Surface NMR using quantum sensors in diamond

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Characterization of the molecular properties of surfaces under ambient or chemically reactive conditions is a fundamental scientific challenge. Moreover, many traditional analytical techniques used for probing surfaces often lack dynamic or molecular selectivity, which limits their applicability for mechanistic and kinetic studies under realistic chemical conditions. Nuclear magnetic resonance spectroscopy (NMR) is a widely used technique and would be ideal for probing interfaces due to the molecular information it provides noninvasively. However, it lacks the sensitivity to probe the small number of spins at surfaces. Here, we use nitrogen vacancy (NV) centers in diamond as quantum sensors to optically detect nuclear magnetic resonance signals from chemically modified aluminum oxide surfaces, prepared with atomic layer deposition (ALD). With the surface NV-NMR technique, we are able to monitor in real-time the formation kinetics of a self assembled monolayer (SAM) based on phosphonate anchoring chemistry to the surface. This demonstrates the capability of quantum sensors as a new surface-sensitive tool with sub-monolayer sensitivity for *in-situ* NMR analysis with the additional advantage of a strongly reduced technical complexity.

INTRODUCTION

Characterization of surface processes at the molecular level is important for understanding fundamental processes that are key to industrial catalysis, energy conversion, electronic circuits, targeted drug delivery, and biosensing [1]. However, many analytical techniques used in surface science are inaccessible under ambient or chemically relevant conditions. Therefore, it remains challenging to perform chemical analysis under the conditions in which these processes occur [2, 3]. Typical surface-sensitive methods, such as Xray photoelectron spectroscopy (XPS), Auger electron spectroscopy and secondary ion mass spectroscopy (SIMS) can perform chemical analysis but require ultra-high vacuum conditions and expensive equipment [4]. To address this gap, great efforts have been devoted to extending XPS analysis to near ambient conditions [2]. Indeed, both near ambient pressure XPS and extended X-ray absorption fine structure (EXAFS) have greatly extended the applicability of these techniques for understanding reaction mechanisms at chemically active interfaces [2, 5]. However, to achieve high sensitivity and resolution both methods require intense synchrotron radiation which limits their practical accessibility and increases their cost. Ultrafast spectroscopy techniques, such as sum frequency generation (SFG) and second harmonic generation (SHG) techniques, can perform analysis under ambient conditions but require technically demanding equipment, such as femtosecond lasers. Additionally, the extracted data are often difficult to interpret [6]. Even with all these techniques available, molecular dynamics or chemical reaction kinetics at surfaces are still difficult to probe experimentally [7].

Nuclear magnetic resonance (NMR) spectroscopy is one of the major tools for chemical and structural analysis in chemistry and material science. Solid-state NMR in particular [8] has advanced understanding of a range of systems including metalorganic frameworks (MOFs) [9], batteries [10], and catalysts [11]. However, sensitivity remains a challenge for traditional NMR spectroscopy, making studies at surfaces difficult due to the limited numbers of nuclear spins. Even in highly porous materials with greater than $1000 \text{ m}^2/\text{g}$ surface area, the concentration of NMR active nuclei of interest often remains low (e.g. 1 mmol of surface atoms/g) [12], which requires long averaging times to obtain solid-state NMR spectra with reasonable signal-to-noise ratio (SNR). As a consequence, the technique is not broadly suitable for in-situ kinetic studies of diverse materials over typical reaction time scales. Recently, surface enhanced NMR spectroscopy relying on hyperpolarization, such as dynamic nuclear polarization ([12, 13] or xenon based techniques [14] gained interest. However, these methods are still limited to porous materials or require powders which can be difficult to prepare [12].

Here, we demonstrate the use of quantum sensors in diamond as a new surface-sensitive spectroscopy technique that works at ambient conditions and can probe planar interfaces on the microscopic length-scale with far greater sensitivity than conventional NMR in a totally non-invasive manner and at much lower costs and reduced technical complexity. The spectroscopic technique relies on the nitrogen vacancy (NV) point defect, consisting of a nitrogen impurity (N) and an adjacent vacancy (V) in the carbon lattice of diamond. These spin-1 defects allow for optical detection of magnetic resonance (ODMR) and have been established as highly sensitive nanoscale magnetic field sensors [15, 16]. Shallow NV-centers are sensitive to magnetic fields from the Larmor precession of nuclei at the surface. This enables nanoscale NMR detection - even down to a single molecule [17] or spin [18, 19]. The measurement volume of such NV sensors [20, 21] corresponds to a hemisphere whose radius is roughly their depth below the surface in the diamond lattice (e.g. 5-10 nm). At this small length scale, the thermal polarization of the nuclear spins can be neglected, since spin noise dominates for a small number of spins [22, 23]. For that reason the NMR signal strength is independent of the magnetic field B₀, reducing experimental complexity and costs, which makes the technique accessible to a broader community [24]. In most nanoscale NV-NMR experiments the sample (typically viscous oil) is placed directly on the diamond surface. In this work, we demonstrate that NV-centers not only can probe diamond surfaces, but can be used as a general tool to probe chemistry on a variety of surfaces. We used atomic layer deposition (ALD), a technology that can be applied to synthesize a wide variety of films with high thickness precision, to coat the diamond with amorphous aluminum oxide (Al₂O₃). Al₂O₃ provides an exemplary surface of high technical relevance in optoelectronic applications and acts as structural support in a variety of catalytic processes [25]. In a proof-of-concept study for this new surfacesensitive spectroscopic technique, we probe the chemical modification of the Al₂O₃ surface with phosphonate anchoring during the formation of a self-assembling monolayer (SAM) [26]. We demonstrate that this technique offers the non-invasive quality of NMR with the advantages of sub-monolayer sensitivity without the cost or technical demands of other state-of-the-art surface analysis techniques.

RESULTS

Surface NV-NMR. The surface NV-NMR technique is performed on a diamond chip in which 15 N was implanted with a particle flu-



Figure 1. Surface NV-NMR spectroscopy on a functionalized Al_2O_3 surface. a) Schematic of the experiment. Near-surface NV-centers in a 2 mm x 2 mm x 0.5 mm diamond chip are excited with 532 nm laser in a total internal reflection geometry. The resulting spin-dependent photoluminescence from the NV defects is detected with a photodiode. The microwave pulses for quantum control of the spin state of the defects are delivered through a small wire loop. b) NV-centers aligned with the magnetic field have sensing volumes with a radius determined by their distance to the surface, which is between 5-12 nm in our case. Inset: Schematic of organic monolayer formed from 12-pentafluorophenoxydodecylphosphonic acid (PFPDPA) on 1 nm Al₂O₃ deposited on the diamond surface by ALD. c) Correlation spectroscopy pulse sequence. Two blocks of dynamic decoupling XY8-N sequences are correlated by sweeping the time between them (t_{corr}). The time spacing τ between the π pulses is set to half the period of the Larmor frequency of the nuclear spin being sensed. The NV spin state is initialized with a 532 nm laser pulse and PL detection with a photodiode occurs after the microwave pulse sequence.

ence of $2x10^{12}/\text{cm}^2$ and with an energy of 2.5 keV (Fig. 1 a) (see Supplementary Note 1 and 2). This results in a distribution of nearsurface NV-centers ~ 5-12 nm below the surface [27]. For these implant conditions, we estimate an NV density of ~ 50-100 NVs/ μ m² [28–30], corresponding to 2-4x10⁵ NV-centers for the ~ 4000 μ m² spot used in our experiments as shown in Fig. 2 c (see Supplementary Note 3). These defects can have four different orientations in the tetrahedral diamond lattice; so one in four is aligned with the external magnetic field B₀. Consequently, an effective ensemble of ~ 0.5-1x10⁵ NV-centers allows for random NMR sampling of the diamond surface within the laser spot. The NMR detection volume of each NV center is determined by its depth, indicated schematically as blue hemispheres in Fig. 1 b.

The quantum sensing scheme for the detection of NMR signals with NV-centers in diamond has been described in detail before [22, 24, 31, 32]. In brief, the electronic ground state of the NV center is a spin triplet with the Zeeman states $m_s = 0$ and ± 1 , which are separated by approximately 2.87 GHz at zero magnetic field. The degenerate $m_s = \pm 1$ states are typically split by a static external magnetic field and transitions between the Zeeman split states can be addressed by microwave fields. The spin state of the NV center can be initialized in the $m_s = 0$ state with laser excitation at a wavelength of 532 nm and optically read out due to spin-dependent photoluminescence (PL), which is weaker for the $m_s = \pm 1$ states compared to the $m_s = 0$ state. After optical excitation nearly all the NV spins are in the $m_s = 0$ state. Subsequently, transitions between the $m_s = 0$ and $m_s = \pm 1$ can be coherently controlled with microwave pulses. For nuclear spin detection, dynamic decoupling sequences (such as XY8-N) are used. These sequences are sensitive to frequencies corresponding to $1/(2\tau)$, where τ is the spacing between the π pulses (Fig. 1 c). Sweeping the time t_{corr} between two XY8-N sequences correlates oscillating magnetic signals, such as spin noise, from nanoscale nuclear spin ensembles. The detected spin noise appears as oscillations in the PL readout as a function of t_{corr} and resembles the free induction decay in traditional NMR spectroscopy. However, this nanoscale NMR spectroscopy can be performed without expensive magnets due to the detection of spin noise. Importantly, there is no need to excite the nuclear spins with radiofrequency pulses or to wait for nuclear spin-lattice relaxation.

Characterization of the Al_2O_3 **layer.** In previous work, NMR signals from samples directly on the diamond surface have been detected [22, 29]. Here, the goal is to probe non-diamond surfaces and interfaces. This requires the preparation of a material of interest on top of the diamond substrate. In our proof-of-concept study we use an Al_2O_3 film prepared by ALD [33], whose surface modification with organophosphonate chemistry shall be investigated through the surface NV-NMR technique. First, we optimized the thickness of the ALD layer by keeping it as thin as possible allowing the NV-centers to sense the surface modification, while also ensuring that it was thick enough to create a closed film onto which a dense molecular monolayer could be bound. The formation of the fluorinated monolayer on the Al_2O_3 surface increases the surface hydrophobicity,



Z (nm)

ALD Al₂O₃ on diamond

c Surface NV-NMR

Figure 2. Surface NV-NMR and validation with complimentary analytical surface techniques. a) Diamond coated with a Al_2O_3 layer. The thickness of the ALD deposited Al_2O_3 layer was determined by AFM scratching measurements. The height profile along the segment indicated by a blue arrow was used to determine the Al_2O_3 film thickness of 0.9 nm \pm 0.1 nm. b) Functionalized Al_2O_3 surface on diamond. The presence of PFPDPA molecules on the surface is confirmed with XPS by the appearance of F 1s and P 2s peaks (blue), which are absent on the clean diamond (yellow). c) Surface NV-NMR spectroscopy. Top: Image of the laser spot (~ 4000 μ m²) on the diamond and time domain correlation signal of ¹⁹F. Bottom: Surface NV-NMR spectrum of ³¹P detected from the monolayer measured at 174 mT and ¹⁹F nuclei detected at 31 mT. The clean diamond reference is shown in yellow.



a Atomic Force Microscopy

Figure 3. Influence of molecular structure on ¹⁹F resonance linewidth. The ¹⁹F linewidth of a monolayer made from PFOPA is \sim 4 times broader than that of PFPDPA. This is likely caused by local dynamics of the fluorinated phenolic moeity which leads to line narrowing of the NMR signal.

which can be investigated with static water contact angle (SWCA) measurements. The minimal ALD layer thickness required to facilitate the formation of a dense SAM was therefore determined using SWCA measurements as a function of Al_2O_3 thickness from 0.5 nm to 3 nm. Saturation of the SWCA signal appears for ALD layers of

1 nm and beyond (see Supplementary Note 4), which indicated that we reached the minimal thickness required for the organic monolayer to be fully formed. The hardness of the diamond chip allows for a scratching experiment to corroborate the thickness of the Al₂O₃ layer on the diamond using atomic force microscopy (AFM) (Fig. 2 a). Removing the Al_2O_3 with the AFM tip revealed two different surfaces - the Al₂O₃ with a root-mean-square (RMS) roughness of 0.71 nm and the underlying diamond with a roughness of 0.25 nm. A vertical cut showed a step height of 0.9 nm \pm 0.1 nm, confirming the thickness expected for the number of ALD cycles [34]. Second, we ensure that the material preserves the NV center properties by quantifying the coherence times before and after depositing the Al₂O₃ film on the diamond. We observe a small reduction in the spin-lattice relaxation T_1 (≤ 10 percent) and spin-spin relaxation T_2 (≈ 25 percent) times (Table 1), with only a minor influence on the sensitivity.

Table 1: Influence of Al₂O₃ on NV center relaxation properties

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	$T_2 (\mu s)$	T_1 (ms)
Clean diamond $(N = 5)$	5.59 ± 0.13	0.80 ± 0.28
After ALD $(N = 8)$	4.19 ± 0.24	0.73 ± 0.30

Chemical characterization of the functionalized Al_2O_3 support. Following thickness optimization, we analyzed the chemical composition of the functionalized Al_2O_3 layer. We selected a C_{12} chain phosphonic acid (PA) terminated with a fluorinated phenolic ring (12-pentafluorophenoxydodecylphosphonic acid (PF-PDPA)) for chemical modification of the Al_2O_3 surface. Fig. 1 a illustrates the bridged bi-dentate binding motif. We note, however, that multiple binding modes might be present on the surface [35]. These flourinated monolayers can be easily prepared by soaking the

4



Figure 4. **Probing surface chemistry** *in-situ.* a) Schematic of monolayer formation on Al_2O_3 in phosphonic acid (PA) solution. b) Individual spectra of ¹⁹F showing the time evolution for a 1 μ M solution. The peak around 1.35 MHz can be assigned to protons (¹H) which are known to be ubiquitous on or within the diamond and are not characteristic of the surface chemistry. c) Real-time monitoring of the ¹⁹F NMR signal amplitude growth in solution for three different PA concentrations shows a decrease in formation rate for lower concentrations. Small markers are measured data points of repeated experiments connected with a vertical line, averaged data are shown with the large points. Data points are fit with a single exponential. Background signal of a clean, non - Al_2O_3 coated diamond in 10 μ M PA solution shows no ¹⁹F signal. Note that due to the absolute value of the Fourier transform, the noise floor is always positive.

 Al_2O_3 /diamond in the PA solution. The functionalization occurs via the binding of phosphonic acid groups to the hydroxy groups of the Al_2O_3 surface [35]. X-ray photoelectron spectroscopy (XPS) confirmed the presence of the F 1s peak and P 2s peaks from the monolayer (Fig. 2 b). Corresponding spectra from the bare substrate prior to functionalization did not contain and detectable F or P. Thus, the phosphorous and fluorine peaks originate from the PFPDPA self-assembled monolayer bound to the Al_2O_3 surface.

The surface NV-NMR technique is capable of providing chemical information from the SAM-functionalized Al_2O_3 surface with high sensitivity, much like XPS but under ambient conditions. The ¹⁹F correlation spectroscopy data provides a time domain NMR signal (Fig. 2 c), clearly showing a oscillation at the ¹⁹F Larmor frequency that decays. The Fourier transform of these data results in the ¹⁹F NMR spectrum with a resonance at 1.247 MHz, which agrees with the theoretical Larmor frequency at 31 mT. Similarly, we can detect the ³¹P signal from the monolayer, which results in a peak at 3 MHz at 174 mT (see Supplementary Note 5). The signal is weaker since the number of spins per molecule and the gyromagnetic ratio are lower, compared to ¹⁹F. Both signals were taken from the same monolayer and laser spot.

We note that other NMR active nuclei are present in this system, most notably ²⁷Al and ¹H. However, the strong ¹³C signal naturally occurring in the diamond overlaps with the ²⁷Al resonance, precluding detection in the current experiment. This can be overcome in the future by using an isotopically enriched ¹²C diamond. The ¹H signal is ubiquitous which is also observed on a clean diamond and cannot be unambiguously attributed to a characteristic of our system. Therefore, it has been excluded from the present analysis.

Characterization of molecular dynamics at the surface. From our measurement of the PFPDPA monolayer a resonance linewidth of ~ 3 kHz was observed, which is narrower than what has been commonly seen in previous NV-NMR experiments of solid samples [29, 36]. For that reason we performed a second set of experiments in which the Al₂O₃ surface was functionalized with a shorter, but perfluorinated PA molecule (1H,1H,2H,2H-perfluoroctanephosphonic acid (PFOPA)) (Fig. 3). The resonance linewidth is much broader (~ 12 kHz) than for the case of the monolayer made from PFPDPA. In the solid state, the linewidth is typically limited by dipolar broadening, which can be minimized by local molecular dynamics such as rotations. The fluorinated phenolic moeity attached to a long carbon chain is more mobile than the ¹⁹F nuclei in the perfluorinated chain of PFOPA, which likely reduces the linewidth [37–39].

Monitoring surface chemistry in real-time. In contrast to other surface sensitive techniques, surface NV-NMR allows for measurements under chemically relevant conditions, for example at the solid-liquid interface. In the present case, this enables the observation of the binding or the phosphonate anchoring group to the Al_2O_3 support at the solid/liquid interface, which is depicted in Fig. 4 a. In particular, the chemical reaction kinetics were directly detected by addition of PFPDPA solution onto a freshly prepared Al_2O_3 layer on a diamond and measurement of the surface NV-NMR signal as a

function of time. Fig. 4 b shows individual surface NV-NMR spectra at different times after adding the PFPDPA solution. The ¹⁹F resonance signal grows on a time scale of tens of minutes and plateaus after approximately half an hour. The broad resonance at 1.325 MHz next to the ¹⁹F signal originates from ¹H, which is present from the beginning and is not characteristic of the surface chemistry. We repeated the experiment using solutions of different concentrations of the PFPDPA, which showed an increase in the monolayer formation rate at higher concentration (Fig. 4 c). The kinetics of the ¹⁹F signal can be modelled with an exponential growth function (see Supplementary Note 6) [40–42]. In a reference experiment, we repeated the experiment with a clean diamond without a Al₂O₃ layer. In this case we did not see any signal at the expected ¹⁹F resonance which indicates that the monolayer is inefficiently formed on diamond.

CONCLUSIONS

The surface NV-NMR technique has been successfully demonstrated to enable the detection of microscopic NMR signals down to sub-monolayer coverages, including with real-time formation of molecular monolayer assembly on Al₂O₃. The use of a diamondbased sensor that is chemically inert and can withstand high temperatures and high pressures is advantageous especially, for chemical applications and catalysis. It will not only bridge the pressure gap in surface science but has the resilience to probe chemical reactions even under harsh conditions *in-situ* [3]. Yet, the technique is still in its infancy and has many possible avenues for improvement. The main challenge is the lack of molecular structural information such as chemical shifts in comparison to traditional NMR spectroscopy. The main reason for the broad (a few kHz) resonances is the dipolar broadening in the solid state. Magic angle spinning, typically used in solid state NMR, may be one way to reduce the linewidth but comes with significant engineering challenges [43]. Alternatives are decoupling pulse sequences, which reduce sensitivity and cannot fully recover high resolution spectra [36]. Another approach is the use of quadrupolar nuclei such as (¹⁴N, ²D, etc.) which have been shown to convey detailed structural information despite their broad lines making them ideal targets for surface NV-NMR [39].

Although the sensitivity did not limit this study, future analysis of sparsely modified surfaces or the detection of nuclei with low gyromagnetic ratios would benefit from further improvement. Advanced NV generation techniques [44, 45] or the growth of preferentially orientated NV-centers [46] can improve the sensitivity significantly. Additionally, an improved readout scheme of the NV quantum state has been shown to increase the sensitivity for single NV-centers by over an order of magnitude [17]. Finally, it should be noted that NV-centers are strongly influenced by magnetic noise. Consequently, para-magnetic materials on the diamond surface will deteriorate the NV coherence times and preclude surface NV-NMR studies of this type. However, other quantum sensing schemes are available to study these types of materials [47].

In summary, the results from this study introduce a new surfacesensitive technique apt for probing the chemical composition of surfaces and chemical reaction kinetics at interfaces. Using the ability of NV-centers to detect NMR signals selectively from nanoscale layers with well-established and versatile ALD techniques, make this a powerful and broadly applicable. As a proof-of-concept, we demonstrate the detection of NMR signals from a phosphonate functionalized Al_2O_3 surface. In contrast to other surface techniques, submonolayer signals can be detected under ambient conditions. This enables us to monitor the chemical formation of the monolayer in real-time in solution at the solid-liquid interface on the microscopic level. We believe that this will facilitate further understanding of a variety of surface phenomena. Not only does this method offer noninvasive benefit of NMR spectroscopy, its functionality under chemically relevant conditions and the low technical complexity make it a practical and sensitive technique for advanced studies in important areas of catalysis, materials science, biological sensing or 2D materials research.

METHODS

Diamond preparation. An electronic grade diamond $(1.1\% {}^{13}\text{C}$ abundance, Element Six) was implanted with ${}^{15}\text{N}$ at an energy of 2.5 keV with an off axis tilt of 7 degrees and with a fluence of $2x10^{12}/\text{cm}^2$ by Innovion. Then, the implanted sample was annealed under vacuum in a home built oven for over 32 hours (see Fig. S1) with a Tectra BORALECTRIC[®] sample heater. The diamond was cleaned before every Al₂O₃ deposition with a tri-acid cleaning protocol involving equal parts boiling sulfuric, nitric, and perchloric acid according to Brown *et al.* [48].

Atomic layer deposition. ALD was performed using a Veeco Fiji G2 system. Prior to deposition, the diamond surface was cleaned insitu by 5x (0.15)s cycles of ozone, which was generated via electrical discharge in O₂. For deposition of Al₂O₃ thin films, the ozonetreated diamond substrates were sequentially exposed to trimethyl aluminum (98-1955, STREM Chemicals) followed by H₂O at 200° C cyclically. Each ALD cycle followed the sequence: TMA pulse/Ar purge/H₂O pulse/Ar purge. This was repeated for 10 cycles to achieve a final thickness of approximately 1 nm (nominal growth per cycle of 0.1 nm/cycle). To achieve a reproducible surface hydroxyl surface termination suitable for monolayer assembly by phosphonate chemistry, the sample was exposed to a remote oxygen plasma within the ALD system. In particular, the RF (13.64 MHz) inductively coupled plasma (ICP) source of the Fiji G2 system was operated at 300 W for a total of 1 min exposure. For repeated use of diamond substrates, the Al₂O₃ was removed by soaking overnight in a 5% NaOH solution (see Supplementary Note 7) before the next ALD deposition.

Phosphonic acid surface functionalization. For the fully formed monolayer shown in Fig. 2 and Fig. 3, the following procedure was used. After the ALD process, the diamond is immersed in a 10 mM solution of either 12-pentafluorophenoxydodecylphosphonic acid (CAS-Number 1049677-16-8, Sigma-Aldrich) or 1H,1H,2H,2H-perfluorooctanephosphonic acid (CAS-Number 252237-40-4, Sigma-Aldrich) in ethanol for 2 days to allow the monolayer formation to equilibrate. Then the diamond is sonicated for 5 minutes in ethanol to remove all physiadsorbed molecules, after which the sample was dried with flowing nitrogen.

Surface NV-NMR. The experiment is based on a modified version of the setup described in Bucher et al. [24] (see details in Supplementary Note 8). Correlation spectroscopy was performed using XY8-4 blocks (a total of 32 π pulses) with t_{corr} swept starting from 2 μ s to obtain the spectra. For ¹⁹F detection t_{corr} was swept until 160 μ s in 801 points. The time domain data was then Fourier transformed and the absolute value plotted using MATLAB. Each spectrum shown in Fig. 2 c is zero-filled with 801 points. For the ¹⁹F signal shown in Fig. 2 c, only 1 average was necessary to obtain a SNR of 95, as calculated by dividing the signal value by the standard deviation of the noise floor within a region without signal. For ³¹P detection, t_{corr} was swept until 80 μ s in 801 points. The ³¹P signal was averaged 2 times for an SNR of 15. The ¹⁹F and ³¹P NMR signals were obtained in 25 minutes and 32 minutes, respectively. For linewidth measurements as shown in Fig. 3, each monolayer was measured to 2501 points and t_{corr} swept to 0.5 ms. After zero-filling to 5001 points, the linewidth of the resonance was fit with a modified

Lorentzian model [49].

In-situ kinetics of monolayer formation. The tri-acid cleaned diamond was first coated with 1 nm of Al_2O_3 and activated with oxygen plasma, as described above. Subsequently, the diamond was glued down to a water-tight liquid sample holder made from a 30 mm cage plate (CP4S, Thorlabs) with a thin round cover slide (100493-678, VWR) glued to the bottom and the top fitted with a threaded lens tube (SM30L03, Thorlabs). The cage plate was then mounted to the surface NV-NMR experiment (see Supplementary Note 8). ¹⁹F is detected with 24 time traces, which were continuously acquired by sweeping t_{corr} to 40 μ s with 201 points. Each data point in Fig. 4 b

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was an average of 4 time traces, which were then Fourier transformed and the ¹⁹F signal amplitude plotted, resulting in 6 points for the kinetic dataset. Each of these experiments are repeated 3 times (1 mM, 10 μ M) and 2 times (1 μ M) and the signal amplitudes are averaged and then fit. Each time trace is normalized to 1 for the time point at 96 minutes. The first point at t = 0 minutes is set to the mean of the noise floor of the Fourier transformed spectra. The background signal is the value at the ¹⁹F frequency within spectra obtained with a clean diamond measured in the intermediate concentration 10 μ M solution. As there is no ¹⁹F, the background data is normalized to the final ¹⁹F signal of a 10 μ M growth kinetics data set.

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Supplementary Note 1: Annealing of implanted diamond for conversion to NV-centers



Figure S1. **Diamond annealing:** Temperature profile of N-implanted diamond annealing.

After implantation, the diamond was annealed in vacuum to form NV-centers using gradual ramp rates and precisely controlled temperatures. The time- and pressure-traces during the annealing process is shown in Fig. S1. The specifications of the oven can be found in the Methods section. The annealing procedure is critical for forming shallow NV-centers, while preserving their sensitivity and brightness and avoiding graphitization and/or damage to the surface [24].

Supplementary Note 2: Properties of NV-centers after annealing



Figure S2. **Example of NV ensemble properties** a) ESR lineshape and B_0 field b) Typical Rabi contrast dry and in ethanol c) XY8-*12* dips of the ¹H peak on the left and the ¹⁹F peak from drop cast PFOPA on the surface on the right. d) Spin-lattice relaxation T_1 E) Spin-spin relaxation T_2 .

The properties of our shallow NV-ensemble in diamond is depicted in Fig. S2. We used the NV resonance for the $m_s = 0$ to -1 transition to determine the magnetic field strength. The contrast of the Rabi oscillations is typically around 7 % for dry state (shown in blue), but is reduced to 4 % in the case of liquid samples on the diamond (shown in yellow). An alternative to the detection of NMR signals on the nanoscale with correlation spectroscopy are dynamic decoupling sequences, where the the time between the π pulses is

swept [29]. Two dips, corresponding to the ¹⁹F and ¹H can be observed in this experiment. The spin-lattice relaxation time T_1 , is measured by increasing the time between optical initialization and readout. Fitting to an exponential gives a range of 700 to 800 μ s for clean diamond and after ALD. The spin-spin relaxation time T_2 , is measured with a Hahn Echo sequence. The dips arise from coupling with ¹³C in the diamond. The data are fit to a stretched exponential function resulting in a range from 4-6 μ s for clean diamond and after ALD deposition.

Supplementary Note 3: Gaussian fitting of laser spot size

The laser spot was imaged using a Basler a2A1920-160umBAS camera (Fig. 2 c). The entire diamond was imaged with a known size of 2 mm as our reference. Fitting a vertical cut through the center of the laser spot to a Gaussian lineshape allowed for the determination of a full width at half maximum (FWHM) of 60.8 μ m. The horizontal line had a FWHM of 82.9 μ m. Taking these values as the maximal extent of an elliptical area yields a total of ~ 4000 μ m².



Figure S3. Area of optical excited NV-centers. The PL spot size of our NV-centers was imaged and the size was determined by vertical and horizontal cuts which were fit to a Gaussian function.

Supplementary Note 4: Determining optimal Al_2O_3 layer thickness for monolayer assembly



Figure S4. Static water contact angle as a function of monolayer growth on different Al_2O_3 layer thicknesses.

As NMR signal decreases with the cubed distance from the NV center, it is crucial to keep the Al₂O₃ layer as thin as possible. However, it must also be sufficiently thick to form a closed layer that enables assembly of dense molecular monolayers based on phosphonate linking chemistry. We used static water contact angle measurements to determine the thinnest layer of Al₂O₃ that reaches saturation, as well as to verify dense, ordered monolayer growth from PF-PDPA. As a reference, a clean diamond without an Al₂O₃ layer was soaked in a PA solution without undergoing the ALD process. The low contact angle indicates low reactivity. By contrast, for an Al₂O₃ layer of 1 nm with a contact angle of approximately 91 degrees the signal indicates that the monolayer reaches a saturation value. Therefore, a thickness of 1 nm was chosen as the thickness of the Al_2O_3 for surface NV-NMR due to the fact that the nuclear spins are as close as possible to the NV layer for sensing and thick enough to form a dense monolayer.

Supplementary Note 5: Magnetic field sweep of ³¹P signal

The ³¹P from the PFPDPA monolayer was detected at 3 different magnetic fields. A linear fit of the resonance frequency as a function of magnetic field is used to calculate the gyromagnetic ratio. This corroborates that the source of the signal arises from ³¹P spins. The gyromagnetic ratio of 17.24 (17.24, 17.25) MHz/T matches the known value for ³¹P.



Figure S5. Magnetic field sweep of ³¹P signal from PFPDPA monolayer.

Supplementary Note 6: Fitting of *in-situ* growth kinetics experiments

The rate of monolayer formation can be influenced by changing the concentration of the PFPDPA solution for the *in-situ* growth kinetics experiments. The averaged data points for each concentration were fit with a single exponential: $y = a(1 - x^{-b}) + c$. We observe an increase in the rate with an increase in concentration (Table S1).

Table S1:	Fitting	parameters	of the	monolaver	growth	kinetics

[Conc.]	a	b	c
1 mM	0.81	0.10	0.13
$10 \ \mu M$	0.084	0.052	0.13
$1 \mu M$	0.093	0.039	0.027

Supplementary Note 7: Diamond surface cleaning for redeposition of $\mbox{Al}_2\mbox{O}_3$



Figure S6. XPS of the surface before and after Al_2O_3 removal with NaOH

To be able to reproduce the ALD and monolayer growth process, removal of the Al_2O_3 is crucial. XPS measurements of F 1s and Al 2s peaks before (blue) and after (yellow) overnight soaking in 5% NaOH shows that NaOH is effective at completely removing the monolayer and Al_2O_3 .

Supplementary Note 8: Materials and Methods

Surface NV-NMR setup. The diamond was positioned in the middle of two neodymium magnets, which were rotated and tilted for alignment of the B₀ field with one of the four possible NV center orientations. For quantum control of the NV-centers, microwave frequencies were generated with a signal source (SynthHD, Windfreak Technologies, LLC., New Port Richey) and fed into a phase shifter (ZX10Q-2-27-S+, Mini-Circuits) and two switches (ZASWA-2-50dRA+, Mini-Circuits) to generate X and Y pulses, then combined (ZX10-2-442-s+) and amplified by a microwave amplifier (ZHL-16W-72+, Mini-Circuits). Initialization of the NV ensemble was achieved by using a 532 nm laser (Verdi G5, Coherent) at a power of around 200 mW at the diamond. The laser pulses were controlled by an acousto-optic modulator (Gooch and Housego, model 3260-220) with pulse durations of 5 μ s. The diamond was glued to a thin glass slide (48393-026, VWR). The photoluminescence (PL) was collected and collimated by 2 condenser lenses, the bottom one placed right above a large area avalanche photodiode (A-CUBE-S3000-10, Laser Components GmbH) below the diamond. In order to increase the light collection efficiency, a 6 mm glass hemisphere (TECHSPEC[®] N-BK7 Half-Ball Lenses, Edmund Optics) was glued to the other side of the cover slide. This assembly was then glued to a 30 mm cage plate (CP4S, Thorlabs) and mounted onto the experiment 1.2 cm above the top condenser lens (ACL25416U-B, Thorlabs). This coupled in the focused laser (LA1986-A-M, Thorlabs) in a total internal reflection geometry. The excitation wavelength was removed from the PL light with a long-pass filter (Edge Basic 647 Long Wave Pass, Semrock) placed immediately between the bottom condenser lens and the photodiode. The photo-voltage was digitised with a data acquisition unit (USB-6229 DAQ, National Instruments). The electron spin resonance (ESR) frequency measured from the dip in PL was used to determine the magnetic field strength and the NV resonance frequency to perform a Rabi experiment, which then determined the π and $\pi/2$ pulse durations for the correlation spectroscopy pulse sequences (Fig. S2). The magnetic field strength B₀ can be adjusted by changing the distance between the magnets. We chose to work at 31 mT for ¹⁹F and at 174 mT for ³¹P detection. The lower magnetic field was suited for higher frequency detection because the spacing between the microwave pulses must be $1/(2\tau)$ which was limited by finite π pulse durations.

SWCA measurements. SWCA measurements were performed on an OCA 15Pro contact angle system (DataPhysics Instruments). Data acquisition and evaluation were realized with SWCA 20 - contact angle (DataPhysics Instruments, version 2.0). For quantifying an average Youngs contact angle $(\vartheta \gamma)$, 2 μ L of deionized H₂O (18.2 M Ω cm at 25 °C, Merck Millipore) was dispensed with a rate of 0.2 μ Ls⁻¹ from a 500 μ L Hamilton syringe onto the sample surface. After allowing the droplet to settle for ~ 3 s, an image was acquired for further processing. The procedure was repeated at least 3 times on different spots on the surface, and the standard deviation (error) was calculated.

X-ray Photoelectron Spectroscopy. X-ray photoelectron spectroscopy (XPS) measurements were performed with an Axis Supra (Kratos, UK) spectrometer. The monochromatized Al K_{α} (1486.9 eV) X-ray tube source was operated at an emission current of 15 mA. The data were recorded with a circular acceptance area of Alsource and the analysed area 700 x 300 μ m in diameter. Spectra were processed with CasaXPS (Casa Software Ltd, version 2.3.17).