

# Enhancing electrocatalytic CO<sub>2</sub> reduction using a system-integrated approach to catalyst discovery

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**Electrocatalytic CO<sub>2</sub> reduction has the dual-promise of neutralizing carbon emissions in the near future, while providing a long-term pathway to create energy-dense chemicals and fuels from atmospheric CO<sub>2</sub>. The field has advanced immensely in recent years, taking significant strides towards commercial realization. Catalyst innovations have played a pivotal role in these advances, with a steady stream of new catalysts providing gains in CO<sub>2</sub> conversion efficiencies and selectivities of both C1 and C2 products. Comparatively few of these catalysts have been tested at commercially-relevant current densities (~200 mA/cm<sup>2</sup>) due to transport limitations in traditional testing configurations and a research focus on fundamental catalyst kinetics, which are measured at substantially lower current densities. A catalyst's selectivity and activity, however, have been shown to be highly sensitive to the local reaction environment, which changes drastically as a function of reaction rate. As a consequence of this, the surface properties of many CO<sub>2</sub> reduction catalysts risk being optimized for the wrong operating conditions. The goal of this article is to communicate the substantial impact of reaction rate on catalytic behaviour and the CO<sub>2</sub> reduction reaction. In brief, this work motivates high current density catalyst testing as a necessary step to properly evaluate materials for electrochemical CO<sub>2</sub> reduction, and to accelerate the technology toward its envisioned application of neutralizing CO<sub>2</sub> emissions on a global scale.**

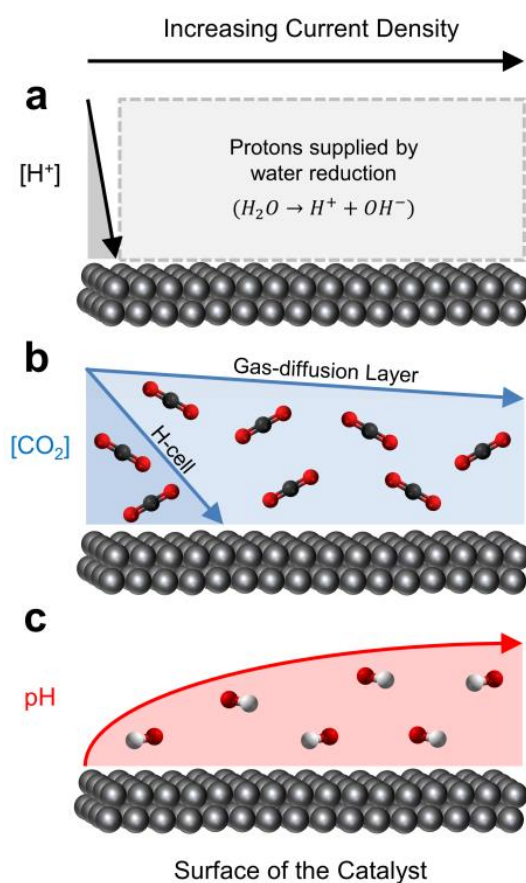
30 Under an applied potential and in the presence of an appropriate catalyst, carbon-dioxide  
31 ( $\text{CO}_2$ ) and water can be electrocatalytically converted into syngas ( $\text{CO} + \text{H}_2$ ), ethylene ( $\text{C}_2\text{H}_4$ ),  
32 methane ( $\text{CH}_4$ ), ethanol ( $\text{C}_2\text{H}_5\text{OH}$ ) and formate ( $\text{HCOOH}$ ) among other products. The  
33 collective market size of these reduction products is  $>500$  Mton/year, indicating the potential  
34 scale of a commercially competitive  $\text{CO}_2$  electrolyzer;<sup>1</sup> in the process, and of utmost  
35 importance and urgency, this process may aid in reducing fossil fuels by supplanting current  
36 production routes. Excitingly,  $\text{CO}_2$  electroreduction catalysts have shown enough promise that  
37 we are beginning to see the first steps towards commercial application of the technology,  
38 including more and more start-ups (see OPUS12, CERT, Dioxide Materials) and established  
39 companies (Siemens) focusing on system design and engineering. As a result, researchers are  
40 now targeting lower overall cell potentials by improving other parts of the conversion unit,<sup>2-5</sup>  
41 while looking to more efficiently integrate electrochemical  $\text{CO}_2$  conversion units with  
42 upstream and downstream processes.<sup>6</sup> Future devices will also need to demonstrate stable  
43 long-term operation ( $>20,000$  hours) at substantial current densities ( $>200$   $\text{mA}/\text{cm}^2$ ) to  
44 minimize the capital-expenditure of a conversion unit to economically-compelling levels.<sup>1,7,8</sup>  
45 Due to the low solubility of  $\text{CO}_2$  in aqueous-fed systems that limits  $\text{CO}_2$  conversion to current  
46 densities of  $\sim 35$   $\text{mA}/\text{cm}^2$ ,<sup>9-11</sup> researchers have turned to pressurized electrolytes and  
47 gas-diffusion layer-based systems to supply enough  $\text{CO}_2$  to the catalyst layer to sustain higher  
48 current densities. Despite these capabilities, an overwhelming percentage ( $>95\%$ )<sup>12,13</sup> of  
49 fundamental studies and catalytic materials are still developed, tested and characterized in  
50 classical H-cell configurations, where current densities are limited. The local catalytic  
51 environment, and subsequently the energetics of the reactions occurring on a catalyst's  
52 surface, are known however to be highly sensitive to changes in reaction rate. Therefore a  
53 fundamental question remains: how representative are the conclusions from aqueous-fed  
54 systems that are constrained to  $\sim 35$   $\text{mA}/\text{cm}^2$  when the goal is to achieve  $>200$   $\text{mA}/\text{cm}^2$ ?

55 This article seeks to shed light on this question by summarizing how the local reaction  
56 environment is known to vary as a function of current density, and how these changes may  
57 impact reactions occurring on a catalyst's surface when pushed to commercial current  
58 densities. To aid in the analysis we draw upon recent literature findings from electrochemical  
59 experiments, transport phenomena and Density-Functional Theory (DFT) modelling.

60 Due to the promise of electrochemical CO<sub>2</sub> reduction technology, and a lack of selective and  
61 efficient cathode materials, a large fraction of the field has undergone a global,  
62 multidisciplinary effort over the last decade to find new and better catalysts. The search is  
63 complicated by the large number of surface factors impacting activity including intermediate  
64 binding energy,<sup>14,15</sup> (via coordination<sup>16,17</sup> and site availability<sup>18–20</sup>), packing,<sup>21,22</sup> kinetic supply  
65 of reactants,<sup>23</sup> desorption of products,<sup>24</sup> adsorbate-adsorbate interactions,<sup>25</sup> etc. The urgency  
66 of these efforts is illustrated by the large number of material-centric review papers published  
67 in the last year alone on catalyst development/optimization.<sup>10,26–30</sup> Here, we define the catalyst  
68 as the surface on which CO<sub>2</sub> is reduced. As with any catalytic process, however, the local and  
69 system reaction environments play equally important roles in efficiently driving specific  
70 reactions, while suppressing unwanted competing ones. Many researchers have reported the  
71 extreme sensitivity of the reaction to changes in local pH,<sup>31–35</sup> electrolytes<sup>36,37</sup> and cations<sup>38–41</sup>  
72 (easily illustrated by replacing K<sup>+</sup> with Na<sup>+</sup>). The 'catalyst' that notably reduces the energy  
73 barrier for CO<sub>2</sub> reduction processes is then very much a combined material and environmental  
74 effect, rather than that of the catalyst's surface (composition, coordination,  
75 nano/mesostructure) alone.

76 The above distinction, while central to any catalytic process, warrants particular attention here  
77 due to the unique peculiarities of electrocatalytic CO<sub>2</sub> reduction in aqueous solvents.  
78 *Specifically, while the local environment directly influences reaction pathways and kinetics,*  
79 *the reduction reaction itself greatly disturbs the local environment.* At the root of this

80 reaction-driven sensitivity is the requirement for both CO<sub>2</sub> and protons (H<sup>+</sup>) to participate in  
 81 the CO<sub>2</sub> reduction process. The ever-present, and in many cases more thermodynamically  
 82 favourable, hydrogen evolution reaction (HER) then simultaneously competes with CO<sub>2</sub>  
 83 reduction for protons and electrons. At extremely low current densities (<1 mA/cm<sup>2</sup>), these  
 84 protons can be supplied to either reaction directly by hydronium contained within the local  
 85 electrolyte (Fig. 1a).<sup>42</sup> As hydronium is depleted, water reduction fills the role as a hydrogen  
 86 source while the unused hydroxide molecule generated as a by-product, rapidly increases the  
 87 local electrolyte pH (Fig. 1c).<sup>43,44</sup> At slightly more moderate CO<sub>2</sub> reduction current densities  
 88 in aqueous-fed systems (~35 mA/cm<sup>2</sup> for C1 products, ~100 mA/cm<sup>2</sup> for C2 products<sup>32,36</sup>),  
 89 CO<sub>2</sub> ultimately becomes depleted by a combination of diffusion limitations from the bulk  
 90 electrolyte, and the now unfavourable bicarbonate-equilibrium conditions as a result of the



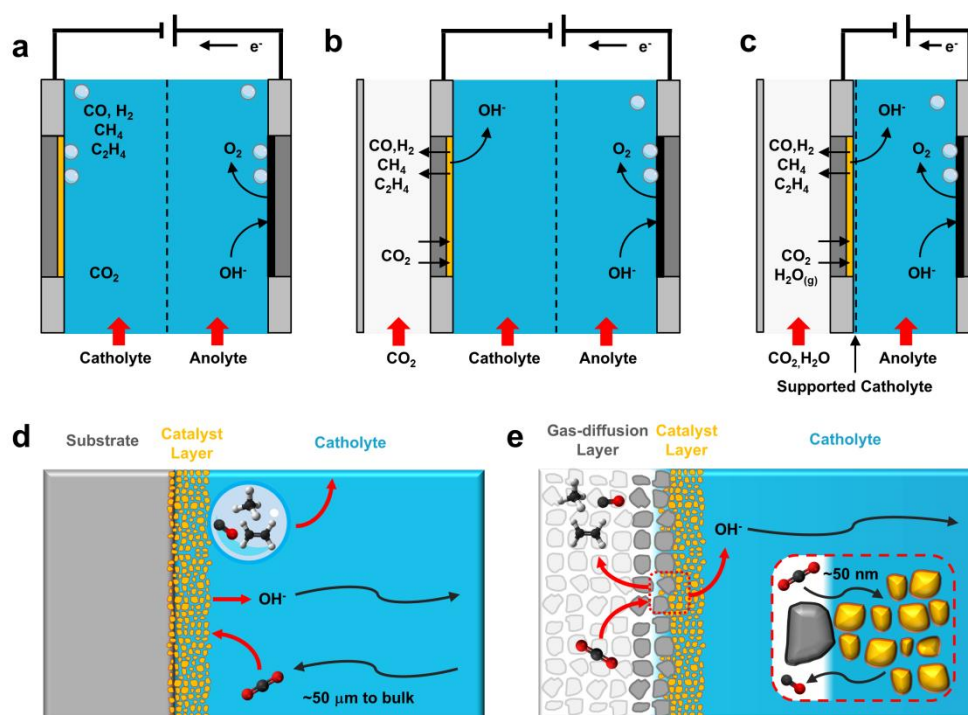
**Fig. 1:** The relative effect of current density on the reaction (a) proton source, (b) concentration of CO<sub>2</sub> and (c) pH at the surface of a CO<sub>2</sub> reduction catalyst.

91 increased local pH.<sup>45,46</sup> Thus, as the reaction proceeds from 0 mA/cm<sup>2</sup> to CO<sub>2</sub>-depletion (Fig.  
92 1b), the surface coverage and binding energies of key intermediates on a catalyst's surface are  
93 ultimately impacted through these changes in the local environment;<sup>44,47,48</sup> not dissimilar from  
94 the surface effects incurred by varying material composition, structure or morphology. The  
95 activity of a catalyst is then identified by the environment around it, as much as its physical  
96 make-up.

97 The inherent sensitivity of the reaction to changing local conditions is seen in literature to  
98 directly and indirectly drive many of the experimental choices in electrolyte type and  
99 concentration. Unsurprisingly, a catalyst can perform exceptionally well or poorly depending  
100 on the medium in which it is tested. The importance of the electrolyte medium on catalytic  
101 behaviour is most clearly displayed through the near ubiquitous use of low KHCO<sub>3</sub>  
102 concentrations in reports of high selectivity C<sub>2</sub> production on nanostructured<sup>49-51</sup> and oxide-  
103 derived Cu<sup>52-55</sup> in H-cells. Here the poor buffering capacity of the electrolyte causes the pH  
104 close to the electrode to quickly increase at low current densities (<20 mA/cm<sup>2</sup>), helping to  
105 promote C<sub>2</sub> products and suppress the competing CH<sub>4</sub> and H<sub>2</sub> reactions.<sup>31,56</sup> The morphology  
106 of the catalyst can be used to provide similar cumulative effects,<sup>52</sup> which ultimately makes  
107 properly separating the direct contributions of intrinsic catalytic activity and the local  
108 environment even more difficult. To this point, the field has collectively learned to manipulate  
109 both catalysts and local electrochemical conditions as needed to optimize CO<sub>2</sub> reduction  
110 performance metrics at primarily indiscriminate current densities. Given this widely-  
111 implemented knowledge that the environment is critical to catalytic behaviour, and that these  
112 conditions change as a function of current density, it is our perspective that the environment at  
113 commercially-relevant current densities (>200 mA/cm<sup>2</sup>) should be used as the criterion for  
114 assessing catalytic performance and suitability.

115

117 The number of CO<sub>2</sub> electroreduction experiments reported at elevated current densities is  
118 rapidly increasing, owed in part to the maturity of the field as well as the observed  
119 performance benefits. Recent work has reviewed various electrochemical architectures  
120 capable of delivering enough CO<sub>2</sub> to the catalyst to reach current densities >200 mA/cm<sup>2</sup>,<sup>12,13</sup>  
121 summarizing in detail the impacts of different components and configurations on system  
122 performance. While we refer the reader to these publications for specific technical advances  
123 in the field, we will briefly summarize several commonly-used architectures (Fig. 2) and  
124 aspects of their operation to illustrate how the reaction environment around the catalyst layer  
125 is influenced by elevated current densities.



**Fig. 2:** Comparative mass transport phenomena in commonly-used electrochemical CO<sub>2</sub> reduction configurations. (a) Cell view for an H-cell configuration with a catalyst deposited on a solid substrate, (b) Cell view for a catalyst deposited on a gas-diffusion layer with a flowing catholyte channel, (c) Cell view for a catalyst deposited on a gas-diffusion layer with a non-flowing catholyte, (d) species transport to and from a catalyst layer in which CO<sub>2</sub> is supplied via diffusion from the bulk electrolyte on the microscale (~50 μm). (e) a CO<sub>2</sub> reduction catalyst layer deposited onto a hydrophobic substrate with CO<sub>2</sub> diffusion from a nearby gas-liquid interface (~50 nm). Liquid species diffuse to the ion-exchange membrane through either a bulk flowing electrolyte or a solid-supported electrolyte layer.

126 In each of the described configurations CO<sub>2</sub> is supplied to a catalyst layer that is fully or  
127 partially immersed in a conductive electrolyte. This CO<sub>2</sub> can diffuse through the  
128 hydrodynamic boundary layer of a saturated bulk electrolyte as in a standard H-cell  
129 configuration (Fig. 2a and d),<sup>45,46,57</sup> or from a nearby gas-phase with a much shorter diffusion  
130 pathway (Fig. 2b, c and e).<sup>58-61</sup> Configurations where CO<sub>2</sub> is provided from the gas-phase use  
131 a gas-diffusion layer to form a gas-liquid interface adjacent to the catalyst layer.  
132 Electrochemical CO<sub>2</sub> reactors using gas-diffusion layers have historically been used in diverse  
133 configurations,<sup>13,62-64</sup> but range primarily from a Kenis-type reactor with a flowing  
134 catholyte<sup>58,65-73</sup> (Fig. 2b), to membrane electrode assemblies<sup>2,5,12,74,75</sup> which directly combine  
135 a gas-diffusion layer, catalyst and ion exchange membrane into one unit (Fig. 2c). These latter  
136 configurations are occasionally denoted as ‘catholyte-free’ or gas-phase electrolysis due to the  
137 lack of a flowing electrolyte between the catalyst layer and ion-exchange membrane. Liquid  
138 water, however, is reported to be present in the porous catalyst layer in the majority of cases.  
139 Further, research has shown that without the direct presence of a solid-supported electrolyte,  
140 CO<sub>2</sub> reduction selectivity can be heavily penalized.<sup>2,5,64</sup> The two cathode variations shown in  
141 Fig. 2b and c then similarly involve the diffusion of CO<sub>2</sub> across a gas-liquid interface and  
142 through a thin electrolyte to a porous catalyst layer (Fig. 2e), where evidence suggests that the  
143 reaction occurs primarily in the aqueous phase rather than at a three-phase solid-liquid-gas  
144 interface. Water management is essential to maintain a stable gas-liquid equilibrium as both  
145 flooding of the gas-diffusion layer and evaporation of the catholyte will impact CO<sub>2</sub> transport  
146 to the catalyst layer.<sup>5,70,76</sup> A catalyst can be applied to the gas-diffusion layer via drop-casting,  
147 airbrushing, electrodeposition, compression, or incorporation into the porous layer itself.

148 The primary difference between these architectures is a roughly 3-order of magnitude  
149 reduction in the CO<sub>2</sub> diffusion pathway to the surface of the catalyst: from ~50 μm in an H-  
150 cell to ~50 nm using a gas-diffusion layer (Fig. 2b,c), which allows for the increased

151 maximum current densities reported in literature. Using a simple 1D reaction-diffusion model  
152 developed previously,<sup>45,46,57-59</sup> we can also approximate the similarities and differences in the  
153 local catalytic environments as a function of reaction rate for both cases.

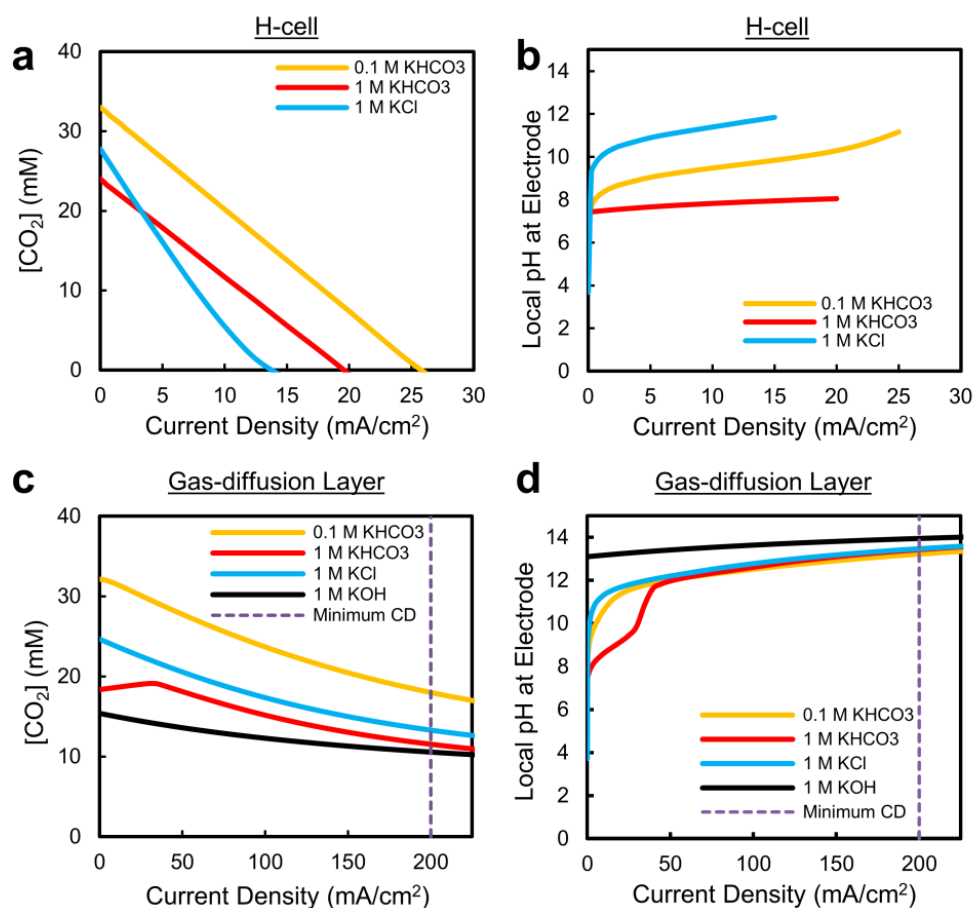
154 The first observation from these curves is that the CO<sub>2</sub> concentration and pH in the electrolyte  
155 at low current densities is relatively similar for both configurations (Fig. 3). This is to be  
156 expected as the decreased CO<sub>2</sub> diffusion pathway does not affect the maximum availability of  
157 CO<sub>2</sub> in an electrolyte, and OH<sup>-</sup> diffusion is moderated by the catholyte. In each case a sudden  
158 increase in local pH is observed for the weakly-buffered electrolytes (1 M KCl,  
159 0.1 M KHCO<sub>3</sub>) as the hydroxide generated as a by-product of water-splitting cannot diffuse  
160 away fast enough or be immediately buffered by the solution (Fig. 3b and d). Shown here, and  
161 in recent experimental work by the Koper group,<sup>42</sup> the electrode pH can in fact shift by as  
162 much as 6 units within the first 1 - 2 mA/cm<sup>2</sup>. Not only could the reaction itself be altered by  
163 this shift, but large pH differences between the reference and working electrodes in this  
164 current density range complicate determination of the equilibrium potential, and subsequently,  
165 product onset potentials and Tafel slopes. Furthermore, for slightly acidic CO<sub>2</sub>-saturated  
166 electrolytes, the solvent itself can also act as the sole proton source (via H<sub>3</sub>O<sup>+</sup>) at current  
167 densities <1 mA/cm<sup>2</sup> until becoming depleted and replaced by water reduction (Fig. 1a). It is  
168 then entirely possible for a CO<sub>2</sub> reduction catalyst to be effective within the Tafel region but  
169 exhibit slower kinetics at higher potentials/current densities if the water reduction step to  
170 liberate protons becomes rate-determining. Within this low current density region, where the  
171 most important electrochemical characterizations of a material take place, the reaction  
172 environment is then extremely sensitive, complicating analysis of intrinsic catalyst behaviour  
173 and the reaction mechanisms at play.

174 In our reaction-diffusion model, the maximum predicted CO<sub>2</sub> reduction current densities in  
175 the aqueous-fed system are again <35 mA/cm<sup>2</sup> for two-electron reduction processes (Fig. 3b).



176 As shown here in Fig. 3d, and in previous gas-diffusion layer modelling work, we can see  
177 however that the maximum current densities are much higher in the case of a gas-diffusion  
178 layer as a result of the reduced CO<sub>2</sub> diffusion distance; here all electrolytes are capable of  
179 sustaining current densities over the proposed 200 mA/cm<sup>2</sup> minimum. This agrees well with  
180 experimental literature where current densities substantially higher than 200 mA/cm<sup>2</sup> have  
181 been reported, including the first report using gas-diffusion layers for CO<sub>2</sub> reduction from  
182 Cook et al. in 1990.<sup>62</sup> In recent work by Dinh et al., for example, current densities of 1 A/cm<sup>2</sup>  
183 were reached in 1 M KOH at 1 atm with an overall CO<sub>2</sub> reduction selectivity over 90%.<sup>70</sup>  
184 While these current densities might not be desired economically due to the corresponding  
185 increase in cell potential required,<sup>1,7</sup> it indicates that substantial CO<sub>2</sub> reduction reaction rates  
186 are possible even if the CO<sub>2</sub> partial pressure is reduced. This is important for future large-  
187 scale devices (>100 cm<sup>2</sup> catalyst areas) where CO<sub>2</sub> partial pressure will decrease as it is  
188 converted in the device. It is also worth noting that the proposed 200 mA/cm<sup>2</sup> minimum  
189 current density is defined primarily for two-electron reduction products. For multi-carbon  
190 products requiring more electrons per CO<sub>2</sub> molecule converted, this current density limit must  
191 be higher to have the same molecular yield per unit area.

192 A final extremely important takeaway from Fig. 3 is that the pH near the electrode will be  
193 inevitably high at commercially-viable current densities (>200 mA/cm<sup>2</sup>) in all of the most  
194 commonly-used electrolytes, regardless of the choice of catalyst or electrolyte (Fig. 3d). Due  
195 to the extremely large quantity of hydroxide generated as a by-product of water reduction and  
196 limitations in transporting hydroxide away from the electrode, we predict the pH within the  
197 catalyst layer will be greater than 12, even for a 1 M KHCO<sub>3</sub> buffered electrolyte. While the use  
198 of a 0.1 M KHCO<sub>3</sub> electrolyte can, and has, been used in an H-cell to approach the local pH  
199 values reached at high current densities (Fig. 2b), these conditions are reached only when CO<sub>2</sub>  
200 is largely depleted and over a small current density/potential range. It is worth noting that the



**Fig. 3:** Simplified predictions of the electrode concentration of CO<sub>2</sub> and pH for commonly-used electrolytes as a function of current density in an (a,b) H-cell with a 50  $\mu\text{m}$  CO<sub>2</sub> diffusion thickness and, (c,d) a gas-diffusion layer with a 50 nm CO<sub>2</sub> diffusion thickness and liquid diffusion layer of 200  $\mu\text{m}$ . A Faradaic efficiency of CO<sub>2</sub>-to-CO of 90% is assumed.

201 locally alkaline conditions could potentially be avoided by using an acidic electrolyte, but the  
 202 reaction kinetics for CO<sub>2</sub> would have to outweigh the heavily-favoured hydrogen evolution  
 203 rate in acidic media. In any of these cases all changes in local pH will also ultimately have to  
 204 be paid for in the overall cell potential, regardless of the locally corrected cathode potential. In  
 205 brief, the results presented in Fig. 3d show a convergence towards local conditions that have  
 206 far-reaching implications for not only catalytic activity, but the stability and maximum  
 207 efficiency of an entire CO<sub>2</sub> reduction system, as discussed in later sections.

208 From existing knowledge about CO<sub>2</sub> reduction, we can infer that the thermodynamics and  
 209 kinetics of reactions on a catalytic surface will be impacted by testing under these elevated  
 210 reaction conditions. Further modelling and direct-measurement experimental studies of the  
 211 local electrolyte environment are of course warranted, however, to even better understand

212 how various factors may influence the reaction (e.g. 2D/3D effects, morphology, partial  
213 pressure of CO<sub>2</sub> in the gas-diffusion layer, etc.). A more detailed representation of CO<sub>2</sub>  
214 diffusion through a porous electrode structure, for instance, may provide further insight but  
215 requires consideration of pore sizes, structures and additives,<sup>77</sup> which are catalyst-specific and  
216 beyond the scope of this article. These studies are particularly needed for zero-gap membrane  
217 electrode assemblies where the extremely-low catholyte volume is expected to make the local  
218 reaction conditions more sensitive to the specific device configuration and the properties of  
219 the ion-exchange membrane.

#### 220 Impact of high current densities on CO<sub>2</sub> reduction catalyst testing

221 In the previous section we discussed the impact that both hydrogen evolution and CO<sub>2</sub>  
222 reduction have on the local environment up to commercially-relevant current densities. This  
223 section discusses the important opposite side of the equation, how does access to surplus CO<sub>2</sub>  
224 and the predicted local environment at higher current densities impact the reactions occurring  
225 on the surface of the catalyst. More importantly, can we use this understanding to design even  
226 better catalysts or conditions that may advance CO<sub>2</sub> electroreduction performance further.

227 In catalyst design the surface of a material is altered as a means of modifying catalytic  
228 behaviour. By changing a surface's composition, nanostructure or even substrate, the binding  
229 energies of molecules to the surface and the reaction rate of different pathways can be tuned  
230 to promote certain reactions, and hopefully, suppress unwanted ones. In electrochemical CO<sub>2</sub>  
231 reduction, the local environment can provide a similar function. Numerous experimental  
232 studies have particularly noted the tendency for higher local pH conditions to favour CO and  
233 multi-carbon products while suppressing H<sub>2</sub> and CH<sub>4</sub> on metals such as Ag and Cu.<sup>32,36</sup> As  
234 shown in Fig. 3d, when operating at higher current densities this effect is pushed to the  
235 extreme, even in highly buffered solutions. An important place to start is then to discuss the  
236 effect of pH on the selectivity of both CO<sub>2</sub> reduction products and hydrogen evolution.

237 Due to its role as the primary competing reaction to CO<sub>2</sub> reduction, H<sub>2</sub> evolution is one of the  
238 most important products to consider. Under basic conditions the H<sub>2</sub> evolution reaction  
239 proceeds through direct water reduction and the Volmer-Tafel or Volmer-Heyrovsky steps  
240 (see EIS Fig. S2). For commonly-used metals such as Cu, Au and Ag, the Volmer step is  
241 particularly sluggish in basic conditions due to both poor water dissociation properties and  
242 weak \*H binding energies, which is pushed even further to the right and away from the peak  
243 of the classical volcano plot under higher pH conditions.<sup>78,79</sup> Simultaneous CO<sub>2</sub> reduction  
244 occurring on a catalyst's surface also tends to further suppress H<sub>2</sub> activity by weakening  
245 hydrogen binding, occupying surface sites and consuming protons.<sup>25,80</sup> For catalysts on the  
246 left side of the hydrogen binding energy volcano curve, however, hydrogen evolution under  
247 locally basic conditions may increase for the same fundamental reasons.

248 For the most commonly-used set of materials, access to >200 mA/cm<sup>2</sup> current densities can  
249 then provide a secondary means of suppressing hydrogen evolution by weakening hydrogen  
250 binding energies under higher local pH conditions. A secondary, non-material approach for  
251 suppressing hydrogen is particularly important for multi-carbon product formation where,  
252 despite achieving 100-fold C<sub>2</sub>:C<sub>1</sub> ratios, many of the best catalysts when operated in an H-  
253 cell are constantly dogged by a persistent 20-30% H<sub>2</sub> selectivity.<sup>62,63,68,69</sup> When these same  
254 catalyst are operated under elevated current densities in the configurations and electrolytes  
255 described here (Fig. 2 and 3), the selectivity towards target products could be increased purely  
256 by penalizing hydrogen evolution, rather than necessarily promoting CO<sub>2</sub> reduction activity.  
257 From Fig. 3 one can expect that any pH-dependent suppression of H<sub>2</sub> would be a function of  
258 current density and buffering strength, with the simultaneous requirement that the  
259 overpotentials needed for CO<sub>2</sub> reduction are also competitive with H<sub>2</sub> evolution.

260 The local pH environment can also separately influence the energetics of different CO<sub>2</sub>  
261 reduction products. For catalysts capable of producing only CO and H<sub>2</sub>, the suppression of H<sub>2</sub>

262 can lead to high CO selectivities. On a Ag catalyst CO formation under alkaline conditions  
263 has been also been observed to be produced almost immediately following the equilibrium  
264 potential of -0.11 V vs RHE<sup>58,67</sup>, indicating that the reaction itself is improved. The behaviour  
265 of CO formation on Cu also differs from that of a more neutral H-cell environment. In H-cell  
266 tests CO selectivities of >20% are observed on Cu and Cu-derivatives only at very low  
267 current densities (<5 mA/cm<sup>2</sup>) before being supplanted by CO<sub>2</sub> reduction to methane,  
268 ethylene and ethanol. Under alkaline conditions performed in a gas-diffusion layer, high CO  
269 selectivities appear over a much broader range (0-200 mA/cm<sup>2</sup>)<sup>70,73</sup>. This suggests that the  
270 binding energy of CO on Cu may be weakened under alkaline conditions, promoting faster  
271 desorption of the formed \*CO intermediate more than under neutral conditions.<sup>70</sup> The  
272 selectivity towards CO does eventually decrease in favour of higher order products similar to  
273 what is observed in an H-cell, but at much higher overall current densities.

274 Similar to H<sub>2</sub> evolution, methane formation on Cu has been shown to be suppressed by  
275 increased local pH on oxide-derived samples. If locally alkaline (pH>12) conditions are  
276 indeed unavoidable with the currently-used electrolytes, then these results suggest that an  
277 alternate reaction mechanism or catalyst may be needed to realize selective methane  
278 formation at elevated current densities. An interesting fundamental result would then be the  
279 demonstration of a catalyst capable of selective methane formation under alkaline or locally  
280 alkaline conditions. Inversely, for multi-carbon product formation on Cu, higher local pH  
281 conditions have been experimentally demonstrated<sup>31,73,83</sup> to be an important factor for  
282 promoting carbon-carbon coupling. The higher observed activity toward multi-carbon species  
283 at lower potentials could be due to the improved CO onset potentials, changes to the binding  
284 energy of \*CO, a direct effect of the local conditions on the energetics of the coupling step  
285 itself, or a cumulative effect of multiple factors. There does not appear to be any strong

286 correlations between pH and product distribution after C-C coupling, however, as most  
287 studies report similar alkane to alkene ratios as in lower current density H-cell experiments.

288 While several studies have operated at elevated current densities using membrane electrode  
289 assemblies or neutral-pH catholytes such as KCl and bicarbonate-based salts,<sup>2,5,12,13,73,84</sup> a  
290 larger number of gas-diffusion layer experiments have used KOH directly as a bulk  
291 catholyte,<sup>58,65-68,70,71,85</sup> with much of the original CO<sub>2</sub> reduction research pioneered by the  
292 Kenis Group. By using an alkaline catholyte directly, the impact of a higher pH environment  
293 on catalyst performance can be probed across both low and high current densities.<sup>65-68,85</sup> In a  
294 1 M KOH environment, CO<sub>2</sub> reduction products have been observed on Cu, Au and Ag  
295 catalysts at earlier overall onset potentials than in neutral conditions, with current densities of  
296 >100 mA/cm<sup>2</sup> having been reached for CO, C<sub>2</sub>H<sub>4</sub> and ethanol at more anodic potentials  
297 than -0.6 V vs. RHE and with <10% H<sub>2</sub> selectivities.<sup>61,65,67,70,86</sup> These experiments, however,  
298 do not pay the same overpotential price associated with the large local pH swing from neutral  
299 to alkaline conditions, which are not taken into account when cathode potential are reported  
300 versus a reversible hydrogen electrode (RHE). Additionally, the interaction between  
301 unreacted CO<sub>2</sub> and hydroxide is problematic for overall stability, as described in the following  
302 section. Using an alkaline electrolyte for testing and characterizing catalyst performance is  
303 however a promising means of mimicking the local environment of high current densities  
304 while being able to measure catalytic activity even at lower current densities.

305 A final consideration for our prediction that many electrolytes will be forced towards high  
306 local pH conditions pertains to catalyst stability. Depending on the properties and composition  
307 of a given catalyst, such conditions may result in the dissolution of metals into solution or the  
308 removal of some species from the surface. While this has not been observed for many of the  
309 Cu and Ag catalysts tested in gas-diffusion layer configurations to date, it should be a  
310 consideration in the design of new catalysts.

311 A second important property of the local reaction environment at commercial-relevant current  
312 densities is access to excess CO<sub>2</sub>. While we have imposed a current density of 200 mA/cm<sup>2</sup> as  
313 a threshold to reach, from Fig. 3c we can see that additional unreacted CO<sub>2</sub> surrounding the  
314 catalyst provides the capacity for even higher current densities. An increased CO<sub>2</sub>  
315 concentration, even at elevated current densities, helps to kinetically ensure that CO<sub>2</sub> rather  
316 than protons are able to populate the reaction surface. More importantly, the reaction will not  
317 be hindered by a deficit of CO<sub>2</sub>, even at higher local pH conditions. While single-carbon CO<sub>2</sub>  
318 reduction products such as CO,<sup>73</sup> HCOOH<sup>88,89</sup> and CH<sub>4</sub><sup>90</sup> can reach relatively high  
319 selectivities even at lower current densities in an H-cell configuration, the highest Faradaic  
320 efficiencies reported for multi-carbon products typically appear at current densities where  
321 CO<sub>2</sub> is almost depleted. This is in part due to the necessity for multi-carbon products to follow  
322 from \*CO (Fig. S2) and C-C coupling which requires both sufficient potentials and current  
323 densities.<sup>48,86,91</sup> Testing novel catalysts at elevated current densities with less CO<sub>2</sub> limitations  
324 would allow the surface coverage of the \*CO reaction intermediate to be maximized over a  
325 wider current and potential range without being kinetically-limited by CO<sub>2</sub> availability. This  
326 is in contrast to current H-cell environments where peak C<sub>2</sub> selectivities are often observed  
327 only at singular operating conditions (i.e. at a specific potential, current density and pH).  
328 Access to a larger operating window then allows more attention to be placed on modifying  
329 catalysts to specifically alter CO<sub>2</sub> reduction product selectivity between higher-order products  
330 such as alkenes vs. alcohols and C<sub>2</sub> vs C<sub>3</sub> products. Finally, operating under an excess CO<sub>2</sub>  
331 environment reduces the overpotential losses associated with transport limitations (sometimes  
332 called concentration polarizations). As a result, plots of voltage versus log(j) have been  
333 observed to remain linear even up to 300 mA/cm<sup>2</sup>, helping to remove one of the barriers to  
334 gauging intrinsic catalyst activity.<sup>17</sup>

335 While the exact implications of high current density catalyst testing will vary slightly with  
336 materials, we can assert that the local environment will differ greatly from the bulk electrolyte  
337 with a substantial chance of changing important surface kinetics and the observed catalytic  
338 activity. Tuning catalysts to optimize morphology or surface binding energies for low current  
339 density characterization may then risk optimizing the catalyst for the wrong environment  
340 unless higher current conditions can be appropriately mimicked. Performing such  
341 experiments adds an additional degree of complexity due to the need for researchers to adopt  
342 either a pressurized system or a gas-diffusion layer to supply additional CO<sub>2</sub> (Fig. 2), in  
343 addition to developing new catalysts. We believe, however, that this to be an essential step to  
344 make the best (highest activity, selectivity and stability) CO<sub>2</sub> reduction catalyst possible.  
345 Fortunately, the range of experimental reports already performed at higher current densities  
346 indicate that CO<sub>2</sub> reduction is typically improved versus operating in the same electrolyte at  
347 lower current densities, at least in part due to increased CO<sub>2</sub> availability and suppression of  
348 pH-independent products.

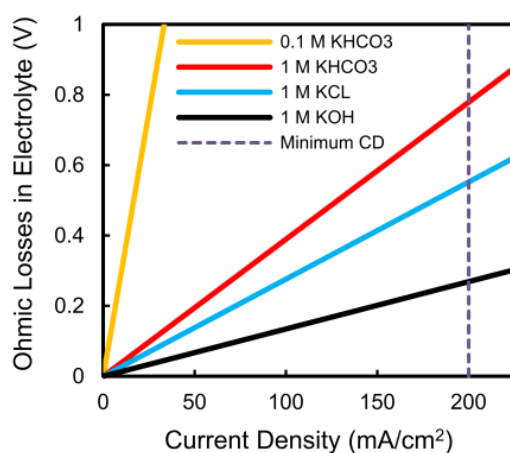
#### 349 Impact of high current densities on system design

350 Up to this point we have discussed the impacts of configuration and reaction conditions on the  
351 CO<sub>2</sub> reduction reaction and vice versa. While understanding the CO<sub>2</sub> reduction reaction and  
352 catalyst over a range of conditions is important, our preferred choice of catalyst and catholyte  
353 will have implications and constraints for the system as a whole. Further catalyst testing in a  
354 more commercial-type reactor will help to bring operational issues to the forefront of the field  
355 and may result in new ingenious scientific and engineering solutions to these issues.

356 One such issue pertains to the twist of fate that the most efficient CO<sub>2</sub> reduction conditions  
357 appear to occur when a highly-alkaline electrolyte is used as a catholyte. While an alkaline  
358 catholyte may provide optimal cathode performance, it comes at the cost of system stability  
359 due to the interaction between unreacted CO<sub>2</sub> and hydroxide in the electrolyte, particularly at



360 current densities less than the CO<sub>2</sub> limiting case where higher concentrations of CO<sub>2</sub> exist in  
361 solution. Our reaction-diffusion model shows that this interaction can decrease the pH within  
362 the porous catalyst layer by 1-2 units at lower current densities depending on the  
363 replenishment rate from a bulk liquid phase (Fig. 3d).<sup>17</sup> While this pH change can be  
364 accounted for when trying to ascertain the intrinsic activity of the catalyst, a portion of the  
365 CO<sub>2</sub> crossing the gas-liquid interface will be converted to bicarbonate upon interacting with  
366 hydroxide and then carbonate.<sup>61</sup> Not only does this decrease CO<sub>2</sub> utilization, but over a long  
367 enough operating time it will destroy the expensive KOH catholyte, itself energy-intensively  
368 produced through electrochemical reactions. At the moment there is no engineering solution  
369 to completely mitigate this effect even at small scales, let alone a more complex >100 cm<sup>2</sup>  
370 reactor design. We may then be resigned to the use of neutral-pH catholytes which to date  
371 would represent an increase in expected cathode overpotentials. Further, overall cell potentials  
372 will be higher due to the need for the oxygen evolution reaction to occur in a similar pH  
373 electrolyte, or by using a bipolar membrane to maintain an alkaline anolyte. While CO<sub>2</sub>-  
374 hydroxide interactions are typically only considered as a critical issue for alkaline catholytes  
375 such as KOH, systems using neutral electrolytes should also aim to balance the generated  
376 hydroxide ions with protons generated by the anode reaction.<sup>92</sup> Even in a neutral-pH medium



**Fig. 4:** Expected ohmic losses as a function of current density for commonly-used electrolytes in an electrochemical cell with a combined 3 mm catholyte and anolyte thickness at 25 °C.

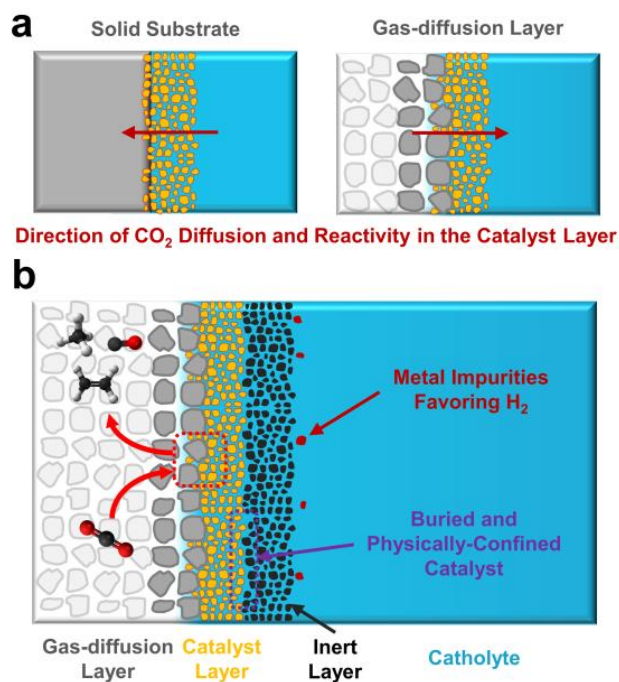
377 the system's electrolyte balance would similarly be steadily shifted away from the initial  
378 condition due our CO<sub>2</sub> reagent that can influence pH; in this case, however, the electrolyte  
379 could be externally regenerated without additional energy input.

380 Another cell design issue with using alkaline electrolytes is the need for anion exchange  
381 membranes, which have comparatively slower ion transport than proton exchange membranes  
382 and overall limited mobility for both bicarbonate and carbonate anions.<sup>11,81</sup> For these reasons  
383 a large amount of research is being undertaken to improve OH<sup>-</sup>, HCO<sub>3</sub><sup>-</sup> and CO<sub>3</sub><sup>2-</sup> transport  
384 through anion exchange membranes, with a fair amount of work done by Dioxide Materials'  
385 and their Sustainion<sup>®</sup> membrane which has shown 1000's of hours of stability, albeit under  
386 specific operating conditions.<sup>25,82</sup> Without sufficient bicarbonate/carbonate transport through  
387 the membrane, the concentration of buffering ions will increase over time, forcing co-ion  
388 transport through the membrane and resulting in electrolyte precipitation, destroying the  
389 membrane and/or the gas-diffusion layer. Without solutions to these issues it will be difficult  
390 for either membrane electrode assemblies or alkaline catholytes to be paired with anion  
391 exchange membranes in a practical CO<sub>2</sub> reduction device.

392 An additional reaction constraint brought on by the need for >200 mA/cm<sup>2</sup> operation pertains  
393 to the practical choice of electrolytes, independent of their impact on catalytic activity. Using  
394 this minimum current density as a target threshold we can approximate the expected ohmic  
395 losses of commonly-used electrolytes at 25°C, regardless of the catalyst or substrate used.  
396 Assuming a combined catholyte and anolyte thickness of 3 mm, for example, it is clear that  
397 certain electrolytes will cause prohibitive ohmic losses (Fig. 4). The 0.1 M KHCO<sub>3</sub> electrolyte  
398 used in the majority of the highest selectivity C<sub>2</sub>H<sub>4</sub> studies, for instance, results in heating  
399 losses of 6 V at 200 mA/cm<sup>2</sup>, five-fold larger than the thermodynamic cell potential when  
400 using an oxygen-evolving anode (Fig. S3). Such a low conductivity electrolyte can then never  
401 be used in a functioning system unless the electrolyte pathway between the anode and cathode

402 is eliminated or greatly reduced, as in the case of membrane electrode assemblies (Fig. 2c).<sup>2,5</sup>  
403 These ohmic drops also put into perspective how other cell losses may be more influential  
404 than further decreases to cathode overpotentials. It is worth noting that the result in Fig. 4  
405 becomes even more pronounced when the presumed even higher current densities needed for  
406 multi-carbon products such as C<sub>2</sub>H<sub>4</sub> are considered. Finally, separate from our discussion  
407 motivating the testing of catalysts at elevated current densities, this result highlights the need  
408 to test catalysts in higher conductivity electrolytes to see the effect that a higher concentration  
409 of supporting ions may or may not impact a catalyst's activity.

410 A final notable difference to performing CO<sub>2</sub> electroreduction in an H-cell versus a gas-  
411 diffusion layer configuration is subtler. As has been recently demonstrated, under high current  
412 density operation the CO<sub>2</sub> reduction reaction is confined to a relatively small portion of a  
413 porous catalyst layer and takes place on the backside of the material (Fig. 5a).<sup>70</sup> This is in  
414 contrast to an H-cell where CO<sub>2</sub> diffuses to the catalyst layer from the front-side bulk  
415 electrolyte (Fig. 5a). Much like the effects of varying morphology and porosity in H-cell  
416 catalyst studies allowed for performance to be improved and varied, the reversed flow  
417 direction of CO<sub>2</sub> transport allows for unique gas-diffusion and catalyst layer engineering to  
418 take place. One recent example is placing an inactive material on top of the catalyst layer  
419 (Fig. 5a) to provide an additionally conductive layer. While this layer is used as a current  
420 collector, it may also help to prevent both catalyst restructuring that can change catalytic  
421 behaviour as well as the deposition of contaminants on the active catalyst surface.  
422 Contaminants are a problem largely unique to CO<sub>2</sub> reduction due to the tendency for most  
423 electrolytes to contain minute concentrations of Ni, Fe or Co, metals capable of preferentially  
424 promoting hydrogen evolution even at low potentials and are a primary source of catalyst  
425 deactivation.<sup>2,95,96</sup> One could further imagine layering catalyst structures on gas-diffusion  
426 layers for either unique catalyst designs or supportive layers that provide new functionalities.



**Fig. 5:** (a) Schematic of the directional change in CO<sub>2</sub> transport for fully-aqueous and gas-diffusion layer CO<sub>2</sub> reduction catalysts. (b) Potential advantages of a change in the CO<sub>2</sub> transport and reaction direction inside a catalyst layer. A buried catalyst layer may be inherently more stable and protected from both contaminants and restructuring while still having access to CO<sub>2</sub>.

427 While many possibilities have yet to be explored for CO<sub>2</sub> reduction, a gas-diffusion layer  
 428 provides a porous electrode structure which fundamentally changes both reagent and product  
 429 transport pathways compared to the impermeable supports traditionally used in fully-aqueous  
 430 reactors. As adoption increases substantial opportunity then exists for further innovative  
 431 catalyst-support integrations to improve either stability or performance. The subtle operational  
 432 difference also means the traditionally described benefits of catalyst nanostructuring (for  
 433 increased surface area, mass transport, facet exposure, etc.) have to be somewhat reassessed,  
 434 which may be difficult as the primary active surface is no longer easily accessible to surface  
 435 characterization techniques.

### 436 Conclusion

437 In summary, in this article we have shown that the minimum current density requirements for  
 438 future commercial systems will ultimately drive CO<sub>2</sub> reduction away from the operating  
 439 conditions where much of the cutting-edge catalyst research has been performed.

440 Acknowledging these realities and testing state-of-the-art catalysts under these more realistic  
441 operating conditions will be important not only for further improving performance metrics  
442 such as selectivity, activity and stability, but to accelerate research towards commercial  
443 applications that are exceedingly needed sooner rather than later. It is encouraging that there  
444 has been a recent noticeable shift in literature towards more system-integrated testing  
445 platforms for electrochemical CO<sub>2</sub> reduction, and we hope that this new perspective further  
446 motivates adoption and helps incite new discoveries.

#### 447 **Conflicts of Interest**

448 There are no conflicts of interest to declare.

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#### 456 **References**

- 457 1 M. Jouny, W. Luc and F. Jiao, *Ind. Eng. Chem. Res.*, 2018, **57**, 2165–2177.
- 459 2 Y. C. Li, D. Zhou, Z. Yan, R. H. Gonçalves, D. A. Salvatore, C. P. Berlinguette and T. E. Mallouk,  
460 *ACS Energy Lett.*, 2016, **1**, 1149–1153.
- 461 3 H.-R. “Molly” Jhong, F. R. Brushett and P. J. A. Kenis, *Adv. Energy Mater.*, 2013, **3**, 589–599.
- 462 4 T. Li, Y. Cao, J. He and C. P. Berlinguette, *ACS Cent. Sci.*, 2017, **3**, 778–783.
- 463 5 D. A. Salvatore, D. M. Weekes, J. He, K. E. Dettelbach, Y. C. Li, T. E. Mallouk and C. P.  
464 Berlinguette, *ACS Energy Lett.*, 2018, **3**, 149–154.
- 465 6 T. Haas, R. Krause, R. Weber, M. Demler and G. Schmid, *Nat. Catal.*, 2018, **1**, 32–39.
- 466 7 S. Verma, B. Kim, H.-R. “Molly” Jhong, S. Ma and P. J. A. Kenis, *ChemSusChem*, 2016, **9**, 1972–  
467 1979.
- 468 8 J. M. Spurgeon and B. Kumar, *Energy Environ. Sci.*, 2018, **11**, 1536–1551.
- 469 9 K. Jiang, S. Siahrostami, T. Zheng, Y. Hu, S. Hwang, E. Stavitski, Y. Peng, J. Dynes, M.  
470 Gangisetty, D. Su, K. Attenkofer and H. Wang, *Energy Environ. Sci.*, DOI:10.1039/C7EE03245E.
- 471 10 W. Zhang, Y. Hu, L. Ma, G. Zhu, Y. Wang, X. Xue, R. Chen, S. Yang and Z. Jin, *Adv. Sci.*, 2018,  
472 **5**, n/a-n/a.

473 11 F. Li, L. Chen, G. P. Knowles, D. R. MacFarlane and J. Zhang, *Angew. Chem. Int. Ed.*, **56**, 505–  
474 509.

475 12 D. M. Weekes, D. A. Salvatore, A. Reyes, A. Huang and C. P. Berlinguette, *Acc. Chem. Res.*, 2018,  
476 **51**, 910–918.

477 13 B. Endrődi, G. Bencsik, F. Darvas, R. Jones, K. Rajeshwar and C. Janáky, *Prog. Energy Combust.*  
478 *Sci.*, 2017, **62**, 133–154.

479 14 D. Kim, J. Resasco, Y. Yu, A. M. Asiri and P. Yang, *Nat. Commun.*, 2014, **5**, 4948.

480 15 X. Liu, J. Xiao, H. Peng, X. Hong, K. Chan and J. K. Nørskov, *Nat. Commun.*, 2017, **8**, 15438.

481 16 H. Mistry, R. Reske, Z. Zeng, Z.-J. Zhao, J. Greeley, P. Strasser and B. R. Cuenya, *J. Am. Chem.*  
482 *Soc.*, 2014, **136**, 16473–16476.

483 17 R. Reske, H. Mistry, F. Behafarid, B. Roldan Cuenya and P. Strasser, *J. Am. Chem. Soc.*, 2014,  
484 **136**, 6978–6986.

485 18 W. Zhu, Y.-J. Zhang, H. Zhang, H. Lv, Q. Li, R. Michalsky, A. A. Peterson and S. Sun, *J. Am.*  
486 *Chem. Soc.*, 2014, **136**, 16132–16135.

487 19 W. Zhu, R. Michalsky, Ö. Metin, H. Lv, S. Guo, C. J. Wright, X. Sun, A. A. Peterson and S. Sun, *J.*  
488 *Am. Chem. Soc.*, 2013, **135**, 16833–16836.

489 20 S. Liu, X.-Z. Wang, H. Tao, T. Li, Q. Liu, Z. Xu, X.-Z. Fu and J.-L. Luo, *Nano Energy*, 2018, **45**,  
490 456–462.

491 21 S. Sen, D. Liu and G. T. R. Palmore, *ACS Catal.*, 2014, **4**, 3091–3095.

492 22 Loiudice Anna, Lobaccaro Peter, Kamali Esmail A., Thao Timothy, Huang Brandon H., Ager Joel  
493 W. and Buonsanti Raffaella, *Angew. Chem. Int. Ed.*, 2016, **55**, 5789–5792.

494 23 A. S. Hall, Y. Yoon, A. Wuttig and Y. Surendranath, *J. Am. Chem. Soc.*, 2015, **137**, 14834–14837.

495 24 X. Min and M. W. Kanan, *J. Am. Chem. Soc.*, 2015, **137**, 4701–4708.

496 25 E. R. Cave, C. Shi, K. P. Kuhl, T. Hatsukade, D. N. Abram, C. Hahn, K. Chan and T. F. Jaramillo,  
497 *ACS Catal.*, 2018, 3035–3040.

498 26 B. Khezri, A. C. Fisher and M. Pumera, *J. Mater. Chem. A*, 2017, **5**, 8230–8246.

499 27 X. Duan, J. Xu, Z. Wei, J. Ma, S. Guo, S. Wang, H. Liu and S. Dou, *Adv. Mater.*, 2017, **29**, n/a-n/a.

500 28 J. Wu, Y. Huang, W. Ye and Y. Li, *Adv. Sci.*, 2017, **4**, n/a-n/a.

501 29 He Jingfu, Johnson Noah J. J., Huang Aoxue and Berlinguette Curtis P., *ChemSusChem*, 2017, **11**,  
502 48–57.

503 30 F. Li, D. R. MacFarlane and J. Zhang, *Nanoscale*, , DOI:10.1039/C7NR09620H.

504 31 K. J. P. Schouten, E. Pérez Gallent and M. T. M. Koper, *J. Electroanal. Chem.*, 2014, **716**, 53–57.

505 32 Y. Pang, T. Burdyny, C.-T. Dinh, M. G. Kibria, J. Z. Fan, M. Liu, E. H. Sargent and D. Sinton,  
506 *Green Chem.*, 2017, **19**, 4023–4030.

507 33 Pander James E., Ren Dan, Huang Yun, Loo Nicholas Wei Xian, Hong Samantha Hui Lee and Yeo  
508 Boon Siang, *ChemElectroChem*, 2017, **5**, 219–237.

509 34 Y. Zhang, X. Zhang, A. M. Bond and J. Zhang, *Phys. Chem. Chem. Phys.*, 2018, **20**, 5936–5941.

510 35 C. F. C. Lim, D. A. Harrington and A. T. Marshall, *Electrochimica Acta*, 2017, **238**, 56–63.

511 36 R. Kas, R. Kortlever, H. Yilmaz, M. T. M. Koper and G. Mul, *ChemElectroChem*, 2015, **2**, 354–  
512 358.

513 37 A. S. Varela, M. Kroschel, T. Reier and P. Strasser, *Catal. Today*, 2016, **260**, 8–13.

514 38 J. Resasco, L. D. Chen, E. Clark, C. Tsai, C. Hahn, T. F. Jaramillo, K. Chan and A. T. Bell, *J. Am.*  
515 *Chem. Soc.*, 2017, **139**, 11277–11287.

516 39 O. Ayemoba and A. Cuesta, *ACS Appl. Mater. Interfaces*, 2017, **9**, 27377–27382.

517 40 E. Pérez-Gallent, G. Marcandalli, M. C. Figueiredo, F. Calle-Vallejo and M. T. M. Koper, *J. Am.*  
518 *Chem. Soc.*, 2017, **139**, 16412–16419.

519 41 M. R. Singh, Y. Kwon, Y. Lum, J. W. Ager and A. T. Bell, *J. Am. Chem. Soc.*, 2016, **138**, 13006–  
520 13012.

521 42 H. Ooka, M. C. Figueiredo and M. T. M. Koper, *Langmuir*, 2017, **33**, 9307–9313.  
522 43 J. D. Goodpaster, A. T. Bell and M. Head-Gordon, *J. Phys. Chem. Lett.*, 2016, **7**, 1471–1477.  
523 44 M. R. Singh, J. D. Goodpaster, A. Z. Weber, M. Head-Gordon and A. T. Bell, *Proc. Natl. Acad. Sci.*, 2017, **114**, E8812–E8821.  
524  
525 45 N. Gupta, M. Gattrell and B. MacDougall, *J. Appl. Electrochem.*, 2006, **36**, 161–172.  
526 46 M. R. Singh, E. L. Clark and A. T. Bell, *Phys. Chem. Chem. Phys.*, 2015, **17**, 18924–18936.  
527 47 T. Cheng, H. Xiao and W. A. Goddard, *Proc. Natl. Acad. Sci.*, 2017, **114**, 1795–1800.  
528 48 J. H. Montoya, C. Shi, K. Chan and J. K. Nørskov, *J. Phys. Chem. Lett.*, 2015, **6**, 2032–2037.  
529 49 Ma Ming, Trzeźniowski Bartek J., Xie Jie and Smith Wilson A., *Angew. Chem. Int. Ed.*, 2016, **55**,  
530 9748–9752.  
531 50 D. Kim, C. S. Kley, Y. Li and P. Yang, *Proc. Natl. Acad. Sci.*, 2017, **114**, 10560–10565.  
532 51 P. D. Luna, R. Quintero-Bermudez, C.-T. Dinh, M. B. Ross, O. S. Bushuyev, P. Todorović, T.  
533 Regier, S. O. Kelley, P. Yang and E. H. Sargent, *Nat. Catal.*, 2018, **1**, 103–110.  
534 52 H. Mistry, A. S. Varela, C. S. Bonifacio, I. Zegkinoglou, I. Sinev, Y.-W. Choi, K. Kisslinger, E. A.  
535 Stach, J. C. Yang, P. Strasser and B. R. Cuenya, *Nat. Commun.*, 2016, **7**, 12123.  
536 53 A. D. Handoko, C. W. Ong, Y. Huang, Z. G. Lee, L. Lin, G. B. Panetti and B. S. Yeo, *J. Phys.*  
537 *Chem. C*, 2016, **120**, 20058–20067.  
538 54 D. Ren, Y. Deng, A. D. Handoko, C. S. Chen, S. Malkhandi and B. S. Yeo, *ACS Catal.*, 2015, **5**,  
539 2814–2821.  
540 55 Y. Lum, B. Yue, P. Lobaccaro, A. T. Bell and J. W. Ager, *J. Phys. Chem. C*, 2017, **121**, 14191–  
541 14203.  
542 56 R. Kortlever, J. Shen, K. J. P. Schouten, F. Calle-Vallejo and M. T. M. Koper, *J. Phys. Chem. Lett.*,  
543 2015, **6**, 4073–4082.  
544 57 T. Burdyny, P. J. Graham, Y. Pang, C.-T. Dinh, M. Liu, E. H. Sargent and D. Sinton, *ACS Sustain.*  
545 *Chem. Eng.*, 2017, **5**, 4031–4040.  
546 58 C. M. Gabardo, A. Seifitokaldani, J. P. Edwards, C.-T. Dinh, T. Burdyny, M. G. Kibria, C. P.  
547 O’Brien, E. H. Sargent and D. Sinton, *Energy Environ. Sci.*, , DOI:10.1039/C8EE01684D.  
548 59 A. Seifitokaldani, C. M. Gabardo, T. Burdyny, C.-T. Dinh, J. P. Edwards, M. G. Kibria, O. S.  
549 Bushuyev, S. O. Kelley, D. Sinton and E. H. Sargent, *J. Am. Chem. Soc.*, 2018, **140**, 3833–3837.  
550 60 L.-C. Weng, A. T. Bell and A. Z. Weber, *Phys. Chem. Chem. Phys.*, 2018, **20**, 16973–16984.  
551 61 C.-T. Dinh, F. P. Garcia de Arquer, D. Sinton and E. H. Sargent, *ACS Energy Lett.*, ,  
552 DOI:10.1021/acseenergylett.8b01734.  
553 62 R. L. Cook, R. C. MacDuff and A. F. Sammells, *J. Electrochem. Soc.*, 1990, **137**, 607–608.  
554 63 H. Yano, F. Shirai, M. Nakayama and K. Ogura, *J. Electroanal. Chem.*, 2002, **519**, 93–100.  
555 64 C. Delacourt, P. L. Ridgway, J. B. Kerr and J. Newman, *J. Electrochem. Soc.*, 2008, **155**, B42–B49.  
556 65 S. Ma, R. Luo, J. I. Gold, A. Z. Yu, B. Kim and P. J. A. Kenis, *J. Mater. Chem. A*, 2016, **4**, 8573–  
557 8578.  
558 66 B. Kim, F. Hillman, M. Ariyoshi, S. Fujikawa and P. J. A. Kenis, *J. Power Sources*, 2016, **312**,  
559 192–198.  
560 67 S. Verma, X. Lu, S. Ma, R. I. Masel and P. J. A. Kenis, *Phys. Chem. Chem. Phys.*, 2016, **18**, 7075–  
561 7084.  
562 68 E. J. Dufek, T. E. Lister and M. E. McIlwain, *Electrochem. Solid-State Lett.*, 2012, **15**, B48–B50.  
563 69 T. T. H. Hoang, S. Ma, J. I. Gold, P. J. A. Kenis and A. A. Gewirth, *ACS Catal.*, 2017, **7**, 3313–  
564 3321.  
565 70 C.-T. Dinh, T. Burdyny, M. G. Kibria, A. Seifitokaldani, C. M. Gabardo, F. P. G. de Arquer, A.  
566 Kiani, J. P. Edwards, P. D. Luna, O. S. Bushuyev, C. Zou, R. Quintero-Bermudez, Y. Pang, D.  
567 Sinton and E. H. Sargent, *Science*, 2018, **360**, 783–787.

568 71 T.-T. Zhuang, Z.-Q. Liang, A. Seifitokaldani, Y. Li, P. D. Luna, T. Burdyny, F. Che, F. Meng, Y.  
569 Min, R. Quintero-Bermudez, C. T. Dinh, Y. Pang, M. Zhong, B. Zhang, J. Li, P.-N. Chen, X.-L.  
570 Zheng, H. Liang, W.-N. Ge, B.-J. Ye, D. Sinton, S.-H. Yu and E. H. Sargent, *Nat. Catal.*, 2018, **1**,  
571 421–428.

572 72 E. Irtem, T. Andreu, A. Parra, M. D. Hernández-Alonso, S. García-Rodríguez, J. M. Riesco-García,  
573 G. Penelas-Pérez and J. R. Morante, *J. Mater. Chem. A*, 2016, **4**, 13582–13588.

574 73 J.-J. Lv, M. Jouny, W. Luc, W. Zhu, J.-J. Zhu and F. Jiao, *Adv. Mater.*, **0**, 1803111.

575 74 Z. Liu, R. I. Masel, Q. Chen, R. Kutz, H. Yang, K. Lewinski, M. Kaplun, S. Luopa and D. R. Lutz,  
576 *J. CO2 Util.*, 2016, **15**, 50–56.

577 75 W. Lee, Y. E. Kim, M. H. Youn, S. K. Jeong and K. T. Park, *Angew. Chem. Int. Ed.*, **57**, 6883–  
578 6887.

579 76 P. Jeanty, C. Scherer, E. Magori, K. Wiesner-Fleischer, O. Hinrichsen and M. Fleischer, *J. CO2*  
580 *Util.*, 2018, **24**, 454–462.

581 77 Analysis of Oxygen-Transport Diffusion Resistance in Proton-Exchange-Membrane Fuel Cells,  
582 <http://jes.ecsdl.org/content/158/4/B416.abstract>, (accessed September 17, 2018).

583 78 W. Sheng, M. Myint, J. G. Chen and Y. Yan, *Energy Environ. Sci.*, 2013, **6**, 1509–1512.

584 79 J. K. Nørskov, T. Bligaard, A. Logadottir, J. R. Kitchin, J. G. Chen, S. Pandelov and U. Stimming,  
585 *J. Electrochem. Soc.*, 2005, **152**, J23–J26.

586 80 Y.-J. Zhang, V. Sethuraman, R. Michalsky and A. A. Peterson, *ACS Catal.*, 2014, **4**, 3742–3748.

587 81 K. Jiang, R. B. Sandberg, A. J. Akey, X. Liu, D. C. Bell, J. K. Nørskov, K. Chan and H. Wang, *Nat.*  
588 *Catal.*, 2018, **1**, 111–119.

589 82 D. Ren, J. Fong and B. S. Yeo, *Nat. Commun.*, 2018, **9**, 925.

590 83 M. G. Kibria, C.-T. Dinh, A. Seifitokaldani, P. D. Luna, T. Burdyny, R. Quintero-Bermudez, M. B.  
591 Ross, O. S. Bushuyev, F. P. G. de Arquer, P. Yang, D. Sinton and E. H. Sargent, *Adv. Mater.*, **0**,  
592 1804867.

593 84 A. D. Castillo, M. Alvarez-Guerra and A. Irabien, *AIChE J.*, **60**, 3557–3564.

594 85 S. Ma, R. Luo, S. Moniri, Y. Lan and P. J. A. Kenis, *J. Electrochem. Soc.*, 2014, **161**, F1124–  
595 F1131.

596 86 S. Ma, M. Sadakiyo, R. Luo, M. Heima, M. Yamauchi and P. J. A. Kenis, *J. Power Sources*, 2016,  
597 **301**, 219–228.

598 87 Q. Li, J. Fu, W. Zhu, Z. Chen, B. Shen, L. Wu, Z. Xi, T. Wang, G. Lu, J. Zhu and S. Sun, *J. Am.*  
599 *Chem. Soc.*, 2017, **139**, 4290–4293.

600 88 X. Zheng, P. De Luna, F. P. García de Arquer, B. Zhang, N. Becknell, M. B. Ross, Y. Li, M. N.  
601 Bani, Y. Li, M. Liu, O. Voznyy, C. T. Dinh, T. Zhuang, P. Stadler, Y. Cui, X. Du, P. Yang and E.  
602 H. Sargent, *Joule*, 2017, **1**, 794–805.

603 89 A. Klinkova, P. De Luna, C.-T. Dinh, O. Voznyy, E. M. Larin, E. Kumacheva and E. H. Sargent,  
604 *ACS Catal.*, 2016, **6**, 8115–8120.

605 90 H. Hashiba, H. K. Sato, S. Yotsuhashi, K. Fujii, M. Sugiyama and Y. Nakano, *Sustain. Energy*  
606 *Fuels*, 2017, **1**, 1734–1739.

607 91 Y. Huang, A. D. Handoko, P. Hirunsit and B. S. Yeo, *ACS Catal.*, 2017, **7**, 1749–1756.

608 92 S. Hernández, M. A. Farkhondehfal, F. Sastre, M. Makkee, G. Saracco and N. Russo, *Green Chem.*,  
609 2017, **19**, 2326–2346.

610 93 J. R. Varcoe, P. Atanassov, D. R. Dekel, A. M. Herring, M. A. Hickner, P. A. Kohl, A. R.  
611 Kucernak, W. E. Mustain, K. Nijmeijer, K. Scott, T. Xu and L. Zhuang, *Energy Environ. Sci.*,  
612 2014, **7**, 3135–3191.

613 94 J. J. Kaczur, H. Yang, Z. Liu, S. D. Sajjad and R. I. Masel, *Front. Chem.*, ,  
614 DOI:10.3389/fchem.2018.00263.

615 95 Y. Hori, H. Konishi, T. Futamura, A. Murata, O. Koga, H. Sakurai and K. Oguma, *Electrochimica*  
616 *Acta*, 2005, **50**, 5354–5369.



617 96 A. Wuttig and Y. Surendranath, *ACS Catal.*, 2015, **5**, 4479–4484.  
618