Assessing the Role of Inter-Molecular Interactions in a Perylene-Based Nanowire Using First-Principles Many-Body Perturbation Theory

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Abstract

We present a first-principles many-body perturbation theory study of the role of inter-molecular coupling on the optoelectronic properties of a one-dimensional π -stacked nanowire composed of perylene-3,4,9,10-tetracarboxylic diimide (PTCDI) molecules on a DNA-like backbone. We determine that strong inter-molecular electronic coupling results in large bandwidths and low carrier effective masses, suggesting a high electron mobility material. Additionally, by including the role of finite temperature phonons on optical absorption via a newly presented approach, we predict that the optical absorption spectrum at room temperature is significantly altered from room temperature due to allowed indirect transitions, while the exciton delocalization and binding energy, a measure of inter-molecular electronic interactions, remains constant. Overall, our studies indicate that strong inter-molecular coupling can dominate the optoelectronic properties of π -conjugated 1D systems even at room temperature.

TOC Graphic



Keywords: GW/BSE; density functional theory; density functional perturbation theory; organic nanowire; electron-phonon interactions; excitons

Organic semiconductors are promising next-generation optoelectronic materials due to their strong light absorption, abundance, flexibility, cheap processing, and chemical tunability.¹⁴ In particular, one-dimensional (1D) organic nanowires, consisting of stacked molecular building blocks, have been incorporated as components in field-effect transistors,⁹⁴¹ chemical sensors,¹²⁴³ optical waveguides,¹⁴⁴⁵ lasers,¹⁶ light emitting diodes,¹⁷ and excitonic circuits.¹⁸⁴⁹ These 1D nanowires can be grown via molecular self-assembly with significant control over their intermolecular structure. This in turn has allowed the design of these materials for high carrier mobility, and efficient light absorption and emission at controllable wavelengths.²⁰⁻²⁶

 π -stacked 1D assemblies, in particular, display higher charge carrier mobility than their thin film counterparts,⁷⁷²⁸ and enhanced exciton diffusion lengths.^{23,29,30} Importantly, their onedimensional nature allows exciton motion to be directed in a particular direction,²⁹ with longrange energy propagation in the form of exciton-polariton migration.²⁷ To further improve the functionality of these materials, it is necessary to develop a quantitative understanding of the role of inter-molecular interactions on their optoelectronic properties, which can be achieved via theoretical modeling.

First-principles and semi-empirical simulations of excited-states in 1D π -conjugated molecular stacks have shown that inter-molecular orientation strongly influences electronic coupling, and therefore the optoelectronic properties.³¹⁻³⁶ Additionally, an interplay between Coulomb and charge-transfer-mediated coupling can result in a wide range of photophysical behaviors in dimers,^{31,22} and extended polymers.^{35,36} Computational studies have also shown that vibrational excitations significantly influence the optoelectronic properties of 1D extended molecular stacks,³⁷⁻⁴⁰ with strongly temperature-dependent charge carrier mobilities³⁷ and strong vibronic coupling in the absorption spectrum.³⁸⁻⁴⁰ However, a quantitative description of how electronic

interactions in 1D π -conjugated stacks are influenced by finite temperature and long-range intermolecular phonons, is still lacking.



Figure 1. (a) A single Perylene-3,4,9,10 tetracarboxylic diimide (PTCDI) molecule on a phosphoalkane backbone, (b) A nanowire composed of an infinite array of these molecules. The black parallelepiped indicates the unit cell.

In this Letter, we utilize density functional theory (DFT), density functional perturbation theory (DFPT) and many-body perturbation theory (MBPT) to study the optoelectronic properties, including the role of phonons, of a nanowire composed of a biologically inspired perylene-3,4,9,10-tetracarboxylic diimide (PTCDI) derivatives along a DNA-like backbone (Figure 1). PTCDI are a well-studied class of organic materials,^{31,41} with a highly tunable intermolecular spacing and orientation⁴² and high electron and exciton mobility.³⁴ Previous computational and experimental studies of finite stacks of these specific PTCDI derivatives have determined that they possess strong inter-molecular coupling along the π -stack direction with measurements suggesting efficient coherent transport along the stacks.^{43,44} In this work, we investigate an extended system (Figure 1b) based on the stacks of Ref.⁴ (Figure 1a). For this artificially constructed but potentially synthesizable nanowire, we determine that there is strong inter-molecular electronic coupling that result in high electron and hole mobility. Additionally, we apply a newly presented approach⁶⁵ accounting for the role of finite temperature phonons on the optical properties of materials. Such an approach, to our knowledge, has never been applied to organic systems previously. We determine that at room temperature, long-range inter-molecular phonon-assisted excitations result in a significantly different optical excitation spectrum, with little impact on the nature or binding energy of the exciton. Our study indicates that strong inter-molecular electronic coupling lasts at finite temperature, suggesting that organic nanowires synthesized with significant π -stacking can retain their electronic coherence at finite temperature.



Figure 2. The optical and electronic structure of the nanowire and molecule at T = 0K, excluding zero-point vibrations. (a) Band structure of the nanowire within the GW approximation with valence states in red and conduction states in blue. (b) The optical absorption spectrum of the gas-phase molecule and the nanowire. (c) Exciton distribution for the lowest energy excitation at 2.3 eV with the hole placed slightly above a carbon atom within the PTCDI plane.

First, we examine the electronic and optical properties of the nanowire at zero temperature, with the optical spectrum compared to the gas-phase. The gas-phase molecule is taken from those previously synthesized (Ref.⁴); while the extended nanowire (Figure 1b) was created such that the inter-molecular spacing and stacking is consistent with the synthesized dimer but that periodic boundary conditions are applied. Due to the presence of the backbone, the spacing between molecules is confined to ~ 3.5 Å, which is quite close for van der Waals-interacting molecules. The strong π - π overlap in this structure results in strong electronic interactions as is evident by the dispersive bandwidths as shown in Figure 2a.

For the nanowire, the fundamental gap, defined as the difference between the electron affinity and ionization potential, is indirect; the computed valence band maximum (VBM) is at the Gamma point while the conduction band minimum (CBM) is at the edge of the first Brillouin zone. The magnitude of the gap is predicted to be 2.7 eV in the nanowire, which is much smaller than that predicted for the gas-phase molecule (4.8 eV). The lowering of the gap in the wire can be partially explained by increased screening in the condensed phase;⁴⁶ however, this is a weakly screening 1D system with a calculated average dielectric constant ~ 1.5. The dominant reason for the lower fundamental gap is explained by the formation of dispersive bands; the predicted valence band (VB) and conduction band (CB) bandwidths are both 0.8 eV.

The strong inter-molecular electronic interactions suggest that this material may display high carrier mobility. In order to assess the electron and hole mobilities, we compute the effective mass for electron and hole as $m_{e/h}^* = \frac{\hbar^2}{\frac{\partial^2 E_{c/v(k)}}{\partial k^2}}$. The carriers are found to be quite light with both

approximately $0.08m_{e}$ for the hole and electron, where m_{e} is the free electron mass. Such small effective masses are consistent with previous studies showing high carrier mobility in other π -stacked nanowires.²⁷²⁸

Figure 2b presents the optical absorption spectrum of the nanowire compared with the isolated molecule. For the gas-phase molecule, there is one peak in absorption at 2.8 eV, consisting of

three excited-state transitions. For the nanowire, there are two low-energy peaks at around 2.6 eV and 3 eV, consisting of two states (labeled peak 1) and three states (labeled peak 2). The peaks associated with the nanowire display a reduced intensity compared to the gas-phase, indicating a reduction of overlap between electron and hole in the solid-state. Additionally, the lowering of the onset of absorption from 2.8 eV for the gas-phase to 2.3 eV for the nanowire, can be explained by solid-state screening as well delocalization of the excited-state. The latter is illustrated in Figure 2c, which show the exciton distribution for the lowest energy excited-state within the nanowire, with the hole confined to a high probability location and the electron density plotted. The electron density is delocalized over multiple molecules, as expected for a system with strong π -orbital interactions^e.



Figure 3. (a) Comparison of the IR spectrum of the gas-phase molecule and nanowire. (b) Zoomed IR spectra from 0 to 300 cm⁻¹. (c-d) Two low energy vibrational modes that dominate the ZG geometry.

In order to understand whether the strong electronic interactions and excited-state delocalization hold at finite temperature, we study the role of room temperature atomic vibrations and long-range phonons on the properties of the nanowire. We first perform a comparison of the phonon normal modes for the extended system and isolated molecule (Figure

3a-b) to understand the changes in vibrational spectra upon formation of the extended system. The IR spectrum for the gas-phase molecule displays peaks characteristic of intra-molecular C=C and C-H stretching modes, as well as C-H rocking, at 850-1050 cm⁴ and 1400-1600 cm⁴, and the characteristic C-H stretch at 3000 cm⁻¹, consistent with other studies of PTCDI derivatives.^{48,49} The IR spectrum of the nanowire is similar to that of the isolated molecule for wavenumber greater than 1400 cm⁻¹. Below this frequency, the phonon modes of the nanowire can be decomposed into two regions: intra-molecular vibrations modulated by the backbone ($\sim 800-1000$ cm⁴) and inter-molecular vibrations (below 500 cm³). In the intra-molecular vibration dominant region, the peak shapes for the nanowire are similar to that of the molecule, but the peaks are shifted to lower energies by (~ 60 cm³). This may be explained by a weakened intra-molecular C=C bond due to the inter-molecular π - interactions; the inter-molecular π -orbital interaction weakens the double bond such that it requires less energy for intra-molecular bond-stretching.⁹⁰ In the lower energy region below 300 cm⁻¹, the vibrational modes include out-of-plane motion of the carbon atom, as well as molecular rotation and backbone motion, with many inter-molecular modes in the nanowire that do not exist in the gas-phase, resulting in higher IR intensities in this region for the wire.

We then compute the optoelectronic properties of the nanowire at T = 300K computed via the method introduced by Zacharias and Giustino (ZG).⁴⁵ As described in the Computational Details, by this method, phonons are accounted for in the calculated optical spectrum by considering a structural re-organization of the nanowire due to phonons. The optical spectrum associated with the distorted structure approximates the finite temperature optical spectrum. Using this approach, we predict a slight geometry re-organization of the nanowire, with an average displacement of 0.2 Å per atom; this structural reorganization is dominated by low-energy, long-range phonon

modes, indicating that these modes will couple to the excited state. The two modes with the highest contribution are shown in Figure 3c-d. Both modes are low energy inter-molecular modes with energies less than 67 cm⁻¹. The displacement associated with all other phonon modes is smaller than these two by a factor of 1.5 or larger, suggesting that the optically excited-state couples most strongly to these inter-molecular phonons, rather than intra-molecular C=C stretch modes, as would be expected for an isolated molecule.



Figure 4. The optical and electronic structure of the nanowire at T = 300K. (a) GW-calculated band structure with valence states in red and conduction states in blue. (b) GW/BSE calculated optical absorption spectrum. (c) Exciton distribution for the lowest energy excited-state at 1.8 eV, with the hole placed slightly above a carbon atom within the distorted PTCDI plane.

The role of room temperature on excitations within the PTCDI nanowire is summarized in Figure 4 and Table 1. The predicted band structure for the ZG structure (Figure 4 a) is direct, with a fundamental gap of 2.8 eV. We note here that the band structure of the nanowire is not direct at finite temperature but the ZG structure accounts for the presence of phonons, which allow the indirect gap to be traversed. Additionally, as shown in Figure 4a, due to the breaking of symmetry between equivalent molecules in the wire, the VB and CB each split into two nearly-degenerate bands. The energy range covered by both bands is 0.9 (1.0) eV for the VB (CB),

consistent with the bandwidths of the 0K structure, suggesting that the strong electronic interactions are preserved.

Figure 4b shows that the optical spectrum of the nanowire at 300K is shifted to lower energies, with multiple peaks within the 0-4 eV energy window. The onset of absorption is at 1.8 eV, 0.5 eV lower than at 0K because the presence of phonons provides the momentum necessary for indirect excitations to occur. Analysis of the exciton wavefunction for the lowest energy state indicates an excitonic state that is delocalized over several molecules, consistent with that of the T = 0K structure. Moreover, the exciton binding energy, calculated as the difference in onset of absorption between the system with electron-hole interactions and one without, is predicted to be 1.2 eV at 0K and 1.0 eV at 300K. The large value of the exciton binding energy is consistent with the weak screening of electron-hole Coulomb interactions within the material. The preservation of the exciton binding energy and exciton extent at finite temperature suggests that the strong inter-molecular electronic interactions are not significantly disrupted at finite temperature.

Table 1. Optical gaps and exciton binding energies^{*} of the nanowire at T = 0K and 300K.

	Optical Gap (eV)	Exciton Binding Energy (eV)
$\mathbf{T} = 0 \mathbf{K}$	2.3	1.1
T= 300K	1.8	1.0

* The exciton binding is calculated as the energy difference between the onset of absorption with and without electron-hole interactions and is not in reference to the fundamental gap.

In summary, we investigated the optoelectronic properties of a nanowire composed of PTCDI derivatives using density functional theory and many-body perturbation theory, including a novel approach for calculating temperature-dependent optical absorption spectra. At T = 0K, we

predict that strong inter-molecular electronic interactions due to π -orbital stacking lead to high bandwidth and low effective mass, suggesting high conductivity nanowires. Additionally, we examined the change in fundamental and optical gaps upon formation of a one-dimensional condensed phase, and showed that the weak screening of the Coulomb interaction in 1D results in higher exciton binding energies than for typical bulk organic materials. Lastly, we quantitatively determined the change of the nanowire properties under the influence of atomic vibrations and phonons, applying a new approach to electron-phonon interactions that has not been previously applied to organic materials. This analysis revealed that at T = 300 K, the presence of phonons allows indirect transitions to occur, resulting in a lowered optical gap, while the exciton delocalization and binding energy remained constant. Our studies indicate that strong inter-molecular coupling can dominate the optoelectronic properties of organic π -stacked materials even at room temperature.

Computational Details

DFT calculations were performed using the Quantum Espresso package⁴⁴ with the Perdew-Burke-Ernzerhof (PBE) approximation to the exchange-correlation.⁴² Norm-conserving Troullier-Martins pseudopotentials⁴⁵ describe the core and nuclei of atoms with 1, 4, 5, 5, and 6, electrons explicitly considered valence for H, C, N, P, and O, respectively. For the gas-phase molecule (Figure 1 a) the wave function cutoff energy for geometry optimization was 120 Ry, while for subsequent MBPT calculations it was set to 80 Ry, converging the total energy to 2.9 meV/atom and 7 meV/atom, respectively. The isolated molecule was placed in a cubic box with side length of 26.5 Å and the geometry optimized within DFT with a force convergence threshold of 3.4 x 10⁴ eV/atom. For the nanowire, the Grimme-D2 van der Waals correction⁴⁴ was utilized for geometry optimization, with a force convergence threshold of 5 x 10^{2} eV/Å. A wave function cutoff energy of 120 Ry and k-point sampling of 1 x 1 x 3 were used, ensuring that the total DFT energy was converged to within 2 meV/atom. The relaxed unit cell size for the nanowire was 6.0 Å x 39.1 Å x 13.9 Å, with the molecular stacks being approximately 3.5 Å apart in a face-to-face orientation.

The vibrational normal modes of both molecule and nanowire were computed using density functional perturbation theory $(DFPT)^{s}$ within the Quantum Espresso package, with a q-point grid of 1 x 1 x 2 used for the nanowire.

MBPT calculations within the GW/BSE approximation^{9,57} for both isolated molecule and nanowire were performed using the BerkeleyGW⁸ package with a PBE starting point. The static dielectric matrix was calculated within the random phase approximation (RPA) and extended to finite frequency with a generalized plasmon pole (GPP) approximation.³⁷ Due to the imposed periodic boundary conditions, the molecule was placed in a supercell, with lattice vectors along the non-periodic directions that are twice the size necessary to contain 99% of the charge density. In order to minimize the spurious interactions between periodic images, the Coulomb potential was truncated at half of the supercell length. The number of empty states for the molecule and the nanowire in the GW sum was set to 4880 and 4000, respectively, with convergence described in the SI. The neutral excitation energies solved by Bethe-Salpeter equation (BSE) use the Tamm-Dancoff approximation³⁷ with 24 valence and 24 conduction bands included in the summation for both the molecule and the nanowire. For the nanowire, GW and BSE calculations were performed on a k-point grid of 1x1x8.

To understand the temperature-dependence of optical absorption of the nanowire, we utilized a recently introduced approach by Zacharias and Giustino (ZG),⁴⁵ which describes the role of finite

temperature on the dielectric response of materials via an approximation to the Williams[®] and Lax[®] (WL) theory. In short, the imaginary component of the temperature-dependent dielectric function can be approximated as $\epsilon_2(\omega; T) = \prod_{\nu} \int dx_{\nu} \frac{\exp(-x_{\nu}^2/2\sigma_{\nu,T}^2)}{\sqrt{2\pi}\sigma_{\nu,T}} \epsilon_2(\omega; x)$ where x_{ν} is the normal mode coordinate, $\sigma_{\nu,T}^2$ is a Gaussian broadening, $\frac{\exp(-x_{\nu}^2/2\sigma_{\nu,T}^2)}{\sqrt{2\pi}\sigma_{\nu,T}}$ is the importance function for Monte Carlo integration, and $\epsilon_2(\omega; x)$ is evaluated at clamped nuclei. In the limit of a large supercell, one single distorted atomic configuration is sufficient to produce the temperaturedependent dielectric function;⁴⁵ the atomic displacements for this configuration are determined from normal mode analysis $\Delta \tau_{\kappa\alpha} = (M_P/M_\kappa)^{\frac{1}{2}} \sum_{\nu} (-1)^{\nu-1} e_{\kappa\alpha,\nu} \sigma_{\nu,T}$ where M_P as the proton mass and M_{κ} the mass of the κ th nucleus.

This analysis is performed within a modified q2r.x subroutine of Quantum Espresso obtained from the authors.⁴⁵ For our study, the perturbed average-geometry at room temperature was obtained by skipping the first 22 normal modes in the ZG analysis. Subsequently, the electronic band structure and optical spectrum were obtained by performing GW/BSE calculations and converged with respect to the q-mesh considered. We determined that a 1x1x2 q-mesh is sufficient as shown in the SI.

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Supporting Information

Supporting Information Available: GW convergence plots of the nanowire, the effective mass fit for the hole and electron, and the density of states plots at valence band edge for convergence of the ZG method.

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