

High-Throughput Screening Single-Atom Alloy for Electroreduction of Dinitrogen to Ammonias

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Abstract

Exploring electrocatalyst with high activity, selectivity and stability is essential for development of applicable electrocatalytic ammonia synthesis technology. By performing density functional theory calculations, we systematically investigated a series of transition-metal doped Au-based single atom alloys (SAAs) as promising electrocatalysts for nitrogen reduction reaction (NRR). For Au-based electrocatalyst, the first hydrogenation step ($*N_2 \rightarrow *NNH$) normally determines the limiting potential of the overall reaction process. Compared with pristine Au(111) surface, introducing single atom can significantly enhance the binding strength of N_2 , leading to decreased energy barrier of the key step, i.e., $\Delta G(*N_2 \rightarrow *NNH)$. According to simulation results, three descriptors were proposed to describe $\Delta G(*N_2 \rightarrow *NNH)$, including $\Delta G(*NNH)$, d -band center, and $\frac{d}{\sqrt{E_m}}$. Eight doped elements (Ti, V, Nb, Ru, Ta, Os, W, and Mo) were initially screened out with limiting potential ranging from -0.75V to -0.30 V. Particularly, Mo- and W-doped systems possess the best activity with limiting potentials

of -0.30 V, respectively. Then the intrinsic relationship between structure and the potential performance was further analyzed by using machine-learning. The selectivity, feasibility, stability of these candidates were also evaluated, confirming that SAA containing Mo, Ru, Ta, and W could be outstanding NRR electrocatalysts. This work not only broadens the understating of SAA application in electrocatalysis, but also devotes to the discovery of novel NRR electrocatalysts.

KEYWORDS: Single-atom alloy, Nitrogen reduction reaction, Electrocatalysis, High-throughput calculations, Machine-learning

1. Introduction

Ammonia (NH_3) is one of the world's most synthesized chemicals, which is the key building block for fertilizer currently and an ideal carrier for renewable energy in future.^{1,2,3} To date, synthesis of ammonia highly relies on the Haber-Bosch process that occurs at high temperatures and high pressures in the presence of iron-based catalysts.^{4,5} Electrochemical nitrogen reduction reaction (NRR) at ambient conditions is an attractive alternative to the Haber-Bosch process. Developing active, selective, and stable electrocatalysts is still challenged.⁶ The competition to hydrogenation evolution reaction (HER) significantly limits the efficiency of NRR process. Because of the inertness to HER and high stability, several noble metals have been reported as promising electrocatalysts, such as Au, Pt, and Ru.⁷⁻¹⁰ Among them, Au exhibits the most efficient performance owing to the low activity of competitive HER.^{11,12} Previous studies demonstrated that the lack of empty *d*-orbitals to accept lone-pair electrons from N_2 may result in extremely weak adsorption to N_2 , and herein difficulty in breaking triple N-N bond.^{13,14} The potential determining step (PDS) is normally the first hydrogenation step from $^*\text{N}_2$ to $^*\text{NNH}$ with potential barrier over 2.0 eV.^{15,16} Can we improve the activity of Au-based materials through a simple strategy?

Since single-atom catalyst (SAC) was first proposed by Zhang et al,¹⁷ SACs have been emerging in various reactions, such as traditional catalysis, photocatalysis, and

electrocatalysis.^{18–20} SACs can be constructed based on a variety of substrates, such as transition metal oxides^{21–26} carbon-based materials,^{27,28} transition metal chalcogenide^{29–31}, and MXenes.^{32,33} Particularly, when the substrate is metal, the SAC is also called as single-atom alloy (SAA).³⁴ Numerous SAA catalytic systems have been reported for their well-defined active site and excellent catalytic performance.^{34–36} For example, CuPd SAA shows high selective for the photocatalytic conversion of CO₂ to CH₄. The *d*-band center of isolated Cu was raised in Pd substrate, leading to enhanced binding with key intermediate.³⁷ On the other hand, PdCu SAA enhances the selectivity toward alcohols for electrochemical CO₂ reduction, compared to common Cu-based catalysts.³⁸ Hydrogen oxidation reaction (HOR) activity can be regulated by isolating single Ni atom into Ru nanosheet.³⁹ Inspired by these works, we proposed a single-atom alloying strategy to improve the performance of Au-based materials for NRR.

To validate our hypothesis, the NRR performance of a series of M/Au(111) systems (M=3*d*, 4*d*, and 5*d* transition metals) were investigated by using density functional theory (DFT) calculation. Au(111) is the most stable facet that has been widely studied.^{40–42} Our theoretical results indicate that many SAAs can enhance the binding strength of N₂ and reduce the potential barrier of the first hydrogenation step ($\Delta G(*N_2 \rightarrow *NNH)$). Three descriptors were established to describe $\Delta G(*N_2 \rightarrow *NNH)$, i.e., $\Delta G(*NNH)$, *d*-band center of the doped metal, and $\frac{d}{\sqrt{E_m}}$ (*d* is the number of *d* electron in valence orbital of doped metal; *E_m* is the electronegativity of doped metal). $\Delta G(*NNH)$ exhibits a linear relationship with $\Delta G(*N_2 \rightarrow *NNH)$, while the *d*-band center and $\frac{d}{\sqrt{E_m}}$ show a volcano relationship. Moreover, combining with machine-learning, we unveiled the intrinsic correlations between structure and $\Delta G(*N_2 \rightarrow *NNH)$. Through the computation of whole reaction paths, eight M/Au(111) systems (M=Ti, V, Nb, Mo, Ru, Ta, W, and Os) were suggested to possess high activity with limiting potential ranging from -0.75 V to -0.30 V. According to $\frac{d}{\sqrt{E_m}}$, we further predicted Mo- and W/Ag(111) SAAs could be highly active. After considering selectivity, feasibility, electrochemical and thermal stability, four M/Au(111) candidates (M=Mo, W, Ru, and

Ta) were finally screened out as excellent catalysts for NRR.

2. Computational Details

Spin-polarized DFT computations were employed by using Vienna Ab initio Simulation Package (VASP).⁴³ The Perdew-Burke-Erzenhof (PBE) functional in the generalized gradient approximation (GGA) was applied to describe the exchange-correlation energy.⁴⁴ The ion-electron interaction was described by Projector Augmented Wave (PAW) with cutoff energy of 450 eV.⁴⁵ The convergence criteria for energy and force were set to be 10^{-5} eV and 0.03 eV/Å, respectively. The van der Waals interaction was considered by the empirical correction of the Grimme's scheme (DFT-D2).⁴⁶ The $4 \times 4 \times 1$ supercell of Au(111) facet with four layers was taken as the substrate model. To reproduce the bulk properties, two bottom layers were fixed. The Brillouin zone was sampled by Gamma centered $3 \times 3 \times 1$ Monkhorst-Pack grid for structural optimizations.⁴⁷ The density of states was calculated by using a $5 \times 5 \times 1$ Monkhorst-Pack grid. The *d*-band center was computed by VASPKIT code.⁴⁸ The figure of the model structures were generated by using ASE.⁴⁹

The NRR process was studied based on the computational hydrogen evolution model (CHE).⁵⁰ The Gibbs free energy change (ΔG) was calculated by the following equation:

$$\Delta G = \Delta E_{\text{DFT}} + \Delta E_{\text{zpe}} - T \Delta S$$

where ΔE_{DFT} is the total energy difference obtained from DFT calculation; ΔE_{zpe} and ΔS are zero-point correction and the variation of entropy computed from vibrational analysis by VASPKIT code.⁴⁸ T is temperature (298.15 K in this work). The implicit solvation model, VASPsol, was used to consider the effect of H₂O solvation. The dielectric constant was 78.4.⁵¹ To evaluate the activity of NRR, the limiting potential (U_{L}) was calculated by

$$U_{\text{L}} = - \Delta G_{\text{max}} / e$$

where ΔG_{max} represents the ΔG of the PDS and e is the number of transferred electron. The less negative U_{L} is, the higher activity the catalyst possesses.

The formation energy, the binding energy of single atom and the dissolution potential are further calculated to evaluate feasibility of SAA system. The formation energy was obtained by the following equation:

$$E_{form} = E_{SAA} - E_{Au(111)} - \mu_M + \mu_{Au}$$

The E_{SAA} , $E_{Au(111)}$, μ_M , and μ_{Au} represent the energy of SAA model, Au(111) substrate, the chemical potential of the doped metal, and that of Au, respectively. The binding energy (E_b) was computed as:

$$E_b = -(E_{SAA} - E_{dAu(111)} - E_M)$$

where $E_{dAu(111)}$ and E_M are the energies of defective Au(111) with one vacancy and the energy of isolated doped metal in vacuum box, respectively. The dissolution potentials (U_{diss}) of SAA were from the following equations:

$$U_{diss} = U_{diss}^0 M - \left(\frac{E_{SAA} - E_{dAu(111)} - \mu_M}{ne} \right)$$

where $U_{diss}^0 M$ and n represent the standard dissolution potentials of doped metal M in the bulk phase and the number of transferred electrons, respectively.⁵²

3. Result and discussion

3.1. N₂ adsorption

Adsorption of N₂ is the key step to wreck inert N≡N bond. There are three possible adsorption sites around doped metal, including top, bridge, and hollow sites (**Figure 1b**). Different adsorption configurations are corresponding to three possible mechanisms, i.e., distal, alternating, and enzymatic pathways.^{52,53} We considered two typical adsorption configurations of N₂ on SAAs, namely end-on and side-on (**Figure 1c**). Among the SAAs with Au(111) substrates, Ag- and Zn-doped systems interact with N₂ quite weakly, similar to pure Au(111) surface. N₂ molecule lies over the doped metal, with a distance more than 3.0 Å, showing as physisorption. The adsorption free energies of N₂, $\Delta G(*N_2)$, are 0.36, 0.36, and 0.39 eV for Ag/Au(111), Zn/Au(111) and Au(111), respectively. In the other systems, top site on the heteroatom is preferable due to the extremely weak Au-N interaction. As shown in **Table S1**, the end-on configurations are more stable than side-on configurations with relatively more negative adsorption free

energies. We focus on the discussion based on end-on configuration. The $\Delta G(^*N_2)$ s range from 0.14 eV (Cu/Au(111)) to -1.14 eV (Os/Au(111)), indicating that most SAAs significantly enhance the N_2 binding compared to pristine Au(111).

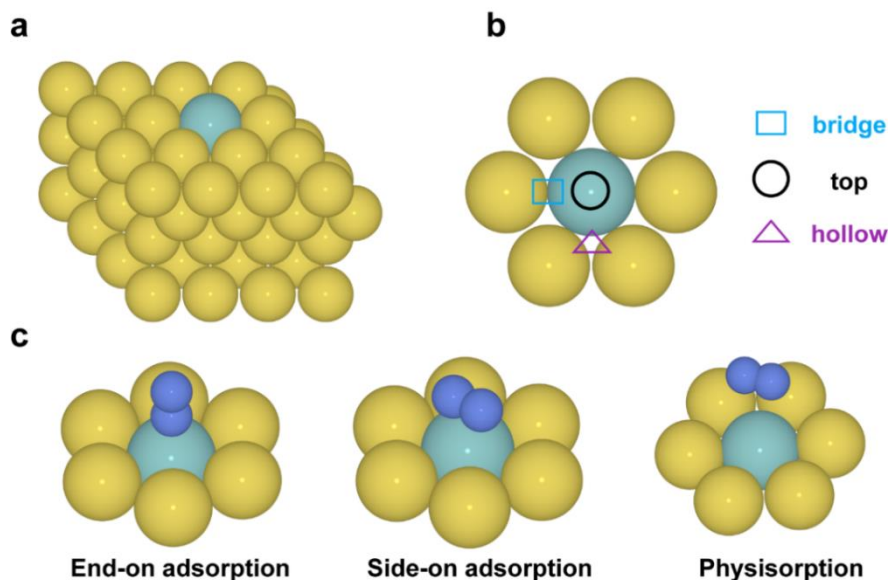


Figure 1.(a) The model of M/Au(111) SAA (M: doped metal); (b) The three possible adsorption sites around doped metal; (c) The two configurations of chemisorption (side-on and end-on) and physisorption configurations of N_2 on SAAs.

There are two types of interactions between transition metal (TM) and adsorbed N_2 .^{13,52} One is σ -donation that the d -orbital of metal accepts the lone-pair of N_2 . The other is π -back donation that the d -electrons of transition metal return back into antibonding orbitals of N_2 to wreck inert $N\equiv N$. Taking N_2 adsorbed on Mo/Au(111) as an example, σ -donation leads to electron accumulation between Mo and N while π -back donation results in electron depletion of $N\equiv N$ bond (**Figure 2a**). In **Figure 2b**, the project density of states (pDOS) reveals that the π^* orbital of the adsorbed N_2 becomes partially occupied, herein N_2 bond is weakened and elongated. The N_2 bond lengths in various systems are summarized in **Table S2**, ranging from 1.118 Å (Y/Au(111)) to 1.137 Å (Os/Au(111)) (vs. 1.115 Å in the free gas molecular). In comparison, there is no apparent electron π -back donation in Zn/Au(111) system (**Figure 2b**), and the bond length of the adsorbed N_2 is almost the same as that of N_2 in gas phase. These results

can be confirmed by the Bader analysis⁵⁴ that the adsorbed N₂ obtained 0.32e from Mo/Au(111). Similarly, other SAAs except Ag- and Zn/Au(111) also donate electron to adsorbed N₂. (Table S3)

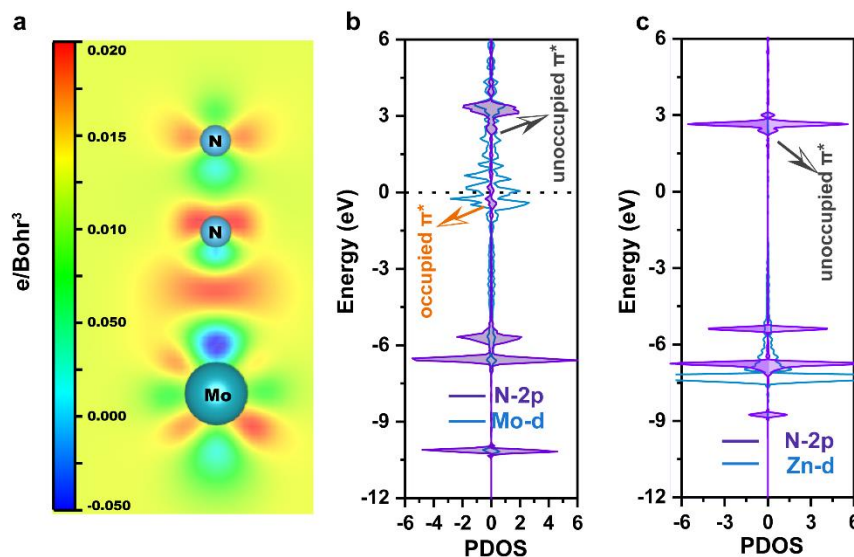


Figure 2. (a) The charge density difference of N₂ adsorbed on Mo/Au(111). The positive and negative regions represent accumulation and depletion of electron, respectively. The project density of states (pDOS) of N₂ adsorbed on (b) Mo/Au(111) and (c) Zn/Au(111). The Fermi level is set to be zero.

3.2. Reaction pathways of eight SAAs candidates

The first hydrogenation step ($*N_2 \rightarrow *NNH$) follows N₂ adsorption on the active site. As listed in Table S4, most SAAs reduce potential barriers of the first hydrogenation step $\Delta G(*N_2 \rightarrow *NNH)$ compared to Au(111). For eight M/Au(111) systems (Ti, V, Nb, Mo, Ru, Ta, W, and Os), $\Delta G(*N_2 \rightarrow *NNH)$ s decrease by more than 1.0 eV. Moreover, we established relationship between binding strengths of intermediates ($*N_2$ and $*NNH$) and $\Delta G(*N_2 \rightarrow *NNH)$. As shown in Figure 3a, $\Delta G(*N_2 \rightarrow *NNH)$ exhibits a better linear correlation with $\Delta G(*NNH)$ ($R^2=0.92$) than $\Delta G(*N_2)$ ($R^2=0.71$), implying that the stability of $*NNH$ determines $\Delta G(*N_2 \rightarrow *NNH)$ (Figure 3b).^{52,55}

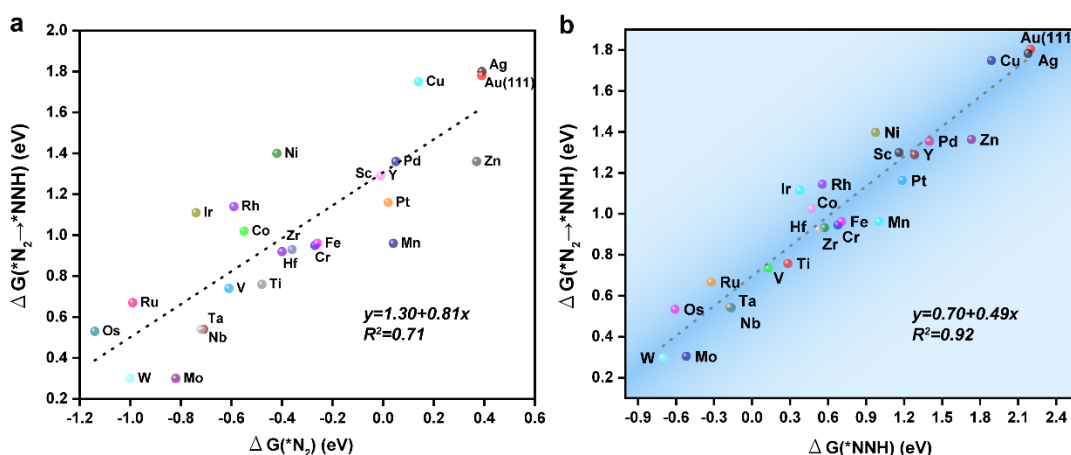


Figure 3. The relationships between $\Delta G(*N_2 \rightarrow *NNH)$ and the adsorption free energies of (a) $*N_2$ ($\Delta G(*N_2)$) and (b) $*NNH$ ($\Delta G(*NNH)$).

According to $\Delta G(*N_2 \rightarrow *NNH)$, eight M/Au(111) systems (M=Ti, V, Nb, Mo, Ru, Ta, W, and Os) are supposed to possess outstanding performance, with relatively small $\Delta G(*N_2 \rightarrow *NNH)$ s ranging from 0.30 eV to 0.75 eV. Subsequently, the full reaction pathways were investigated. Here, we only considered commonly discussed distal and alternating reaction mechanisms, since N_2 adsorptions are all in end-on configurations (**Figure 4a**). For the distal path, the first three H^+/e^- pairs attack the terminal N continuously, and another three H^+/e^- pairs attack the N atom connecting to the surface. In many previous studies, the N-N bond-breaking takes place at the third hydrogenation step ($*NNH_2 \rightarrow *N + NH_3$).^{56,57} This phenomenon can also be observed in W, Os, and Ru/Au(111). While for Mo/Au(111), the N-N bond is cleaved at the fourth hydrogenation step ($*NNH_3 \rightarrow *NH + NH_3$). The N-N bond-breaking is even at the fifth hydrogenation step ($*NHNH_3 \rightarrow *NH_2 + NH_3$) on Nb and Ta/Au(111), or the last hydrogenation step on Ti and V/Au(111) (**Figure 4b**). Despite the first ammonia molecule desorbs at various steps, the PDSs are all the first hydrogenation steps ($*N_2 \rightarrow *NNH$) (**Figure 4d**).

In comparison, six H^+/e^- pairs attack two N atoms alternatively in the alternating mechanism. However, compared to distal mechanisms, the second hydrogenation step was hindered in alternating mechanism ($*NNH \rightarrow *NHNH$) for those SAAs, owing to the more positive free energy change (**Table S5**). For example, the $\Delta G(*NNH \rightarrow *NNH_2)$

(-0.72 eV) are much lower than the $\Delta G(*\text{NNH} \rightarrow *\text{NHNH})$ (0.52 eV) on W/Au(111). Thus, the aforementioned eight SAAs show high activities with limiting potential (U_L) ranging from -0.30 V to -0.75 V via distal mechanism (Figure 4c), which are better than Ru(0001) surface ($U_L = -0.98$ V) with highest activity among bulk metals.^{16,58} Moreover, the activities of Mo and W/Au(111) are better than most single-atom catalysts (SACs) reported in experiments^{59–62} and theory simulations^{56,57,63,64} with limiting barriers of 0.30 eV.

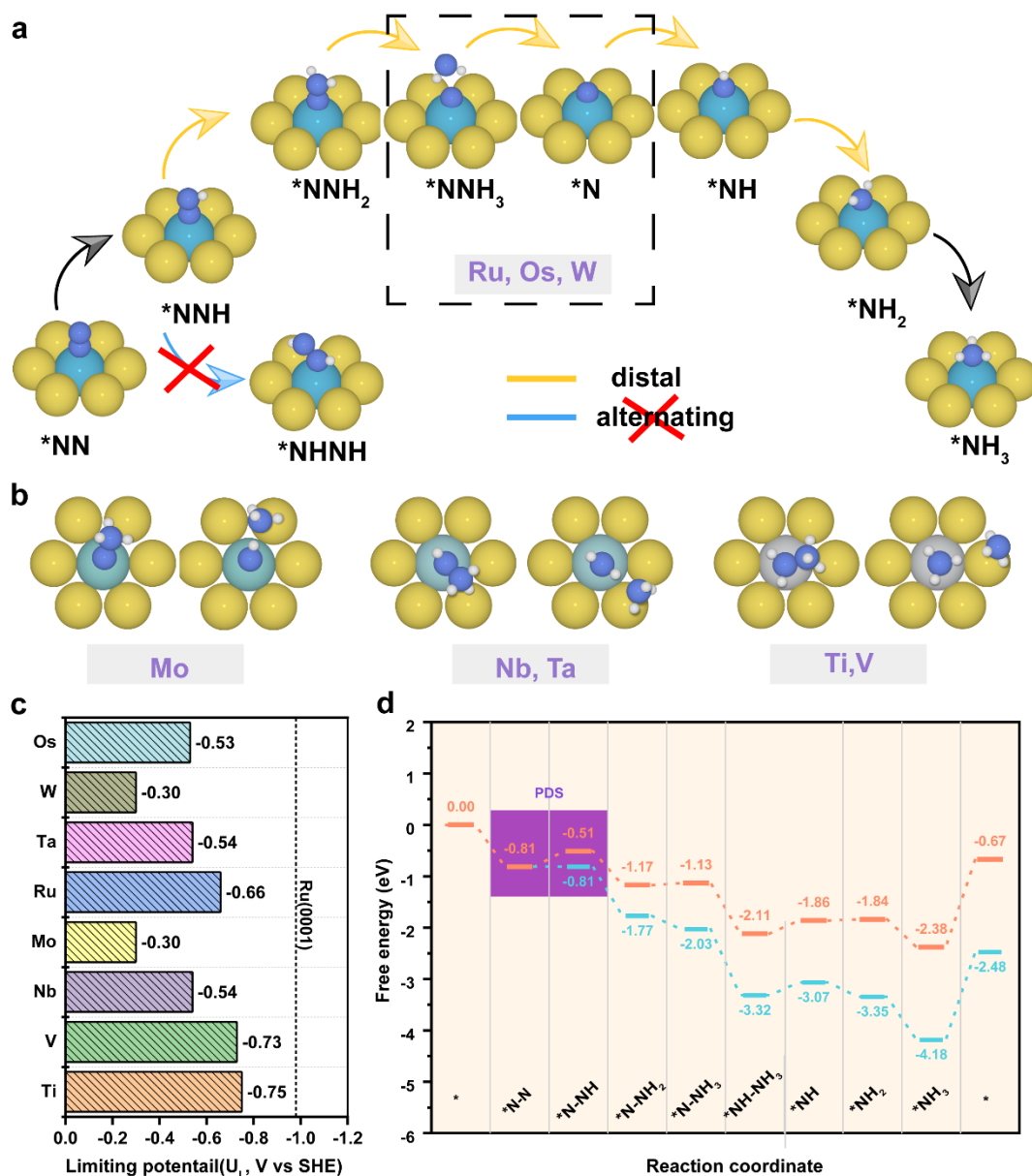


Figure 4. The reaction pathways of W/Au(111) through (a) distal mechanisms and (b) The N-N bond-breaking steps on typical SAAs. (c) The limiting potentials (U_L) of

SAAAs and (d) The Free-energy diagram of Mo/Au(111).

3.3. The relationship between $\Delta G(*N_2 \rightarrow *NNH)$ and intrinsic properties of doped Metal.

Based on previous analysis, the potential determining step of most promising eight candidates SAAAs for NRR all are the first hydrogenation step ($*N_2 \rightarrow *NNH$). Thus, to further understand the relationship between $\Delta G(*N_2 \rightarrow *NNH)$ and electronic structures of SAAAs, we analyzed the ***d*-band center** of doped metal in each SAA, which is a commonly used descriptor in catalytic field.⁶⁵ **Figure 5(a)** exhibits a volcano relationship between $\Delta G(*N_2 \rightarrow *NNH)$ and the ***d*-band center**. The W- and Mo- doped systems are on the top, which possess the least positive $\Delta G(*N_2 \rightarrow *NNH)$ s. The *d*-band center is still not an intrinsic descriptor. Inspired by previous studies,^{66–68} we purposed a new descriptor ($\frac{d}{\sqrt{E_M}}$) that only contains elemental physical parameters of dopant, namely electronegativity (E_M) and the number of *d*-electron in valence orbital (***d***). From **Figure 5(b)**, one can see that $\frac{d}{\sqrt{E_M}}$ also shows a volcano relationship with $\Delta G(*N_2 \rightarrow *NNH)$. Thus we suggest that one can simply predict the performance based on this intrinsic descriptor. For example, we predicted that Mo and W/Ag(111) may also exhibit high activity with low energy barriers of $\Delta G(*N_2 \rightarrow *NNH)$ owing to similar electronic structures of Ag and Au. The further simulation shows that $\Delta G(*N_2 \rightarrow *NNH)$ for Mo and W/Ag(111) are 0.19 and 0.09 eV, respectively. The distal mechanism is preferred, similar to the Au-based system. Whereas the PDSs are the third hydrogenation step ($*NNH_2 \rightarrow *NNH_3$) with free energy changes of 0.46 and 0.47 eV for Mo and W/Ag(111), respectively. Herein these Ag-based alloys are also promising for NRR. These findings indicate that building a simple intrinsic descriptor can effectively design catalysts to reduce large computational resources.

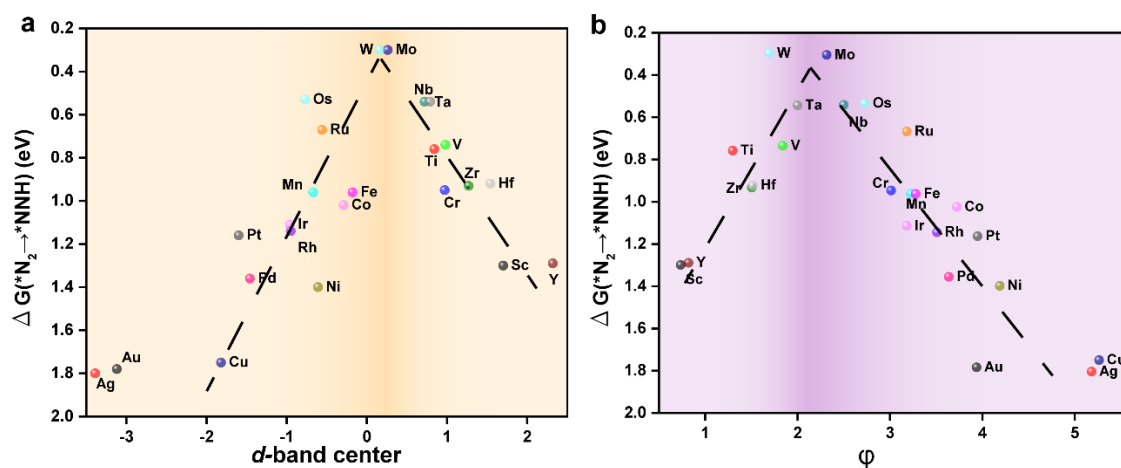


Figure 5. The volcano relationships between $\Delta G(*N_2 \rightarrow *NNH)$ and **(a)** d-band center and **(b)** ϕ that equals to $\frac{d}{\sqrt{E_m}}$.

Moreover, we applied the Random Forest Algorithm (RFA) in Scikit-learn to analyze the relationship between properties of dopant metal and $\Delta G(*N_2 \rightarrow *NNH)$.⁶⁹ Inspired by previous research,^{70–72} five parameters of doped metal were identified as the input data, including the number of *d*-electron, radius, the group number, electronegativity, and electron affinity. As shown in **Figure 6a**, the RFA model performs well with a training score of 0.96 and a testing score of 0.90. The result shows that the group number and *d*-electron are the two most important features (**Figure 6b**). They decide the electron distribution in the outer shell of dopant, which directly affects the interactions (σ -donation and π -backdonation) between dopant and adsorbate. We suggest that the machine-learning technology can be used to accelerate the discovery of efficient SAA for NRR.

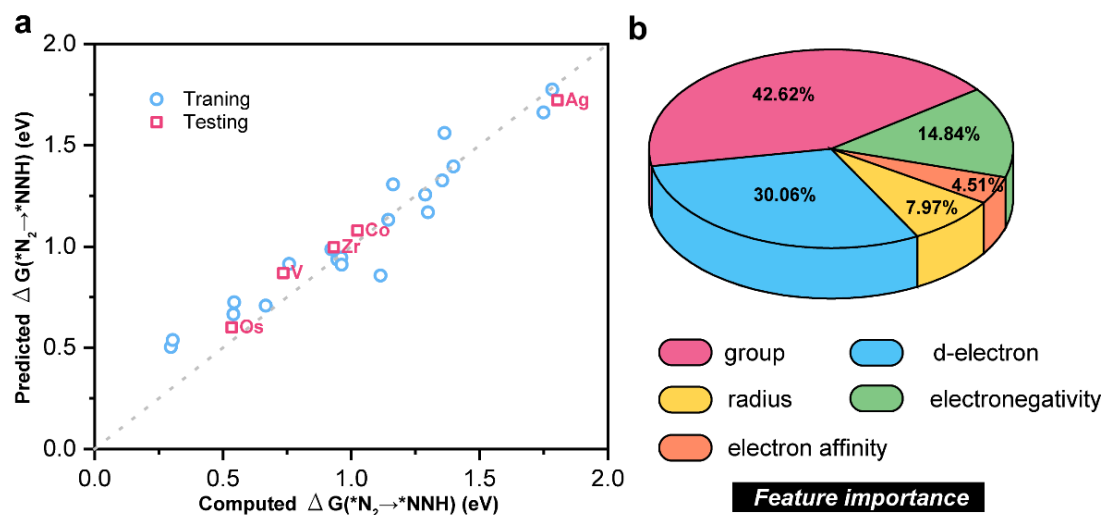


Figure 6. (a) Comparison between DFT and predicted $\Delta G(*N_2 \rightarrow *NNH)$. (b) Importance of features from the Random Forest model for $\Delta G(*N_2 \rightarrow *NNH)$.

3.4 The potential application of SAAs on NRR.

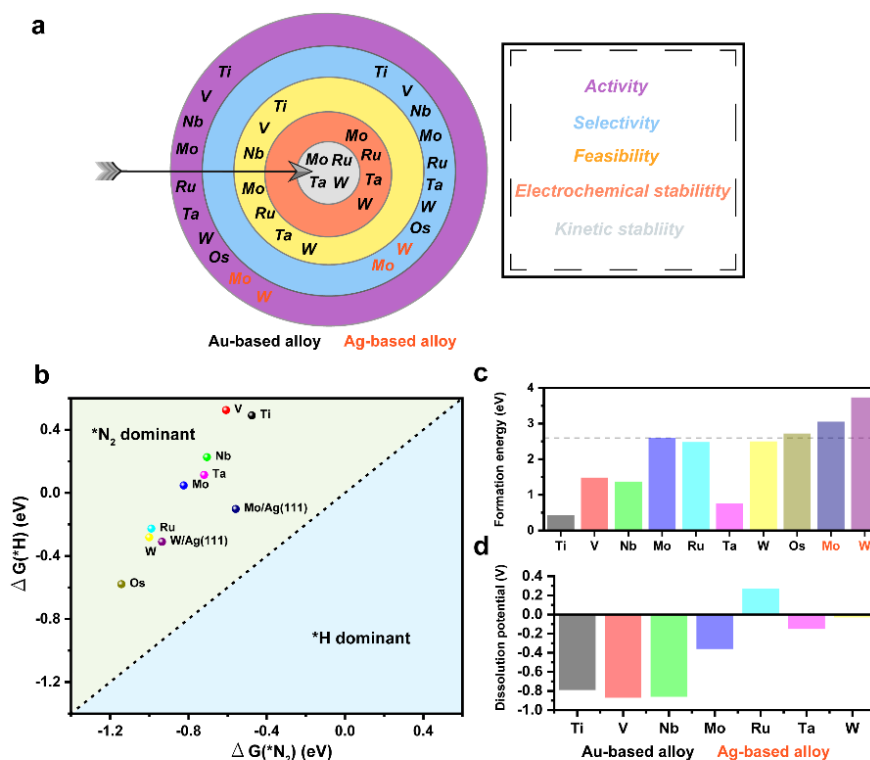


Figure 7. (a) The illustration of the evaluating process. (b) The calculated $\Delta G(*N_2)$ and $\Delta G(*H)$. The left upper corner is the $*N_2$ dominant region where $\Delta G(*H) > \Delta G(*N_2)$.

(c) The formation energies of SAAs. (d) The dissolution potentials of SAAs.

Selectivity. Besides activity, an ideal NRR electrocatalyst should be able to suppress the hydrogenation evolution reduction (HER) to achieve high faradaic efficiency.⁷³⁻⁷⁵ To evaluate the selectivity of the eight M/Au(111) and two M/Ag(111) SAAs, we compared $\Delta G(*H)$ and $\Delta G(*N_2)$. As shown in **Figure 7b**, one can see that these eight SAAs possess positive values of $(\Delta G(*H) - \Delta G(*N_2))$, indicating that these surfaces tend to capture nitrogen rather than hydrogen and then lead to the expected NRR selectivity.^{60,64}

Feasibility. Moreover, an applicable electrocatalyst should be facile in experiment. To investigate the synthetic accessibility of selected candidates, we calculated their formation energies (E_{formS}) (**Figure 7c**). Mo/Au(111) was taken as a reference since it has been synthesized.⁷⁶ Except for Os/Au(111), Mo/Ag(111), and W/Ag(111), other five Au-based SAAs show less positive E_{form} values than that of Mo/Au(111) (2.59 eV) (**Figure 7c**), implying that these materials are more readily synthesized in experiment.

Electrochemical stability. To evaluate the electrochemical stability in electrolyte, the dissolution potential (U_{diss}) of the seven M/Au(111) SAAs (M=Ti, V, Nb, Mo, Ru, Ta, and W) were calculated to compare with that of prepared Mo/Au(111), which exhibits long-term stability for effective CO₂ electrochemical reduction.^{13,52} From **Figure 7d**, Ru, Ta, and W /Au(111) systems possess more positive U_{diss} than Mo/Au(111), corresponding to their long-term stability in electrochemical reductive environment, namely the high electrochemical stability.⁷⁶

Kinetic stability. Now Mo, Ru, Ta, and W/Au(111) are picked out as candidates through the evaluation of feasibility and electrochemical stability. Then the aggregation of a single atom on the substrate is checked based on the difference between binding energy (E_b) and cohesive energy (E_c) of the doped transition metal. The $(E_b - E_c)$ values of Ta, Mo, W, and Ru are 1.34, -0.49, -0.40, and -0.38 eV, respectively. The positive $(E_b - E_c)$ value of Ta-containing system is related to highly thermal stability. Beyond thermal stability, single-atom system can still be kinetically stable, such as the

synthesized Mo/Au(111).^{20,77,78} In comparison, Ru- and W-containing systems possess more positive values of E_b (7.40 and 8.46 eV) than that of Mo/Au(111) (7.08 eV), hence they can be kinetically stable in preparation and electrocatalysis.

After evaluating selectivity, feasibility, and stability, the Mo, Ru, Ta, and W/Au(111) SAAs are finally screened out as the potential electrocatalysts for NRR (**Figure 7a**).

4. Conclusion

In summary, a series of SAAs were investigated as candidates for NRR by using DFT calculation. Simulation results show that the alloying strategy effectively enhances the strength of N_2 adsorption and thus reduces the free energy barriers of the first hydrogenation step. Compared to Au(111) surface, eight M/Au(111) systems (M=Ti, V, Nb, Mo, Ru, Ta, W, and Os) reduce the barrier of PDS ($\Delta G(*N_2 \rightarrow *NNH)$) over 1.0 eV, corresponding to high activity with limiting potential ranging from -0.30 V to -0.75 V. Three descriptors were proposed to describe $\Delta G(*N_2 \rightarrow *NNH)$, including $\Delta G(*NNH)$, d -band center, and $\frac{d}{\sqrt{E_m}}$. Since the new descriptor is only related to the intrinsic properties of doped elements, we predict that Mo and W/Ag(111) possess high activity as well. Random Forest algorithm in machine learning was used to investigate the relationship between intrinsic properties of doped metal and $\Delta G(*N_2 \rightarrow *NNH)$. Finally, the selectivity over HER, feasibility, electrochemical, and kinetic stability were computed to evaluate the potential application of SAAs. Four SAAs (Mo, W, Ru, and Ta/Au(111)) were screened out as promising electrocatalysts for NRR. We hope our work can inspire both theoretical and experimental investigations on SAAs for NRR.

Reference

- 1 G. Qing, R. Ghazfar, S. T. Jackowski, F. Habibzadeh, M. M. Ashtiani, C.-P. Chen, M. R. Smith and T. W. Hamann, *Chem. Rev.*, 2020, **120**, 5437–5516.
- 2 J. Li, S. Chen, F. Quan, G. Zhan, F. Jia, Z. Ai and L. Zhang, *Chem*, 2020, **6**, 885–901.
- 3 V. Kyriakou, I. Garagounis, A. Vourros, E. Vasileiou and M. Stoukides, *Joule*, 2020, **4**, 142–158.
- 4 W. Guo, K. Zhang, Z. Liang, R. Zou and Q. Xu, *Chem. Soc. Rev.*, 2019, **48**, 5658–5716.
- 5 S. L. Foster, S. I. P. Bakovic, R. D. Duda, S. Maheshwari, R. D. Milton, S. D. Minteer, M. J. Janik, J. N. Renner and L. F. Greenlee, *Nat. Catal.*, 2018, **1**, 490–500.
- 6 B. H. R. Suryanto, H.-L. Du, D. Wang, J. Chen, A. N. Simonov and D. R. MacFarlane, *Nat. Catal.*, 2019, **2**, 290–296.
- 7 V. Kordali, G. Kyriacou and C. Lambrou, *Chem. Commun.*, 2000, 1673–1674.
- 8 H. K. Lee, C. S. L. Koh, Y. H. Lee, C. Liu, I. Y. Phang, X. Han, C.-K. Tsung and X. Y. Ling, *Sci. Adv.*, 2018, **4**, eaar3208.
- 9 Y.-J. Mao, L. Wei, X.-S. Zhao, Y.-S. Wei, J.-W. Li, T. Sheng, F.-C. Zhu, N. Tian, Z.-Y. Zhou and S.-G. Sun, *Chem. Commun.*, 2019, **55**, 9335–9338.
- 10 J. Li, G. Zhan, J. Yang, F. Quan, C. Mao, Y. Liu, B. Wang, F. Lei, L. Li, A. W. M. Chan, L. Xu, Y. Shi, Y. Du, W. Hao, P. K. Wong, J. Wang, S.-X. Dou, L. Zhang and J. C. Yu, *J. Am. Chem. Soc.*, 2020, **142**, 7036–7046.
- 11 H.-J. Yin, J.-H. Zhou and Y.-W. Zhang, *Inorg. Chem. Front.*, 2019, **6**, 2582–2618.
- 12 Z. W. Seh, J. Kibsgaard, C. F. Dickens, I. Chorkendorff, J. K. Nørskov and T. F. Jaramillo, *Science*, 2017, **355**, eaad4998.
- 13 M.-A. Légaré, G. Bélanger-Chabot, R. D. Dewhurst, E. Welz, I. Krummenacher, B. Engels and H. Braunschweig, *Science*, 2018, **359**, 896–900.
- 14 C. Ling, Y. Ouyang, Q. Li, X. Bai, X. Mao, A. Du and J. Wang, *Small Methods*, 2019, **3**, 1800376.
- 15 J. H. Montoya, C. Tsai, A. Vojvodic and J. K. Nørskov, *ChemSusChem*, 2015, **8**, 2180–2186.
- 16 E. Skúlason, T. Bligaard, S. Gudmundsdóttir, F. Studt, J. Rossmeisl, F. Abild-Pedersen, T. Vegge, H. Jónsson and J. K. Nørskov, *Phys Chem Chem Phys*, 2012, **14**, 1235–1245.
- 17 B. Qiao, A. Wang, X. Yang, L. F. Allard, Z. Jiang, Y. Cui, J. Liu, J. Li and T. Zhang, *Nat. Chem.*, 2011, **3**, 634–641.
- 18 Z. Li, S. Ji, Y. Liu, X. Cao, S. Tian, Y. Chen, Z. Niu and Y. Li, *Chem. Rev.*, 2020, **120**, 623–682.
- 19 W. Zang, Z. Kou, S. J. Pennycook and J. Wang, *Adv Energy Mater*, 2020, 18.
- 20 Y. Wang, J. Mao, X. Meng, L. Yu, D. Deng and X. Bao, *Chem. Rev.*, 2019, **119**, 1806–1854.
- 21 Z. Zhang, Y. Zhu, H. Asakura, B. Zhang, J. Zhang, M. Zhou, Y. Han, T. Tanaka, A. Wang, T. Zhang and N. Yan, *Nat. Commun.*, 2017, **8**, 16100.
- 22 M. Moses-DeBusk, M. Yoon, L. F. Allard, D. R. Mullins, Z. Wu, X. Yang, G. Veith, G. M. Stocks and C. K. Narula, *J. Am. Chem. Soc.*, 2013, **135**, 12634–12645.
- 23 F. Maurer, J. Jelic, J. Wang, A. Gänzler, P. Dolcet, C. Wöll, Y. Wang, F. Studt, M. Casapu and J.-D. Grunwaldt, *Nat. Catal.*, 2020, 1–10.
- 24 L. Nie, D. Mei, H. Xiong, B. Peng, Z. Ren, X. I. P. Hernandez, A. DeLaRiva, M. Wang, M. H. Engelhard, L. Kovarik, A. K. Datye and Y. Wang, *Science*, 2017, **358**, 1419–1423.
- 25 W. Qu, X. Liu, J. Chen, Y. Dong, X. Tang and Y. Chen, *Nat. Commun.*, 2020, **11**, 1532.
- 26 J. Lin, A. Wang, B. Qiao, X. Liu, X. Yang, X. Wang, J. Liang, J. Li, J. Liu and T. Zhang, *J. Am. Chem. Soc.*, 2013, **135**, 15314–15317.
- 27 H. Fei, J. Dong, Y. Feng, C. S. Allen, C. Wan, B. Voloskiy, M. Li, Z. Zhao, Y. Wang, H. Sun, P.

- An, W. Chen, Z. Guo, C. Lee, D. Chen, I. Shakir, M. Liu, T. Hu, Y. Li, A. I. Kirkland, X. Duan and Y. Huang, *Nat. Catal.*, 2018, **1**, 63–72.
- 28 Y. Peng, B. Lu and S. Chen, *Adv. Mater.*, 2018, **30**, 1801995.
- 29 P. Vancsó, Z. I. Popov, J. Pető, T. Ollár, G. Dobrik, J. S. Pap, C. Hwang, P. B. Sorokin and L. Tapasztó, *ACS Energy Lett.*, 2019, **4**, 1947–1953.
- 30 Y. Shi, W.-M. Huang, J. Li, Y. Zhou, Z.-Q. Li, Y.-C. Yin and X.-H. Xia, *Nat. Commun.*, 2020, **11**, 4558.
- 31 G. Liu, A. W. Robertson, M. M.-J. Li, W. C. H. Kuo, M. T. Darby, M. H. Muhieddine, Y.-C. Lin, K. Suenaga, M. Stamatakis, J. H. Warner and S. C. E. Tsang, *Nat. Chem.*, 2017, **9**, 810–816.
- 32 J. Zhang, Y. Zhao, X. Guo, C. Chen, C.-L. Dong, R.-S. Liu, C.-P. Han, Y. Li, Y. Gogotsi and G. Wang, *Nat. Catal.*, 2018, **1**, 985–992.
- 33 D. Zhao, Z. Chen, W. Yang, S. Liu, X. Zhang, Y. Yu, W.-C. Cheong, L. Zheng, F. Ren, G. Ying, X. Cao, D. Wang, Q. Peng, G. Wang and C. Chen, *J. Am. Chem. Soc.*, 2019, **141**, 4086–4093.
- 34 R. T. Hannagan, G. Giannakakis, M. Flytzani-Stephanopoulos and E. C. H. Sykes, *Chem. Rev.*, 2020, acs.chemrev.0c00078.
- 35 G. Kyriakou, M. B. Boucher, A. D. Jewell, E. A. Lewis, T. J. Lawton, A. E. Baber, H. L. Tierney, M. Flytzani-Stephanopoulos and E. C. H. Sykes, *Science*, 2012, **335**, 1209–1212.
- 36 G. Giannakakis, M. Flytzani-Stephanopoulos and E. C. H. Sykes, *Acc. Chem. Res.*, 2019, **52**, 237–247.
- 37 R. Long, Y. Li, Y. Liu, S. Chen, X. Zheng, C. Gao, C. He, N. Chen, Z. Qi, L. Song, J. Jiang, J. Zhu and Y. Xiong, *J. Am. Chem. Soc.*, 2017, **139**, 4486–4492.
- 38 J. Li, A. Xu, F. Li, Z. Wang, C. Zou, C. M. Gabardo, Y. Wang, A. Ozden, Y. Xu, D.-H. Nam, Y. Lum, J. Wicks, B. Chen, Z. Wang, J. Chen, Y. Wen, T. Zhuang, M. Luo, X. Du, T.-K. Sham, B. Zhang, E. H. Sargent and D. Sinton, *Nat. Commun.*, 2020, **11**, 3685.
- 39 J. Mao, C.-T. He, J. Pei, Y. Liu, J. Li, W. Chen, D. He, D. Wang and Y. Li, *Nano Lett.*, 2020, **20**, 3442–3448.
- 40 J. M. P. Martirez and E. A. Carter, *ACS Nano*, 2016, **10**, 2940–2949.
- 41 J. M. P. Martirez and E. A. Carter, *Sci. Adv.*, 2017, **3**, eaao4710.
- 42 Y. Wang, L. Cao, N. J. Libretto, X. Li, C. Li, Y. Wan, C. He, J. Lee, J. Gregg, H. Zong, D. Su, J. T. Miller, T. Mueller and C. Wang, *J. Am. Chem. Soc.*, , DOI:10.1021/jacs.9b05766.
- 43 G. Kresse and J. Furthmüller, *Phys. Rev. B*, 1996, **54**, 11169–11186.
- 44 J. P. Perdew, K. Burke and M. Ernzerhof, *Phys. Rev. Lett.*, 1996, **77**, 3865–3868.
- 45 G. Kresse and D. Joubert, *Phys. Rev. B*, 1999, **59**, 1758–1775.
- 46 S. Grimme, *J. Comput. Chem.*, 2006, **27**, 1787–1799.
- 47 H. J. Monkhorst and J. D. Pack, *Phys. Rev. B*, 1976, **13**, 5188–5192.
- 48 [1908.08269] VASPKIT: A Pre- and Post-Processing Program for VASP code, <https://arxiv.org/abs/1908.08269>, (accessed December 11, 2019).
- 49 The atomic simulation environment—a Python library for working with atoms - IOPscience, <https://iopscience.iop.org/article/10.1088/1361-648X/aa680e>, (accessed February 20, 2020).
- 50 J. K. Nørskov, J. Rossmeisl, A. Logadottir, L. Lindqvist, J. R. Kitchin, T. Bligaard and H. Jónsson, *J. Phys. Chem. B*, 2004, **108**, 17886–17892.
- 51 K. Mathew, R. Sundararaman, K. Letchworth-Weaver, T. A. Arias and R. G. Hennig, *J. Chem. Phys.*, 2014, **140**, 084106.
- 52 X. Guo, J. Gu, S. Lin, S. Zhang, Z. Chen and S. Huang, *J. Am. Chem. Soc.*, 2020, **142**, 5709–

- 5721.
- 53 X. Cui, C. Tang and Q. Zhang, *Adv. Energy Mater.*, 2018, **8**, 1800369.
- 54 M. Yu and D. R. Trinkle, *J. Chem. Phys.*, 2011, **134**, 064111.
- 55 S. Tang, T. Liu, Q. Dang, X. Zhou, X. Li, T. Yang, Y. Luo, E. Sharman and J. Jiang, *J. Phys. Chem. Lett.*, 2020, **11**, 5051–5058.
- 56 L. Li, X. Wang, H. Guo, G. Yao, H. Yu, Z. Tian, B. Li and L. Chen, *Small Methods*, 2019, **3**, 1900337.
- 57 L. Cai, N. Zhang, B. Qiu and Y. Chai, *ACS Appl. Mater. Interfaces*, 2020, **12**, 20448–20455.
- 58 X. Guo, S. Lin, J. Gu, S. Zhang, Z. Chen and S. Huang, *Adv. Funct. Mater.*, **n/a**, 2008056.
- 59 Y. Ma, T. Yang, H. Zou, W. Zang, Z. Kou, L. Mao, Y. Feng, L. Shen, S. J. Pennycook, L. Duan, X. Li and J. Wang, *Adv. Mater.*, 2020, 2002177.
- 60 H. Tao, C. Choi, L.-X. Ding, Z. Jiang, Z. Han, M. Jia, Q. Fan, Y. Gao, H. Wang, A. W. Robertson, S. Hong, Y. Jung, S. Liu and Z. Sun, *Chem*, 2019, **5**, 204–214.
- 61 Y. Chen, R. Guo, X. Peng, X. Wang, X. Liu, J. Ren, J. He, L. Zhuo, J. Sun, Y. Liu, Y. Wu and J. Luo, *ACS Nano*, 2020, **14**, 6938–6946.
- 62 Z. Geng, Y. Liu, X. Kong, P. Li, K. Li, Z. Liu, J. Du, M. Shu, R. Si and J. Zeng, *Adv. Mater.*, 2018, **30**, 1803498.
- 63 S. Wang, W. Wei, X. Lv, B. Huang and Y. Dai, *J. Mater. Chem. A*, 2020, 10.1039.C9TA10935H.
- 64 C. Choi, S. Back, N.-Y. Kim, J. Lim, Y.-H. Kim and Y. Jung, *ACS Catal.*, 2018, **8**, 7517–7525.
- 65 E. Santos and W. Schmickler, *ChemPhysChem*, 2006, **7**, 2282–2285.
- 66 H. Guo, L. Li, X. Wang, G. Yao, H. Yu, Z. Tian, B. Li and L. Chen, *ACS Appl. Mater. Interfaces*, 2019, **11**, 36506–36514.
- 67 W. Gao, Y. Chen, B. Li, S.-P. Liu, X. Liu and Q. Jiang, *Nat. Commun.*, 2020, **11**, 1196.
- 68 H. Xu, D. Cheng, D. Cao and X. C. Zeng, *Nat. Catal.*, 2018, **1**, 339–348.
- 69 F. Pedregosa, G. Varoquaux, A. Gramfort, V. Michel, B. Thirion, O. Grisel, M. Blondel, P. Prettenhofer, R. Weiss, V. Dubourg, J. Vanderplas, A. Passos and D. Cournapeau, *J. Mach. Learn. Res.*, 2011, **12**, 2825–2830.
- 70 X. Guo, S. Lin, J. Gu, S. Zhang, Z. Chen and S. Huang, *ACS Catal.*, 2019, 11042–11054.
- 71 L. Wu, T. Guo and T. Li, *J. Mater. Chem. A*, , DOI:10/ghc8d3.
- 72 H. Yuan, Z. Li, X. C. Zeng and J. Yang, *J. Phys. Chem. Lett.*, 2020, **11**, 3481–3487.
- 73 A. R. Singh, B. A. Rohr, M. J. Statt, J. A. Schwalbe, M. Cargnello and J. K. Nørskov, *ACS Catal.*, 2019, **9**, 8316–8324.
- 74 S. D. Minter, P. Christopher and S. Linic, *ACS Energy Lett.*, 2019, **4**, 163–166.
- 75 A. R. Singh, B. A. Rohr, J. A. Schwalbe, M. Cargnello, K. Chan, T. F. Jaramillo, I. Chorkendorff and J. K. Nørskov, *ACS Catal.*, 2017, **7**, 706–709.
- 76 K. Sun, Y. Ji, Y. Liu and Z. Wang, *J. Mater. Chem. A*, 2020, **8**, 12291–12295.
- 77 X.-F. Yang, A. Wang, B. Qiao, J. Li, J. Liu and T. Zhang, *Acc. Chem. Res.*, 2013, **46**, 1740–1748.
- 78 J. Liu, *ACS Catal.*, 2017, **7**, 34–59.

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