysts (a-c) and respective particle
287 size distributions (d-f). NaOH/Pt molar ratios for the Pt nanoparticle synthesis were 10
288 (a, d), 5.5 (b, e), and 5.0 (c, f), respectively. The insets show HR-TEM micrographs of
289 unsupported Pt nanoparticles. N, d_{MN} , and d_{MA} stand for number of analyzed Pt
290 nanoparticles, mean number diameter, and mean area diameter, respectively.

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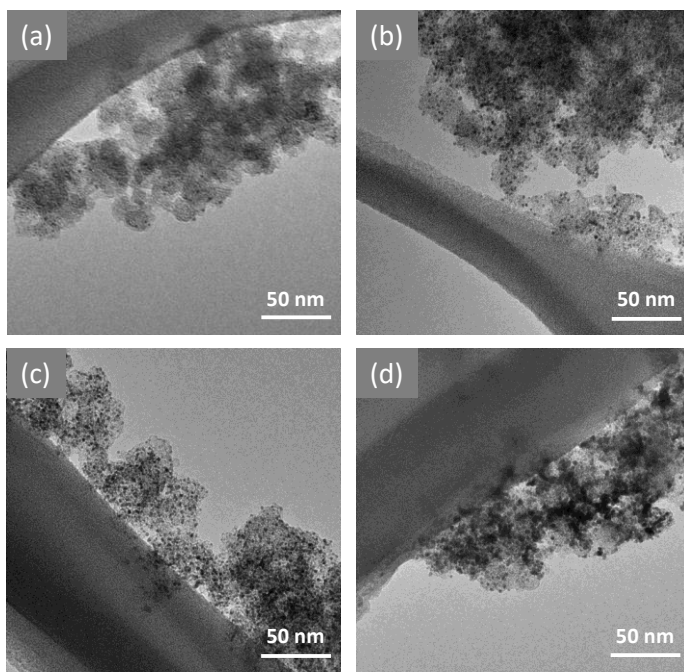
292 Varying the inter-particle distance at constant particle size is a more challenging exercise.

293 By mixing the colloidal nanoparticle suspensions with different amounts of Vulcan
294 carbon support we varied the nominal metal loading between 10 – 70 wt. % Pt/Vulcan for

295 each colloidal nanoparticle suspension. Figure 2 displays TEM micrographs of 10, 30, 50,

296 and 70 wt. % Pt/Vulcan catalysts prepared from the colloidal suspension of 1.8 nm sized
297 Pt nanoparticles. It is seen that the Pt nanoparticles are highly dispersed on the carbon
298 support up to the nominal Pt loading of 50 wt. % (Figure 2 (a) – (c)), whereas some minor
299 agglomeration of Pt nanoparticles can be observed at very high nominal metal loadings
300 like 70 wt. % Pt/Vulcan (Figure 2 (d)). The mean diameter of the Pt nanoparticles is
301 almost independent of the metal loading. Only a small increase in the number of slightly
302 larger Pt particles (> 2.5 nm) is observed for very high loading (70 wt. %, see Figure S1).

303



304

305 Figure 2. TEM micrographs of Pt/Vulcan catalysts prepared by using 1.8 nm sized Pt
306 nanoparticles (NaOH/Pt molar ratio = 10). The nominal Pt loadings are (a) 10 wt. %, (b)
307 30 wt. %, (c) 50 wt. %, and (d) 70 wt. %, respectively.

308

309 Varying the metal loading with constant particle size leads to a systematic change in inter-
310 particle distance on the carbon support. The successful variation of the inter-particle
311 distance at constant particle size without agglomeration is best scrutinized by determining
312 the electrochemically active surface area (ECSA) as function of the metal loading. If the
313 ECSA is independent of the metal loading the absence of particle agglomeration is
314 confirmed. The average inter-particle distance can then be calculated from the BET
315 surface area of the carbon support, the number of Pt nanoparticles on the support as well
316 as the diameter of Pt nanoparticles, see experimental section. In Figure 3, the results of
317 the electrochemical characterization via rotating disk electrode (RDE) measurements are
318 summarized (representative cyclic voltammograms and CO stripping voltammograms of
319 the Pt/C catalysts and the analysis of the cyclic voltammograms are displayed in the
320 supplementary information in Figures S2, S3, and S4, respectively.). In Figure 3, each
321 color represents a different Pt particle size, i.e., 1.8, 2.8, and 3.9 nm in average diameter.
322 It is seen that the ECSA is indeed almost constant for each Pt particle size independent of
323 the metal loading. Only with a very large number of nanoparticles immobilized on the
324 support (high metal loadings and small nanoparticles) a substantial decrease in ECSA is
325 observed, indicating particle agglomeration during the particle immobilization step of the
326 catalyst preparation. In these cases, particle size and inter-particle distance effects are

327 interwoven due to the limitations in catalyst synthesis. In the other cases, the ECSA values
328 for each sample are consistent with the individual particle size determined from the TEM
329 data assuming that 67 % of the particle surface of the supported Pt nanoparticles is
330 electrochemically accessible (see Figure S5).

331 The influence of particle size and inter-particle distance on the ORR activity is
332 demonstrated in Figures 3(b) and 3(c). It is seen that the Pt surface area-based specific
333 ORR activity (SA) (i.e., the reaction rate normalized to the number of catalytically active
334 sites) depends on the particle size as well as on the inter-particle distance. For smaller
335 (1.8 nm) Pt nanoparticles the influence of the inter-particle distance on the SA is most
336 prominent. The SA at 0.9 V_{RHE} increases by ca. 220 % (all values are rounded) when the
337 average inter-particle distance is decreased from 11.5 nm to 1.4 nm. This trend is opposite
338 to the one reported by Watanabe et al. in the 1980's where the SA decreases with a
339 decrease in the inter-particle distance between 14-20 nm²⁸. It should be noted that the
340 inter-particle distance values are not exact when Pt agglomeration is suspected and the
341 ECSA is decreased (60 and 70 wt.% for the 1.8 nm sized Pt particles) because its influence
342 is not considered in the inter-particle distance calculation. However, in the present work,
343 even if the data points where Pt agglomeration is suspected are discarded, the SA still
344 increases by 170 % with decreasing inter-particle distance from 11.5 nm to 2.7 nm. The

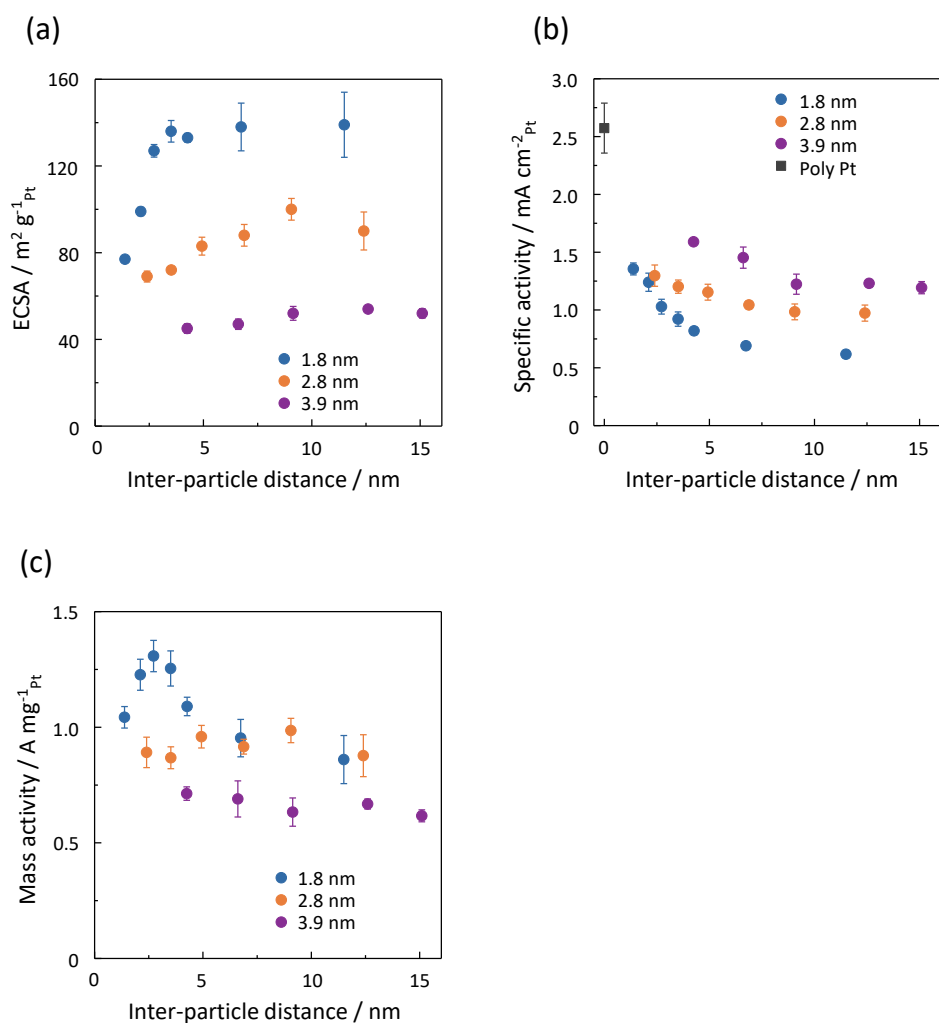
345 increase in SA with the particle size can be extracted based on the activity values at high
346 inter-particle distance. Here, the nanoparticles can be considered as individual, supported
347 particles. Based on the data at 0.9 V_{RHE}, the SA increases by 190 % when increasing the
348 average particle size from 1.8 nm to 3.9 nm. This trend of the SA with the particle size is
349 in good agreement with the results obtained from the measurements of sputtered Pt/GC
350 (glassy carbon) model electrodes²³ and Nafion-free measurements of carbon supported
351 Pt/C catalysts⁵¹ reported in literature, see Figure 4. In general, this “intrinsic” particle size
352 effect can be related to a shift in the reduction potential of Pt oxide species, see ref.¹⁹ and
353 Figure 5. The results demonstrate that the inter-particle distance is equally important as
354 the particle size for the SA of the ORR. Analyzing the Tafel plots of the SA with changing
355 the inter-particle distance at same loadings, see Figure S6, demonstrates that the Tafel
356 slopes of the respective catalysts are identical which confirms that these results are not
357 due to experimental artifacts in the measurement but due to changes in reaction kinetics.

358

359 The trend for the mass activity (MA), i.e., the reaction rate per mass of Pt, is more
360 complicated as the MA depends on the SA and the ECSA (MA = SA · ECSA). At large
361 inter-particle distance (low metal loading) the MA of the 1.8 and 2.8 nm Pt nanoparticle
362 catalysts is the same, while the 3.9 nm nanoparticle catalyst has a slightly lower MA.

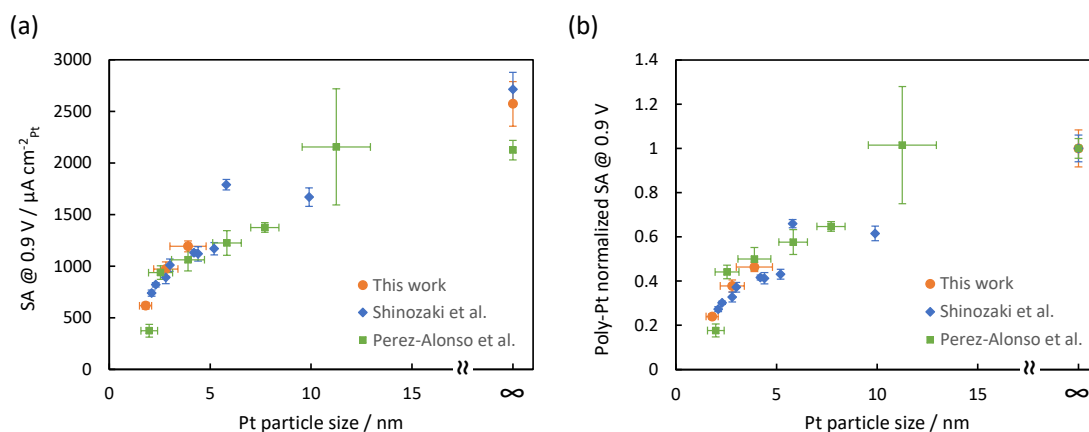
363 However, decreasing the inter-particle distance (increasing metal loading), the MA of the
364 1.8 nm Pt nanoparticle catalyst exhibits a maximum due to particle agglomeration (see
365 Figure 2d). First, the MA increases due to the particle proximity effect and then decreases
366 due to a loss in ECSA because of agglomeration. In summary, the optimization of particle
367 size and inter-particle distance in the here prepared catalyst leads to a maximum MA of
368 1300 A g^{-1} at $0.9 V_{\text{RHE}}$ for the Pt/Vulcan catalyst, which is more than twice of the MA we
369 determined for standard commercial Pt/C catalysts¹⁵.

370



371
 372 Figure 3. ECSA (a), ORR specific activity (SA) at 0.9 V_{RHE} (b) and ORR mass activity
 373 (MA) at 0.9 V_{RHE} (c) of the Pt/Vulcan catalysts with various Pt loadings and Pt
 374 nanoparticle sizes determined from the RDE measurements. Each measurement was
 375 repeated more than 4 times and error bars show 95 % confidence interval.

376

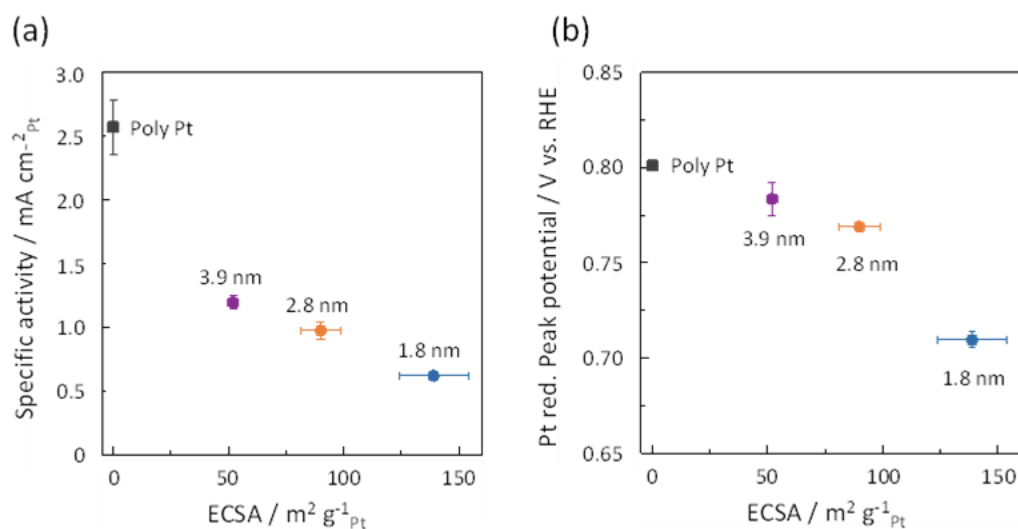


377

378

379 Figure 4. (a) Particle size dependence of the specific ORR activity (SA) at 0.9 V vs. RHE
 380 in 0.1 M HClO₄ electrolyte measured in this study in comparison with literature values.
 381 The SA values for this study were taken from Figure 5a. (b) Normalized SA dependence
 382 on the particle size obtained in this study as well as found in literature. Literature values
 383 were taken from Shinozaki et al.²⁵ and Perez-Alonzo et al.²⁴.

384



385

386

387 Figure 5. “Intrinsic” particle size effects on (a) ORR specific activity and (b) Pt-oxide
 388 reduction peak potential. Measured values of the 3 different sized Pt nanoparticles at the
 389 lowest Pt loading (10, 20, and 30 wt. % in nominal Pt loading for 1.8, 2.8, and 3.9 nm

390 sized Pt nanoparticles, respectively) are plotted as a function of ECSA. The experimental
391 data for polycrystalline Pt is added to the plot at an ECSA value of 0.

392

393 As mentioned above, different effects can cause the observed particle proximity effect. In

394 literature effects ranging from changes in the EDL structure at the catalyst-electrolyte

395 interface¹⁴ to changes in the intrinsic electronic properties of the nanoparticles due to a

396 screening of the charge transfer barrier due to electron tunneling³⁸ are discussed.

397 Previously, it has been reported that for smaller particle sizes the Pt-OH reduction peak

398 potential in cyclic voltammograms (CVs) of Pt/C measured in Ar purged electrolyte shifts

399 to more positive values with decreasing inter-particle distance³⁶. This behavior which is

400 often associated with higher ORR activity¹⁹ is here confirmed with the prepared Pt/Vulcan

401 catalysts, see Figure 6. In Figure 6, it is seen that the reduction peak potential value of the

402 Pt nanoparticles approaches that of bulk Pt with a decrease in the inter-particle distance.

403 The observed trends in oxophilicity suggest that the surface properties of the Pt

404 nanoparticles become more “bulk-like” with decreasing the inter-particle distance.

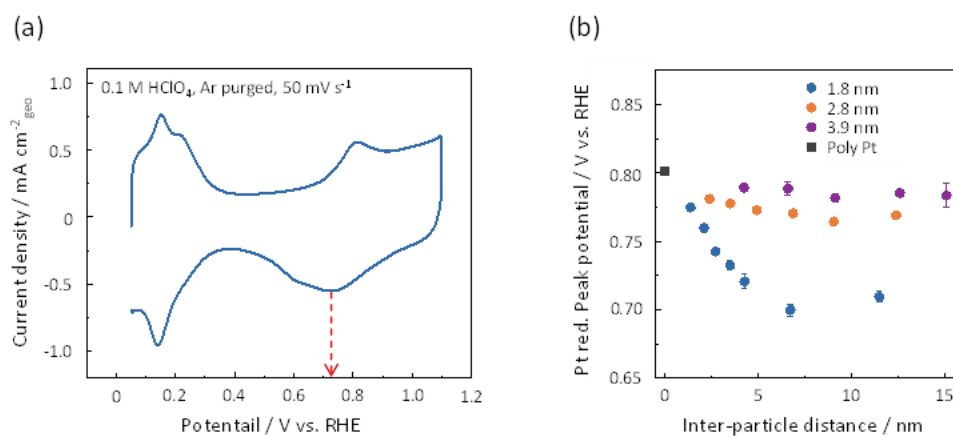
405 Interestingly, the proximity effect on the Pt reduction peak potential as well as the increase

406 in SA are the most prominent for small (1.8 nm) Pt nanoparticles. This could be related

407 to the fact that the “larger” Pt nanoparticles are more “bulk-like” to begin with (i.e., at

408 large inter-particle distance). A similar trend, i.e., indications that the surface properties

409 of the Pt nanoparticles become more “bulk-like” with decreasing the inter-particle
410 distance can be seen by an analysis of the CO oxidation peaks in the CO stripping
411 voltammograms (Figure S7). While the peak position of the CO oxidation peaks is
412 independent of the inter-particle distance, its width decreases with decreasing inter-
413 particle distance.



414 Figure 6. (a) Typical cyclic voltammogram of Pt/Vulcan catalyst measured in Ar-purged
415 0.1 M HClO₄ electrolyte. In this case, Pt reduction peak is seen at around 0.75 V. (b)
416 Reduction peak potential of the Pt/Vulcan catalysts with various metal loadings and
417 nanoparticle sizes.
418

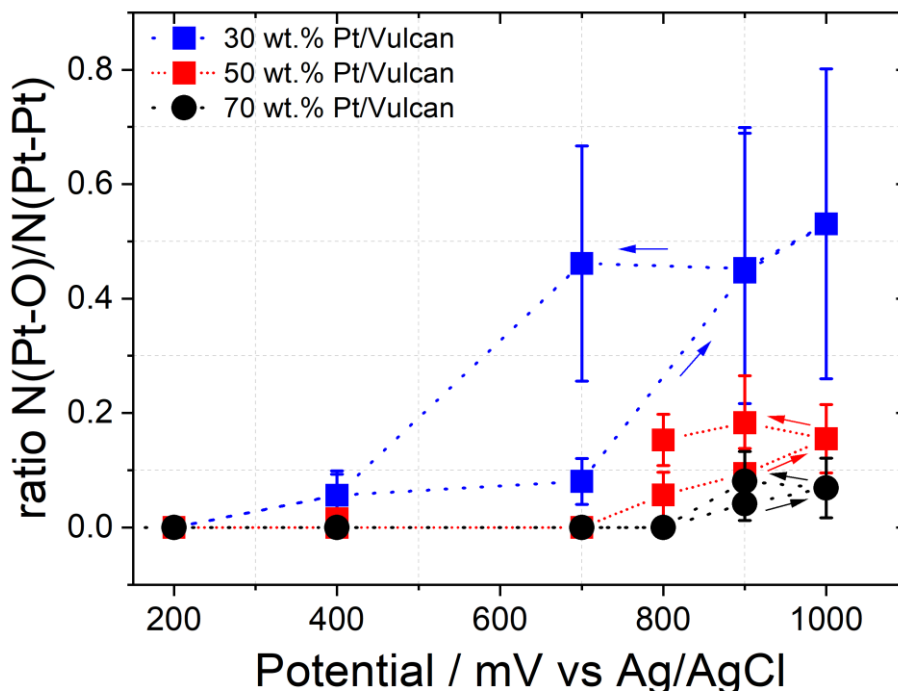
419
420 The trend in oxophilicity can also be observed in X-ray photoelectron spectroscopy (XPS)
421 and X-ray absorption spectroscopy (XAS). As previously reported¹² *ex situ* Pt 4f XPS
422 spectra, Figure S8, demonstrate that after preparing Pt/Vulcan catalysts with the same
423 nanoparticle size (1.8 nm), but different inter-particle distance and exposing them to air,
424 their degree of oxidation state decreases with decreasing inter-particle distance

425 (increasing metal loading). The same trend in reduced oxophilicity with decreasing inter-
426 particle distance is observed in *ex situ* recorded extended X-ray absorption fine structure
427 (EXAFS) and X-ray absorption near edge structure (XANES) measurements, see Figures
428 S9, S10 and Table S2. Decreasing the inter-particle distance (increasing metal loading)
429 the determined Pt-O coordination numbers decreases, while the Pt-M coordination
430 numbers stay almost constant with slight variations.)¹². Concurrently, the white line
431 intensity in the Pt L₃ edge XANES spectra decreases¹². One could argue that the
432 established trend in reduced oxophilicity with decreasing inter-particle distance is
433 observed *ex situ* after exposure to air and therefore not relevant for the electrochemical
434 performance.

435 Therefore, considerable effort was undertaken to establish *operando* spectroscopic
436 evidence of the particle proximity effect in addition to the trends of the specific ORR
437 activity and the shift in the Pt-OH reduction peak potential obtained from the CVs. Here,
438 to the best of our knowledge we demonstrate for the first time such evidence using
439 *operando* electrochemical XAS. In Figure 7, it is shown that XAS spectra recorded under
440 potential control reveal the same trend in oxophilicity with inter-particle distance as the
441 *ex situ* data. Plotting the ratio of the Pt-O and Pt-Pt coordination numbers extracted from
442 the *operando* Pt L₃ edge EXAFS spectra as a function of the electrode potential for the

443 different Pt/Vulcan catalysts it can be seen that the degree of oxidation (i.e., at fixed
 444 electrode potential) changes with decreasing the inter-particle distance (increasing the
 445 metal loading). Furthermore, the hysteresis^{52,53}, i.e., difference in oxide coverage when
 446 stepping the electrode potential in positive or negative direction, is less pronounced with
 447 the highly loaded Pt/Vulcan catalyst. The data clearly demonstrate that a small inter-
 448 particle distance of the Pt nanoparticles supported on the Vulcan leads to a lower oxide
 449 coverage, which in turn leads to an improved specific ORR activity.

450



451

452 Figure 7. *Operando* electrochemical EXAFS data of 1.8 nm Pt/Vulcan catalysts with
 453 different metal loadings, 30 wt. %, 50 wt. %, and 70 wt. %, that results in different inter-
 454 particle distances. The plots show the calculated ratio of both coordination numbers N(Pt-

455 O) and N(Pt-Pt) as a function of the applied electrode potential. The arrows indicate if the
456 spectra were recorded coming from a lower or higher electrode potential. Measurements
457 with 10 wt. % were not feasible due to too thick catalyst layers required for good quality
458 XAS spectra. The error bars indicate the standard error obtained from the fit analysis
459 using the EXAFS equation.

460

461 **4. Conclusions**

462 In an extensive effort of varying particle size and inter-particle distances for Pt based
463 ORR catalysts independent of each other, we manage to disentangle the influence of the
464 inter-particle distance and the particle size on the specific ORR activity. It is found that
465 both effects are equally important when optimizing the specific ORR activity for a
466 catalyst. Larger Pt nanoparticles exhibit a higher specific ORR activity than smaller ones,
467 when the inter-particle distance is large. However, the specific ORR activity of Pt
468 nanoparticles can be improved when immobilizing them at high coverage on a carbon
469 support. This influence of the inter-particle distance on the catalytic ORR activity is most
470 pronounced for smaller nanoparticles as compared to larger ones. Although high metal
471 loadings also favour particle agglomeration during synthesis, in particular for small
472 nanoparticles a clear increase in activity with the inter-particle distance is seen when
473 limiting the discussion to the metal loadings where the ECSA proves constant dispersion
474 (and thus constant particle size). As a consequence, we observe the highest mass
475 normalized ORR activity for small nanoparticles at high coverage.

476 As compared to the long-discussed particle size effect, inter-particle distance effects are
477 a fairly recent topic in catalysis and the underlying mechanisms have been addressed by
478 only few research groups. Essentially two explanation models have been suggested in
479 literature. One focussing on the effect of the inter-particle distance on the EDL
480 structure^{14,37} and one focusing on electronic cross-coupling effects as a result of changes
481 in the screening of charge transfer barriers between the particles³⁸.

482 Concomitantly with the increase in ORR activity with decreasing inter-particle distance,
483 a decrease in oxophilicity is observed. In the present work we could for the first time
484 demonstrated the correlation of the particle proximity effect to a reduced oxophilicity of
485 the Pt nanoparticles by *operando* electrochemical XAS. Although none of the explanation
486 models of the particle proximity effect can be excluded. Two findings point towards the
487 importance of electronic cross coupling effects. First, the particle proximity effect is
488 clearly more pronounced at small nanoparticles. Second, the same trend of reduced
489 oxophilicity at low inter-particle distance is observed in *operando* XAS and *ex situ* XAS
490 and XPS spectroscopy. Therefore, we propose that the electronic properties of
491 nanoparticles are influenced by the inter-particle distance in a more fundamental way than
492 only by a change in the EDL. As a consequence, mesoscopic properties such as the inter-
493 particle distance are an important design parameter for catalytic processes.

494

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504

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506

507 **Author contributions:**

508 M.I., A.Z, and M.A. designed the electrochemical measurements. M.I. performed and
509 analysed the RDE and TEM measurements. S.B.S. and J.Q. performed and analysed the
510 HR-TEM measurements. M.O., A.D., J.Q., A.Z., F.B. performed the XAS measurements,
511 that were analysed by M.O. and A.D. C.D. recorded the XPS measurements that were

512 analysed by A.Z. M.I. and M.A. wrote the first draft of the paper. All authors participated
513 in writing the final draft of the paper.

514

515 **Statement on competing financial interest's:**

516 There are no financial interests of the authors involved in this work.

517

518 **Additional information:**

519 Supplementary information accompanies this paper.

520 Relationship between particle diameter and electrochemically active surface area, Tafel
521 plots, Pt reduction peak potentials, *Ex situ* Pt 4f XPS spectra, *Ex situ* Pt L3 XAS spectra,
522 FT-transformed magnitudes of the k²-weighted EXAFS spectra, and results from fitting
523 of the *ex situ* EXAFS spectra

524

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