# Assessment of Constant-Potential Implicit Solvation Calculations of Electrochemical Energy Barriers for H<sub>2</sub> Evolution on Pt

Maxime Van den Bossche,<sup>†,‡</sup> Egill Skúlason,<sup>‡</sup> Christoph Rose-Petruck,<sup>†</sup> and Hannes Jónsson<sup>\*,‡</sup>

†Department of Chemistry, Brown University, Providence, RI, United States ‡Science Institute and Faculty of Physical Sciences, University of Iceland, 107 Reykjavík, Iceland

E-mail: hj@hi.is

## <sup>1</sup> Abstract

Theoretical estimation of the activation energy 2 of electrochemical reactions is of critical impor-3 tance but remains challenging. In this work, we 4 address the usage of an implicit solvation model 5 for describing hydrogen evolution reaction steps 6 on Pt(111) and Pt(110), and compare with 7 the 'extrapolation' approach as well as single-8 crystal measurements. We find that both meth-9 ods yield qualitatively similar results, which are 10 in fair agreement with the experimental data. 11 Care should be taken, however, in addressing 12 spurious electrostatic interactions between pe-13 riodically repeated slabs in the VASPsol im-14 plementation. Considering the lower computa-15 tional cost and higher flexibility of the implicit 16 solvation approach, we expect this method to 17 become a valuable tool in electrocatalysis. 18

# <sup>19</sup> Introduction

Electrocatalytic reactions involve electron 20 transfer to or from an electrode surface and 21 are evidently crucial for all electrochemical 22 production processes, as well as for corrosion 23 chemistry and the electronics industry. De-24 tailed insights into the reaction mechanism and 25 kinetics, however, are difficult to obtain and 26 generally require a combination of experimen-27

tal and computational work.

While in classical, "gas phase" heterogeneous 29 catalysis the field of theoretical modeling is 30 rather well established, further methodologi-31 cal development is still required when it comes 32 to electrocatalysis. Two major challenges in 33 this field are given by (i) the description of 34 metal/electrolyte interface, and (ii) the require-35 ment of constant electrode potential over the 36 course of an electron transfer reaction. These 37 aspects complicate, in particular, the evalua-38 tion of activation energies of electron trans-39 fer reactions. As a result, the vast majority 40 of computational electrocatalysis studies have 41 been limited to thermodynamic properties only, 42 such as reaction energies.<sup>1,2</sup> It is, however, well 43 recognized that the proper study of reaction 44 kinetics (in electrocatalysis as in conventional 45 catalysis) requires knowledge of transition state 46 energies.<sup>3,4</sup> 47

28

Several approaches have so far been proposed 48 to address these challenges, which differ espe-49 cially in the manner the constant-potential re-50 quirement is handled. This requirement stip-51 ulates that the electrode potential w.r.t. the 52 reference electrode remains constant over the 53 course of an elementary reaction, also when it 54 is accompanied by electron transfer. As only a 55 finite number of electrons  $N_{\rm e}$  can be included in 56 the electronic structure calculations, however, 57

the potential will change significantly over the course of an electron transfer reaction, if  $N_{\rm e}$  is held constant.<sup>5-7</sup> Extended electrode surfaces are furthermore most conveniently modeled using periodically repeated supercells, which must satisfy overall charge neutrality.

In this context, the arguably most rigor-64 ous method is the so-called 'extrapolation' ap-65 proach,<sup>5,6</sup> where energy barriers are evaluated 66 at successively larger lateral cell dimensions and 67 extrapolated to the infinite cell limit where the 68 potential drop is zero. The number of electrons 69  $N_{\rm e}$  is determined by the charge neutrality con-70 dition, which entails that e.g. hydrogen atoms 71 must be added to or withdrawn from the elec-72 trolyte to vary the electrode potential. This 73 method has so far exclusively been applied to 74 FCC(111) surfaces in conjunction with ice-like 75 hexagonal water structures.<sup>5,6,8,9</sup> 76

In a second, related method,  $^{10,11}$  the compu-77 tationally costly extrapolation is avoided by as-78 suming that the required correction to the tran-79 sition state energy has a purely capacitive char-80 acter. This correction is based on changes in 81 the charges of the metal atoms at the electrode 82 surface, and hence an additional ambiguity lies 83 in the choice of the charge density partitioning 84 scheme. 85

Thirdly, in the 'double reference' ap-86  $proach^{12,13}$  the electrode potential is controlled 87 by varying  $N_{\rm e}$ , with introduction of a homo-88 geneous compensating background to maintain 89 overall charge neutrality. It is not quite clear, 90 at this point, if this method applies sound cor-91 rections to the obtained total energies, which is 92 known to be a non-trivial problem in the case 93 of charged slabs in homogeneous compensating 94 backgrounds.<sup>14,15</sup> 95

Methods have also been proposed where 96 charge neutrality is maintained by localiz-97 ing the compensating charge in a 'counter-98 electrode' at some distance away from the 99 electrode/electrolyte interface.<sup>16,17</sup> In reality, 100 however, charge compensation is mediated by 101 changes in the concentration profiles of the elec-102 trolyte's ions near the electrode surface. 103

<sup>104</sup> In yet another approach<sup>18,19</sup> electrochemi-<sup>105</sup> cal barriers are estimated in a two-step pro-<sup>106</sup> cedure inspired by Markus theory. First, the activation energy of the corresponding non-107 electrochemical reaction is calculated (e.g. re-108 action with adsorbed H in the case of a proton-109 coupled electron transfer (PCET) reaction) and 110 it is assumed that, at the same potential, the 111 electrochemical process possesses the same acti-112 vation energy. The potential dependence of the 113 activation energy is then described using e.g. 114 Butler-Volmer theory, requiring further simpli-115 fying assumptions regarding the symmetry fac-116 tor of the reaction. 117

Finally, a more recently developed approach, 118 which we will be focusing on in this work, is 119 one where (part of) the electrolyte is approx-120 imated by a polarizable dielectric continuum 121 an implicit solvation model).<sup>20–23</sup> The (i.e. 122 constant-potential requirement is here obeyed 123 by varying  $N_{\rm e}$  with concomitant changes in 124 the concentration of counterions in the sur-125 rounding electrolytes (frequently described by 126 Poisson-Boltzmann theory). The main assump-127 tion here, then, is that the ionic distribution in 128 the electrolyte is equilibrated also at the saddle 129 point, and not only at the reactant and product 130 states. 131

Although several studies have applied the 132 above constant-potential approach with im-133 plicit solvation models (using e.g. the VASP-134 sol,<sup>24–27</sup> JDFTx,<sup>28</sup> SIESTA<sup>29–33</sup> and GPAW 135  $codes^{34}$ ), we are not aware of attempts to com-136 pare its results to those of the extrapolation 137 method, or to detailed single-crystal measure-138 ments. 139

The present work offers such a comparison, focusing on the various steps of the hydrogen evolution reaction (HER) on Pt(111) and Pt(110). There are three types of reactions relevant to the HER:

Volmer:	$\mathrm{H^{+} + e^{-} +^{*} \rightarrow H^{*}}$	(1)
---------	--	-----

- Tafel:  $2 \operatorname{H}^* \to \operatorname{H}_2 + 2^*$  (2)
- Heyrovský:  $\mathrm{H}^+ + \mathrm{e}^- + \mathrm{H}^* \to \mathrm{H}_2 + {}^*$  (3)

Adsorbed H and vacant adsorption sites are denoted H<sup>\*</sup> and <sup>\*</sup>, respectively. <sup>141</sup>

The (111) facet is chosen to compare with the <sup>142</sup> extrapolation approach, which has so far been <sup>143</sup> exclusively applied to FCC(111) surfaces due to <sup>144</sup> the commensurability with the hexagonal ice bi- <sup>145</sup>



Figure 1: Side and top views of the obtained transition states at 0  $V_{SHE}$  for the Volmer reaction on Pt(111) at an initial hydrogen coverage of 0.92 ML. The considered water structure are a protonated hexagonal ice bilayer [Panel (a)] and a H<sub>3</sub>O<sup>+</sup>(H<sub>2</sub>O)<sub>3</sub> ('Eigen') cation [Panel (b)].

layer structure used to model the metal/water 146 interface. The structure of this adlayer is illus-147 trated in Figure 1, in the transition state for 148 hydrogen deposition via the Volmer reaction. 149 As the implicit solvation approach is less re-150 stricted in terms of water structures, we will 151 also compare with results from a water cluster 152 model, corresponding to the 'Eigen' cation in its 153 protonated form  $(H_3O^+(H_2O)_3)$ . As the equi-154 librium H coverage on Pt(111) at 0  $V_{SHE}$  has 155 been calculated to be circa 0.9 ML using the 156 RPBE functional,<sup>6</sup> we will focus on HER reac-157 tions involving H atoms adsorbed in threefold 158 hollow sites at a coverage of up to 1 ML. Also, 159 the HER steps on (110) facets are included, as 160 experimental Tafel slopes are here easier to an-161 alyze and indicate a Volmer-Tafel mechanism 162 with the Tafel reaction as the rate-determining 163 step (RDS).<sup>35</sup> 164

It should be noted that the above considerations apply not only to activation energies but
also to reaction energies. The latter have, however, typically been calculated using the compu-

tational hydrogen electrode (CHE) formalism,<sup>1</sup> 169 which makes use of the equilibrium relation of 170 the proton/hydrogen redox-couple of the standard hydrogen electrode (SHE). The focus of 172 this work will hence lie on the estimation of 173 barrier heights. 174

175

## Computational methods

To facilitate the comparison to results from 176 previous publications with the extrapolation 177 approach, $^{6}$  the same computational setup is 178 used here. The bulk of the electronic struc-179 ture calculations are performed with the VASP 180 code<sup>36–39</sup> using Kohn-Sham density functional 181 theory (DFT).<sup>40,41</sup> Core electrons are treated 182 via standard projector augmented wave (PAW) 183 setups with the following valences: H (1), O (6), 184 Pt (10). The basis set consists of plane waves 185 with a kinetic energy up to 350 eV. 186

Additional calculations are carried out using the JDFTx code<sup>21</sup> employing GBRV ultrasoft pseudopotentials<sup>42</sup> with the recommended cutoffs for the wave functions (20 Hartree) and the electron density (100 Hartree).<sup>1</sup>

Electronic exchange and correlation are 192 described using the revised Perdew-Burke-193 Ernzerhof (RPBE) functional<sup>43</sup> which modifies 194 the exchange enhancement factor of the origi-195 nal PBE expression<sup>44</sup> to improve the descrip-196 tion of adsorption energies on transition metal 197 surfaces. In JDFTx the RPBE functional is 198 implemented through the LibXC library.<sup>45</sup> 199

Three-layer slabs of  $c(3 \times 4)$  geometry are em-200 ployed for the Pt(111) surface, with the RPBE 201 crystal lattice constant of 4.025 Å. For the 202 Pt(110) slab, six metal layers are used with a 203  $p(2 \times 4)$  cell size and a  $(1 \times 2)$  missing-row re-204 construction. In both cases the first Brillouin 205 zone is sampled using  $(4 \times 3 \times 1)$  Monkhorst-206 Pack grids.<sup>46,47</sup> Local minimizations and saddle 207 point searches with the dimer method  $^{48,49}$  are 208 pursued until the largest force components are 209 less than 0.05 eV/Å in magnitude. All metal 210 layers except the topmost layer are constrained 211 to their bulk lattice positions. The distances 212

<sup>&</sup>lt;sup>1</sup>In the GBRV pseudopotential set the  $5p^6$  semicore states are included in the valence for Pt.

separating periodically repeated slabs will beaddressed in the Results section.

Implicit solvation calculations of the aqueous electrolyte are performed using the GLSSA13 solvent model,<sup>50</sup> which is available in VASP through the VASPsol extension.<sup>22,23</sup> Like other models such as SCCS,<sup>51</sup> the method puts forward a dielectric profile  $\epsilon$  as a functional of the Kohn-Sham electron density distribution  $\rho$ :

$$\epsilon[\rho] = 1 + (\epsilon_{\text{bulk}} - 1) S[\rho], \qquad (4)$$

with the following form for the 'shape function' S:

$$S[\rho] = \frac{1}{2} \operatorname{erfc}\left(\frac{\log\left(\rho/\rho_{\mathrm{cut}}\right)}{\sigma\sqrt{2}}\right).$$
 (5)

The dielectric function therefore changes grad-224 ually from 1 to the bulk dielectric constant  $\epsilon_{\text{bulk}}$ 225 (78.4, the experimental value for liquid water at 226 298  $K^{52}$ ). The particular shape of this transi-227 tion is governed by the  $\sigma$  and  $\rho_{\rm cut}$  parameter, for 228 which we have applied the default values (0.6)229 and 0.0025 Å<sup>-3</sup>, respectively). The free energy 230 required to create a solvent cavity around the 231 solute is calculated as 232

$$A_{\text{cavity}} = \tau \int |\nabla S| d\mathbf{r}, \qquad (6)$$

where the  $\tau$  parameter (0.525 meV/Å<sup>2</sup>) has been optimized to reproduce the solvation energies of a series of organic molecules.<sup>22</sup>

In case the electrode potential is different
from the potential of zero charge (PZC), a counterion charge distribution is included in the implicit solvent region through the use of the generalized Poisson-Boltzmann equation:

$$\nabla \cdot \epsilon(\mathbf{r}) \nabla \phi_{tot}(\mathbf{r}) = \left(\frac{2c_0 q^2 e^2}{k_B T}\right) \phi_{tot}(\mathbf{r}) - \rho_{solute}(\mathbf{r}),$$
(7)

$$c_{\pm}(\mathbf{r}) = c_0 \exp\left(\frac{\pm q \, e \, \phi_{tot}(\mathbf{r})}{k_B T}\right) S\left[\rho(\mathbf{r})\right]. \tag{8}$$

We used  $\kappa$  values corresponding to a Debye length of 3 Å, corresponding to a 1 M concentration of a 1:1 electrolyte ( $c_0 = 1$  M). This mimics the experimental situation where 1 M of a strong acid is used (U<sub>SHE</sub> = 0 for HER and  $\mathbf{pH}=0).$ 

Though grand-canonical SCF algorithms ex-247 ist<sup>21</sup> which vary the number of electrons  $N_e$  so 248 as to match the targeted chemical potential, 249 currently only conventional (canonical) meth-250 ods are available in VASPsol, making it nec-251 essary to adjust  $N_e$  in between successive SCF 252 loops instead. For local minimization runs, we 253 use a simple iterative approach: 254

246

$$N_e(i+1) = N_e(i) - a \cdot [\mu_e(i) - \mu_{e,\text{target}}].$$
 (9)

We find an *a* value of  $1.0 \text{ V}^{-1}$  to be appropriate 255 for the structures considered in this work. In 256 this manner, the electrode potential converges 257 to the targeted value as the local optimization 258 proceeds. For saddle point searches, we find it 259 more convenient to adapt  $N_e$  after a completed 260 dimer search, iterating until the target potential 261 is matched within 15 mV. 262

The electronic chemical potential  $\mu_e$  is calculated by comparing the Fermi level with the potential in the bulk electrolyte, which in an implicit solvent equals that of an electron in vacuum:<sup>23</sup> 267

$$\mu_e = \epsilon_{\text{Fermi}} - V_{\text{bulk}}.$$
 (10)

It should be noted that the use of Equation (10)268 is not restricted to symmetric slab structures, 269 as the electrolyte screens the dipole moment of 270 the solute, resulting in a zero net dipole mo-271 ment and therefore also in a flat electrostatic 272 potential in the bulk solvent region. If, then, a 273 certain potential U is to be attained w.r.t. the 274 standard hydrogen electrode (SHE), the target 275 chemical potential is given by: 276

$$\mu_{e,\text{target}} = -\Phi_{\text{SHE}} - U, \qquad (11)$$

where  $\Phi_{\rm SHE}$  is the work function of the SHE. 277 We have taken  $\Phi_{\rm SHE}$  equal to 4.43 V, which lies 278 within the experimental value of the SHE work 279 function compared to vacuum which is usually 280 measured to be 4.44 (+/- 0.02) V.<sup>53</sup> 281

To compare energy differences of two structures at different  $N_e$  values (but identical electrode potential U), the VASPsol and JDFTx 284 output energies need to be corrected by the cost 285 286 (gain) of removing (adding) electrons:

$$\Omega = E_{\rm DFT} + N_{\rm e}\,\mu_{\rm e}(U). \tag{12}$$

Even though  $E_{\text{DFT}}$  here includes entropy terms from the solvation model, we will refer to  $\Delta\Omega$  as (electronic) grand-canonical energy differences, as no entropic contributions are included in the DFT calculations.

<sup>292</sup> The symmetry factor  $\beta$  for an elementary re-<sup>293</sup> action at a given U corresponds to the deriva-<sup>294</sup> tive of the activation energy w.r.t. U, which <sup>295</sup> we evaluate using a central difference scheme <sup>296</sup> ( $\Delta U = 0.1$ V):

$$\beta(U) = \frac{\partial \Delta \Omega_{\text{act}}}{e \, \partial U}, \qquad (13)$$
$$\simeq \frac{\Delta \Omega_{\text{act}}(U + \Delta U) + \Delta \Omega_{\text{act}}(U - \Delta U)}{2e\Delta U}. \qquad (14)$$

### <sup>297</sup> Results and Discussion

In the following paragraphs we present our results for Volmer, Tafel, and Heyrovský energy barriers on Pt(111) and Pt(110) at U = 0 V on the SHE scale.

#### $_{302}$ H<sub>2</sub> evolution on Pt(111)

#### 303 Convergence w.r.t. slab separation

The activation energies calculated with VASP-304 sol (following the procedure outlined in the pre-305 vious Section) are found to converge only slowly 306 with respect to the distance  $L_z$  between period-307 ically repeated slabs. More precisely, the acti-308 vation energies are a linear function of  $1/L_z$ , 309 as shown in Figure 2. The slopes of the fit-310 ted regression lines appear to be anti-correlated 311 to the difference in  $N_e$  between the initial and 312 transition states. The slope is namely close to 313 zero for the Tafel reactions (for which  $\Delta N_e \simeq$ 314  $-0.05 \text{ e}^{-}$ ) and negative for the Volmer and Hey-315 rovský reactions (where  $\Delta N_e \simeq 0.6 \text{ e}^-$ ). Impor-316 tantly, however, the calculated electrode poten-317 tial is found to remain constant upon variation 318 of  $L_z$ , with deviations of at most 30 mV. 319



Figure 2: Activation energies calculated at 0  $V_{SHE}$  using VASPsol for HER reactions on Pt(111) using Eigen ions as a proton model for the PCET reactions. The values are plotted as function of the inverse of the cell length perpendicular to the plane of the slab. The three types of barriers have been evaluated at two different hydrogen coverages in the reactant state, i.e. at/below 1 ML and above 1 ML.

The  $1/L_z$  dependence suggests the presence of 320 spurious charge interactions in the conventional 321 3D-periodic implementation in VASPsol. This 322 behavior is possibly connected to the observa-323 tion that standard Poisson-Boltzmann models 324 do not necessarily enforce strict charge neutral-325 ity.<sup>54</sup> We therefore performed single-point cal-326 culations using the JDFTx code which imple-327 ments a truncated Coulomb scheme to fully de-328 couple periodically repeated slabs.<sup>55</sup> As shown 329 in Table 1, the JDFTx activation energies in-330 deed agree well with the VASPsol results ex-331 trapolated to  $L_z = \infty$ . The small differences 332 that remain can be attributed to the different 333 treatments of core-valence interactions and ki-334 netic energy cutoffs. 335

It should furthermore be noted that the  $_{336}$  $L_z$ -dependence seems not to have been taken  $_{337}$ into account in several previous studies  $^{24,25,27}$   $_{338}$ 

Table 1: Comparison of VASPsol activation energies (obtained at 0 V<sub>SHE</sub> and extrapolated to  $L_z = \infty$ ) and JDFTx single-point calculations performed at the VASPsol geometries using a Coulomb truncation scheme. Each barrier has been evaluated at two different initial hydrogen coverages  $\theta_{\rm H}$  as well as two different water structures (a water cluster corresponding to the 'Eigen' cation as well as the (protonated) ice bilayer ('IBL') structure).

		$\Delta\Omega_{\rm act}~({\rm eV})$			
	$ heta_{ m H}$	VASPsol	JDFTx	VASPsol	JDFTx
Reaction	(ML)	(Eigen)	(Eigen)	(IBL)	(IBL)
Volmer	0.92	0.477	0.479	0.692	0.693
	1.00	0.655	0.629	0.805	0.838
Tafel	1.00	0.865	1.007	0.807	0.830
	1.08	0.545	0.407	0.420	0.416
Heyrovský	1.00	1.408	1.364	1.618	1.625
	1.08	1.110	0.995	1.393	1.388

of PCET reactions using the VASPsol code. 339 Without extrapolation to  $L_z = \infty$ , values of 340  $L_z$  of over 400 Å would be have been required 341 in the present work to converge the Volmer and 342 Heyrovský barriers on Pt(111) within 0.1 eV. 343 Moreover, without extrapolation one may even 344 obtain negative activation energies if too low  $L_z$ 345 values are employed, as shown in Figure 2). 346

#### 347 Water model

Figure 3 furthermore shows the barrier heights 348 when an ice bilayer is used for the water struc-349 ture (red bars), compared to the results with an 350 Eigen ion (blue bars).<sup>2</sup> The two sets are in fair 351 agreement, though it can be noticed that the 352 barriers for the PCET reactions (i.e. Volmer 353 and Heyrovský steps) are consistently increased 354 by about 0.2 eV when employing the ice bilayer 355 model. We attribute this difference, in part, 356 to the relative rigidity of the bilayer structure, 357 leading to a more pronounced loss (compared to 358 the more 'flexible' water cluster model) of hy-359 drogen bonds in the transition state than in the 360 initial state. The changes in H-bond distances 361 are reported in Table 2 for the Volmer reac-362 tion (the corresponding transition state struc-363 tures are shown in Figure 1). For the water 364 cluster, the coordination of the central  $H_3O^+$ 365

ion is qualitatively similar in the initial and 366 transition states: 2 H-bonds are donated, and 367 1 H-bond is accepted. Using the ice bilayer, 368 however, the  $H_3O^+$  ion at the saddle point is 369 significantly less well solvated compared to the 370 initial state: two fewer H-bonds are donated, 371 and only one more (stretched) H-bond is ac-372 cepted. Another difference between the two 373 water models is that the next-nearest neigh-374 bor water molecules are described explicitly in 375 the bilayer model, and are substituted by im-376 plicit solvent in the cluster model, which may 377 also contribute to the difference in activation 378 energies. Though the structural properties of 379 the water/Pt(111) interface remains an active 380 field of research <sup>56,57</sup>, several theoretical and ex-381 perimental works have provided support for ice 382 bilayer formation on Pt(111).<sup>58–61</sup> An interest-383 ing yet unanswered question, in this regard, is 384 how fast or slow the dielectric properties of the 385 electrolyte approach that of the bulk in case 386 an ice-like (bi)layer is formed at the surface. 387 The choice of the bulk dielectric constant in the 388 present implicit solvent calculations presumes 389 that the convergence rate is similar to that at 390 a liquid water/metal interface. 391

This suggests that the energy barrier obtained 392 with the bilayer structure would be more accurate than with the cluster model. As will be 394 discussed further below, however, the Volmer 395 and Heyrovský steps are not kinetically relevant 396

<sup>&</sup>lt;sup>2</sup>Note that the Tafel barriers reported in the 'water cluster' approach are calculated without explicit water molecules.



Figure 3: Activation energies for HER reactions on Pt(111) calculated at 0 V<sub>SHE</sub> using the implicit solvation scheme (VASPsol) at different initial hydrogen coverages and with two different water structures. The values obtained via the extrapolation approach are also indicated (see Ref. 6)

for HER on Pt(111) at 0  $V_{SHE}$ , and therefore the relative accuracy of the two water models cannot be compared in this context.

Table 2: Hydrogen-bond distances in Å for the  $H_3O^+$  ion at the initial state (IS) and transition state (TS) for the Volmer step on Pt(111) with an initial H coverage of 0.92 ML at 0  $V_{SHE}$ .

		Acce-
Water structure	Donor	$\operatorname{ptor}$
Water cluster (IS)	1.39, 1.39	2.09
Water cluster (TS)	1.75, 1.80	1.94
Ice bilayer (IS)	1.45, 1.59, 1.58	-
Ice bilayer (TS)	1.70	2.46

#### 400 Comparison with the extrapolation ap-401 proach

<sup>402</sup> The calculated barrier heights are now com-<sup>403</sup> pared with literature results obtained with the

extrapolation at the same electrode potential, 404 hydrogen coverage, and water model (the ice bi-405 layer structure, see Ref. 6) as also shown in Fig-406 ure 3.<sup>3</sup> The agreement is remarkably good, con-407 sidering the pronounced differences in method-408 ology. Both approaches therefore agree that hy-409 drogen is evolved via a Volmer-Tafel mechanism 410 with the Tafel step as the RDS at 0 V w.r.t. 411 SHE. 412

Additionally, Figure 3 shows that with the 413 implicit solvation approach the barriers change 414 as the coverage is increased beyond 1 ML, in 415 ways which are similar to the extrapolation ap-416 proach.<sup>6</sup> Increasing the coverage beyond 1 ML 417 requires the occupation of the energetically less 418 favorable atop sites. Following a Bell-Evans-419 Polanyi principle, the activation energy for the 420 Volmer step increases, whereas those of the 421 Tafel and Heyrovský steps decrease. 422

Lastly, the symmetry factors  $\beta$  calculated at 423  $0 V_{SHE}$  with the ice bilayer model amount to 424 0.62 (Volmer,  $\theta_{\rm H} = 0.92$  ML), -0.04 (Tafel,  $\theta_{\rm H}$ 425 = 1 ML), and 0.76 (Heyrovský,  $\theta_{\rm H} = 1$  ML). 426 The symmetry factors are mainly determined 427 by the amounts of charge transfer from the ini-428 tial state to the saddle point, which are calcu-429 lated to be 0.58, -0.07 and 0.62 electrons, re-430 spectively. The activation energies of the two 431 PCET reactions will therefore indeed decrease 432 as the applied potential is lowered, while the 433 barrier for the non-electrochemical Tafel step 434 is nearly potential-independent. Skúlason and 435 coworkers<sup>6</sup> obtained values of 0.44,  $\sim 0$ , and 436 1.07, respectively, using the extrapolation ap-437 proach. These values are, however, not directly 438 comparable to ours, as these also include the 439 potential dependence of the hydrogen coverage 440 and of the transition state geometry and are 441 obtained by linear regression over a fairly wide 442 potential range of 1 V or more. 443

<sup>&</sup>lt;sup>3</sup>In Ref. 6 the Heyrovský barrier at an initial H coverage of 1 ML is given at -0.2  $V_{\rm SHE}$ . We therefore used a typical symmetry factor of 0.5 to estimate the corresponding value at 0  $V_{\rm SHE}$ .

# 444 Comparison with experimental measure-445 ments

Experimental measurements of the HER ki-446 netics on Pt(111) surfaces are available in the 447 works by Marković et al.<sup>35</sup> and He et al.<sup>62</sup> Al-448 though both groups measure similar current 449 densities at 0 V, different temperature depen-450 dencies are reported. We follow the explana-451 tion offered in Ref. 62 that the Pt(111) sub-452 strate used in Ref. 35 may have contained low 453 concentrations of highly active defect sites. In 454 this view, the higher apparent activation en-455 ergy measured by He et al., 0.67 eV, should 456 be closer to that of a pristine Pt(111) sur-457 face. The magnitude of the corresponding pre-458 exponential factor (circa  $10^{10} \text{ mA/cm}^2$ ) is char-459 acteristic of a process involving only surface 460 adsorbates, supporting the Tafel reaction as 461 the RDS. These findings therefore qualitatively 462 agree with the computational results described 463 in the previous paragraphs, where the same 464 Tafel RDS is found, though with a somewhat 465 higher activation energy (0.80-0.85 eV). As the 466 barrier for the Tafel step is not sensitive to the 467 water structure at the interface, this does not, 468 unfortunately, allow to discriminate between 469 different types of water models. 470

# 471 Comparison with other implicit solvent472 calculations

Fang et al. have previously  $addressed^{31}$  the 473 HER on Pt(111) using a similar method im-474 plemented in the SIESTA code.<sup>32,33</sup> At 0 V<sub>SHE</sub> 475 and a hydrogen coverage at or below 1 ML, the 476 Tafel barrier reported by Fang et al. (0.92 eV)477 agrees well with our calculations, while the re-478 ported Volmer and Heyrovský barrier heights 479 are significantly lower than ours (< 0.2 eV and 480 0.93 eV, respectively). In attempting to locate 481 the origins of this difference for the Volmer re-482 action, we find that the inclusion of zero-point 483 vibrational energy corrections lowers the calcu-484 lated barrier by 0.10 eV, while applying the 485 PBE functional only decreases the barrier by 486 0.03 eV compared to RPBE. The use of a water 487 trimer cluster with the PBE functional, how-488 ever, increases the barrier by 0.16 eV compared 489 to the tetramer. This suggests that the influ-490

ence of e.g. the different basis set types and 491 implicit solvent models needs to be investigated 492 to elucidate the remaining discrepancies. 493

An alternative constant-potential implicit sol-494 vation approach has been recently proposed<sup>34</sup> 495 where essentially the countercharge is not de-496 scribed by a Poisson-Boltzmann equation, but 497 is instead homogeneously spread out over a part 498 of the bulk solvent region. Using this 'sol-499 vated jellium method' (SJM), the barrier for 500 the Volmer step on Pt(111) is found to be sig-501 nificantly lower (circa 0.13 eV at 0  $V_{\rm SHE}$  using 502 the ice bilayer model and the PBE functional). 503 For this water structure, we do observe a larger 504 difference between RPBE and PBE, with acti-505 vation energies of respectively 0.69 and 0.50 eV 506 according to our calculations. Further investi-507 gation will be needed to explain the remaining 508 difference, e.g. whether or not it is due to the 509 simplified description of the countercharge dis-510 tribution. Describing this distribution using a 511 jellium slab close to the electrode (with a width 512 equal to the Debye length) would be expected 513 to be appropriate only in the limit of very high 514 ionic strengths. 515

## H<sub>2</sub> evolution on Pt(110)-(1 $\times$ 2) 516

We now turn to the missing-row reconstructed 517 Pt(110) surface, with a hydrogen coverage of 1 518 ML where all ridge and (micro-)facet sites are 519 occupied (see Panel (b) in Figure 4). Although 520 there is a weak thermodynamic preference for 521 the adsorption of additional hydrogen atoms in 522 the trough sites at  $U_{\rm SHE} = 0$  V (with differ-523 ential binding energies of circa 0.1 eV, <sup>6,63,64</sup> it 524 will be argued below that trough-adsorbed hy-525 drogen atoms, if present, would not contribute 526 significantly to the hydrogen evolution rate at 527 0 V. 528

Panel (a) in Figure 4 shows the calculated 529 barrier heights for several relevant Volmer and 530 Tafel steps at and below 1 ML coverage of hy-531 drogen. Various attempts at locating saddle 532 points for the Heyrovský reactions converged 533 to Volmer-like saddle points, suggesting that 534 these reactions are inoperable on this surface. 535 On both types of sites, the Tafel reaction pro-536 ceeds via an intermediate corresponding to a 537



Figure 4: Panel (a): barrier heights calculated at 0 V<sub>SHE</sub> for the Volmer and Tafel processes on the Pt(110)-(1×2) surface. Panel (b): structural model of the transition state at 0 V<sub>SHE</sub> for the Volmer reaction on a (micro-)facet.

Kubas complex,<sup>65</sup> consisting of a stretched hy-538 drogen molecule on top of a platinum atom (see 539 Figure 5). Desorption of  $H_2$  via the Tafel re-540 action is faster on the facet (0.75 eV) com-541 pared to the ridge (0.84 eV). This is in line 542 with the higher binding energy of the ridge-543 bound adatoms. These results are comparable 544 to calculations in the literature using the same 545 functional and hydrogen coverage, but without 546 solvent and at the potential of zero charge.<sup>6,64</sup> 547



Figure 5: Structural models of the Kubas complex as an intermediate of the Tafel reaction at 0 V<sub>SHE</sub> on ridges (a) and microfacets (b) of the Pt(110)-(1×2) surface.

Similarly, at an initial hydrogen coverage of 548 0.89 ML, we find a higher barrier for the Volmer 549 step on the facets (0.94 eV) compared to the 550 ridge (0.33 eV). This change exceeds the differ-551 ence in hydrogen binding energy between the 552 two site types (calculated to be only 0.23 eV). 553 Hence, even though hydrogen desorption will 554 take place on the facets, replenishment of the 555



Figure 6: Calculated energy diagram for HER on Pt(110)- $(1\times2)$  at at 0 V<sub>SHE</sub> for a steady-state hydrogen coverage of 1 ML. The reaction energies have been evaluated using the CHE formalism.

coverage happens most rapidly via diffusion of 556 H atoms from the ridge to the vacant facet sites 557 followed by Volmer discharge on the now-vacant 558 ridge site. This process is illustrated in the en-559 ergy diagram in Figure 6. As the diffusion bar-560 rier is low (0.3 eV), the Volmer step will dic-561 tate the apparent activation energy, which then 562 amounts to 0.56 eV. This entails that hydrogen 563 deposition is fast relative to the Tafel reaction, 564 making the latter the RDS. This is in quali-565 tative agreement with the measurements on a 566 Pt(110) single-crystal by Marković and cowork-567 ers,<sup>35</sup> where the same RDS is found. The re-568 ported activation energy (0.1 eV) is much lower 569 than suggested by our calculations (0.75 eV), 570 but is likely to be an underestimate due to a 571 low concentration of highly active defect sites, 572 as has been argued in the case of Pt(111).<sup>62</sup> The 573 relative activities of the (111) and (110)- $(1 \times 2)$ 574 surfaces are, however, in good agreement, as the 575 apparent activation energy is circa 0.1 eV lower 576 on Pt(110)-(1×2) than on Pt(111) according to 577 the experiments as well as the present calcula-578 tions. 579

We now return to the question whether hydrogen coverages beyond 1 ML (with occupations of trough sites) could be relevant at 582

U = 0 V<sub>SHE</sub>. Firstly, the relatively high bar-583 rier obtained for the Volmer reaction on the 584 facet suggests that the corresponding barrier on 585 the trough will be even higher, requiring a sim-586 ilar diffusion-assisted mechanism as described 587 above. However, moving a ridge-bound H atom 588 to a trough position requires more energy (0.46)589 eV) than to a facet site. Achieving a signif-590 icant occupation of the trough sites is, there-591 fore, kinetically difficult. Additionally, even if a 592 partial occupation of the trough sites would be 593 reached, the Tafel reaction on the facets would 594 remain the main desorption pathway for hydro-595 gen. Previous studies have namely found that, 596 despite the lower binding energy, the barrier for 597 hydrogen desorption from trough sites is larger 598 compared to the facet sites when the hydro-599 gen coverage is at or slightly beyond 1 ML.<sup>63,64</sup> 600 It is therefore sufficient to only consider ridge-601 and facet-bound hydrogen atoms for HER on 602  $Pt(110)-(1\times 2)$  at 0 V<sub>SHE</sub>. 603

# 604 Conclusions

We have evaluated the performance of a re-605 cently developed constant-potential implicit 606 solvation approach to electrochemical barriers 607 for one of the classical processes in catalysis, the 608 hydrogen evolution reaction on platinum. The 609 relevant Volmer, Tafel, and Heyrovský kinet-610 ics on Pt(111) and Pt(110) at 0 V versus SHE 611 compare well with the previously developed 'ex-612 trapolation' approach and with available exper-613 imental data. Importantly, the inclusion of an 614 implicit solvent only moderately increases the 615 computational cost, whereas calculating barrier 616 heights with the extrapolation approach is more 617 expensive by at least one order of magnitude. 618 Care should be taken, however, in dealing with 619 spurious electrostatic interactions between peri-620 odically repeated slabs when using the VASPsol 621 code. 622

Although further testing for other types of reactions and materials is warranted, we have so far found the accuracy to be satisfactory and expect implicit solvation approaches to receive increasing attention in future research. The application and continued development of costeffective methods is most welcome, as the previous lack of such approaches has frequently 630 led to kinetic aspects being disregarded in firstprinciples electrocatalysis research. 632

Acknowledgement The authors thank Vitae Industries for providing computational resources, and Javed Hussain for valuable discussions. This material is based upon work supported by the National Science Foundation under Grant No. CHE-1665372.

# Supporting Information Avail<sup>533</sup> able <sup>634</sup>

Coordinate files pertaining to the hydrogen evolution reactions on Pt(111) and Pt(110). 636

637

# References

- Nørskov, J. K.; Rossmeisl, J.; Logadottir, A.; Lindqvist, L.; Kitchin, J. R.; Bligaard, T.; Jónsson, H. Origin of the Overpotential for Oxygen Reduction at a Fuel-Cell Cathode. J. Phys. Chem. B 2004, 642 108, 17886–17892. 643
- (2) Nørskov, J. K.; Bligaard, T.; Logadottir, A.; Kitchin, J. R.; Chen, J. G.; Pandelov, S.; Stimming, U. Trends in the Exchange Current for Hydrogen Evolution. *J. Electrochem. Soc.* 2005, 152, J23–J26. 648
- (3) Skúlason, E.; Jónsson, H. Atomic Scale 649 Simulations of Heterogeneous Electro-650 catalysis: Recent Advances. Adv. Phys. X 651 2017, 2, 481–495. 652
- (4) Exner, K. S.; Over, H. Kinetics of Electrocatalytic Reactions from First-Principles: 654 A Critical Comparison with the Ab Initio Thermodynamics Approach. Acc. Chem. 656 Res. 2017, 50, 1240–1247. 657
- (5) Rossmeisl, J.: Skúlason. E.; 658 Björketun, М. E.; Tripkovic, V.; 659 Nørskov, J. K. Modeling the Electri-660 fied Solid-Liquid Interface. J. Chem. 661 Phys. Lett. 2008, 466, 68–71. 662

(6) Skúlason, E.; Tripkovic, V.; 663 Björketun, M. E.; Gudmundsdóttir, S.; 664 Karlberg, G.; Rossmeisl, J.; Bligaard, T.; 665 Jónsson, H.; Nørskov, J. K. Modeling the 666 Electrochemical Hydrogen Oxidation and 667 Evolution Reactions on the Basis of Den-668 sity Functional Theory Calculations. J. 669 *Phys. Chem. C* **2010**, *114*, 18182–18197. 670

- (7) Skúlason, E.; S. Karlberg, G.; Ross-671 meisl, J.; Bligaard, T.; Greeley, J.; 672 Jónsson, H.; K. Nørskov, J. Density Func-673 tional Theory Calculations for the Hy-674 drogen Evolution Reaction in an Elec-675 trochemical Double Layer on the Pt(111)676 Electrode. Phys. Chem. Chem. Phys. 677 **2007**, *9*, 3241–3250. 678
- (8) Hussain, J.; Jónsson, H.; Skúlason, E.
  Faraday Efficiency and Mechanism of
  Electrochemical Surface Reactions: CO<sub>2</sub>
  Reduction and H<sub>2</sub> Formation on Pt(111). *Faraday Discuss.* 2016, 195, 619–636.
- (9) Hussain, J.; Jónsson, H.; Skúlason, E. Calculations of Product Selectivity in Electrochemical CO<sub>2</sub> Reduction. ACS Catal.
  2018, 5240–5249.
- (10) Chan, K.; Nørskov, J. K. Electrochemi cal Barriers Made Simple. J. Phys. Chem.
   Lett. 2015, 6, 2663–2668.
- (11) Chan, K.; Nørskov, J. K. Potential Dependence of Electrochemical Barriers from ab Initio Calculations. J. Phys. Chem. Lett.
  2016, 7, 1686–1690.
- (12) Filhol, J.-S.; Neurock, M. Elucidation of
  the Electrochemical Activation of Water
  over Pd by First Principles. Angew. Chem. *Int. Ed.* 2006, 45, 402–406.
- (13) Taylor, C. D.; Wasileski, S. A.; Filhol, J.S.; Neurock, M. First Principles Reaction Modeling of the Electrochemical Interface: Consideration and Calculation of a Tunable Surface Potential from Atomic and Electronic Structure. *Phys. Rev. B* 2006, 73, 165402.

- (14) Makov, G.; Payne, M. C. Periodic Boundary Conditions in Ab Initio Calculations. 707 *Phys. Rev. B* 1995, *51*, 4014–4022. 708
- (15) Komsa, H.-P.: Rantala, Τ. T.: 709 Pasquarello, А. Finite-Size Supercell 710 Correction Schemes for Charged Defect 711 Calculations. Phys. Rev. B 2012, 86, 712 045112. 713
- (16) Otani, M.; Hamada, I.; Sugino, O.; 714 Morikawa, Y.; Okamoto, Y.; Ikeshoji, T. 715 Structure of the Water/Platinum Interface – a First Principles Simulation Under 717 Bias Potential. *Phys. Chem. Chem. Phys.* 718 2008, 10, 3609–3612. 719
- (17) Surendralal, S.; Todorova, M.; Fin-720 nis, M. W.; Neugebauer, J. First-721 Principles Approach to Model Electro-722 chemical Reactions: Understanding the 723 Fundamental Mechanisms behind Mg 724 Corrosion. Phys. Rev. Lett. 2018, 120, 725 246801. 726
- (18) Akhade, S. A.; Bernstein, Ν. J.; 727 Esopi, M. R.; Regula, M. J.; Janik, M. J. 728 Method to Approximate Simple А 729 Electrode Potential-Dependent Activa-730 tion Energies Using Density Functional 731 Theory. Catal. Today 2017, 288, 63–73. 732
- (19) Akhade, S. A.; Nidzyn, R. M.; Ros-733 tamikia, G.; Janik, M. J. Using Brønsted-734 Evans-Polanyi Relations to Predict Elec-735 trode Potential-Dependent Activation Energies. *Catal. Today* 2018, *312*, 82–91.737
- (20) Sundararaman, R.; Goddard, W. A.; 738
  Arias, T. A. Grand Canonical Electronic 739
  Density-Functional Theory: Algorithms 740
  and Applications to Electrochemistry. J. 741
  Chem. Phys. 2017, 146, 114104. 742
- (21) Sundararaman, R.; Letchworth-Weaver, K.; Schwarz, K. A.; Gunceler, D.; 744
  Ozhabes, Y.; Arias, T. A. JDFTx: Software for Joint Density-Functional Theory. 746
  SoftwareX 2017, 6, 278-284. 747
- (22) Mathew, K.; Sundararaman, R.; 748 Letchworth-Weaver, K.; Arias, T. A.; 749

Hennig, R. G. Implicit Solvation Modelfor Density-Functional Study of Nanocrys-

for Density-Functional Study of Nanocrys tal Surfaces and Reaction Pathways. J.

- 753 Chem. Phys. **2014**, 140, 084106.
- Mathew, K.; Hennig, R. G. Implicit
  Self-Consistent Description of Electrolyte
  in Plane-Wave Density-Functional Theory. arXiv:1601.03346 [cond-mat] 2016, arXiv: 1601.03346.
- (24) Goodpaster, J. D.; Bell, A. T.; Head-Gordon, M. Identification of Possible
  Pathways for C-C Bond Formation during
  Electrochemical Reduction of CO<sub>2</sub>: New
  Theoretical Insights from an Improved
  Electrochemical Model. J. Phys. Chem.
  Lett. 2016, 7, 1471–1477.
- (25) Singh, M. R.; Goodpaster, J. D.; Weber, A. Z.; Head-Gordon, M.; Bell, A. T.
  Mechanistic Insights into Electrochemical Reduction of CO<sub>2</sub> Over Ag Using Density Functional Theory and Transport Models. *PNAS* 2017, 114, E8812–E8821.
- (26) Garza, A. J.; Bell, A. T.; Head-Gordon, M. Is Subsurface Oxygen Necessary for the Electrochemical Reduction of CO<sub>2</sub> on Copper? *J. Phys. Chem. Lett.* **2018**, *9*, 601–606.
- 777 (27) Garza, A. J.; Bell, A. T.; Head-778 Gordon, M. Mechanism of CO2 Reduction779 at Copper Surfaces: Pathways to C2 Prod-780 ucts. ACS Catal. 2018, 8, 1490–1499.
- (28) Schwarz, K. A.; Sundararaman, R.; Moffat, T. P.; Allison, T. C. Formic Acid Oxidation on Platinum: A Simple Mechanistic Study. *Phys. Chem. Chem. Phys.* **2015**, *17*, 20805–20813.
- (29) Wang, H.-F.; Liu, Z.-P. Formic Acid Oxidation at Pt/H<sub>2</sub>O Interface from Periodic DFT Calculations Integrated with a Continuum Solvation Model. J. Phys. Chem. C 2009, 113, 17502–17508.
- (30) Fang, Y.; Liu, Z. Electrochemical Reactions at the Electrode/Solution Interface: Theory and Applications to Water

Electrolysis and Oxygen Reduction. *Sci.* 794 *China Chem.* **2010**, *53*, 543–552. 795

- (31) Fang, Y.-H.; Wei, G.-F.; Liu, Z.-P. Catalytic Role of Minority Species and Minority Sites for Electrochemical Hydrogen Evolution on Metals: Surface Charging, Coverage, and Tafel Kinetics. J. Phys. Chem. C 2013, 117, 7669–7680.
- (32) Fang, Y.-H.; Wei, G.-F.; Liu, Z.-P. Theoretical Modeling of Electrode/Electrolyte 803
  Interface from First-Principles Periodic 804
  Continuum Solvation Method. Catal. Toreta 2013, 202, 98–104. 805
- (33) Fang, Y.-H.; Liu, Z.-P. Tafel Kinetics <sup>807</sup> of Electrocatalytic Reactions: From Experiment to First-Principles. ACS Catal. <sup>809</sup> 2014, 4, 4364–4376.
- (34) Kastlunger, G.; Lindgren, P.; Peter-<sup>811</sup> son, A. A. Controlled-Potential Simula-<sup>812</sup> tion of Elementary Electrochemical Re-<sup>813</sup> actions: Proton Discharge on Metal Sur-<sup>814</sup> faces. J. Phys. Chem. C 2018,<sup>815</sup>
- (35) Marković, N. M.; Grgur, B. N.; Ross, P. N.
  Temperature-Dependent Hydrogen Electrochemistry on Platinum Low-Index
  Single-Crystal Surfaces in Acid Solutions.
  J. Phys. Chem. B 1997, 101, 5405-5413.
- (36) Kresse, G.; Hafner, J. Ab Initio MolecularDynamics Simulation of the Liquid-MetalAmorphous-Semiconductor Transition in
  Germanium. Phys. Rev. B 1994, 49, 824
  14251–14269.
- (37) Kresse, G.; Furthmüller, J. Efficiency 826
  of Ab-Initio Total Energy Calculations 827
  for Metals and Semiconductors Using a 828
  Plane-Wave Basis Set. Comp. Mater. Sci. 829
  1996, 6, 15–50. 830
- (38) Kresse, G.; Furthmüller, J. Efficient Iterative Schemes for Ab Initio Total-Energy Calculations Using a Plane-Wave Basis
  Set. Phys. Rev. B 1996, 54, 11169–11186.
- (39) Kresse, G.; Joubert, D. From Ultrasoft Pseudopotentials to the Projector 836

Augmented-Wave Method. *Phys. Rev. B* **1999**, *59*, 1758–1775.

- (40) Hohenberg, P.; Kohn, W. Inhomogeneous
  Electron Gas. *Phys. Rev.* 1964, 136,
  B864–B871.
- Kohn, W.; Sham, L. J. Self-Consistent
  Equations Including Exchange and Correlation Effects. *Phys. Rev.* 1965, 140,
  A1133–A1138.
- (42) Garrity, K. F.; Bennett, J. W.; 846 Rabe, K. M.: Vanderbilt, D. Pseu-847 dopotentials for High-Throughput DFT 848 Calculations. Comp. Mater. Sci. 2014, 849 81, 446-452. 850
- L. (43) Hammer, B.; B.; Hansen, 851 Nørskov, J. K. Improved Adsorption 852 Energetics Within Density-Functional 853 Theory Using Revised Perdew-Burke-854 Ernzerhof Functionals. Phys. Rev. B 855 **1999**, *59*, 7413–7421. 856
- <sup>857</sup> (44) Perdew, J. P.; Burke, K.; Ernzerhof, M. Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **1996**, 77, 3865–3868.
- (45) Marques, M. A. L.; Oliveira, M. J. T.;
  Burnus, T. Libxc: A Library of Exchange
  and Correlation Functionals for Density Functional Theory. *Computer Physics Communications* 2012, 183, 2272–2281.
- (46) Monkhorst, H. J.; Pack, J. D. Special Points for Brillouin-Zone Integrations.
   *Phys. Rev. B* 1976, *13*, 5188–5192.
- <sup>869</sup> (47) Pack, J. D.; Monkhorst, H. J. 'Special
  Points for Brillouin-Zone Integrations' a
  Reply. *Phys. Rev. B* **1977**, *16*, 1748–1749.
- (48) Henkelman, G.; Jónsson, H. A Dimer Method for Finding Saddle Points on High Dimensional Potential Surfaces Using Only First Derivatives. J. Chem. Phys. 1999, 111, 7010–7022.
- (49) Olsen, R. A.; Kroes, G. J.; Henkelman, G.;
  Arnaldsson, A.; Jónsson, H. Comparison

of Methods for Finding Saddle Points 879 Without Knowledge of the Final States. 880 J. Chem. Phys. **2004**, 121, 9776–9792. 881

- (50) Gunceler, D.; Letchworth-Weaver, K.; 882
  Sundararaman, R.; Schwarz, K. A.; 883
  Arias, T. A. The Importance of Nonlinear Fluid Response in Joint Density-Functional Theory Studies of Battery Systems. *Modelling Simul. Mater. Sci. Eng.* 887
  2013, 21, 074005. 888
- (51) Andreussi, O.; Dabo, I.; Marzari, N. Revised Self-Consistent Continuum Solvation in Electronic-Structure Calculations.
  J. Chem. Phys. 2012, 136, 064102.
- (52) Murrell, J.; Jenkins, A. Properties of Liquids and Solutions, 2nd ed.; Wiley & Sons:
   Chichester, England, 1994.
- (53) Trasatti, S. The Absolute Electrode Potential: An Explanatory Note (recommendations 1986). Pure Appl. Chem. 1986, 898 58, 955–966.
- (54) Melander, M.; Kuisma, M.; Chris-900 tensen, T.; Honkala, K. Grand-Canonical 901 Approach to Density Functional Theory of 902 Electrocatalytic Systems: Thermodynam-903 ics of Solid-Liquid Interfaces at Constant 904 Ion and Electrode Potentials; ChemRxiv, 905 2018. 906
- (55) Sundararaman, R.; Arias, T. A. Regularization of the Coulomb singularity in exact exchange by Wigner-Seitz truncated interactions: Towards chemical accuracy in nontrivial systems. *Phys. Rev. B* 2013, 87, 165122.
- (56) Steinmann, S. N.; Ferreira De Morais, R.; 913 Götz, A. W.; Fleurat-Lessard, P.; Iannuzzi, M.; Sautet, P.; Michel, C. Force 915 Field for Water over Pt(111): Development, Assessment, and Comparison. J. 917 Chem. Theory Comput. 2018, 14, 3238-3251.
- (57) Le, J.; Iannuzzi, M.; Cuesta, A.; Cheng, J. 920 Determining Potentials of Zero Charge 921

- 922of Metal Electrodes versus the Stan-923dard Hydrogen Electrode from Density-924Functional-Theory-Based Molecular Dy-925namics. Phys. Rev. Lett. 2017, 119,926016801.
- 927 (58) Meng, S.; Xu, L. F.; Wang, E. G.; Gao, S.
  928 Vibrational Recognition of Hydrogen929 Bonded Water Networks on a Metal Sur930 face. *Phys. Rev. Lett.* **2002**, *89*, 176104.
- (59) Ogasawara, H.; Brena, B.; Nordlund, D.;
  Nyberg, M.; Pelmenschikov, A.; Pettersson, L. G. M.; Nilsson, A. Structure and Bonding of Water on Pt(111). *Phys. Rev. Lett.* 2002, *89*, 276102.
- (60) Schnur, S.; Groß, A. Properties of MetalWater Interfaces Studied from First Principles. New J. Phys. 2009, 11, 125003.
- (61) Groß, A.; Gossenberger, F.; Lin, X.; Naderian, M.; Sakong, S.; Roman, T. Water Structures at Metal Electrodes Studied by Ab Initio Molecular Dynamics Simulations. J. Electrochem. Soc. 2014, 161, E3015–E3020.
- (62) He, Z.-D.; Wei, J.; Chen, Y.-X.; Santos, E.; Schmickler, W. Hydrogen Evolution at Pt(111) Activation Energy, Frequency Factor and Hydrogen Repulsion. *Electrochim. Acta* 2017, 255, 391–395.
- 950 (63) Gudmundsdóttir, S.; Skúlason, E.; 951 Jónsson, H. Reentrant Mechanism for As-952 sociative Desorption:  $H_2/Pt(110)-(1\times 2)$ . 953 Phys. Rev. Lett. **2012**, 108, 156101.
- (64) Gudmundsdóttir, S.; Skúlason, E.; Weststrate, K.-J.; Juurlink, L.; Jónsson, H.
  Hydrogen Adsorption and Desorption at the Pt(110)-(1×2) Surface: Experimental and Theoretical Study. *Phys. Chem. Chem. Phys.* 2013, 15, 6323–6332.
- 960 (65) Kubas, G. J. Molecular Hydrogen Com-961 plexes: Coordination of a  $\sigma$  Bond to Tran-962 sition Metals. Acc. Chem. Res. **1988**, 21, 963 120–128.

# **TOC** Graphic

